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Phase equilibria and solubility limits in the B-Ce-Fe-Nd system

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Context

- Powerful permanents magnets are polycrystalline material of Nd₂Fe₁₄B phase (T_1) .
- Ce-substituted Nd-Fe-B permanent magnets present poor magnetic properties.
- Magnetic performance of Ce-substituted Nd-Fe-B magnets can be enhanced through a **better control of grain boundary (GB)** compositions.

Results

Calculations

- To support the development of the CALPHAD database, DFT simulations were performed to calculate heat capacities of some complex compounds.
- Calculation features:
- CASTEP software
- PAW pseudo-potential method
- GGA PBE Monkhorst-Pack scheme Quasi-harmonic statistical model [2] Electronic contribution : Electron gas model Validation: Comparaison with literature Low temperature experimental heat capacities

- Especially in the Multi Main Phase (MMP) approach:
- Two type of τ_1 phases are synthetized separately : one with 100 at.% Nd and another which is a mixture 50 at.% Nd and 50 at.% Ce. They are crushed, the two powders are then mixed in various proportions and sintered.
- Several phases are present at GB including CeFe₂ and a RE-rich phase. Annealing the CeFe₂ phase allows to obtain more continuous GB, screening short range coupling between adjacent grains [1].

Objectives

- Develop a thermodynamic database of the (Ce-Nd)-Fe-B elements by using the CALPHAD method to optimize the fabrication process This will required on and GB phases (ReFe₂, RE-rich, etc.) to:
- Study their stabilities and range of solubilities
- Measure their transition temperatures
- Compute their specific heats and formation enthalpies



Phase diagram of Ce-Fe-Nd at 1023 K



Isopleth of Ce-Fe-Nd at 80 at.% Fe

Thermodynamic model: The CALPHAD approach

General expression for the Gibbs energy of a φ phase:



- Liquid phase : Ionic liquid model $(C^{\nu_c})_P(A^{\nu_A}, N^0, Va^{Q^-})_O$
- End-member interaction parameters : Redlich-Kister-Muggianu

Thermal analysis

Measured solubilities in the intermetallics after heat treatment (10 days at 1023 K):

- Solid solution (Ce,Nd)₂Fe₁₇ \rightarrow RE₂Fe₁₇
- 43.6 at.% Nd in CeFe₂

Comparaison DTA/calculation : Good results at 80 % at. Fe

Application Solidification without diffusion in solids Atomic fraction of elements in REFe₂ Solidification at thermodynamic $B_6Ce_{12}Fe_{70}Nd_{12}$ of а below 1300 K at thermodynamic equilibrium equilibrium 1600 -1600 -1600 $\begin{array}{c} -- \operatorname{RE}_{2}\operatorname{Fe}_{14}\operatorname{B} \\ -- \operatorname{REFe}_{4}\operatorname{B}_{4} \end{array}$ RE₂Fe₁₄B REFe₄B₄ — Liquid + x(Ce,REFe₂) — Liquid B₆Ce₁₂Fe₇₀Nd₁₂ B₆Ce₁₂Fe₇₀Nd₁₂ $B_6Ce_{12}Fe_{70}Nd_{12}$ equilibrium Thermodynamic (a) REFe₂ $x(Fe, REFe_{2})$ REFe 1500 1500 1500 BCC(Fe) RE-rich x(Nd,REFe₂) BCC(Fe) (Lever conditions) 1400 1400 1400 -(b) Without diffusion in solids below D(solides)=+∞ 1300 1300 K (Scheil-Gulliver conditions) 1300 1300 Ś D(solides)=0 $\widehat{\mathbf{X}}$ nperature (K) 1200 1200 -1200 nperature Three phase after cooling: atu (a) 1100 1100 -1100 - $RE_{2}Fe_{14}B$, $REFe_{4}B_{4}$ and $REFe_{2}$ (a) Nd-enrichment in REFe₂ which $\begin{bmatrix} 0 \\ 0 \end{bmatrix}$ 1000 1000 -1000 -

Solidification sample using the database:

Results of simulation:

- will reduce magnetic performance
 - (b) Disparition of the BBC(Fe) phase above 1300 K under thermodynamic equilibrium conditions. Formation of **REFe₂ and a RE-rich phase** during cooling, (b) closer to GB magnet effective compositions



Perspectives

Adding to the database :

- Other RE elements Dy, La, Pr
- Oxygen (traces in permanents magnets)
- Hydrogen (use in decrepitation processes)
- Other GB control elements : Co, Cu, Ga

Y. Zhang, T. Ma, J. Jin, J. Li, C. Wu, B. Shen, M. Yan, Effects of REFe₂ on microstructure and magnetic properties of Nd-Ce-Fe-B sintered magnets, Acta Materialia. 128 (2017) 22–30. [1] I.L. Fleche, Thermodynamical functions for crystals with large unit cells such as zircon, coffinite, fluorapatite, and iodoapatite from *ab initio* calculations, Phys. Rev. B. 65 (2002) 245116. [2]

Conclusions A thermodynamic database of the B-Ce-Fe-Nd system is under development. This computational tool will allow to

predict stable phases and their thermodynamic properties for any compositions and temperatures.

The database can be associated with other multi-physics codes to simulate composition evolutions within the main $RE_2Fe_{14}B$ grain and the GB during successive heat treatments. Other physical phenomena are controlling this evolution such as: diffusion, GB type, interfacial interactions, etc. and will required further dedicated modelling.