

Evidence for a build-in remnant field in symmetrically contacted MAPbB_3 X-ray detectors

Ferdinand Ledee, Javier-Alejandro Mayen Guillen, Stephanie Lombard, Julien

Zaccaro, Jean-Marie Verilhac, Eric Gros d'Aillon

▶ To cite this version:

Ferdinand Ledee, Javier-Alejandro Mayen Guillen, Stephanie Lombard, Julien Zaccaro, Jean-Marie Verilhac, et al.. Evidence for a build-in remnant field in symmetrically contacted MAPbB_3 X-ray detectors. Journal of Applied Physics, 2023, 134, pp.195703. 10.1063/5.0170580. cea-04468082

HAL Id: cea-04468082 https://cea.hal.science/cea-04468082

Submitted on 20 Feb 2024

HAL is a multi-disciplinary open access archive for the deposit and dissemination of scientific research documents, whether they are published or not. The documents may come from teaching and research institutions in France or abroad, or from public or private research centers. L'archive ouverte pluridisciplinaire **HAL**, est destinée au dépôt et à la diffusion de documents scientifiques de niveau recherche, publiés ou non, émanant des établissements d'enseignement et de recherche français ou étrangers, des laboratoires publics ou privés.

Evidence for a build-in remnant field in symmetrically contacted MAPbBr₃ X-ray detectors

-

3	
4 5	Ferdinand Lédée ^{1,2,*} , Javier Mayén-Guillen ² , Stéphanie Lombard ² , Julien Zacarro ³ , Jean-Marie Verilhac ² , Eric Gros-Daillon ^{1,*}
6	¹ Grenoble Alpes University, CEA, LETI, DOPT, F38000 Grenoble, France
7	² Grenoble Alpes University, CEA, LITEN, DTNM, F38000 Grenoble, France
8	³ Grenoble Alpes University, CNRS, Grenoble INP, Institut Néel, F38042 Grenoble, France
9	* <u>ferdinand.ledee@cea.fr</u> ; <u>eric.grosdaillon@cea.fr</u>
10	Keywords: X-ray detectors, halide perovskites, sensitivity, simulation, Hecht equation
11	
12 13	1. Abstract

14 Millimeter-thick methylammonium lead tribromide (MAPbBr₃) single crystal X-ray detectors have 15 recently raised attention due their high X-ray attenuation efficiency and good charge transport properties. However, an intriguing feature of the photocurrent response of MAPbBr₃ detectors has 16 17 been largely overlooked in the literature. After biaising, transient sensitivity is measured under X-rays 18 at short-circuit (Bias = 0V), thus revealing a large remnant electric field that build-up under bias. Here, 19 we exploit the X-ray sensitivity of MAPbBr₃ detectors at zero-bias in order to probe the internal built-20 in field, as well as to investigate the charge transport properties of the perovskite material. Our model 21 derived from the Hecht equation is able to fully rationalize the response of the detectors both at short-22 circuit and under moderate applied bias. Moreover, we provide a method for the estimation of the 23 internal electric field, and for the sum of the electrons and holes mobility-lifetime products $\mu_e \tau_e +$ 24 $\mu_h \tau_h$. This general method could extend to any perovskite-based X-ray detector exhibiting transient 25 sensitivity at zero-bias.

26

27 **2. Introduction**

28

29 Metal halide perovskites are the current key players in the field of radiation detection, challenging the 30 performance of classical semiconductors in several applications while maintaining competitive 31 manufacturing costs. In particular, methylammonium lead tribromide (MAPbBr₃) in single crystal form 32 is one of the most intensively studied emerging material for the direct detection of X-rays.[1]-[7] 33 Despite the high quality of these as-grown single crystals,[8]-[10] several features of this semiconducting material remain unexplained. Notably, in some cases the photocurrent response of 34 35 MAPbBr₃ seems to deviate from the Hecht model.[1], [3], [4], [6] Hecht model is the most widely used 36 model to describe the photoresponse of semiconductor-based X-rays detectors such as CdTe, CdZnTe,

37 GaAs; and shall be applied as well to halide perovskites.[11]–[13] In particular for MAPbBr₃, it is often 38 applied in the form displayed in Eq. (2) to fit the experimental sensitivity-voltage (S-V) dataplots in 39 order to extract the mobility-lifetime product $\mu\tau$ of the electronic charge carriers.[1], [2], [14], [15] Yet, 40 previous work from our group revealed that the X-ray photoresponse of Cr/MAPbBr₃/Cr devices is not 41 in accordance with the Hecht model.[16] Moreover, in the recent work of Alvarez et al, it is shown that 42 sizeable X-ray photocurrent can be measured in Cr/MAPbBr₃/Cr devices in short-circuit conditions. 43 Unlike photovoltaic-inspired devices that provide stable photocurrent output under irradiation,[17], 44 [18] the X-ray photocurrent measured in Cr/MAPbBr₃/Cr photoconductors at zero-bias decays over 45 time and follows the relaxation dynamics of the dark current.[19] We propose that this is the outcome 46 of a residual electric field created upon biasing that relaxes when the bias is removed. This residual 47 field would originate from the build-up of a space-charge when the Cr/MAPbBr₃/Cr device is biased, 48 and would result in a diminished net applied electric field under operation. The internal electric field 49 would be responsible for the anomalies listed above, in particular the discrepancy between the 50 experimental sensitivity-voltage (S-V) data and the Eq. (2). 51 This paper aims to provide a detailed picture of the X-ray photoresponse of Cr/MAPbBr₃/Cr single

52 crystal devices under short-circuit conditions. Starting from the assumption that the residual electric 53 field prevalent at zero-bias is also present under an externally applied bias, we developed a model 54 derived from the Hecht model that is able to assess with great precision the internal electric field that 55 opposes the externally applied bias under operation. In addition, we use our model to bring an 56 alternative pathway to establish relevant intervals for the $\mu\tau$ products in MAPbBr₃-based detectors. 57 We believe that this work sheds new light on the X-ray photoresponse of MAPbBr₃ detectors; and will 58 gives new insights into the residual charges that are formed in millimeter-thick lead halide perovskite 59 devices under bias, as well as their aftermath on the detection performances.

60

61

3. Discussion on the internal electric field

62

69

In a semiconductor-based planar detector, the photocurrent measured under X-ray results from the induction of current by the photogenerated charges on the contact electrodes, also referred as the Ramo-Shockley theorem.[20] The integration of the photocurrent over the X-ray pulse duration brings one of the methods that allow to estimate the sensitivity, hereby expressed in μC mGy⁻¹ cm⁻². Without any trapping, the sensitivity reaches a maximum value and depends only on the amount of

68 photogenerated charges in the semiconductor :

$$S_0 = \int s_0(x) dx \tag{1}$$

70 With s_0 the charge generated in an elemental volume dx of the semiconductor (see Methods section). 71 With trapping and for a planar detector with two facing electrodes, the analytical integration of the 72 Ramo theorem over the charge carrier lifetime τ gives the sensitivity-voltage (*S-V*) relationship which 73 reads:

74
$$S(V) = \frac{S_0 \mu \tau V}{L^2} \left[1 - exp\left(-\frac{L^2}{\mu \tau V}\right) \right]$$
(2)

With S_0 the maximum sensitivity, μ the charge carrier mobility, V the applied bias and L the distance between the electrodes. This relation is only valid when the following conditions are met altogether i) the energy is absorbed close to the irradiated electrode ; ii) only one type of charge carrier drifts in the device; iii) the electric field is uniform.[11] When the photocurrent contribution from the second
charge carrier type cannot be neglected, e.g. when the absorption length is not negligible with respect
to *L*, the relation becomes a function of the depth. It translates into the two-carriers Hecht equation
which reads:

$$\eta(x,V) = \frac{\mu_e \tau_e V}{L^2} \left[1 - e^{-\frac{xL}{\mu_e \tau_e V}} \right] + \frac{\mu_h \tau_h V}{L^2} \left[1 - e^{-\frac{L(L-x)}{\mu_h \tau_h V}} \right]$$
(3)

83 Where η is the charge induction efficiency (CIE), x is the position of creation of the charge carriers 84 (0<x<L), and the subscripts e and h denote for electrons and holes, respectively. The *S*-*V* relationship 85 now reads :

86

82

$$S(V) = \int s_0(x)\eta(x,V)dx \tag{4}$$

87 With s_0 the charge generated in an elemental volume dx of the semiconductor (see Methods section). 88 Further details on the fabrication of the detector, on the Hecht model and calculation of S_0 are 89 provided in the Methods section.

90 A common approach in the research applied to perovskite X-ray detectors consists of using equation 91 (2) or (4) to fit the experimental sensitivity-voltage data in order to extract a value for the mobility-92 lifetime product $\mu\tau$. In most of the works, the maximum sensitivity S_0 (or the maximum induced 93 current I_0) is not estimated, and thus is set as a free parameter in the Hecht fits. We propose that this 94 produces a lot of misinterpretation of the S-V data. As an illustration, Fig. 1 displays experimental S-V 95 curve for a MAPbBr₃ single crystal planar detector fitted both with and without a calibrated and fixed 96 S_0 value. The $\mu\tau$ product for electrons extracted from the fit differs by almost an order of magnitude depending on the method. Furthermore, in Figure 1b, the Hecht model is not in good accordance with 97 98 the experimental S-V. The experimental sensitivity measured at higher biases is lower than the value 99 predicted with the Hecht model, and seems to plateau to a much lower value than S_0 (typically S_{max} = 100 $0.25 S_0$). Here the S_0 value has been estimated using Klein's rule (W = 14/5*Eg + 0.5)[21], however 101 choosing a W value equal to that experimentally determined for CsPbBr₃ by He et al. does not change 102 the result of the fit (see Fig. S1 for further details).[22] Furthermore, the Many equation also fails to fit 103 our experimental S-V data (Fig. S4). This behavior has already been observed by our group for all our 104 MAPbBr₃-based devices, as well as for various perovskite compositions.[16] The plateau of the 105 sensitivity to a lower S value in MAPbBr₃-based devices is overlooked in the literature because the 106 maximum sensitivity S_0 (or I_0) is mostly set as a free parameters as in Fig.1a.[1] Note that in that case, 107 the S_0 value is not linked to any macroscopic quantity. Besides, S-V graphs are often plotted in semi-108 log scale, which misleads the reader into thinking that the fit is better than it actually is (see Fig.S2 for 109 an illustration). [1], [2], [14] Remarkably, we carried out a S-V measurement adding a depolarization 110 time (bias =0V) inbetween each bias step. Despite the relatively long (5-10 min) depolarization time 111 and the relatively short (\approx 5s) bias pulse duration, the low S_{max} plateau is still present and the Hecht 112 equation fails to fit the results (see Fig. S3 for further details). We attribute the plateau at lower S_0 113 value to a non-uniform electric field across the detector at higher applied bias. Several phenomena 114 may distort the electric field, including charge trapping in the bulk of the semiconducting material [23], 115 surface polarization, [24] depletion region due to Schottky contacts [25] or ion accumulation at the 116 vicinity of the contact electrodes.[26] All of these potential mechanisms may lead to mis-estimation 117 of the values of $\mu\tau$ products in MAPbBr₃-based detectors using Eq. 2. To date, the hypothesis of ion 118 accumulation seems to be the prevailing one in the literature on perovskite detectors.[26], [27] 119 Considering the numerous works on the migration of ionic species in MAPbBr₃,[28]–[31] it seems to 120 be the most possible explanation to the best of our knowledge.



122Fig. 1. Sensitivity-voltage plots of a MAPbBr3 device with corresponding two-carriers Hecht fits with (a)123 S_0 set as a free parameter and (b) S_0 set as the total charge generated in the device. See Methods for124more details on the calculation of S_0 .

121

Another intriguing feature of our MAPbBr₃-based devices arises from the so-called bias memory effect 126 in short-circuit conditions. As notified in previous works, after biasing, MAPbBr₃ and other halide 127 128 perovskites devices exhibit transient dark current relaxation with inverted sign at 0V.[19], [32] In 129 addition the dark current transient is accompanied with non-negligible sensitivity under X-rays during the relaxation window at 0V, as shown in Fig.2(c). Notably, the sensitivity at 0V decays over time with 130 131 slow dynamics akin to the dark current transient (Fig.S5). Fig.2(d-e-f) provides a depiction of the bias memory effect. Several works suggest that upon biasing, the displacement of mobile ions toward the 132 contact electrodes builds up a space-charge that distorts the electric field (Fig.2d and 2e).[27], [33] 133 Consequently the net electric field in the bulk region of the device is reduced with respect to the 134 135 uniform case (dotted blue line in Fig. 2e). [19][27] In the present work, we approximate the electric field in the bulk to $\frac{V-Vi}{L-wi}$ where Vi is the potential drop associated to the space-charge formation, wi 136 the total space-charge width. When the bias is subsequently released, the space-charge built-up by 137 the accumulations of ions prevails and creates a residual electric field that we approximate to $\frac{Vi}{L-wi}$ in 138 the bulk region (Fig. 2f). Note that during both biasing and recovery steps, we deem the electric field 139 to be flat in the bulk region of the detector and non-uniform in the space-charge regions, in accordance 140 141 with previous works.[19], [26] After bias removal, the residual field during the recovery step gradually fades out as the mobile ions return to equilibrium. It should be noted again that the present work does 142 143 not assume an origin for the internal electric field, which can originate from any space charge build-up 144 in the perovskite material. Nonetheless, the migration of mobile ions provides a comprehensive 145 picture to describe these results, as depicted in the schematics of Fig.2d. In this picture, the dark 146 current recovery measured at 0V would originate from an interplay between ionic and electronic 147 currents[19]; and the photogenerated charge carriers may drift along the residual electric field at OV during a characteristic recovery time, thus generating decaying photocurrent over time (Fig. 2c). 148





Fig. 2. (a) Microscope picture of a typical Cr/MAPbBr₃/Cr single crystal device used in this study. (b) Schematic of the X-ray irradiation setup. (c) Dark current (black curve) and sensitivity-time (red curve) measurements during both biasing et recovery steps. (d-e-f) Proposed model for the electric field profile in the depth of the device during the experiment. The electric field is supposed linear in the bulk and is approximated by $\frac{V-Vi}{L-wi}$ during the biasing step (e), and $\frac{Vi}{L-wi}$ during the recovery step (f).

155 Fig.3 gives additional insights about the residual electric field for a Cr/MAPbBr₃/Cr single crystal device. 156 For the moderate biasing time range (400-2000 seconds) and applied bias (10V) used in this experiment, the sensitivity at 10V decreases slightly over time (Fig.3a). We attribute this to the space-157 158 charge build-up that gradually reduces the net electric field in the bulk region of the device, with a 159 consequent reduction of the sensitivity.[27] When the bias is released, sensitivity is measured at OV 160 owing to the residual electric field. Notably, the sensitivity measured at 0V immediately after removing 161 the bias is the same regardless of the biasing duration in the [400-1800s] bias duration range 162 investigated in this work (Fig.3a). It suggests that, for this 2-mm thick MAPbBr₃ device, the space-163 charge build-up is rather fast (≤400 seconds) and remains constant. Thus, the sensitivity of the device at 0V is initially the same for biasing duration ≥400s. In addition, the residual electric field varies with 164 the applied bias. As seen in Fig 3b, the sensitivity at 0V measured immediately after removing the bias 165 166 scales with the previous bias value accordingly, indicating that the space-charge build-up is a function 167 of the applied electric field; that could indicate the drifting of mobile ionic species.[30], [31] However, 168 we noted only a 30% increase in the residual sensitivity at 0V with bias increasing from 10V to 150V, 169 which would suggest that most of the mobile ions have already reach the electrode for biases $\geq 10V$ 170 and biasing duration ≥400 seconds. Markedly, this experiment could partially explain why the 171 experimental S-V is not well fitted with the Hecht equation in Fig.1b. Along with the rising space-charge 172 that produces distortion of the electric field at high bias, the net electric field in the bulk is progressively 173 reduced by the internal field build-up as the applied bias is increased. As a consequence, the net 174 electric field in the device can be underestimated, as it will be discussed below. Finally, the dynamics 175 of the transient sensitivity at OV may provide further information on the mobility and the concentration 176 of mobile ionic species in the material. This will be discussed in a future work. In the present work, we 177 will focus on the sensitivity at 0V measured immediately after removing the bias, and develop a model 178 that accords accurately the above-mentioned results.





Fig. 3. (a) Sensitivity-time measurements for different biasing duration at +10V (b) Sensitivity measured
 immediately after removing the bias for different previously applied bias. The biasing duration is 400
 seconds. Blue dotted line is an eye guideline.

184 **4. Model**

185

186 In the following, we propose to model the sensitivity of a 2-mm thick MAPbBr₃ planar device 187 considering the following assumptions : (i) the electric field is mainly uniform throughout the device; 188 (ii) an internal electric field is created upon biasing, and is equivalent to the residual electric field at OV 189 immediately after removing the bias; (iii) the internal electric field opposes the externally applied electric field (see Fig.3a). Regarding (i), the simulations carried out by Alvarez et al. show that the 190 191 accumulation of mobile ions forms a depletion region at the very vicinity of the electrodes, with a 192 depletion width only up to a few micrometers when the device is previously biased at 10V.[19] Besides, the electric field as function of the depth is mostly flat in the bulk, [19], [26] and for a 2-mm thick device 193 wi<<L, thus the internal field in the bulk region is approximated by $\frac{Vi}{L}$ (Fig. 4). We can consider a uniform 194 electric field in the sensitivity model provided that the penetration depth of the X-rays is much greater 195 196 than the depletion width. In the present case, the hard RQA5 X-rays with a high mean energy of 53 keV 197 generate charges throughout the volume of the 2-mm thick MAPbBr₃ device, as evidenced in Fig.4c 198 (dotted line). In addition, we limit our model to rather moderate applied bias (10V), for which the 199 electric field distortion is limited. Finally, the assumptions (ii) and (iii) are justified by the dark current 200 and sensitivity-time measurements displayed in Fig.2 and Fig.3.

201 Due to high penetration depth of the X-rays used in the present study, using the two-carriers Hecht 202 equation (3) is compulsory. Considering the assumptions listed above, the model reads as follow. The 203 sensitivity under bias measured a fraction of time before removing the bias writes:

204
$$S(V - Vi) = \int s_0(x)\eta_{bias}(x, V - Vi)dx$$
(5)

205
$$\eta_{bias}(x, V - Vi) = \frac{\mu_e \tau_e (V - Vi)}{L^2} \left[1 - e^{-\frac{L(L-x)}{\mu_e \tau_e (V - Vi)}} \right] + \frac{\mu_h \tau_h V}{L^2} \left[1 - e^{-\frac{xL}{\mu_h \tau_h (V - Vi)}} \right]$$
(6)

206 The sensitivity at 0V measured immediately after removing the bias writes :

$$S(Vi) = \int s_0(x)\eta_{short}(x,Vi)dx$$
⁽⁷⁾

208

207

209
$$\eta_{short}(x,Vi) = \frac{\mu_e \tau_e Vi}{L^2} \left[1 - e^{-\frac{xL}{\mu_e \tau_e V}} \right] + \frac{\mu_h \tau_h Vi}{L^2} \left[1 - e^{-\frac{L(L-x)}{\mu_h \tau_h Vi}} \right]$$
(8)

210

Where V is the externally applied bias, Vi is the potential associated to the internal electric field $=\frac{Vi}{I}$, 211 η_{bias} is the CIE before removing the bias and η_{short} is the CIE after removing the bias and x is the 212 213 position of interaction of the X-ray in the MAPbBr₃ device. Note that the equations (6) and (8) 214 correspond to the case where a positive bias is applied to the irradiated electrode. For a 2-mm thick 215 MAPbBr₃ device and borderline hard ~50 keV X-rays, the response in biasing conditions is slightly 216 dominated by hole transport (Fig.4a), and the response in short-circuit conditions is slightly dominated 217 by electron transport (Fig.4b). Fig. 4c displays the results of the model calculation for the following input parameters : $\mu_e \tau_e = 1.10^{-4} \text{ cm}^2 \text{ V}^{-1}$; $\mu_h \tau_h = 2.10^{-4} \text{ cm}^2 \text{ V}^{-1}$; L = 0.2 cm ; V = 10V ; Vi = 1V. It can be 218 seen that the total induced charge under X-rays - or in other word, the sensitivity S - is close to the 219 experimental value shown in Fig.3a (S \approx 0.2 μ C mGy⁻¹ cm⁻² at 10V and S \approx 0.02 μ C mGy⁻¹ cm⁻² at 0V). 220 221 Notably, it implies that the internal electrical field at 10V could be as high as 5 V/cm; which 222 corresponds to 10 percent of the externally applied electric field (50 V/cm). It indicates that there could 223 be a 10% uncertainty on the value of the applied bias that could partially explain the discrepancy 224 between the S-V plot and the Hecht model, as depicted in Fig.1b.

225



Fig. 4. Schematics of the device during (a) the biasing condition and (b) the short-circuit condition. The internal and external electric fields are deemed uniform throughout the device (c) Results of the calculation. Blue and red lines are the CIE as function of distance at 10V and 0V respectively. Black dotted line is the photogenerated charge. The x-axis is the distance in the device from the irradiated electrode (x=0) to the back electrode (x=L=2mm). Calculated for RQA5 X-rays.

232

5. Results and discussion

234

In this part, it will be demonstrated how the model allows to estimate precisely the internal electric field in a Cr/MAPbBr₃/Cr planar device, as well as reasonable ranges for the $\mu_e \tau_e$ and $\mu_h \tau_h$ products.

- The set of equation (5-8) has 3 unknown parameters : $\mu_e \tau_e$; $\mu_h \tau_h$ and Vi. Only two experimental 237 measurement of sensitivity are used as inputs for the model : S at 10V before removing the bias, and 238 239 S at 0V immediately after removing the bias. Hence it may provide only a set of solution ($\mu_e \tau_e$, $\mu_h \tau_h, Vi$). Nonetheless, we will see that the set of possible solution is limited and allow to establish a 240 241 credible range for the above cited parameters.
- 242 The model was applied to the measurements displayed in Fig.3a for a MAPbBr₃ single crystal planar 243 device with 2-mm thickness, 5 x 5 mm² chromium electrodes and RQA5 X-rays. The picture of the 244 device is shown in Fig. 2a. Further details on the fabrication of the device and the experimental setup are supplied in the Methods section. The experimental measurement of sensitivity gives : S(10-Vi) = 245 0.17 μ C mGy⁻¹ cm⁻² and S(Vi) = 0.021 μ C mGy⁻¹ cm⁻². We calculated all the possible ($\mu_e \tau_e, \mu_h \tau_h, Vi$) for 246 which S(10-Vi) and S(Vi) are both equal to the above cited values at the same time. The Fig.5a. displays 247 248 the results of the calculation in the ($\mu_e \tau_e$, $\mu_h \tau_h$) space. For all Vi, S(10-Vi) and S(Vi) form two straight lines that only cross for a limited number of Vi values ranging from 0.9 V to 1.2 V. The intersection of 249 the two S(10-Vi) and S(Vi) lines implies that it exists a set of $(\mu_e \tau_e, \mu_h \tau_h, Vi)$ that allows to fully describe 250 251 the experimental sensitivity measured under X-ray, or in other word, it provides a solution of the equation system. Hence, all the possible solutions (intersections) for this experiment are displayed in 252 Fig. 5b. It allows us to establish that the internal electric field $Ei = \frac{Vi}{L}$ is in the range [4.4 – 5.6] V/cm, 253 or is equal to 5.1 V/cm with 15% uncertainty. Regarding the possible values for ($\mu_e \tau_e$, $\mu_h \tau_h$), the range 254 255 of solutions is much broader. $\mu_e \tau_e$ spans from 5 x 10⁻⁶ cm² V⁻¹ to 3 x 10⁻⁴ cm² V⁻¹ and $\mu_h \tau_h$ spans from 1 x 10⁻⁷ cm² V⁻¹ to 2.5 x 10⁻⁴ cm² V⁻¹. The uncertainty on the individual $\mu_e \tau_e$ and $\mu_h \tau_h$ values is therefore 256 257 very high with this model. Nonetheless, more notable is the fact that this experiment computes quite 258 precisely a value for the sum $\mu_e \tau_e + \mu_h \tau_h$, as displayed by the linear fit in Fig. 5b. All of these solutions are aligned, and the sum $\mu_e \tau_e + \mu_h \tau_h$ has a near constant value of ~2.5 x 10⁻⁴ cm² V⁻¹. Here, the range 259 of solutions is a straight line because we are dealing with small electric fields, which means that the 260 261 mean free path of the charge carriers is small compared to the thickness of the device. In that peculiar case, the exponential term in the Hecht equation becomes negligible, and the CIE becomes $\simeq \frac{\mu_e \tau_e V}{r^2} +$ 262 $\frac{\mu_h \tau_h V}{L^2} = \frac{(\mu_e \tau_e + \mu_h \tau_h) V}{L^2}$. Substituting this into equation 7 gives 2.6 x 10⁻⁴ cm² V⁻¹ for Vi =1V, which is 263 consistent with the above results. The information on the sum $\mu_e \tau_e + \mu_h \tau_h$ provides new insights into 264 265 the charge transport properties of the material. Using other complementary measurements, it is then 266 possible to fully rationalize the transport of the material. For instance, the rather symmetric S-V plot displayed in Fig.1. indicate that the respective $\mu_e \tau_e$ and $\mu_h \tau_h$ values probably have the same order of 267 magnitude. Otherwise, the S-V data would be very asymmetrical for the X-ray exposure conditions and 268 device thickness used in this work. (see supplementary Fig.S6). This is consistent with photocurrent 269 270 measurements made under the AM-241 source, which indicate that the $\mu_e \tau_e$ and $\mu_h \tau_h$ could be
- 271 separated only by a factor of 2.[34], [35]



Fig.5. Calculations of the $(\mu_e \tau_e, \mu_h \tau_h, Vi)$ solutions for S(10-Vi) = 0.17 μ C mGy⁻¹ cm⁻² (solid lines) and S(Vi) = 0.021 μ C mGy⁻¹ cm⁻² (dash lines). (a) S(10-V) and S(Vi) lines as function of Vi. Each color corresponds to a different Vi input value. (b) Intersections of the S(10-V) and S(Vi) lines with Vi varying from 0.85 to 1.2V.

272

278 To assess the robustness of the method, we simulated the response of MAPbBr₃ devices for 279 hypothetical ($\mu_e \tau_e$, $\mu_h \tau_h$, Vi) and thickness values. Several of these simulations are displayed in Fig.6. 280 Remarkably, we noticed that it does not seem to exist a case where the two S(10-Vi) and S(Vi) lines 281 doesn't cross in one point in the $(\mu_e \tau_e, \mu_h \tau_h)$ space for a single Vi value. As seen in Fig. 6a and 6b, the 282 lines are almost parallel when the $(\mu_e \tau_e, \mu_h \tau_h)$ values are moderate ($\leq 1 \times 10^{-4} \text{ cm}^2 \text{ V}^{-1}$). When the 283 charge transport is better ($\approx 1 \times 10^{-3} \text{ cm}^2 \text{ V}^{-1}$) as in Fig 6c, the model still provides a single solution for a 284 single Vi value. In the latter case however, the S(V-Vi) line is no longer parallel to the S(Vi) line. It implies 285 that the error on Vi is much higher than in the moderate charge transport case since the position of the S(Vi) line varies sizably with the Vi value (Fig. 6a-b). It can be understood by considering that high 286 287 sensitivity increases the total number of possible solutions for the model. If the transport is high, the 288 sensitivity is high accordingly, and the model struggles to resolve accurately a value for Vi. If such a 289 case were to be met, a quick fix would consist of reducing the applied bias V in order to reduce the 290 charge carrier mean free path. As seen in Fig. 6d, it results in a reduction of the curvature of the S(V-291 Vi), and an increased precision on the estimation of Vi. In addition, having straight S(V-Vi) and S(Vi) 292 lines is associated to a low mean free path for the charge carriers, which allows to establish accurately 293 a value of the sum $\mu_e \tau_e + \mu_h \tau_h$ (see Fig. 5b). Lowering the bias during the biasing step to reduce the mean free path is thus interesting to increase the precision for both the internal potential Vi, and the 294 295 sum of the mobility lifetime products $\mu_e \tau_e + \mu_h \tau_h$.





Fig. 6. Calculation of the S(V-Vi) and S(Vi) lines for different ($\mu_e \tau_e, \mu_h \tau_h, Vi$) input values. The thickness of the device is 2 mm. The intersection of the S(V-Vi) and S(Vi) lines provides the solution of the equation system.

300 Finally, we can use this model to estimate an order of magnitude for the space-charge density generated upon biasing. First, we considered the parallel plate capacitor approximation which reads 301 $Q = C \times V$ and $Q = \frac{\varepsilon \varepsilon_0 S}{L}$ with Q the charge, C the capacity, ε the relative permittivity of the 302 perovskite, ε_0 the vacuum permittivity, S the electrode surface area and L the device thickness. From 303 304 the Fig.5, we can assume a value for Vi \approx 1V, corresponding to a charge of 50 nC. If we assume that this 305 charge was extracted from the bulk of the crystals, e.g. via migration of charged ionic species, and that 306 the totality of the charges reached the electrode, we can establish that the charge density in the bulk was initially 1 µC cm⁻³. Now if we consider that this charge originate from moving ionic species, it would 307 represent a density of 10⁺¹³ ions cm⁻³, which is in accordance with ionic concentrations previously 308 309 measured in similar MAPbBr₃ single crystal devices.[19], [36]

310

6. Conclusion

312

In this paper, we provide robust experimental evidences for the presence of an internal electric field in Cr/MAPbBr₃/Cr X-ray detectors that builds up upon biasing. This internal electric field generates a significant but decaying response under X-rays at OV; and diminish the net electric field in the device under bias. We propose that the discordance between the experimental *S-V* plots and the Hecht model is due to the internal electric field that alters the *S-V* relationship. Starting from a modified version of the Hecht equation, we developed a model for the sensitivity that involves an internal electric field with opposed sign to the externally applied electric field. We found that the model can accurately

- 320 predict a value for the internal electric field, even without prior information about the charge transport
- properties of the material. Notably, the actual internal electric field in the bulk of our 2-mm MAPbBr₃ device exclude by about 10% at 50% and 50%
- device could be overestimated by about 10% at 50V/cm. In addition, we exploit our model as an
- 323 alternative pathway for the evaluation of the $\mu_e \tau_e$ and $\mu_h \tau_h$ products, and more precisely for the sum 324 $\mu_e \tau_e + \mu_h \tau_h$. We believe that our model rationalize some intriguing electrical properties of MAPbBr₃
- $\frac{1}{2}$ detectors under bias such as dark current relaxation and zero-bias sensitivity; and could be extended
- to other perovskite composition exhibiting dark current relaxation at zero-bias such as CH₃NH₃Pbl₃ and
- 327 CsPbBr₃.[32], [37] Future work will bring further information on the remnant charge density in the
- 328 MAPbBr₃ material by exploiting the dynamics of sensitivity decay at zero-bias.
- 329

330 Methods

331

332 **Device Fabrication**

333 Millimeter-sized methylammonium lead tribromide (MAPbBr₃) single crystals were grown using the 334 Modified Inverse Temperature Crystallization (MITC) method described in details in previous works.[9], [16], [19] Briefly, the precursor solutions were prepared in an argon-filled glovebox by 335 336 dissolving PbBr₂ (ultra dry, 99.999 % metal basis) and MABr (≥99 %) in anhydrous N,N-337 Dimethylformamide with 1:1 molar ratio. The growth was carried out using a crystal seed, as described 338 in [9]. Both the large facets of the crystals were mechanically polished with SiC disks and isopropanol, 339 then with fabric disks and diamond paste in order to obtain a mirror grade quality. 100-nm Cr 340 electrodes were subsequently thermally evaporated on both facets. We found that proper polishing 341 and chromium electrodes enhance the electrical stability of the devices under bias, as also noted in 342 previous works.[15], [38]

343 X-ray characterization

Dark current and sensitivity measurements were conducted on the same setup using a Keithley 428 344 345 current amplifier and a Keithley 487 power supply. The samples were irradiated with short X-ray pulses 346 (4 Hz repetition rate, 100 ms duration) using a medical-grade pulsed X-ray tube with tungsten anode 347 and 70 kVp accelerating voltage. The incident X-rays were filtrated through a 0.8 mm beryllium window and 23.5 mm additionnal aluminium filters to yield a RQA5 spectrum, following the IEC 62220 348 349 International Standard. The incident dose was calibrated with a PTW Unidos dosimeter, yielding 24 ± 0.6 µGy_{air} per pulse. The X-rays were collimated through a 12.6 mm² hole pierced in a ≈5mm-thick lead 350 351 plate. Due to long-range diffusion of the X-ray generated charge carriers in the area not covered by 352 the electrodes, the collimation is necessary to properly evaluate the absorption surface area of the 353 incident X-rays, hence the sensitivity. Each sensitivity measurement consisted of 10 X-ray pulses that 354 were averaged to reduce experimental error (see Fig. S7). Typical uncertainty on the sensitivity 355 measurement was 3%.

356 Numerical simulation

The elemental charge deposited in the MAPbBr₃ material s_0 and the maximum theoretical sensitivity S₀ were calculated using Eq. 9 and Eq. 10 that read:

359
$$S_0(x) = \frac{e \times E^*(x)}{W_{\pm}}$$
 (9)

360
$$E^*(x) = \int N(E) \times E \times A(x, E) \, dE \tag{10}$$

361 Where e is the elemental charge ; E^* is the energy deposited by the X-rays in the perovskite material 362 (Fig. S8), W_{\pm} is the electron-hole pair creation energy that was calculated using Klein's rule $W_{\pm} =$ 363 $\frac{14}{5}E_g + 0.5$ with $E_g = 2.3$ eV [16], [21] ; D is the dose per surface area ; N is the number of incident 364 photon emitted by the X-ray tube as function of photon energy E; A is the absorbed fraction as function 365 of the position of interaction of the X-rays in the MAPbBr3 device. The latter is computed as for Eq. 11 366 :

$$A(x,E) = T(x,E) - T(x + dx,E) = e^{-\sigma(E)\rho x} - e^{-\sigma(E)\rho(x+dx)}$$
(11)

368

367

369 Where *T* is the transmitted fraction; σ is the photon cross section; ρ is the density (3.8 g cm⁻³ for 370 MAPbBr₃)[39]. The photon cross section was obtained from NIST photon cross sections database 371 considering only the photoelectric interaction, assuming that it is the predominant interaction 372 mechanism in the 20-70 keV energy range for MAPbBr₃.[40] The number of incident photons *N* was 373 computed for a RQA5 spectrum using the SpekCalc software.[41]–[43] It is given as function of incident 374 X-ray dose in mGy_{air} and per unit area (Fig. S9).

375 Supplementary Material

376 The Supplementary Material is available free of charge at:

Additionnal Hecht equation fitting plots, computations of the energy deposited by the X-rays and the
 number of incident photons, examples of X-ray-induced photocurrent pulses under bias and at short circuit.

380

381 Author Declaration section

382

383 Conflict of Interest Statement

384 The authors have no conflicts to disclose.

385 Author Contributions

F.L.: conceptualization, X-ray characterizations, computations, writing-original draft ; J.M-G. and S.L.:
 crystal growth, device fabrication ; J.Z. and J-M.V.: investigations, writing-review and editing , project
 administration ; E.G.-D.: conceptualization, methodology, writing-review and editing.

389 Data Availability Statement

The data that support the findings of this study are available from the corresponding authors upon reasonable request.

392 Acknowledgement

This work has received funding from the European Union's Horizon 2020 research and innovation program under the Photonics Public–Private Partnership (www.photonics21.org) with the project PEROXIS under grant agreement N° 871336.

397 **References**

- 398
- 399 [1] H. Wei *et al.*, « Sensitive X-ray detectors made of methylammonium lead tribromide perovskite
 400 single crystals », *Nature Photon*, vol. 10, n° 5, p. 333-339, mai 2016, doi:
 401 10.1038/nphoton.2016.41.
- 402 [2] W. Wei *et al.*, « Monolithic integration of hybrid perovskite single crystals with heterogenous
 403 substrate for highly sensitive X-ray imaging », *Nature Photon*, vol. 11, nº 5, p. 315-321, mai 2017,
 404 doi: 10.1038/nphoton.2017.43.
- 405 [3] X. Wang *et al.*, « PIN Diodes Array Made of Perovskite Single Crystal for X-Ray Imaging »,
 406 *physica status solidi (RRL) Rapid Research Letters*, vol. 12, n° 10, p. 1800380, oct. 2018, doi: 10.1002/pssr.201800380.
- 408 [4] Q. Xu *et al.*, « High-Performance Surface Barrier X-ray Detector Based on Methylammonium
 409 Lead Tribromide Single Crystals », *ACS Appl. Mater. Interfaces*, vol. 11, nº 10, p. 9679-9684,
 410 mars 2019, doi: 10.1021/acsami.8b21605.
- 411 [5] L. Li *et al.*, « Enhanced X-ray Sensitivity of MAPbBr3 Detector by Tailoring the Interface-States
 412 Density », *ACS Appl. Mater. Interfaces*, vol. 11, nº 7, p. 7522-7528, févr. 2019, doi:
 413 10.1021/acsami.8b18598.
- 414 [6] X. Geng *et al.*, « High-Quality Single Crystal Perovskite for Highly Sensitive X-Ray Detector »,
 415 *IEEE Electron Device Lett.*, vol. 41, nº 2, p. 256-259, févr. 2020, doi:
 416 10.1109/LED.2019.2960384.
- W. Zhang, H. Wang, H. Dong, F. Li, Z. Wang, et Y. Shao, « Mechanical polishing with chemical passivation of perovskite single crystals for high-performance X-ray detectors », *J. Mater. Chem.*C, vol. 10, n° 45, p. 17353-17363, 2022, doi: 10.1039/D2TC03850A.
- [8] M. I. Saidaminov *et al.*, « High-quality bulk hybrid perovskite single crystals within minutes by inverse temperature crystallization », *Nat Commun*, vol. 6, nº 1, p. 7586, nov. 2015, doi: 10.1038/ncomms8586.
- [9] S. Amari, J.-M. Verilhac, E. Gros D'Aillon, A. Ibanez, et J. Zaccaro, « Optimization of the
 Growth Conditions for High Quality CH₃ NH₃ PbBr₃ Hybrid Perovskite Single Crystals », *Crystal Growth & Design*, vol. 20, n° 3, p. 1665-1672, mars 2020, doi: 10.1021/acs.cgd.9b01429.
- [10]F. Yao *et al.*, « Room-temperature liquid diffused separation induced crystallization for highquality perovskite single crystals », *Nat Commun*, vol. 11, nº 1, p. 1194, mars 2020, doi:
 10.1038/s41467-020-15037-x.
- [11]K. Hecht, « Zum Mechanismus des lichtelektrischen Primärstromes in isolierenden Kristallen »,
 Z. Physik, vol. 77, n° 3-4, p. 235-245, mars 1932, doi: 10.1007/BF01338917.
- [12]T. E. Schlesinger et R. B. James, *Semiconductors for Room Temperature Nuclear Detector Applications*, Academic Press., vol. 43. in Semiconductors and Semimetals, vol. 43. 1995.
- 433 [13]G. Bertolini et A. Coche, *Semiconductor Detectors*, Eds. Interscience. Wiley, 1968.
- [14] J. Song *et al.*, « Facile Strategy for Facet Competition Management to Improve the Performance
 of Perovskite Single-Crystal X-ray Detectors », *J. Phys. Chem. Lett.*, vol. 11, nº 9, p. 3529-3535,
 mai 2020, doi: 10.1021/acs.jpclett.0c00770.
- [15]R. Tan *et al.*, « Characterization of solution grown 3D polycrystalline methylammonium lead
 tribromide for x-ray detection », *Journal of Applied Physics*, vol. 132, n° 20, p. 204503, nov.
 2022, doi: 10.1063/5.0100362.
- [16] J. M. Guillén *et al.*, « MAPb(Br1–xClx)3 Hybrid Perovskite Materials for Direct X-ray
 Detection », mars 2023, doi: https://doi.org/10.1021/acsaelm.3c00114.
- [17]Z. Gou *et al.*, « Self-Powered X-Ray Detector Based on All-Inorganic Perovskite Thick Film with
 High Sensitivity Under Low Dose Rate », *Phys. Status Solidi RRL*, vol. 13, nº 8, p. 1900094, août
 2019, doi: 10.1002/pssr.201900094.
- [18]K. Sakhatskyi *et al.*, « Stable perovskite single-crystal X-ray imaging detectors with single-photon sensitivity », *Nat. Photon.*, mai 2023, doi: 10.1038/s41566-023-01207-y.

396

- [19] A. O. Alvarez *et al.*, « Ionic Field Screening in MAPbBr3 Crystals Revealed from Remnant
 Sensitivity in X-ray Detection », *ACS Phys. Chem Au*, p. acsphyschemau.3c00002, mai 2023, doi:
 10.1021/acsphyschemau.3c00002.
- [20] S. Ramo, « Currents Induced by Electron Motion », *Proc. IRE*, vol. 27, nº 9, p. 584-585, sept.
 1939, doi: 10.1109/JRPROC.1939.228757.
- [21]C. A. Klein, « Bandgap Dependence and Related Features of Radiation Ionization Energies in
 Semiconductors », *Journal of Applied Physics*, vol. 39, nº 4, p. 2029-2038, mars 1968, doi:
 10.1063/1.1656484.
- [22]Y. He *et al.*, « High spectral resolution of gamma-rays at room temperature by perovskite
 CsPbBr3 single crystals », *Nat Commun*, vol. 9, nº 1, p. 1609, déc. 2018, doi: 10.1038/s41467018-04073-3.
- [23] M. Zanichelli, A. Santi, M. Pavesi, et A. Zappettini, « Charge collection in semi-insulator
 radiation detectors in the presence of a linear decreasing electric field », *J. Phys. D: Appl. Phys.*,
 vol. 46, nº 36, p. 365103, sept. 2013, doi: 10.1088/0022-3727/46/36/365103.
- [24]F. Principato, A. A. Turturici, M. Gallo, et L. Abbene, « Polarization phenomena in Al/p-CdTe/Pt
 X-ray detectors », *Nuclear Instruments and Methods in Physics Research Section A: Accelerators, Spectrometers, Detectors and Associated Equipment*, vol. 730, p. 141-145, déc. 2013, doi:
 10.1016/j.nima.2013.05.157.
- [25]H. Meng *et al.*, « Simulation of the Charge Collection in a Nonuniform Electric Field for MAPbI3
 Semiconductor Nuclear Detector With Schottky Contact », *IEEE Trans. Nucl. Sci.*, vol. 69, n° 9,
 p. 1990-1998, sept. 2022, doi: 10.1109/TNS.2022.3194317.
- 468 [26] S. Jia *et al.*, « Ion-Accumulation-Induced Charge Tunneling for High Gain Factor in P–I–N469 Structured Perovskite CH ₃ NH ₃ PbI ₃ X-Ray Detector », *Adv. Mater. Technol.*, p. 2100908, déc.
 470 2021, doi: 10.1002/admt.202100908.
- [27] O. Baussens *et al.*, « An insight into the charge carriers transport properties and electric field distribution of CH₃ NH₃ PbBr₃ thick single crystals », *Appl. Phys. Lett.*, vol. 117, nº 4, p. 041904, juill. 2020, doi: 10.1063/5.0011713.
- [28] Y. Luo *et al.*, « Direct Observation of Halide Migration and its Effect on the Photoluminescence of Methylammonium Lead Bromide Perovskite Single Crystals », *Adv. Mater.*, vol. 29, n° 43, p. 1703451, nov. 2017, doi: 10.1002/adma.201703451.
- [29] J. Pospisil *et al.*, « Reversible Formation of Gold Halides in Single-Crystal Hybrid-Perovskite/Au
 Interface upon Biasing and Effect on Electronic Carrier Injection », *Adv. Funct. Mater.*, vol. 29, n°
 32, p. 1900881, août 2019, doi: 10.1002/adfm.201900881.
- 480 [30] M. García-Batlle *et al.*, « Moving Ions Vary Electronic Conductivity in Lead Bromide Perovskite
 481 Single Crystals through Dynamic Doping », *Adv. Electron. Mater.*, vol. 6, nº 10, p. 2000485, oct.
 482 2020, doi: 10.1002/aelm.202000485.
- [31]M. García-Batlle *et al.*, « Coupling between Ion Drift and Kinetics of Electronic Current
 Transients in MAPbBr3 Single Crystals », *ACS Energy Lett.*, vol. 7, nº 3, p. 946-951, mars 2022,
 doi: 10.1021/acsenergylett.1c02578.
- [32] M. García-Batlle *et al.*, « Mobile Ion-Driven Modulation of Electronic Conductivity Explains
 Long-Timescale Electrical Response in Lead Iodide Perovskite Thick Pellets », *ACS Appl. Mater. Interfaces*, vol. 13, nº 30, p. 35617-35624, août 2021, doi: 10.1021/acsami.1c06046.
- [33]O. Almora, D. Miravel, G. Ilario, et G.-B. Germà, « Long-Term Field Screening by Mobile Ions
 in Thick Metal Halide Perovskites: Understanding Saturation Currents », *physica status solidi* (*RRL*) *Rapid Research Letters*, nº 16, p. 2200336, 2022, doi: DOI: 10.1002/pssr.202200336.
- 492 [34]S. H. Bennett *et al.*, « Charge transport comparison of FA, MA and Cs lead halide perovskite
 493 single crystals for radiation detection », *Front. Detect. Sci. Technol.*, vol. 1, p. 1249892, sept.
 494 2023, doi: 10.3389/fdest.2023.1249892.
- [35] X. Liu, H. Zhang, B. Zhang, J. Dong, W. Jie, et Y. Xu, « Charge Transport Behavior in SolutionGrown Methylammonium Lead Tribromide Perovskite Single Crystal Using α Particles », *J. Phys. Chem. C*, vol. 122, n° 26, p. 14355-14361, juill. 2018, doi: 10.1021/acs.jpcc.8b03512.
- 498 [36]E. A. Duijnstee, V. M. Le Corre, M. B. Johnston, L. J. A. Koster, J. Lim, et H. J. Snaith,
- 499 « Understanding Dark Current-Voltage Characteristics in Metal-Halide Perovskite Single
- 500 Crystals », *Phys. Rev. Applied*, vol. 15, nº 1, p. 014006, janv. 2021, doi:
- 501 10.1103/PhysRevApplied.15.014006.

- [37]B.-B. Zhang *et al.*, « Defect proliferation in CsPbBr3 crystal induced by ion migration », *Appl. Phys. Lett.*, vol. 116, nº 6, p. 063505, févr. 2020, doi: 10.1063/1.5134108.
- [38]R. Tan, B. Dryzhakov, J. Charest, B. Hu, M. Ahmadi, et E. Lukosi, « Improved Radiation Sensing
 with Methylammonium Lead Tribromide Perovskite Semiconductors », *Nuclear Instruments and Methods in Physics Research Section A: Accelerators, Spectrometers, Detectors and Associated Equipment*, vol. 986, p. 164710, janv. 2021, doi: 10.1016/j.nima.2020.164710.
- 508 [39]D. Weber, « CH3NH3PbX3, ein Pb(II)-System mit kubischer Perowskitstruktur / CH3NH3PbX3,
 509 a Pb(II)-System with Cubic Perovskite Structure », *Zeitschrift für Naturforschung B*, vol. 33, n°
 510 12, p. 1443-1445, déc. 1978, doi: 10.1515/znb-1978-1214.
- 511 [40]NIST, « XCOM: Photon Cross Sections Database ». [En ligne]. Disponible sur:
- 512 https://physics.nist.gov/PhysRefData/Xcom/html/xcom1.html
- [41]G. G. Poludniowski et P. M. Evans, « Calculation of x-ray spectra emerging from an x-ray tube.
 Part I. Electron penetration characteristics in x-ray targets », *Medical Physics*, vol. 34, n° 6Part1,
 p. 2164-2174, juin 2007, doi: 10.1118/1.2734725.
- 516 [42]G. G. Poludniowski, « Calculation of x-ray spectra emerging from an x-ray tube. Part II. X-ray
 517 production and filtration in x-ray targets », *Medical Physics*, vol. 34, n° 6Part1, p. 2175-2186, juin
 518 2007, doi: 10.1118/1.2734726.
- [43]G. Poludniowski, G. Landry, F. DeBlois, P. M. Evans, et F. Verhaegen, « SpekCalc : a program to calculate photon spectra from tungsten anode x-ray tubes », *Phys. Med. Biol.*, vol. 54, nº 19, p.
- 521 N433-N438, oct. 2009, doi: 10.1088/0031-9155/54/19/N01.