

As-fabricated Nitride and Silicide Coated U-Mo Atomized Particles Complementary Microstructural Analyses

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H. Palancher, X. Iltis, H. J. Ryu, K. H. Lee, Y. J. Jeong, et al.. As-fabricated Nitride and Silicide Coated U-Mo Atomized Particles Complementary Microstructural Analyses. RERTR 2015- The 36th International Meeting on Reduced Enrichment for Research and Test Reactors, Oct 2015, Seoul, South Korea. hal-02445717

HAL Id: hal-02445717 https://cea.hal.science/hal-02445717v1

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AS-FABRICATED NITRIDE AND SILICIDE COATED U-MO ATOMIZED PARTICLES: COMPLEMENTARY MICROSTRUCTURAL ANALYSES

H. Palancher, X. Iltis

CEA, DEN, DEC, F-13108 Saint Paul Lez Durance Cedex - France

H.J. Ryu



Korea Advanced Institute of Science and Technology, 291 Daehakro, Yuseong, Daejeon 305-701 - Republic of Korea

K.H. Lee, Y.J. Jeong, J.M Park



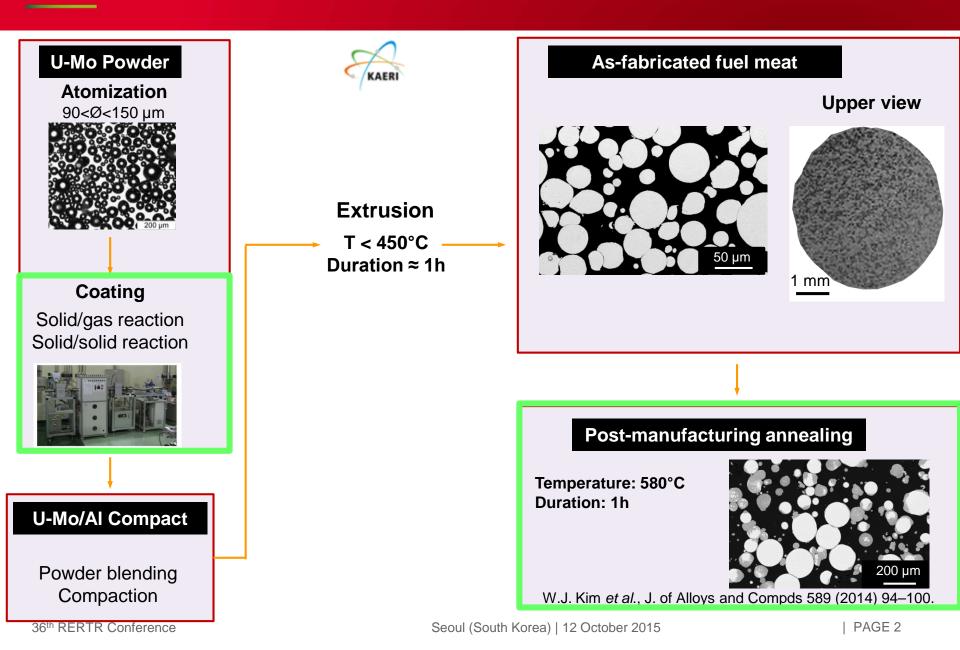
Research Reactor Fuel Development Division

Korea Atomic Energy Research Institute, 989-111 Daedeokdaero, Yuseong, 305-353 Daejeon – Republic of Korea

M. Delpech CEA, DEN, DISN, 91191 Gif sur Yvette– France

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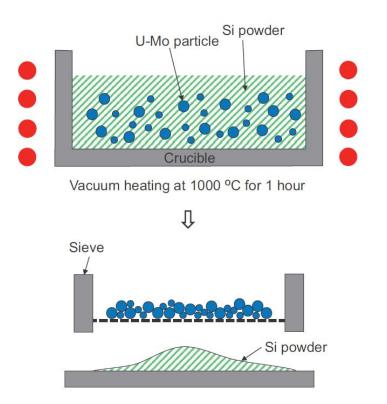
U-MO/AL SAMPLE MANUFACTURING (NO-CLADDING)

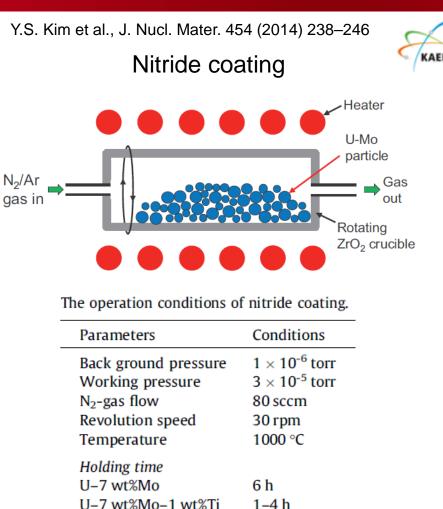


COATING CONDITIONS FOR THE U-MO POWDER (KOMO-5 TEST)

H.J. Ryu et al., Nucl. Eng. Technol. 43 (2011) 159-166.

Si coating



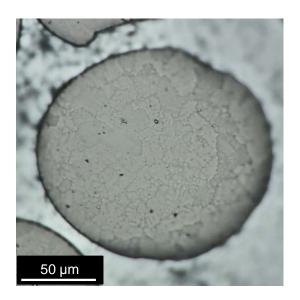


Goals of the characterisation: coating composition but also U-Mo grain size modification

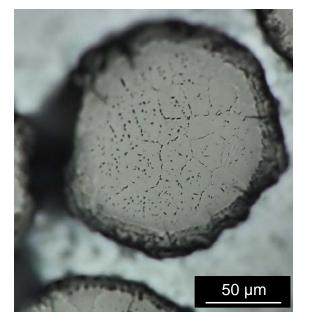


GRAIN SIZE EVALUATION USING OPTICAL MICROSCOPY

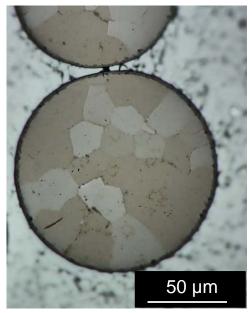
As-atomised



U-Mo(Si) (1000°C 1h)



U-Mo(N) (1000°C 6h)



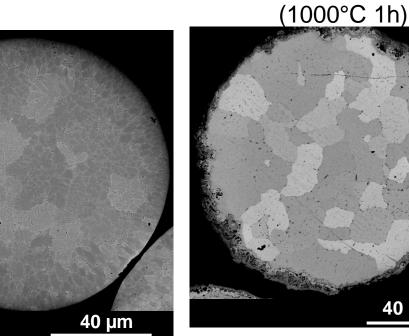
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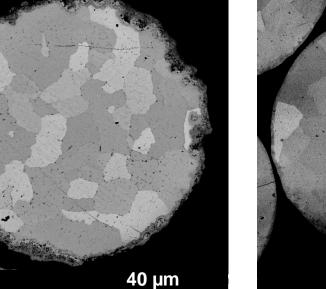


GRAIN SIZE EVALUATION USING SEM

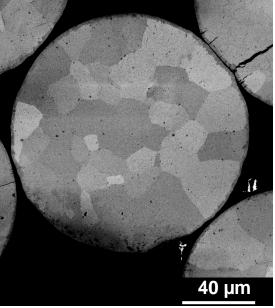
U-Mo(Si)

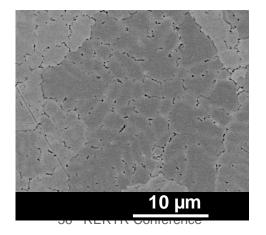
As-atomised





U-Mo(N) (1000°C 6h)



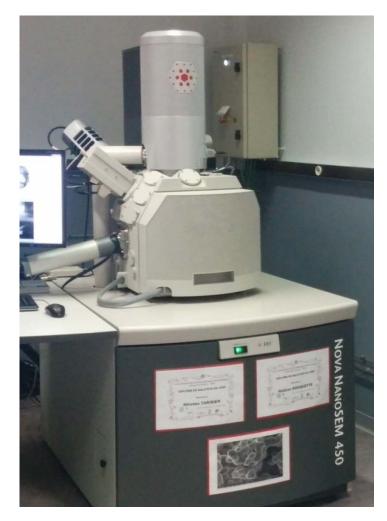


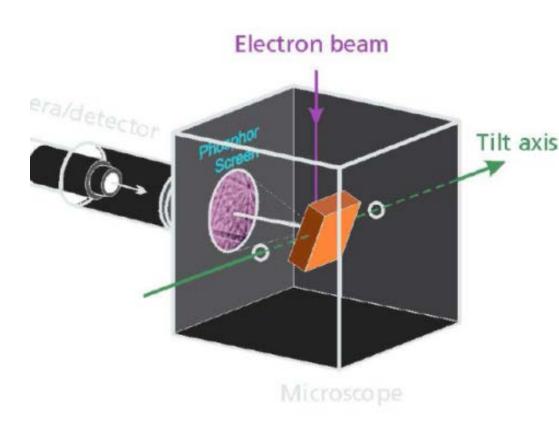
Even with high quality SEM characterisation combined with image analysis, grain size is difficult to determine accurately.

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ELECTRON BACK-SCATTER DIFFRACTION (EBSD)





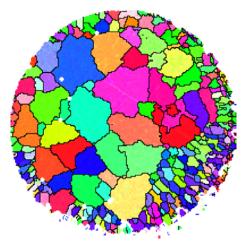
For further technical details on the application of this technique on as-atomised UMo nuclear fuel: See: *G. Champion, Ph.D. Thesis (Rennes University) 2013 (France).*

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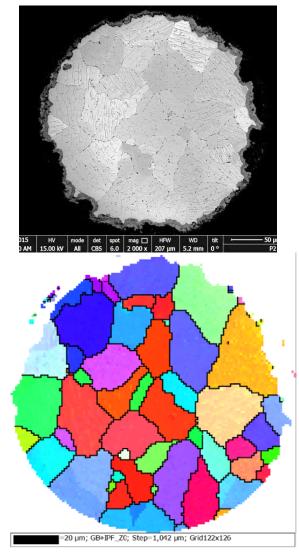
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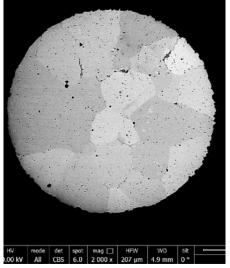
As-atomised

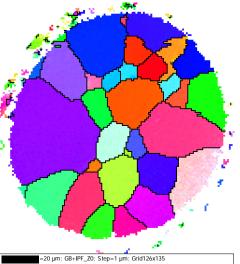


U-Mo(Si) (1000°C 1h)

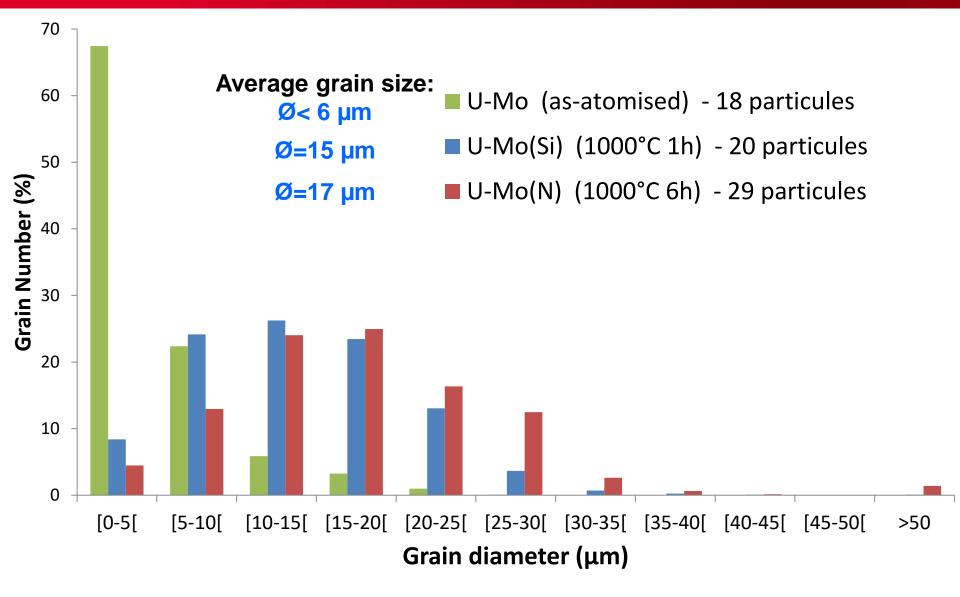


U-Mo(N) (1000°C 6h)





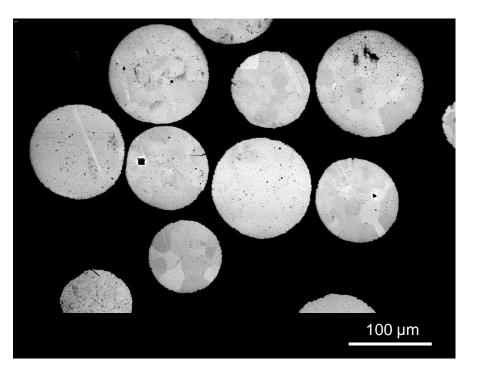






U-MO MICROSTRUCTURE AFTER TT AT 1000°C DURING 6H: LARGE EBSD MAPS

SEM



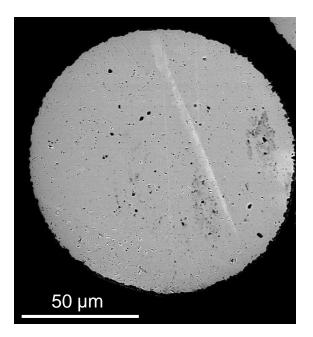
EBSD

Evidence for the presence of « single crystal » U-Mo particles

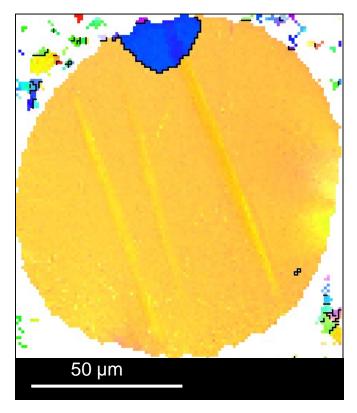
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MICROSTRUCTURE AFTER TT AT 1000°C DURING 6H: SINGLE CRYSTAL PARTICLES

SEM



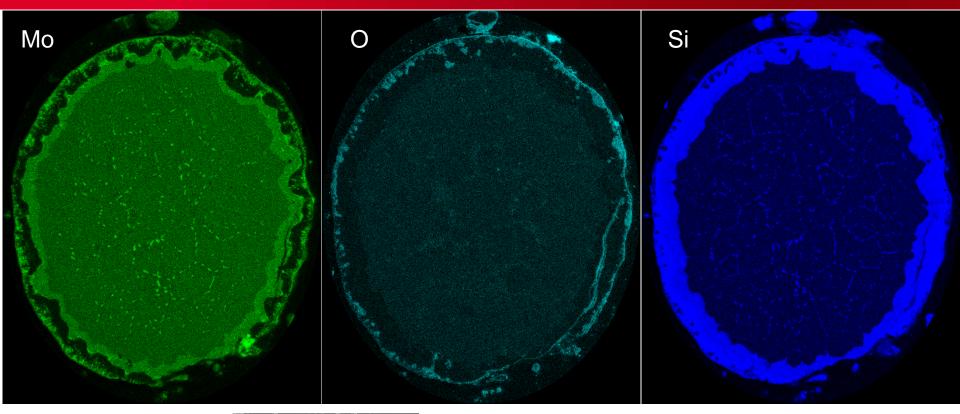
EBSD

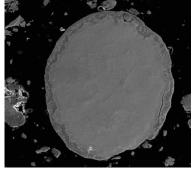


Evidence for the presence of « single crystal » U-Mo particles

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PRECIPITATION AS A RESULT OF THE POWDER ANNEALING



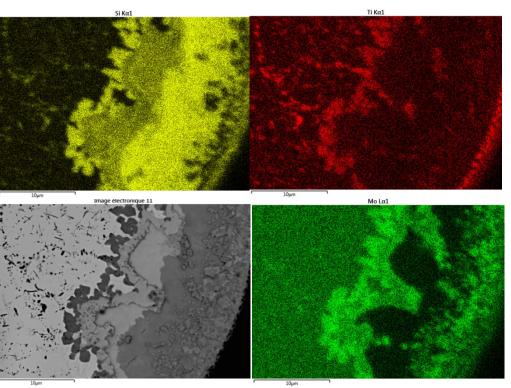


In U-Mo (Si) particles, intergranular Mo-Si rich precipitates can be observed



ADDITION ELEMENTS PRECIPITATION: THE CASE OF U-MO-TI PARTICLES

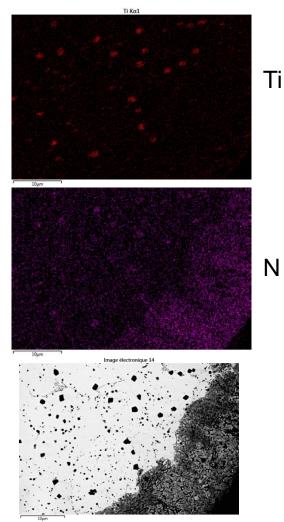
U-Mo-Ti(Si) (1000°C 1h)



Precipitation of Ti after thermal treatments :

- into Ti,Si and probably Mo rich precipitates in U-Mo-Ti(Si)
- into TiN in U-Mo-Ti(N)

U-Mo-Ti (N) (1000°C 6h)

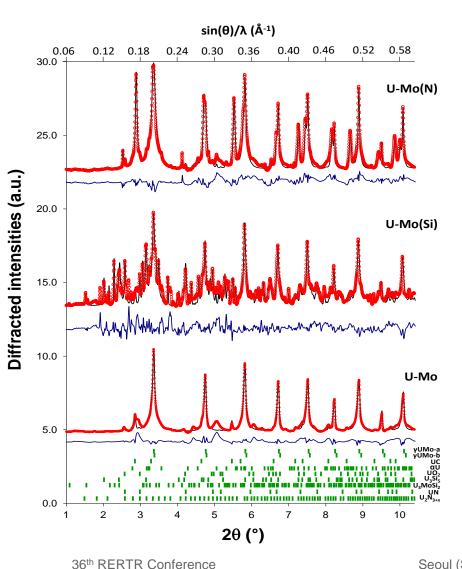


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X-RAY DIFFRACTION AT HIGH ENERGY (87 KEV)

• Ln (Yobs) — Ln(Ycalc) — Ln(Yobs)-Ln(Ycalc)







- Two γU-Mo lattice constants in the as-atomised case only
- Additional tetragonal superlattice
- Quantification of impurities: U(C,O) UO₂



X-RAY DIFFRACTION AT HIGH ENERGY (87 KEV)

									Weight	fraction (%
	U-	Mo cor	е	Pollution	N Co	ating	Si coa	ating		
	γ- UMo	α-U	UC	UO ₂	UN	U_4N_7	USi _{1.88}	U_3Si_2	U₃Si₅	U ₃ MoSi ₂
Powder UMo	98.11	0	1.6	0.29	0	0	0	0	0	0
powder UMo(Si)	51.62	0	0	7.11	1.3	0	5.05	2.95	15.48	16.49
powder UMo(N)	59.94	1.72	0	0.91	37.4	0.02	0	0	0	0

Conclusion :

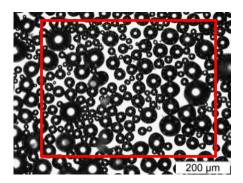
- No destabilization of the γ U-Mo phase (cooling rate high enough)
- Very high concentration of UN phase in U-Mo(N) (not consistent with 2-3 µm thick coating)
- Very complex crystallographic composition of the UMo(Si) coating

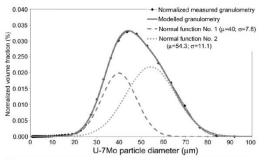
	U-Mo As-atomised	U-Mo(Si) (1000°C 1h)	U-Mo(N)/AI
Destabilisation ratio	0	<5%	<5%

Cesa

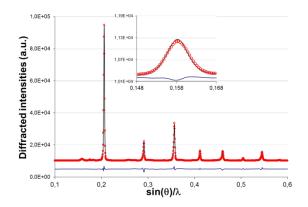
MEASUREMENTS OF THE AVERAGE COATING THICKNESS AT A MACROSCOPIC SCALE

HE-XRD measurements on **powder sample**





$V_{ m Shell}^{ m Powder}$		$\int_{0}^{\infty} f_{V}(D) V_{\text{Shell}}^{\text{Particle}} \mathrm{d}D$	
V ^{Powder} Core	$+ V_{\text{Shell}}^{\text{Powder}}$	$\int_{0}^{\infty} \int_{0}^{\infty} f_{V}(D) V_{\text{Shell}}^{\text{Particle}} \mathrm{d}D + \int_{0}^{\infty} f_{V}(D) V_{\text{Core}}^{\text{Particle}} \mathrm{d}D,$	
	e = 0	$0.0004V_{\text{ratio}}^2 + 0.0902V_{\text{ratio}} + 0.0013$	



Validation of the method

Method firstly validated on 1 μ m thick UO₂ coated U-Mo particles using tomography

H. Palancher et al., J. Appl. Cryst. (2012). 45, 906-913

Method firstly validated on 0,4 µm thick ZrN coated U-Mo particles using FIB/SEM (INL measurements)

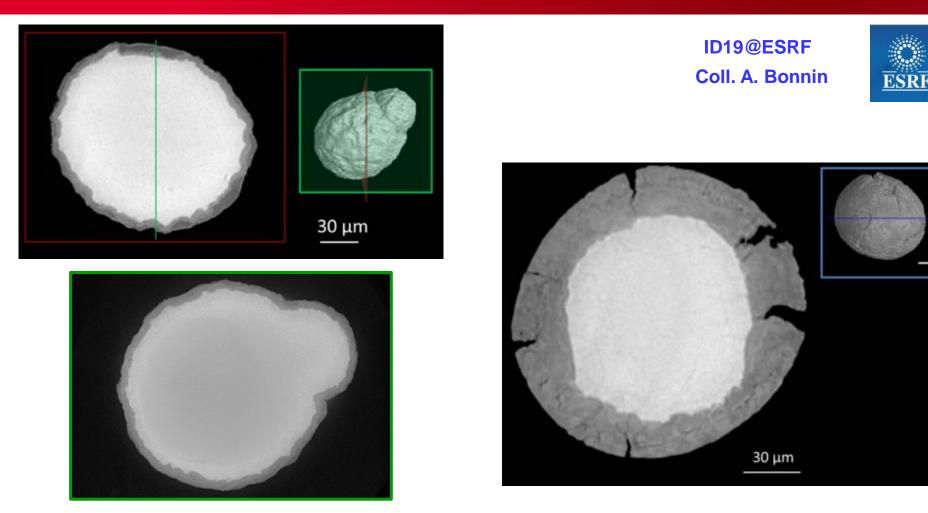
T. Zweifel et al., J. Nucl. Mater. (2013). 442, 124-132.

The UMo(N) case

Phases from the coating represent more than 37 wt% of the material which is not consitent with a 2-3 μ m thick layer measured with SEM: N diffusion deep into UMo particles has damaged the U-Mo long-range crisytallinity



COATING: HE-MCT ON U-MO(SI)



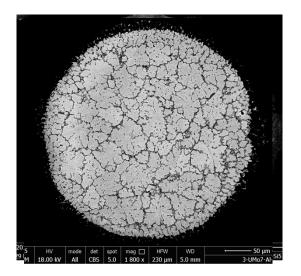
µ-CT helps evidencing:

- The good quality of the coating even around particles with non-spherical shapes
- cracks that can not be due to sample preparation contrary to SEM for example.

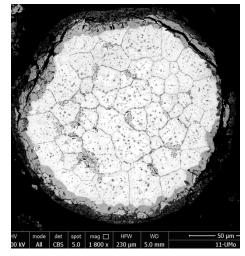
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U-MO MICROSTRUCTURE EVOLUTION AFTER MEAT EXTRUSION AND TT : SEM AND HE-XRD ANALYSES

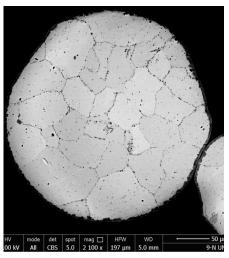
U-Mo(Si)/AI,Si5



U-Mo(Si)/Al (1000°C 1h)



U-Mo(N)/Al (1000°C 6h)



SEM examinations:

much higher destabilization of the γ UMo phase (starting at grain boundaries) in U-Mo(Si)/AI than in U-Mo(N)/AI

HE-XRD:		U-Mo/Al	U-Mo(Si)/Al	U-Mo(N)/AI
	Destabilisation ratio	15.1	23.7	2.1

Good agreement between both analyses !!!



CONCLUSIONS

A wide range of cutting edge measurements have been performed to investigate both the coating and the microstructure of U-Mo particles

	As-atomised U-Mo	U-Mo(Si) (1000°C 1h)	U-Mo(N) (1000°C 6h)
Average grain size (µm)	<6	15	17
Grain shape	dendritic	Sharpe edges	Sharpe edgesSome « single crystal particles »
Consequences of the coating		 Precipitation of Mo, Si, O High oxygen pollution 	Lower long range order in U-Mo phase
Destabilisation ratio In the annealed meat	15	24	2



CONCLUSIONS

A wide range of cutting edge measurements have been performed to investigate both the coating and the microstructure of U-Mo particles

	As-atomised U-Mo	U-Mo(Si) (1000°C 1h)	U-Mo(N) (1000°C 6h)
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Consequences of the coating		 Precipitation of Mo, Si, O High oxygen pollution 	Lower long range order in U-Mo phase
Destabilisation ratio In the annealed meat	15	24	2
Coating crystallographic composition		Composed of wide range of crystallographic phases	Mainly UN
Presence of cracks		Some particles with radial cracks	Too thin to exhibit large cracks



A. Bonnin, post-doc (2010-2012)

Development of a non destructive approach for coating layers and their evolution at the different steps of the manufacturing process

Alight ESRF	į į z · y Ξμπ	т. Тат., 5 µm	ζ , 5μm	
X-ray beam size	0.1 µm	0.1 µm	20-120 µm	~ CM ²
Cructellegrephi	Nano- XRD	Nano XRD-CT	Far field 3D- XRD	HE-XRD
Crystallographic composition:	C X	×	×	×
Thickness:	×	×	×	×
Other coating c (Density, shape coating/core int	1	Porosities VU-Mo	Δ Δ Δ Δ Δ Δ Δ Δ Δ Δ Δ Δ Δ	HE-μCT ×

H. Palancher *et al.*, *J. Appl. Cryst.* (2012). **45**, 906-913

Ce	22	3D MULTI SC	ALE ANALYSIS USING RADIATIONS	G SYNCHROTRON	
A. Bonr	nin, post-doc (2010-2012)	As-atomis	ed U-Mo particle micros		F
_	ta, sum 0.1 μm	ریم دیم 0.1 μm	5μm 20-120 μm	~ cm ²	
	Nano- XRD	Nano XRD-CT	Far field 3D- XRD	HE-XRD	•
	z y 5μm	 U(C, O) grain orientation re with U-Mo U(C,O) precip 	elationships	 Two γU-Mo lattice constants Additional tetragonal superlattice Quantification of impurities: U(C,O) =1 wt.% 	

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24th March 2015 A. Bonnin, J. Wright, R. Tucoulou, <u>H. Palancher</u>, Appl. Phys. Lett. **105** (2014), 084103. UO₂ =0.3 wt.%





THANK YOU FOR YOUR ATTENTION