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## Three method comparison used for calibration of Drucker-Prager Cap constitutive model for ceramic powder compaction

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Contents ufacturing fuel pellets in nuclear industry is composed of three main steps : Powder grinding, Pressing and Sintering. Powder during uniaxial pressing step is stressed with a lot of stages such filling, compaction, discharge in die and e ejection with a springback. Multiple phases occurs in the powder during the compaction stage and the aggiomerates inside the powder undergo successively rearrangement, strain and breakage. Ceallsec is conducting research to trol manufacturing processes in order to minimise defects and rejects in the polet, and to take into account the history of the powder and its rheology when shaping a pellet, with control of the density card that precedes the final ering stage. In addition, the characteristics of powders are constantly evolving in order to improve the quality of or reprocessing of the product in the reactor for greater efficiency. So, combined with the multi-scale approach, a great to staft and that characteristics of powders are constantly evolving in order to improve the quality of the unanfacture for predicting. In the present work, three methods for calibrating the elastic and poroach, a great to sted in order to reduce the amount of powder used. The first method is the reference method and has the disadvantage of using the pellet after ejection to characteristics shear failure by means of an axial and radial ture tests. The second method uses less powder and proposes replacing the fracture test with a shear test, but extrapolation of the cohesion and friction angle is not easy at high consolidation pressure. The third method is a numerical simulation of the model is preserved. Drucker fracture tests. The second method uses less powder and proposes replacing the fracture test with a shear test, but extrapolation of the cohesi method that uses a minimum amount of powder and enables to show that the accuracy of the model is preserved. Crack due to the springback during ejection step or Stress gradient in the pellet due to the internal and contact friction after sintering step 1 TT 1 Shrinkage after sintering due to single cycle (a) or double cycle ((b) Double compaction cycle Density gradient due to the cycle Pellet defects before or after sintering \_\_\_\_\_ Modelling : Porous elastic-plastic material behaviour  $q = \sqrt{3J_2} = |\sigma_a - \sigma_r|$ Drucker-Prager Cap (DPC) load surfaces in (p, q) plan: a) For one relative density [7], b) For the evolution of relative density [8]  $\rho = \rho_0 \exp \varepsilon_{vol}^{pla}$ ь) (a) MDPC model 8000 a) solidation depends on : \*  $\varepsilon_{Vol}^{Pla} > 0$ \* The direction of the Damage හ <sup>6000</sup> surface normal The plastic factor value  $d + p_{tan} b$ 4000 Consolidation BF dea detr 2000  $d + p \tan b$ deve eters to be defined : 1.2 tters to be defined : o-cellasticity: Log Bulk modulus k Poisson's coefficier Tensile Limit Initial void ratio e0 tcker-Prager: Cohesion d Friction angle β Transition surface p Eccentricity rdening/densification  $\overline{d\varepsilon_p} = \overline{d\varepsilon_{Vol}} = \overline{2d\varepsilon_r} + \overline{d\varepsilon_a}$ A Load path  $R(d + p_s \tan \beta)$  $\overrightarrow{d\varepsilon_q} = \frac{2}{3} (\overrightarrow{d\varepsilon_a} - \overrightarrow{d\varepsilon_r})$ 1.1 D Hyd tion 6000 4000  $p = \frac{I_1}{3} = \frac{1}{3}(\sigma_a + 2\sigma_r)$ 2000  $f\left(\overline{\bar{\sigma}}, \overrightarrow{k_{\alpha}}\right) = f\left(p, q, \rho, \varepsilon_{v}^{p}\right) = f\left(p, q, \rho\right)$ Hydrostatic Stress D, Da  $f_s = q - p \tan \beta - d = 0$  Damage limit ., & Wassgren, C. (2010). A phase of powder compaction & Huang, H. (2020). A densities the basis of the second second second densities of the second second second second densities of the second second second second second densities of the second second second second second second densities of the second second second second second second densities of the second he sensitivity of Drucker-Prager Cap mo ions. Powder Technology, 198(3), 315-32  $f_c = \sqrt{(p - p_a)^2 + (Rq)^2} - R(d + p_a \tan \beta) = 0$  Consolidation limit n lav  $= f(\varepsilon_{u}^{pla})$ **Model calibration and partial results** □<u>Method 1 (M1) :</u> □ Method 3 (M3) : b) Medium low strain gauge location for calibration (Silicon): • y = 1.3816x Rose • M: Lower part Die M1 (~20 pellets) 1 2D Axisymétric 3D model **3D** model calibration R(d+n tanß model Cohesion d Axisymmetric 2D model/Powder Material/Isight optimisation: Method 2 (M2) : For each parameters, the error is: ricity R Law compo Log bulk Poisson's Tensile li Initial vo Cohesion Friction a 0.0 0.2 0.5 0.2 28 10 0.03 0.25 0.3 0.45 218 40  $\varepsilon_x = \sum (X_i - x_i)^2$ M2 (~15 pellets eq.) with powder FT4 omb <sup>isw</sup> characterise e0 d C) With i = the density level entricity Cap harde X = the experimental value to i point hip for transfer of  $c_{i\phi}$  to x = the calculated value to i point  $tan \beta = \frac{3sin \phi}{\sqrt{3}\cos\theta - \sin\theta \sin\theta}$  $d = \frac{3c\cos\phi}{\sqrt{3}\cos\theta - \sin\theta \sin\phi}$ Û € < Optimization1 Isight optimization work flow → 🚧 — → 🚧 – → 📌 -208.9 → 🔳 Data MatchingData Matching-Data Matching-2 Calculator Abaqus D 5 5 (cm<sup>2</sup> (60,8% of (pm)) N 6,6 Strain vs punch Stress vs strain Stress vs punch Experimental (target) and calculated (RP) match displacement displacement Experimental kN\_6,69g/cm3 (60,7% of pm) Density card computed Double compaction <u>cycle</u> Experimental (target) and alculated (RP) <u>match</u> (target) and alculated (RP) <u>match</u> **DPC** law for different materials **Conclusion/Outlook** ε<sub>Strair</sub> Estress E<sub>Strass\_vs\_Strain</sub>  $\varepsilon_{Total} = \varepsilon_{Strain} + \varepsilon_{Strass} + \varepsilon_{Strass vs Strain}$ 

As the conclusion of this study, three calibration methods of the Drucker Prager cap porous elastic and plastic model have been challenged in order to reduce the powder quantity engaged. The third numerical method has a great interest because ten pellets for ten densities are enough with the conservation of the model representatively. The Hooke-Jeeves optimization algorithm was used and other optimization methods are available in the Isight code for the future investigations. The interest of the numerical method is data matching between Abaqus and Isight, and the future goal will be to achieve a meta model or simplify it for press digital twin include in the main numerical manufacturing chain. Furthermore, we plan to use numerical tools like Al associated to the large nuclear material data base.

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