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Object kinetic Monte Carlo modelling of irradiation microstructures with elastic interactions

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Abstract.

Elastic interactions between point defects and sinks, such as dislocations and cavities, affect the diffusion of point defects and are responsible for some of the features observed in microstructures under irradiation. It is therefore necessary to include elastic interactions in kinetic simulations for a quantitative prediction of material properties. In this work a method is presented to accurately and efficiently evaluate the strain field in object kinetic Monte Carlo simulations. It can handle any strain field which is biharmonic, such as the one generated by a dislocation segment or a cavity in isotropic elasticity. A speed-up of several orders of magnitude is obtained compared to the direct summation over strain sources, so that simulations over experimental time scales can be performed within reasonable computation times. The case of a thin foil containing a high density of loops under irradiation is investigated. Loop growth rates are found to depend on the loop radius, as shown experimentally, but more complex effects due to the surrounding microstructure are also highlighted.

1 Introduction

Microstructure evolution under irradiation is known to be driven by the large 2 supersaturation of vacancies and self-interstitial atoms (SIAs) [1]. These point defects 3 migrate to pre-existing sinks, such as dislocations, grain boundaries and surfaces, or 4 agglomerate with each other, which typically leads to the formation of cavities and 5 dislocation loops. The sustained growth of both vacancy and SIA clusters when 6 only Frenkel pairs are generated by irradiation has been successfully explained by the 7 dislocation bias model [2]. This model relies on the fact that SIAs interact more 8 strongly with elastic fields due to their larger distortion field, and therefore diffuse 9 preferentially towards dislocations and dislocation loops, which create long-range elastic 10 fields compared to cavities. Modelling precisely the diffusion of point defects to the 11 various sinks of the microstructure is therefore important to predict quantitatively the 12 growth rate of dislocation loops and cavities, from which void swelling rates can be 13 deduced [3]. 14

Object kinetic Monte Carlo (OKMC) has been successfully used to model the 15 growth kinetics of vacancy and SIA clusters [4, 5, 6, 7, 8]. However, in most cases 16 elastic fields were not taken into account. They were included in OKMC only in a few 17 studies, considering that sinks do not evolve [9, 10, 11, 12, 13, 14], more rarely with 18 full account of sink evolution [15, 16]. To model the preferential absorption of SIAs 19 by dislocations without including the elastic field, a larger absorption region around 20 dislocations is generally adopted regarding SIAs. Beyond this absorption region, point 21 defects perform a pure random walk, *i.e.* the diffusion is not biased by elastic fields. This 22 approach is very efficient, since all jump frequencies are the same whatever the point 23 defect location in the matrix, but it relies on a parameter, the extent of the absorption 24 region, which must be determined by dedicated elastic bias calculations [10, 13]. Elastic 25 biases depend on the temperature, sink size in the case of dislocation loops and cavities, 26 so in general a tedious parametrization task is necessary to cover all relevant cases. In 27 addition biases are assumed not to depend on the surrounding microstructure, which has 28 proven to be erroneous [17, 16]. Taking into account explicitly the elastic interactions in 29 OKMC appears necessary to obtain results in quantitative agreement with experimental 30 results. 31

Performing efficient OKMC simulations with elastic interactions is not an easy task. 32 OKMC relies on the fast evaluation of jump frequencies, which becomes problematic 33 when they depend on the interaction of point defects with the local elastic field. 34 Within linear elasticity, it is convenient to obtain this local field by summing individual 35 contributions of all point defects and sinks in the system. This approach becomes 36 more and more expensive as the number of sinks in the system increases, or if periodic 37 boundary conditions (PBCs) are used. Indeed, due to the long range of the fields 38 generated by dislocations, it is not possible to use a cutoff distance for their evaluation. 39 In practice, it is nearly impossible to reach realistic simulation times if the field changes 40 after each defect jump. 41

Calculations may become considerably more tractable if the strain field of point 42 defects and point defect clusters which are mobile can be neglected. This is usually 43 the case if these mobile objects remain small. For example, in fcc metals Frank loops 44 are immobile and their strain field, within the dipolar approximation, is proportional 45 to the number of SIAs which form them. For typical sizes of several nanometers, it is 46 orders of magnitude larger than the strain field of SIAs or small mobile SIA clusters. 47 Therefore, if the concentration of mobile species is not too high, their contribution to 48 the field can be safely ignored. This assumption will be made in the rest of this work. 49 Since the field is unchanged after the jump of a defect, it may be advantageous to use 50 techniques which evaluate the field on a grid[‡]. The local field, after each defect jump, 51 can be calculated efficiently by interpolation of the values at the grid points. The field is 52 updated only when the sink microstructure changes, due to the absorption or emission 53 of a mobile species. The calculation of the field on the grid may be costly, so this 54 approach is especially useful when jump events are much more frequent than absorption 55 and emission events by sinks. This is true except at very low temperature (high sink 56 density) and high temperature (high emission rate of defects by sinks). 57

Calculating the field on a grid can be done efficiently by two main techniques. 58 The first one is to solve the problem in Fourier space, using fast Fourier transform 59 (FFT) [18]. It has been done successfully in phase field [19, 20] and discrete dislocation 60 dynamics simulations (DDD) [21, 22]. However this method requires some care to 61 avoid numerical artifacts due to the FFT grid [23]. Since the elastic field can vary 62 quite sharply, in particular close to sinks, a dense grid is required to obtain reliable 63 point defect trajectories in OKMC. This can become unnecessarily expensive for large 64 systems with low sink density. 65

The other technique is the fast multipole method (FMM) [24]. Although it is often used as an order N method to calculate the field at the location of the sources, it gives a local expansion of the field on a grid, which can be used in the present case to evaluate the local field at the position of mobile species. An upper bound for the error on the field can be defined rigorously and an adaptive version of the algorithm [25] permits to reduce the memory footprint and computation time when sources are heterogeneously distributed. Several flavours of the FMM have been used in DDD [26, 27, 28].

In this article a method based on the FMM is presented to calculate efficiently the elastic field in OKMC simulations. It enables the simulation of large elastically isotropic systems with common sinks observed in irradiated microstructures (cavities, dislocations, dislocation loops, stacking fault tetrahedra, etc.). We show that this real space approach is particularly convenient to simulate systems with heterogeneous distributions of sinks, or with free surfaces. As an application we simulate the evolution of dislocation loops in aluminum. We show that interstitial loops do not necessarily all

[‡] One could even envisage to precalculate the field on all atomic sites. Due to memory constraints, this can only be done on small systems. This method is especially useful when the microstructure does not evolve with time [12]. Otherwise an update of the field on each lattice site is necessary after a change in the microstructure, which may become computationally expensive.

grow under irradiation, in contradiction with previous OKMC and rate theory models,
but in agreement with experiments.

⁸² 1. OKMC modelling with elastic interactions

⁸³ 1.1. Jump frequencies of point defects in presence of an elastic field

⁸⁴ OKMC is a method which is well adapted to simulate the evolution of microstructures ⁸⁵ under irradiation. It relies on a database of known events, which are of several types: ⁸⁶ creation of new defects by irradiation, jump of a point defect from a stable position ⁸⁷ to another one, thermal emission of a point defect. Events are performed sequentially ⁸⁸ following the standard residence time algorithm [29, 30]. Elastic interactions between ⁸⁹ point defects and sinks alter the frequency of events. For example, the jump frequency ⁹⁰ of a point defect from $\boldsymbol{x} = (x_1, x_2, x_3)$ to $\boldsymbol{x} + \boldsymbol{h}$ can be written as

$$\nu^{h}(\boldsymbol{x}) = \nu_{0} \exp\left(-\frac{\Delta E^{h}(\boldsymbol{x})}{k_{\rm B}T}\right),\tag{1}$$

⁹¹ where ν_0 is an attempt frequency and ΔE^h is the activation energy, which reads

$$\Delta E^{\boldsymbol{h}}(\boldsymbol{x}) = E^{\mathrm{m}} + E^{\mathrm{int,s}}(\boldsymbol{x} + \boldsymbol{h}/2) - E^{\mathrm{int,e}}(\boldsymbol{x}).$$
(2)

In this expression, $E^{\rm m}$ is the migration energy without elastic interactions, $E^{\rm int,e}$ and $E^{\rm int,s}$ are the interaction energies for stable and saddle configurations, respectively. Here the saddle position is assumed to be located halfway between the two stable positions. A similar dependency of emission rates on elastic interactions exists [31] and will not be detailed here.

To lowest order, the interaction energy of a point defect with a strain field ε_{ij} depends on the elastic dipole of the point defect P_{ij} [32, 33] (summation on repeated indexes is implied):

$$E^{\rm int}(\boldsymbol{x}) = -P_{ij}\varepsilon_{ij}(\boldsymbol{x}). \tag{3}$$

For anisotropic defects, the elastic dipole depends on the defect orientation. It is in
 general different for stable and saddle configurations.

Within linear elasticity, the strain field can be calculated at any point by summing the strain contributions of all objects in the microstructure.

104 1.2. Strain sources

As discussed previously, the interaction between point defects or small mobile point defect clusters is not considered in this work. This is in general justified by the fact that the strain field due to large objects is much more intense. There is no difficulty to add this contribution of point defects as dipole-dipole interactions. However, in general the efficiency of the code would be considerably reduced due to the update of the field after any event affecting a point defect. Expressions of strain fields created by three common objects, a dislocation loop, a dislocation segment and a cavity, are recalled in this section, in the framework of isotropic elasticity. The shear modulus and Poisson's ratio are noted μ and ν , respectively. Only the solution in an infinite medium is given; surface effects are treated in Section 3.2.

¹¹⁵ 1.2.1. Dislocation segment and dislocation loop We follow the approach by Ghoniem ¹¹⁶ and Sun [34] to give explicit expressions for the strain. To derive the strain field, we ¹¹⁷ start from Mura's formula, which expresses the derivative of the displacement field \boldsymbol{u} ¹¹⁸ created by a dislocation loop [35, 36] as a line integral over the perimeter of the loop C¹¹⁹ (see also Fig. 1):

$$\frac{\partial u_i}{\partial x_j} = u_{i,j} = \epsilon_{jnh} b_m C_{pqmn} \oint_{\mathcal{C}} \frac{\partial}{\partial x_q} G_{ip}^{\infty}(\boldsymbol{x} - \boldsymbol{x'}) \, \mathrm{d}x'_h, \tag{4}$$

where ϵ_{ijk} is the permutation tensor, **b** is the Burgers vector, G_{ij}^{∞} is the elastic Green function in an infinite body, and C_{ijkl} are the elastic constants.



Figure 1. Notations used for a curved dislocation C.

For an isotropic material, the elastic Green function is given by $(\bar{x} = x - x')$, $\bar{x} = |\bar{x}|$:

$$G_{ij}^{\infty}(\bar{\boldsymbol{x}}) = \frac{1}{16\pi(1-\nu)\mu} \left(\delta_{ij} \frac{3-4\nu}{\bar{x}} + \frac{\bar{x}_i \bar{x}_j}{\bar{x}^3} \right) = \frac{1}{4\pi\mu} \frac{\delta_{ij}}{\bar{x}} - \frac{1}{16\pi\mu(1-\nu)} \frac{\partial^2}{\partial x_i \partial x_j} \bar{x}.$$
 (5)

124 In this equation δ_{ij} is the Kronecker delta.

125 Elastic constants can be written as

$$C_{pqmn} = \mu \left(\delta_{pm}\delta_{qn} + \delta_{pn}\delta_{qm}\right) + \frac{2\nu\mu}{1 - 2\nu}\delta_{pq}\delta_{mn},\tag{6}$$

and since $\varepsilon_{ij} = 1/2 (u_{i,j} + u_{j,i})$ the strain reads [34]:

$$\varepsilon_{ij}(\bar{\boldsymbol{x}}) = \frac{1}{8\pi} \oint_{\mathcal{C}} \left[-\frac{1}{2} (\epsilon_{jkl} b_i \bar{x}_{,l} + \epsilon_{ikl} b_j \bar{x}_{,l} - \epsilon_{ikl} b_l \bar{x}_{,j} - \epsilon_{jkl} b_l \bar{x}_{,i})_{,pp} + \frac{\epsilon_{kmn} b_n \bar{x}_{,mij}}{1 - \nu} \right] \mathrm{d}x'_k.$$
(7)

127 The parametric form of $\boldsymbol{x}'(v)$ given by Ghoniem and Sun is used:

$$\boldsymbol{x}'(v) = \boldsymbol{q}_i \mathcal{N}_i(v), \tag{8}$$

where $\mathcal{N}_i(v)$ are basis functions, v is a parameter and \mathbf{q}_i are generalized coordinates. The differential reads

$$dx'_{k} = x'_{k,v} dv = q_{ik} \mathcal{N}_{i,v}(v) dv.$$
(9)

Using quadrature to evaluate the integral, the strain can be numerically computed by summing over a finite number of strain sources:

$$\varepsilon_{ij}(\bar{\boldsymbol{x}}) \approx \frac{1}{8\pi} \sum_{\alpha=1}^{N_{q}} w_{\alpha} \left[-\frac{1}{2} (\epsilon_{jkl} b_{i} \bar{x}_{,l} + \epsilon_{ikl} b_{j} \bar{x}_{,l} - \epsilon_{ikl} b_{l} \bar{x}_{,j} - \epsilon_{jkl} b_{l} \bar{x}_{,i})_{,pp}(v_{\alpha}) + \frac{\epsilon_{kmn} b_{n} \bar{x}_{,mij}(v_{\alpha})}{1 - \nu} \right] x_{k,v}'(v_{\alpha})$$

$$\tag{10}$$

where $N_{\rm q}$ is the number of quadrature points and w_{α} are the quadrature weights. In the following Gauss-Legendre quadrature is used.

This expression can be simplified, since some terms are zero and $\bar{x}_{,ijk}$ is invariant by permutation. The expressions can be given, for example, as a function of $\bar{x}_{,111}$, $\bar{x}_{,112}$, $\bar{x}_{,113}$, $\bar{x}_{,123}$, $\bar{x}_{,221}$, $\bar{x}_{,222}$, $\bar{x}_{,223}$, $\bar{x}_{,331}$, $\bar{x}_{,332}$ and $\bar{x}_{,333}$ (similar simplications were performed for the stress in [34]). Explicit expressions of the strain components are given in Appendix A.

For a circular dislocation loop of radius r, located at the origin, and whose normal to its habit plane is n, the following coordinates can be chosen:

$$\boldsymbol{q}_{1} = r \begin{pmatrix} \frac{n_{2}^{2}}{1+n_{3}} + n_{3} \\ -\frac{n_{1}n_{2}}{1+n_{3}} \\ -n_{1} \end{pmatrix} \qquad \boldsymbol{q}_{2} = r \begin{pmatrix} \frac{-n_{1}n_{2}}{1+n_{3}} \\ \frac{n_{1}^{2}}{1+n_{3}} + n_{3} \\ -n_{2} \end{pmatrix}$$
(11)

and the basis functions are $(v \in [-1, 1])$ for Gauss-Legendre quadrature)

$$\mathcal{N}_1(v) = \cos(\pi v)$$
 $\mathcal{N}_2(v) = \sin(\pi v).$ (12)

 $_{142}$ Expression (10) can also be used for a dislocation segment. In this case,

$$\boldsymbol{q}_1 = \boldsymbol{x}_1 \qquad \boldsymbol{q}_2 = \boldsymbol{x}_2, \tag{13}$$

where x_1 and x_2 are the end points of the segment and the basis functions are $(v \in [-1,1])$

$$\mathcal{N}_1(v) = \frac{1}{2}(1-v)$$
 $\mathcal{N}_2(v) = \frac{1}{2}(1+v).$ (14)

145 1.2.2. Cavity Let us consider a cavity of radius r, which contains some gas with 146 pressure p, located at \mathbf{x}' . The strain field created by this cavity in the matrix, at 147 \mathbf{x} ($\bar{\mathbf{x}} > r$), is [37, 38]

$$\varepsilon_{ij}(\bar{\boldsymbol{x}}) = \left(p - \frac{2\hat{\gamma}}{r}\right) \frac{1}{4\mu} \left(\frac{r}{\bar{x}}\right)^3 \left[\delta_{ij} - 3\frac{\bar{x}_i \bar{x}_j}{\bar{x}^2}\right],\tag{15}$$

where $\hat{\gamma}$ is the surface tension of the metal.

It can also be written under the following form, which is more convenient for use in the FMM:

$$\varepsilon_{ij}(\bar{\boldsymbol{x}}) = -\left(p - \frac{2\hat{\gamma}}{r}\right) \frac{1}{4\mu} r^3 \frac{\partial^2}{\partial x'_i \partial x'_j} \left(\frac{1}{\bar{x}}\right).$$
(16)

¹⁵¹ Contrary to the dislocation segment and the dislocation loop, the strain is calculated ¹⁵² exactly by using a single strain source located at x'.

¹⁵³ 2. Fast calculation of the strain field

¹⁵⁴ 2.1. Calculation of the field using local expansions and FMM

Following Eq. (16), any component of the strain field generated by a cavity can be written as a derivative of $1/\bar{x}$, so it is harmonic. A general solution to $\Delta \phi = 0$ is

$$\phi(\boldsymbol{x}) = \sum_{l=0}^{\infty} \sum_{m=-l}^{l} \left(L_{lm} x^{l} Y_{l}^{m}(\theta, \varphi) + M_{lm} \frac{Y_{l}^{m}(\theta, \varphi)}{x^{l+1}} \right)$$
(17)

$$=\sum_{l=0}^{\infty}\sum_{m=-l}^{l}\left(L_{lm}R_{l}^{m}(x,\theta,\varphi)+M_{lm}I_{l}^{m}(x,\theta,\varphi)\right),$$
(18)

where Y_l^m are the spherical harmonics (see Appendix B, Eq. (B.1)), R_l^m and I_l^m are the regular and irregular solid harmonics, respectively. If the strain field is evaluated at \boldsymbol{x} with x < x', multipole expansion coefficients M_{lm} are all zero and only local expansion coefficients L_{lm} are useful. In this case Eq. (18) is called a local expansion. It can be truncated to a given value of l to give an approximate value of the strain field.

The strain field generated by a dislocation segment depends on the derivatives of \bar{x} (Eq. (7)), so it is not harmonic in general. However, \bar{x} is a solution to the biharmonic equation. This is also the case of its derivatives. Any solution ψ to the biharmonic equation can be written as [39]

$$\psi(\boldsymbol{x}) = \phi(\boldsymbol{x}) + x^2 \omega(\boldsymbol{x}), \tag{19}$$

where ϕ and ω are harmonic functions. So for a microstructure containing dislocations and cavities, if all strain sources are such that x' > x, any strain field component can be evaluated by a local expansion of the following type:

$$\psi(\boldsymbol{x}) = \sum_{l=0}^{\infty} \sum_{m=-l}^{l} L_{lm}^{(0)} R_{l}^{m}(x,\theta,\varphi) + x^{2} \sum_{l=0}^{\infty} \sum_{m=-l}^{l} L_{lm}^{(2)} R_{l}^{m}(x,\theta,\varphi).$$
(20)

¹⁶⁹ Compared to a Taylor expansion of the field, this decomposition is more compact: if ¹⁷⁰ the summation is performed up to order p, the number of terms is $O(p^2)$, whereas a ¹⁷¹ cartesian expansion requires $O(p^3)$ terms. Since a fast evaluation of the local expansion ¹⁷² is crucial in the KMC algorithm, using spherical harmonics is particularly interesting.

The FMM offers a rigorous framework to evaluate a field using local expansions with 173 a prescribed accuracy. It is based on a hierarchical set of boxes forming a mesh, which 174 can be uniform or adaptive. In the adaptive version, boxes are sub-divided in child boxes 175 until the number of sources in each box is lower than a threshold n_{max} . The case of a 176 uniform mesh is shown schematically in Fig. 2. The field at \boldsymbol{x} , inside box b, is obtained 177 by summing two terms. The first one is a local expansion as in Eq. (20), which contains 178 the contribution of all sources located in boxes "well separated" from b (in blue). In 179 particular, it contains information on periodic images of the simulation box and more 180 generally on boundary conditions (see section 3). The second term is a direct sum over 181

the other sources (in orange), using expressions (10) and (15). Using local expansions 182 only for boxes which are well separated from b ensures that local expansions converge. 183 Cubic boxes are considered well separated if they do not share a boundary point. In 184 this case, the minimal distance between the centers of boxes is $d_{\min} = 2/\sqrt{3}(r_b + r_b)$, 185 where r_b is the radius of the smallest sphere enclosing b. This worst case can be used to 186 define well-separated boxes when they are not cubic, by imposing that they are distant 187 from each other by at least d_{\min} . It is the criterion used in our FMM implementation, 188 which handles non-cubic boxes. 189



Figure 2. Schematic view of the calculation of the strain field using FMM in a microstructure containing dislocation loops and cavities. The contribution of blue sources, located in boxes which are well-separated from box b, containing the target point, is accounted for in a local expansion whose origin is the center of b. The contribution of other sources (in orange) is calculated with expressions (10) and (15).

The core of the FMM is the fast determination of local expansions. It relies on operations on multipole expansions of the fields. Once they are calculated for childless boxes, multipole expansions are translated and merged at lower levels of the hierarchy (upward pass) and converted into local expansions. Local expansions are then translated until they are expressed with respect to the center of childless boxes (downward pass). Details can be found, for example, in Ref. [25].

The FMM for harmonic fields, based on expansions in spherical harmonics, has 196 been amply discussed [24, 40]. In particular, it has been optimised to speed up the 197 conversion of multipole expansions into local expansions, using rotation of expansions 198 and exponential representations [25]. For biharmonic fields, Gumerov and Duraiswami 199 have proposed a modified version of the FMM, based on the decomposition in harmonic 200 fields given in Eq. (19) [39]. This algorithm uses the classical operations on harmonic 201 fields and includes an additional step due to the x^2 factor in Eq. (19). Since the definition 202 of spherical harmonics in the present work differs from the one they used, translation 203 and conversion formulas are given in Appendix C. The version implemented in this work 204 includes rotation of expansions [41]. 205

²⁰⁶ The FMM requires the calculation of multipole expansions of the field created by

a set of sources. In addition, for the adaptive version, which is used here, calculation of
local expansions directly from sources is also necessary. Coefficients of the expansions
are given for cavities and dislocation segments in the following sections.

210 2.2. Multipole expansion for a cavity

To obtain the multipole expansion of the strain field generated by a cavity, it is convenient to use the following expansion of the coulombic potential [42]:

$$\frac{1}{\bar{x}} = \sum_{l=0}^{\infty} \sum_{m=-l}^{l} x^{\prime l} Y_{l}^{-m}(\theta',\varphi') \frac{Y_{l}^{m}(\theta,\varphi)}{x^{l+1}} = \sum_{l=0}^{\infty} \sum_{m=-l}^{l} R_{l}^{-m}(x',\theta',\varphi') I_{l}^{m}(x,\theta,\varphi).$$
(21)

Inserting this expression in Eq. (16) yields the following multipole expansion for the strain field of a cavity:

$$\varepsilon_{ij}(\boldsymbol{x}) = \sum_{l=0}^{\infty} \sum_{m=-l}^{l} M_{lm} I_l^m(\boldsymbol{x}, \boldsymbol{\theta}, \boldsymbol{\varphi}), \qquad (22)$$

215 with

$$M_{lm} = -\left(p - \frac{2\hat{\gamma}}{r}\right) \frac{1}{4\mu} r^3 \frac{\partial^2}{\partial x'_i \partial x'_j} R_l^{-m}(x', \theta', \varphi').$$
(23)

²¹⁶ Second derivatives of regular solid harmonics are given in Appendix E.

217 2.3. Local expansion for a cavity

²¹⁸ Coefficients of the local expansion are obtained straightforwardly from the previous ²¹⁹ section: the strain field reads

$$\varepsilon_{ij}(\boldsymbol{x}) = \sum_{l=0}^{\infty} \sum_{m=-l}^{l} L_{lm} R_l^m(\boldsymbol{x}, \boldsymbol{\theta}, \boldsymbol{\varphi}), \qquad (24)$$

220 with

$$L_{lm} = -\left(p - \frac{2\hat{\gamma}}{r}\right) \frac{1}{4\mu} r^3 \frac{\partial^2}{\partial x'_i \partial x'_j} I_l^{-m}(x', \theta', \varphi').$$
(25)

²²¹ Second derivatives of irregular solid harmonics are given in Appendix D.

222 2.4. Multipole expansion for a dislocation segment

As can be seen from Eq. (10), the multipole expansion of the strain field of a dislocation segment is readily deduced from the expansion of $\bar{x}_{,ijk}$. Since \bar{x} is a solution to the biharmonic equation, it can be expressed as in Eq. (19) [39]. With our convention for spherical harmonics, we have, for x > x':

$$\bar{x} = \sum_{l=0}^{\infty} \sum_{m=-l}^{l} \frac{1}{2l+3} x'^2 R_l^{-m}(x',\theta',\varphi') I_l^m(x,\theta,\varphi) + x^2 \sum_{l=0}^{\infty} \sum_{m=-l}^{l} \frac{-1}{2l-1} R_l^{-m}(x',\theta',\varphi') I_l^m(x,\theta,\varphi).$$
(26)

²²⁷ The derivatives $\bar{x}_{,ijk} = -\bar{x}_{,i'j'k'}$ can thus be written under the following form:

$$\bar{x}_{,ijk} = \sum_{l=0}^{\infty} \sum_{m=-l}^{l} M_{lm}^{(0)} I_l^m(x,\theta,\varphi) + x^2 \sum_{l=0}^{\infty} \sum_{m=-l}^{l} M_{lm}^{(2)} I_l^m(x,\theta,\varphi),$$
(27)

with

$$M_{lm}^{(0)} = -\frac{1}{2l+3} \frac{\partial^3}{\partial x'_i \partial x'_j \partial x'_k} \left(x'^2 R_l^{-m}(x',\theta',\varphi') \right)$$
(28)

$$M_{lm}^{(2)} = \frac{1}{2l-1} \frac{\partial^3}{\partial x'_i \partial x'_j \partial x'_k} \left(R_l^{-m}(x',\theta',\varphi') \right).$$
⁽²⁹⁾

²²⁸ Derivatives of regular solid harmonics in Appendix E may be used to obtain more ²²⁹ convenient expressions of multipole expansion coefficients.

230 2.5. Local expansion for a dislocation segment

As for multipole expansions, local expansions can be calculated from the following expression of \bar{x} (for x < x'):

$$\bar{x} = \sum_{l=0}^{\infty} \sum_{m=-l}^{l} \frac{-x^{\prime 2}}{2l-1} I_{l}^{-m}(x^{\prime},\theta^{\prime},\varphi^{\prime}) R_{l}^{m}(x,\theta,\varphi) + x^{2} \sum_{l=0}^{\infty} \sum_{m=-l}^{l} \frac{1}{2l+3} I_{l}^{-m}(x^{\prime},\theta^{\prime},\varphi^{\prime}) R_{l}^{m}(x,\theta,\varphi).$$
(30)

²³³ The local expansion of $\bar{x}_{,ijk} = -\bar{x}_{,i'j'k'}$ is

$$\bar{x}_{,ijk} = \sum_{l=0}^{\infty} \sum_{m=-l}^{l} L_{lm}^{(0)} R_l^m(x,\theta,\varphi) + x^2 \sum_{l=0}^{\infty} \sum_{m=-l}^{l} L_{lm}^{(2)} R_l^m(x,\theta,\varphi),$$
(31)

with

$$L_{lm}^{(0)} = \frac{1}{2l-1} \frac{\partial^3}{\partial x'_i \partial x'_j \partial x'_k} \left(x'^2 I_l^{-m}(x',\theta',\varphi') \right)$$
(32)

$$L_{lm}^{(2)} = -\frac{1}{2l+3} \frac{\partial^3}{\partial x_i' \partial x_j' \partial x_k'} \left(I_l^{-m}(x',\theta',\varphi') \right).$$
(33)

Derivatives of irregular solid harmonics in Appendix D may be used to obtain more
 convenient expressions of local expansion coefficients.

236 2.6. Comparison with other FMM approaches for isotropic elasticity

The FMM has already been used to calculate elastic fields generated by dislocation ensembles. The first implementation for three dimensional DDD used expansions in cartesian coordinates [26]. As previously discussed, this approach is not optimal here

10

since local expansions are evaluated with $O(p^3)$ operations. On the contrary, the 240 evaluation of an expansion in regular solid harmonics requires only $O(p^2)$ operations. 241

The FMM based on spherical harmonics, which was initially developed for the 242 harmonic potential $1/\bar{x}$, has been used for other interaction potentials. Fu et al. [43] 243 showed that harmonic FMM can be used for interaction potentials of the form 244

$$\mathcal{F}(\mathcal{Q}(\boldsymbol{x}')/\bar{\boldsymbol{x}},\boldsymbol{x}),\tag{34}$$

where \mathcal{Q} is a function depending on data related to sources only and \mathcal{F} is a linear 245 operator on $\mathcal{Q}(\mathbf{x}')/\bar{\mathbf{x}}$. In general several harmonic FMM calls must be combined to 246 generate the field. In practice, some identities are useful to write potentials depending 247 on derivatives of \bar{x} , such as the elastic Green function and its derivatives, under the form 248 given by Eq. (34) [43, 44]. For example, the third derivatives can be written as [27]: 249

$$\bar{x}_{,ijk} = \left(x_k \frac{\partial^2}{\partial x_i \partial x_j} + \delta_{jk} \frac{\partial}{\partial x_i} + \delta_{ki} \frac{\partial}{\partial x_j}\right) \left(\frac{1}{\bar{x}}\right) - \frac{\partial^2}{\partial x_i \partial x_j} \left(\frac{x'_k}{\bar{x}}\right).$$
(35)

From Eq. (10), the strain field generated by a single source on a dislocation segment 250 can be expressed as 251

$$\varepsilon_{ij}(\bar{\boldsymbol{x}}) = q_{rst}^{ij} \bar{x}_{,rst}.$$
(36)

Inserting Eq. (35) into Eq. (36), we obtain: 252

$$\varepsilon_{ij}(\bar{\boldsymbol{x}}) = \left(x_t \frac{\partial^2}{\partial x_r \partial x_s} + \delta_{st} \frac{\partial}{\partial x_r} + \delta_{rt} \frac{\partial}{\partial x_s}\right) \left(\frac{q_{rst}^{ij}}{\bar{x}}\right) - \frac{\partial^2}{\partial x_s \partial x_t} \left(\frac{q_{rst}^{ij} x_t'}{\bar{x}}\right).$$
(37)

It can be seen that without any optimisation, the calculation of one component of the 253 strain field requires 36 harmonic FMM calls (27 for q_{rst}^{ij}/\bar{x} and 9 for $q_{rst}^{ij}x_t'/\bar{x}$). This 254 is significantly more expensive than the approach adopted here, which requires only 2 255 harmonic calls per component. Recently, Chen et al. [28] used the same decomposition 256 of \bar{x} as Fu et al. [43] to develop an FMM for the calculation of the stress field induced by 257 dislocation segments. However no details concerning the implementation were provided. 258 The number of FMM calls is limited in our approach because the derivatives are 259 taken with respect to the source variables x'_i instead of the target variables x_i (see 260 Eqs. (22)-(23) and (27)-(29)). A similar technique was already used for the calculation of 261 the magnetic field generated by magnetic dipoles [45]. Another advantage of this method 262 is that any harmonic or biharmonic field can be included in the present formalism 263 without requiring additional FMMs. If the derivatives are taken on the target variables 264 and if the fields are not derived the same way, one must keep track of the nature of the 265 sources, thereby increasing the number of FMMs.

3. Periodic boundary conditions and free surfaces 267

266

In OKMC simulations, PBCs may be used for self-defect migration, in order to simulate 268

the microstructure evolution in the bulk of large systems, where the effect of free surfaces 269

or grain boundaries can be locally neglected. Accordingly, the contribution of the elastic fields from the sources in the image boxes must be summed to provide the field due to a periodic layout of strain sources. We describe in section 3.1 how to perform this summation efficiently with FMM and to ensure that it converges to a correct value.

Considering some boundaries of the simulation box as free surfaces is another interesting case. This situation is representative of thin foils observed in transmission electron microscopy, which are a few hundred nanometers thick. The above expressions of the strain field induced by cavities and dislocation segments correspond to isolated defects in an infinite medium. A correction to the elastic field calculated by FMM is necessary to ensure that it corresponds to a system with free surfaces (section 3.2).

280 3.1. Periodic boundary conditions: removing the spurious dipolar contribution

To sum the contributions of periodic images, we extend the upward pass of the FMM to lower "macroscopic" levels of hierarchy and obtain the multipole expansions of larger and larger clusters of the initial simulation box [46]. The downward pass is performed from macroscopic levels down to the highest levels of hierarchy, with interaction lists containing boxes located in image simulation boxes. This simple method permits to add the elastic field of a large number of periodic image boxes at low computation cost.

However, it is known that the field obtained by direct summation over image boxes is 287 not necessarily absolutely convergent. In some cases the value of the sum depends on the 288 way the summation is performed and may not correspond to a periodic solution [47]. For 289 example, let us consider the strain fields created by dislocation loops and cavities, which 290 decay as $1/\bar{x}^3$ for large values of \bar{x} . The summation is only conditionally convergent if it 291 is performed along the three directions of space. Cai *et al.* have shown that the strain 292 field corresponding to a periodic solution $\varepsilon_{ij}^{\text{PBC}}$ differs from the field obtained by direct 293 summation by a constant field ε^0 [47]: 294

$$\varepsilon_{ij}(\boldsymbol{x}) = \varepsilon_{ij}^{\text{PBC}}(\boldsymbol{x}) + \varepsilon_{ij}^{0}.$$
(38)

This means that the displacement field resulting from the same summation procedure has two spurious components, a linear one and a constant one:

$$u_i(\boldsymbol{x}) = u_i^{\text{PBC}}(\boldsymbol{x}) + \boldsymbol{g}_i \cdot \boldsymbol{x} + u_i^0, \qquad (39)$$

²⁹⁷ where u_i^{PBC} is the periodic displacement field, g_i and u_i^0 are a constant vector and a ²⁹⁸ constant scalar, respectively. The constant field ε^0 , which must be subtracted from ε , ²⁹⁹ is deduced from g_i :

$$\varepsilon_{ij}^{0} = \frac{1}{2}(g_{ij} + g_{ji}).$$
(40)

The constant vector g_i can be computed by evaluating the displacement field (39) at one corner of the box and at the three adjacent corners. Although this method is simple and general, it is rather cumbersome in the present context, since with the FMM we only have access to the strain field. The evaluation of the displacement field would require to develop an additional FMM, which we want to avoid.

Recently it has been shown that for a box containing an elastic dipole component but no monopole term, the field ε^0 can be written in the following explicit form [48]:

$$\varepsilon_{ij}^{0}(\boldsymbol{x}) = \frac{1}{2} \int_{\mathcal{S}} p_{lk} n_k \left[G_{il,j}^{\infty}(\boldsymbol{\bar{x}}) + G_{jl,i}^{\infty}(\boldsymbol{\bar{x}}) \right] \mathrm{d}S', \tag{41}$$

where $\boldsymbol{p} = \boldsymbol{P}/V$, with \boldsymbol{P} the elastic dipole tensor of the system and V the volume of 307 the simulation box, n the outward-pointing normal to the integration surface \mathcal{S} . This 308 surface delimits the domain over which the direct summation over strain sources is 309 performed. Actually, since the function to integrate varies as $1/\bar{x}^2$, the integral does not 310 depend on the domain itself, but only on its shape. The summation domain must be 311 sufficiently large to fully account for higher order terms, which are absolutely convergent. 312 In practice, using 2 to 3 macroscopic levels in FMM is enough. The field ε^0 is calculated 313 at the center of the simulation box. In isotropic elasticity, it is convenient to use the 314 analytical form given in Ref. [48] and recalled in Appendix F. 315

Removing ε^0 amounts to simulating a finite macroscopic system under loading given by $\sigma = -p$ [48]. This means that to simulate a physical system with free surfaces, this loading should be removed. Finally the strain in a macroscopic system with free surfaces is obtained from the FMM field ε^{FMM} through

$$\boldsymbol{\varepsilon}(\boldsymbol{x}) = \boldsymbol{\varepsilon}^{\text{FMM}}(\boldsymbol{x}) - \boldsymbol{\varepsilon}^0 + \boldsymbol{S}\boldsymbol{p},$$
(42)

where S is the compliance matrix, *i.e.* the inverse of the stiffness matrix (see Eq. (6)):

$$S_{ijkl} = \frac{1}{4\mu} (\delta_{ik} \delta_{jl} + \delta_{il} \delta_{jk}) - \frac{\nu}{2\mu(1+\nu)} \delta_{ij} \delta_{kl}.$$
(43)

Tensor p remains to be determined. During the upward pass of the FMM, the multipole expansion of the field generated by all objects in the simulation box is calculated, so the strain field can be written as

$$\varepsilon_{ij}(\boldsymbol{x}) = \sum_{l=0}^{p} \sum_{m=-l}^{l} M_{lm}^{ij,(0)} \frac{Y_l^m(\theta,\varphi)}{x^{l+1}} + x^2 \sum_{l=0}^{p} \sum_{m=-l}^{l} M_{lm}^{ij,(2)} \frac{Y_l^m(\theta,\varphi)}{x^{l+1}}.$$
 (44)

Far from the simulation box, one must recover the strain field of an elastic dipole, decaying as $1/x^3$. This means that $M_{2m}^{ij,(0)}$ and $M_{4m}^{ij,(2)}$ are related to the total elastic dipole in the simulation box. The displacement field induced by an elastic dipole \boldsymbol{P} located at the origin in the middle of the box is [49]

$$u_i^{\rm dip}(\boldsymbol{x}) = -G_{ij,k}^{\infty}(\boldsymbol{x})P_{jk} \tag{45}$$

$$= -\frac{1}{16\pi\mu(1-\nu)} \left[-2(1-2\nu)\frac{x_k}{x^3}P_{ik} + \frac{x_i}{x^3}\operatorname{Tr}\left(\boldsymbol{P}\right) - 3\frac{x_ix_kx_l}{x^5}P_{lk} \right], \quad (46)$$

328 from which the strain field is obtained:

$$\varepsilon_{ij}^{\text{dip}}(\boldsymbol{x}) = -\frac{1}{16\pi\mu(1-\nu)} \left[-2(1-2\nu)\frac{P_{ij}}{x^3} + \frac{x^2\delta_{ij} - 3x_ix_j}{x^5} \operatorname{Tr}(\boldsymbol{P}) - 6\nu\frac{x_i}{x^5} (x_k P_{jk}) - 6\nu\frac{x_j}{x^5} (x_k P_{ik}) + 3\frac{5x_ix_j - \delta_{ij}x^2}{x^7} x_k x_l P_{kl} \right]. \quad (47)$$

From this expression, one can evaluate the multipole expansion coefficients $M_{20}^{ij,(0)}$ (see Eq. (B.3))

$$M_{20}^{ij,(0)} = \int_0^{2\pi} \int_0^{\pi} \frac{5}{4\pi} \varepsilon_{ij}^{\text{dip}}(\boldsymbol{x}) x^3 (Y_2^0(\theta,\varphi))^* \sin\theta \,\mathrm{d}\theta \mathrm{d}\varphi.$$
(48)

³³¹ This leads to the following linear system:

$$M_{20}^{11,(0)} = \frac{1}{16\pi\mu(1-\nu)} \left[\frac{4}{7} (4-7\nu)P_{11} - \frac{4}{7}P_{22} + \frac{2}{7}P_{33} \right]$$
(49)

$$M_{20}^{22,(0)} = \frac{1}{16\pi\mu(1-\nu)} \left[-\frac{4}{7}P_{11} + \frac{4}{7}(4-7\nu)P_{22} + \frac{2}{7}P_{33} \right]$$
(50)

$$M_{20}^{33,(0)} = \frac{1}{16\pi\mu(1-\nu)} \left[\frac{2}{7} P_{11} + \frac{2}{7} P_{22} - \frac{8}{7} (4-7\nu) P_{33} \right]$$
(51)

$$M_{20}^{23,(0)} = -\frac{1}{16\pi\mu(1-\nu)}\frac{2}{7}(5-7\nu)P_{23}$$
(52)

$$M_{20}^{13,(0)} = -\frac{1}{16\pi\mu(1-\nu)}\frac{2}{7}(5-7\nu)P_{13}$$
(53)

$$M_{20}^{12,(0)} = \frac{1}{16\pi\mu(1-\nu)} \frac{4}{7} (5-7\nu) P_{12}.$$
(54)

This system is underdetermined for $\nu = 1/2$ or $\nu = 4/7$. In practice $\nu < 1/2$ so it can be safely used to calculate the dipole tensor associated to the simulation box and thus p.

335 3.2. Free surfaces

The elastic Green function given in Eq. (5) is only valid for an infinite medium. If some boundaries of the simulation box correspond to a free surface, the strain field calculated with this Green function may be erroneous. Unphysical forces T may appear at the surface if some objects are too close to it. In order to obtain the correct elastic solution, we adopt the method proposed by van der Giessen and Needleman [50]. They add a field which cancels surface tractions produced by the solution in an infinite medium. This field is the solution of the following equations:

$$\boldsymbol{\nabla} \cdot \boldsymbol{\sigma} = 0 \quad \text{in } \mathcal{V} \tag{55}$$

$$\boldsymbol{\sigma} \cdot \boldsymbol{n} = -\boldsymbol{T} \quad \text{on } \mathcal{S}^{\mathrm{f}},\tag{56}$$

where \mathcal{V} is the domain corresponding to the simulation box, \mathcal{S}^{f} its free surfaces, \boldsymbol{n} is the outward normal vector to \mathcal{S}^{f} . Although surface tractions can vary steeply, the field induced by these forces smoothen with depth. An adaptive mesh is therefore particularly adapted to solve these equations. The finite element method (FEM) can be easily used with adaptive meshes, which makes this method particularly interesting in this context. It is based on the variational formulation of problem (55)-(56) [51]: we must find \boldsymbol{u} such that

$$\int_{\mathcal{V}} \boldsymbol{\sigma}(\boldsymbol{u}) : \boldsymbol{\varepsilon}(\boldsymbol{v}) \, \mathrm{d}V = -\int_{\mathcal{S}^{\mathrm{f}}} \boldsymbol{T} \cdot \boldsymbol{v} \, \mathrm{d}S \qquad \forall \boldsymbol{v} \in H^{1},$$
(57)

with \boldsymbol{u} the displacement field to be found, \boldsymbol{v} a vector test function and H^1 the Sobolev space of vector functions. The colon operator is the inner product between two tensors, *i.e.* $\boldsymbol{\sigma} : \boldsymbol{\varepsilon} = \sigma_{ij} \varepsilon_{ij}$. Since there is no Dirichlet boundary condition (on \boldsymbol{u}), the solution is not unique. If free surfaces are considered in the three directions of space, *i.e.* with pure Neumann boundary conditions, any rigid body motion, given by (with \boldsymbol{a} , \boldsymbol{b} constant vectors)

$$\boldsymbol{u}(\boldsymbol{x}) = \boldsymbol{a} \times \boldsymbol{x} + \boldsymbol{b},\tag{58}$$

can be added to the solution ($\varepsilon(u) = 0$). The space of rigid body motions RM consists of translations and rotations, and for a pure Neumann problem, it is of dimension 6.

A unique solution can be found by setting the displacement of some points to zero [52]. Here we use Lagrange multipliers [51] to impose that the solution is linearly independent of functions in RM. This leads to the following saddle point problem [51, 53]: we must find $(\boldsymbol{u}, \boldsymbol{\lambda})$, with \boldsymbol{u} in H^1 and $\boldsymbol{\lambda}$ in RM, such that

$$\int_{\mathcal{V}} \boldsymbol{\sigma}(\boldsymbol{u}) : \boldsymbol{\varepsilon}(\boldsymbol{v}) \, \mathrm{d}V + \int_{\mathcal{V}} \boldsymbol{v} \cdot \boldsymbol{\lambda} \, \mathrm{d}V = -\int_{\mathcal{S}^{\mathrm{f}}} \boldsymbol{T} \cdot \boldsymbol{v} \, \mathrm{d}S \qquad \forall \boldsymbol{v} \in H^{1}$$
(59)

$$\int_{\mathcal{V}} \boldsymbol{u} \cdot \boldsymbol{\mu} \, \mathrm{d}V = 0 \qquad \forall \boldsymbol{\mu} \in RM.$$
 (60)

The number of rigid body motions to be blocked depends on the number of surfaces. The 362 three translation motions must always be included in the rigid motions to be blocked; 363 however the three rotations must be included only if the simulation box has free surfaces 364 along the three directions of space. If PBCs are used along one direction, only the 365 rotation around this direction must be blocked, so the null space of rigid body motions 366 is of dimension 4. If PBCs are used along two directions, only translations must be 367 considered and the null space is only of dimension 3. In practice, problem (59)-(60) was 368 solved using the open-source software FEniCS [54]. 369

The mesh to be used in FEM calculations can be fixed at the beginning of an OKMC simulation, but this is not always optimal in terms of efficiency and accuracy.

If strain sources remain far from surfaces, spatial variations of surface tractions are 372 gentle and a coarse mesh can be used near surfaces without loss of accuracy, resulting in 373 highly efficient FEM solving. If strain sources get closer to the surface as the simulation 374 proceeds, surface tractions are expected to vary much more steeply and the mesh should 375 be locally refined. Refining the mesh only where it is necessary is important to keep 376 the FEM solving efficient. The adaptive FMM provides a rather natural way to do so. 377 Since the FMM box size depends on the local density of sources, steep variations of the 378 traction field resulting from sources close to the surface are expected to be accompanied 379 with a local decrease of the FMM box size. Accordingly, we set the typical size of 380 FEM elements at the surface proportional to the size of FMM boxes. In addition, the 381 number of surface traction evaluations is set the same for all FMM boxes, resulting in a 382 higher density of evaluations where FMM boxes are smaller. A schematical view of the 383 FMM/FEM coupling is shown in Fig. 3. 384



Figure 3. Coupling of adaptive FMM and FEM to obtain the solution compatible with traction-free upper and lower boundaries. The solution in an infinite medium is first obtained, using FMM (ε^{FMM}). The adaptive grid is used to evaluate traction forces T on surfaces. Then the solution at mechanical equilibrium under prescribed traction forces -T is determined with FEM (ε^{FEM}). The solution of the initial problem is $\varepsilon = \varepsilon^{\text{FMM}} + \varepsilon^{\text{FEM}}$.

We chose the open-source software Gmsh [55] to generate the FEM mesh. This mesh is updated as infrequently as possible, only when the FMM adaptive mesh is modified. Otherwise, only surface tractions are re-evaluated before the FEM solving. Between two successive FEM solvings with the same mesh, the solutions are not expected to vary much and it is advantageous to re-use the solution from a preceding solving as an initial guess if an iterative solver is used. If a direct solver is employed, it is useful to re-use the LU factorization.

³⁹² 4. Numerical results

In this section we present a few test cases, to assess the accuracy and performance of the method, and to highlight its capabilities. All tests are performed in aluminum at room temperature. Parameters, which are taken from Ref. [13], are reminded in Table 1.

Parameter	Symbol	Value
Lattice parameter	a_0	0.405 nm
Atomic volume	Ω	$a_0^3/4 = 0.0167 \text{ nm}^3$
Poisson's ratio	ν	0.35
Shear modulus	μ	$26 { m GPa}$
Burgers vector (Frank loops)	\boldsymbol{b}	$a_0[111]/3$
Capture radius of dislocations	$r_{ m c}$	0.47 nm
Migration energy of vacancies	$E_{\rm v}^{\rm m}$	$0.605 \mathrm{eV}$
Migration energy of SIAs	$E_{\rm i}^{\rm m}$	$0.105 \ \mathrm{eV}$
Attempt frequency	$ u_0$	$10^{13} { m Hz}$
Elastic dipole of vacancies at stable position Elastic dipole of vacancies at saddle position (jump in [110] direction)	$oldsymbol{P}_{\mathrm{v}}^{\mathrm{e}}$	$\begin{bmatrix} -3.238 & 0 & 0 \\ 0 & -3.238 & 0 \\ 0 & 0 & -3.238 \end{bmatrix} eV$ $\begin{bmatrix} -2.866 & -0.080 & 0 \\ -0.080 & -2.866 & 0 \\ 0 & 0 & 1.000 \end{bmatrix} eV$
Elastic dipole of SIAs at stable position ([100] orientation)	$oldsymbol{P}_{\mathrm{i}}^{\mathrm{e}}$	$\begin{bmatrix} 19.652 & 0 & 0 \\ 0 & 18.518 & 0 \\ 0 & 0 & 18.518 \end{bmatrix} eV$
Elastic dipole of SIAs at saddle position (jump in [110] direction)	$oldsymbol{P}^{ m s}_{ m i}$	$ \begin{bmatrix} 19.498 & 1.133 & 0 \\ 1.133 & 19.498 & 0 \\ 0 & 0 & 19.034 \end{bmatrix} eV $

 Table 1. Material parameters for aluminum

396 4.1. Accuracy and performance of the method

The truncation of expansions to order p limits the accuracy of the method. The error on any component of the strain field can be conveniently estimated by the L^2 relative error norm [25, 39]:

$$e_{2} = \left(\frac{\sum_{k=1}^{n} |\varepsilon^{\text{direct}}(\boldsymbol{x}_{k}) - \varepsilon^{\text{FMM}}(\boldsymbol{x}_{k})|^{2}}{\sum_{k=1}^{n} |\varepsilon^{\text{direct}}(\boldsymbol{x}_{k})|^{2}}\right)^{1/2},$$
(61)

where the indexes of the strain component have been dropped for the sake of clarity. The strain field component ε^{FMM} is calculated by FMM, with expansions truncated to order

p. A direct summation over all strain sources is performed to evaluate $\varepsilon^{\text{direct}}$, which is 402 our reference value. The location of the *n* strain field evaluations (with $n = 10^4$ here) is 403 chosen randomly in the matrix, outside the absorption region of sinks. In the test case 404 considered here, sinks are Frank loops of the same radius, equal to 5 nm. Each loop is 405 modelled as a collection of straight dislocation segments of maximum length equal to 406 1 nm. The absorption regions are cylinders of radius $r_{\rm c} = 2b$, where $b = |\mathbf{b}|$, centered 407 on the dislocation segments. A total of 40 loops, randomly placed in a cubic system of 408 edge length 200 nm, generate the strain field. Neither periodic boundary conditions nor 409 coupling with FEM are considered. 410

The relative error on ε_{11} is shown in Fig. 4 as a function of truncation order p. It steadily decreases with increasing order. Since the computation cost increases with p, its value should be chosen as small as possible. In general, a value of e_2 lower than 10^{-2} is unnecessary, so in the following p = 7 is adopted.



Figure 4. L^2 relative error norm on component ε_{11} as a function of the order p of the multipole and local expansions in the FMM. The system is cubic, of size 200 nm. It contains 40 Frank dislocation loops of 5 nm radius. The error is calculated on $n = 10^4$ points located outside the absorption region of loops.

The main purpose of using FMM is to reduce the computation time for the evaluation of the strain field. Computation times obtained with a direct calculation and with FMM, assuming that the expansions are up to date, are shown in Fig. 5-(a). The system is the same as before, but the number of loops, or equivalently the number of strain sources, is varied. The highest number of loops considered (100 loops) corresponds to loop densities typically seen in irradiated materials ($\sim 10^{22} \text{ m}^{-3}$). The computation

time with the direct approach increases linearly with the number of sources and reaches 421 values of more than 10^{-3} s for 100 loops. When the FMM is used, the computation 422 time is reduced by a factor of around 500 in this high density case. With the FMM, 423 the field is calculated as the sum of two contributions. The first one includes sources 424 which are sufficiently far from the box containing the target point. It is evaluated with 425 a local expansion to order p. The associated computation time is independent of the 426 number of sources. The second contribution is due to the remaining sources, which are 427 too close to the box containing the target point to be included in the local expansion. 428 In the adaptive FMM, they are taken into account with a multipole expansion if the 429 box containing the sources is sufficiently far from the target and if it has a large number 430 of sources. Otherwise, a direct summation on sources is performed (orange sources in 431 Fig. 2). This second contribution explains the increase in computation time with the 432 number of loops. If the maximum number of sources per box is reduced from 70 to 30, 433 additional levels are introduced and in general, the number of sources which contribute 434 to the direct sum decreases. That is why the computation time also decreases. The case 435 of a uniform FMM, with a box size equal to 5 nm, is also reported in Fig. 5-(a). If the 436 number of loops is small, boxes containing target points are most often surrounded by 437 boxes with no sources. The computational cost corresponds to the evaluation of a local 438 expansion. Once the number of loops is sufficiently large to populate the boxes around 439 each target point, the computation time increases because of the direct summation. 440

As a first approximation, PBCs can be introduced in the direct approach by 441 summing the contributions of sources located in the first neighbour shell of periodic 442 images. This is not a bad approximation if PBCs are used along at most two directions, 443 since the sum is absolutely convergent in this case (see Section 3.1). The computation 444 time when PBCs are set along two directions corresponds to the dashed black line in 445 Fig. 5-(a). As expected, it is around 9 times higher than without PBCs. With FMM, 446 the computation time only marginally increases, because the number of sources on 447 which the direct summation is performed only increases for target points located close 448 to the boundaries of the system. This is a clear advantage of the FMM over the direct 449 calculation. 450

Up to now it has been assumed that multipole and local expansions are up to date. 451 Each time a sink absorbs or emits a point defect, expansions must be updated. We 452 see in Fig. 5-(b) that with our version of the FMM, which is not fully optimized, the 453 calculation of multipole and local expansions for all strain components can be up to a few 454 seconds for dense microstructures. It increases if $n_{\rm max}$ decreases, since additional levels 455 are introduced. We have seen that on the contrary, reducing n_{max} speeds up the field 456 evaluation. So $n_{\rm max}$ must be adjusted to minimize the sum of the two contributions 457 to the computation time. Empirically, $n_{\rm max} = 70$ appears to be satisfactory with 458 the present implementation. We also see in Fig. 5-(b) that considering PBCs hardly 459 affects the time to update expansions, so that the global computation time is nearly 460 independent of PBCs with FMM. As in Fig. 5-(a), the computation time for the uniform 461 FMM is reported. With such a small box size of 5 nm, the update of expansions takes 462



Figure 5. (a) Computation time for one strain field evaluation, for direct and FMM calculations, as a function of the number of loops in a $(200 \text{ nm})^3$ system. Loops have a radius of 5 nm. Dashed curves refer to periodic calculations along two directions. Periodic images are only partially accounted for in the direct calculation, by including the first neighbour shell (8 images boxes). The adaptive version of the FMM is used with two different criteria for the division of boxes ($n_{\text{max}} = 30$ and $n_{\text{max}} = 70$). The green curve corresponds to a FMM calculation with a 5 nm uniform mesh. (b) Computation time for the initial evaluation of local expansions by FMM.

much more time than with the adaptive FMM. Although the box size can be tuned to obtain satisfactory performance, it becomes difficult to do so when the number of sources evolves appreciably with time. This is typically the case under irradiation, with the nucleation and growth of clusters. So in general the adaptive version should be preferred.

Since the update of FMM expansions takes significant time, the FMM is useful only if defects perform a large number of jumps n between two absorption or emission events. The speed-up is then

$$s = \frac{(z+1)nt_{\text{direct}}}{(z+1)nt_{\text{FMM}} + t_{\text{update}}},\tag{62}$$

where t_{direct} and t_{FMM} are the computation time per evaluation of strain field (Fig. 5-471 (a)) in the direct and FMM approaches, respectively, and t_{update} is the time required 472 to update FMM expansion coefficients (Fig 5-(b)). The factor (z + 1) accounts for the 473 z possible jumps a defect can perform (8 for an SIA, 12 for a vacancy), because an 474 estimation of the field is necessary for each saddle point. In addition the interaction 475 energy for the initial stable configuration must be calculated. To evaluate n, let us 476 consider absorption events only and neglect elastic interactions. The average number of 477 jumps of a defect before absorption is [56] 478

$$n = \frac{12}{k^2 a_0^2},\tag{63}$$

where the sink strength is assumed to be the one of a concentration C of loops of radius r and capture radius r_c [57]:

$$k^2 = \frac{4\pi^2 r}{\ln\left(\frac{8r}{r_c}\right)}C.$$
(64)

For one loop in the simulation box, we obtain around $n = 10^7$ jumps before absorption, while for 100 loops, n is around 10^5 . From results shown in Fig. 5, the resulting speedup for a system with PBCs in two directions is around 250 for one loop and 1700 for 100 loops. The proportion of time taken for the update of FMM expansion coefficients increases with the number of loops, so that $(z+1)nt_{\text{FMM}} \approx t_{\text{update}}$ for 100 loops. However, the time for direct evaluation of the field also increases rapidly with the number of loops. This results in a larger speed-up if the number of loops increases.

A more precise estimation of the speed-up is obtained by measuring the average 488 CPU time per KMC time step directly in a kinetic simulation. This measurement also 489 takes into account a variety of other operations occurring during a KMC time step, such 490 as the computation of energy, the handling of collisions between defects, etc. Previous 491 microstructures are taken as starting configurations. They evolve following the creation 492 of Frenkel pairs at a rate of 10^{-4} dpa/s. Results are shown in Fig. 6. The speed-up is 493 similar to the previous estimation, from two to three orders of magnitude depending on 494 the number of loops in the system. As before, the update of FMM expansion coefficients 495 takes a larger proportion of the computation time as the number of loops increases. We 496



Figure 6. Computation time per KMC step as a function of the number of loops in a $(200 \text{ nm})^3$ system. Loops have a radius of 5 nm. Direct and FMM calculation of the strain field are shown in black solid lines. For FMM, the orange colour area indicates the proportion of time spent in the update of FMM expansion coefficients, other operations being represented in blue.

⁴⁹⁷ note that in these simulations, we avoid recalculating all multipole expansion coefficients
⁴⁹⁸ each time an absorption event occurs. Only boxes whose sources have changed have their
⁴⁹⁹ multipole expansions updated. Then only necessary multipole-to-local operations are
⁵⁰⁰ performed, which saves computation time.

501 4.2. Impact of free surface on dislocation loop growth

In the vicinity of a surface, the growth rate of clusters is altered, because a proportion 502 of point defects is absorbed by this sink instead of contributing to cluster evolution. 503 This leads, for example, to the presence of void and loop denuded zones [58, 59, 60]. 504 This effect has been investigated with standard OKMC [61]. With the present method 505 it is possible to study a more subtle effect, related to the strain field generated by 506 clusters. We have seen in section 3.2 that if a cluster is located close to a surface, the 507 associated strain field does not correspond to the field generated in an infinite medium, 508 since surfaces must be traction-free. The resulting modified energy landscape impacts 509 the diffusion paths of point defects and thus may change the cluster behaviour. 510

In the following we consider a system with free surfaces along e_3 , containing a loop



Figure 7. Visualization of the computation of the strain field produced by a dislocation loop located at 2 nm from the upper surface of a $(30 \text{ nm})^3$ system, at (a,c,e) t = 10 s and (b,d,f) t = 40 s. (a,b) Adaptive FMM mesh. (c,d) Component σ_{33} induced by the presence of the loop, if the solution in an infinite medium is used. This solution leads to significant traction forces T on the upper surface. (e,f) Component σ_{33} when the FEM solution of the mechanical equilibrium with prescribed forces -T is added to the infinite medium solution.

nucleus at a variable distance d from the upper surface. PBCs are used in the two other 512 directions. The loop nucleus is an interstitial cluster containing 11 SIAs (r = 0.5 nm). 513 which is assumed to be immobile and to grow as a Frank loop oriented parallel to 514 the surface. The dimensions of the system are 30 nm \times 30 nm \times 30 nm. SIAs and 515 vacancies are inserted as Frenkel pairs at a dose rate of 5×10^{-3} dpa/s. A view of the 516 FMM adaptive mesh at two different times is given in Fig. 7-(a,b). The region where 517 the mesh is refined correlates with the position of the dislocation line. Following the 518 coupling scheme proposed in Fig. 3, this mesh refinement leads to a fine FEM mesh at 519 the upper surface close to the dislocation line, where the field varies steeply (Fig. 7-520 (c,d)). In the case shown in Fig. 7, which corresponds to a dislocation loop 2 nm away 521 from the upper surface, traction forces created by the solution in a infinite medium can 522 be larger than 1 GPa. When the FEM solution of the mechanical equilibrium with 523 opposite traction forces is added, traction forces are substantially reduced. We see that 524 this induces a significant change in the stress inside the material (Fig. 7-(e,f)). 525

The evolution of loop size with time is shown in Fig. 8, including or not the compensation of traction forces. 20 simulations have been performed for distances to the surface ranging from 2 to 15 nm. One sees that except when the loop is very close to the surface (d = 2 nm), the image field correction has only a small influence on loop growth.



Figure 8. Evolution of loop radius with time for different distances to the surface d. 20 simulations have been performed for each value of d, with and without compensation of traction forces created by the elastic solution in an infinite medium.

531 4.3. Evolution of dislocation loops in thin foils under irradiation

It is known that in thin foils of fcc and bcc metals, interstitial dislocation loops 532 grow under irradiation [62, 63]. This growth is explained successfully by the so-called 533 "dislocation bias model" [2]. Loops interact more strongly with SIAs than vacancies, 534 which results in a preferential absorption of SIAs. The remaining net quantity of 535 vacancies in the matrix is absorbed by sinks which generate shorter range, or less 536 intense, elastic fields. These sinks can be cavities, surfaces or grain boundaries for 537 example. However, shrinkage of interstitial dislocation loops has been observed in some 538 cases under irradiation. In aluminum, it has been documented by Jitsukawa and Hojou 539 in 600 nm thick foils irradiated in a transmission electron microscope [64]. These authors 540 have explained loop shrinkage by a dependency of the elastic bias of loops on their size. 541 Indeed, even though all loops have a positive elastic bias, which means they interact 542 more strongly with SIAs than vacancies, it does not imply that they all grow. Since 543 vacancies and SIAs are produced in equal quantities, loops with the smallest elastic 544 bias necessarily absorb more vacancies than SIAs if no sinks with lower elastic bias are 545 present (for example, surfaces). In general, small loops tend to exhibit lower growth 546



Figure 9. One of the 10 loop configurations considered to investigate loop growth dependency on distance to surface and loop radius.

rates than large loops, which points to an increase of the elastic bias with size. Numerical evaluations of bias of isolated loops are consistent with these findings [65, 66, 67].

Absorption efficiencies of loops depend not only on their size, but also on the 549 surrounding microstructure [17, 16]. To assess more precisely the effect of elastic bias 550 on loop behaviour, we simulate microstructures containing several loops of various sizes. 551 We consider a system of size 100 nm \times 100 nm \times 200 nm, with free surfaces along e_3 552 and PBCs in the two other directions. 50 dislocation loops are randomly placed in 553 the simulation box, which represents a loop density of 2.5×20^{22} m⁻³. Their radius is 554 drawn from a normal distribution of mean 10 nm and of standard deviation 2 nm, and 555 their habit plane is chosen randomly among the four possible ones. A total of 10 loop 556 configurations are generated. Each configuration is simulated 5 times, with different 557 random seeds, and the average growth rate of a loop is deduced from linear fits of the 558 evolution of loop radius over 10 seconds. SIAs and vacancies are produced as Frenkel 559 pairs, at a dose rate of 10^{-4} dpa/s. An example of a loop microstructure is shown 560 in Fig. 9. No other sinks, such as cavities or grain boundaries, are introduced in the 561 simulation. 562

Loop growth rate has a clear dependency on the distance of the loop to the surface 563 d (Fig. 10-(a)). On average, loops close to the surface have the highest growth rate, 564 because vacancies can easily reach the surface. For values of d larger than around 40 nm, 565 surface effects become negligible. Vacancies are mostly absorbed by loops, which results 566 in slow growth and even shrinkage of loops. The growth rate is plotted as a function 567 of the loop radius in Fig. 10-(b), for loops which are more than 40 nm away from both 568 surfaces. On average, the growth rate increases with the loop radius, although values are 569 very scattered. This result is consistent with experiments and with the bias calculations 570 on isolated loops, which show a strong increase of bias for small radii and a plateau 571 at around 10 nm [66, 67]. The large scattering of growth rates points to an additional 572 effect of the surrounding microstructure on the loop behaviour. Recently, it has been 573 shown that to first order, the absorption efficiency of a loop is proportional to $(V/\bar{V})^{1/4}$, 574 where V is the Voronoi volume of the loop and \overline{V} is the average Voronoi volume [16]. 575 The growth rate also seems to follow this trend (Fig. 10-(c)). The scattering is however 576 large, which highlights the complex dependency of loop growth rate on the surrounding 577 microstructure, especially when long range elastic interactions are taken into account. 578 If other microstructural features with lower elastic bias, such as cavities, are present in 579 sufficient number in the bulk, loop growth rates are expected to become all positive but 580 still scattered. 581

582 Conclusion

A method has been proposed to efficiently and accurately estimate the strain field generated by defects in irradiated microstructures, in the framework of OKMC simulations. The strain field of cavities and dislocation segments is evaluated by a local expansion in spherical harmonics if the sources are sufficiently far from the evaluation point. Otherwise, a direct sum over the strain sources is performed. This leads to a drastic reduction in the evaluation time of the field compared to a direct summation over strain sources, especially if periodic boundary conditions are used.

To evaluate efficiently the local expansion coefficients, the FMM for biharmonic kernels, which relies on two calls to FMM for harmonic kernels, is used. The number of FMM calls is reduced with respect to previous approaches and kept independent on the number of field types (cavities, dislocation segments). These improvements could be interesting for other simulation methods which rely on FMM on biharmonic kernels, such as DDD.

In addition to its efficiency, another advantage of FMM is the relatively straightforward treatment of finite systems. If surfaces are located far away in the three directions of space and if the strain field contains a dipolar component, the sum of strain contributions over periodic images is conditionally convergent. The correction to this sum to give a system free from surface tractions is easily obtained from the multipole expansion of the system, which is a by-product of FMM. This situation is quite common in practice: it arises, for example, when the microstructure contains dislocation loops



Figure 10. Loop growth rate in a 200 nm thick aluminum foil, irradiated at 10^{-4} dpa/s at 300 K. The system size is 100 nm in the two other directions. 10 loop configurations are simulated. Initial configurations contain a random arrangement of 50 loops, which corresponds to a density of 2.5×20^{22} m⁻³. The loop radius is chosen randomly from a gaussian distribution of mean value 10 nm and of standard deviation 2 nm. Error bars correspond to the standard error of the mean calculated on 5 independent simulations of the same configuration. (a) Loop growth rate as a function of the loop radius. Only loops which are more than 40 nm away from both surfaces are shown. (c) Loop growth rate as a function of the loop and \overline{V} is the average Voronoi volume.

and cavities. If some surfaces correspond to the boundaries of the simulation box, an elastic problem with prescribed tractions on some of the boundaries must be solved with FEM. The adaptive mesh of the FMM is particularly convenient to build a FEM mesh fitting the variations of traction forces.

Preliminary simulations of loop microstructures with elastic interactions reveal the large scattering of loop growth rates, although correlations with loop radius and with the Voronoi volume of loops have been highlighted. These results show that using elastic bias of isolated defects is probably not sufficient to provide a quantitative description of irradiated microstructures.

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⁶¹⁹ Appendix A. Strain components for a dislocation loop

Using the notations from Section 1.2.1, explicit expressions of the strain components for a dislocation loop are (the dependency on v_{α} is dropped for clarity):

$$\varepsilon_{11}(\bar{\boldsymbol{x}}) = \frac{1}{8\pi} \sum_{\alpha=1}^{N_{q}} w_{\alpha} \left[\frac{1}{1-\nu} \left(b_{3}\bar{x}_{,112} - b_{2}\bar{x}_{,113} \right) x_{1,\nu}' + \left(\frac{\nu}{1-\nu} b_{1}\bar{x}_{,113} - b_{1}\bar{x}_{,223} - b_{1}\bar{x}_{,333} - \frac{\nu}{1-\nu} b_{3}\bar{x}_{,111} + b_{3}\bar{x}_{,221} + b_{3}\bar{x}_{,331} \right) x_{2,\nu}' + \left(-\frac{\nu}{1-\nu} b_{1}\bar{x}_{,112} + b_{1}\bar{x}_{,222} + b_{1}\bar{x}_{,332} + \frac{\nu}{1-\nu} b_{2}\bar{x}_{,111} - b_{2}\bar{x}_{,221} - b_{2}\bar{x}_{,331} \right) x_{3,\nu}' \right]$$
(A.1)

$$\varepsilon_{22}(\bar{\boldsymbol{x}}) = \frac{1}{8\pi} \sum_{\alpha=1}^{N_{q}} w_{\alpha} \left[\left(-\frac{\nu}{1-\nu} b_{2}\bar{x}_{,223} + b_{2}\bar{x}_{,333} + b_{2}\bar{x}_{,113} + \frac{\nu}{1-\nu} b_{3}\bar{x}_{,222} - b_{3}\bar{x}_{,332} - b_{3}\bar{x}_{,112} \right) x_{1,\nu}' \right. \\ \left. + \frac{1}{1-\nu} \left(b_{1}\bar{x}_{,223} - b_{3}\bar{x}_{,221} \right) x_{2,\nu}' \right. \\ \left. + \left(\frac{\nu}{1-\nu} b_{2}\bar{x}_{,211} - b_{2}\bar{x}_{,331} - b_{2}\bar{x}_{,111} - \frac{\nu}{1-\nu} b_{1}\bar{x}_{,222} + b_{1}\bar{x}_{,332} + b_{1}\bar{x}_{,112} \right) x_{3,\nu}' \right]$$
(A.2)

$$\varepsilon_{33}(\bar{\boldsymbol{x}}) = \frac{1}{8\pi} \sum_{\alpha=1}^{N_{q}} w_{\alpha} \left[\left(\frac{\nu}{1-\nu} b_{3}\bar{x}_{,332} - b_{3}\bar{x}_{,112} - b_{3}\bar{x}_{,222} - \frac{\nu}{1-\nu} b_{2}\bar{x}_{,333} + b_{2}\bar{x}_{,113} + b_{2}\bar{x}_{,223} \right) x_{1,\nu}' + \left(-\frac{\nu}{1-\nu} b_{3}\bar{x}_{,331} + b_{3}\bar{x}_{,111} + b_{3}\bar{x}_{,221} + \frac{\nu}{1-\nu} b_{1}\bar{x}_{,333} - b_{1}\bar{x}_{,113} - b_{1}\bar{x}_{,223} \right) x_{2,\nu}' + \frac{1}{1-\nu} \left(b_{2}\bar{x}_{,331} - b_{1}\bar{x}_{,332} \right) x_{3,\nu}' \right]$$
(A.3)

$$\varepsilon_{23}(\bar{\boldsymbol{x}}) = \frac{1}{8\pi} \sum_{\alpha=1}^{N_{q}} w_{\alpha} \left[\frac{1}{1-\nu} \left(b_{3}\bar{x}_{,223} - b_{2}\bar{x}_{,332} \right) x_{1,\nu}' + \frac{1}{2} \left(b_{2}\bar{x}_{,111} + b_{2}\bar{x}_{,221} + b_{2}\bar{x}_{,331} - b_{1}\bar{x}_{,112} - b_{1}\bar{x}_{,222} + \frac{1+\nu}{1-\nu} b_{1}\bar{x}_{,332} - \frac{2}{1-\nu} b_{3}\bar{x}_{,123} \right) x_{2,\nu}' + \frac{1}{2} \left(-b_{3}\bar{x}_{,111} - b_{3}\bar{x}_{,221} - b_{3}\bar{x}_{,331} + b_{1}\bar{x}_{,113} - \frac{1+\nu}{1-\nu} b_{1}\bar{x}_{,223} + b_{1}\bar{x}_{,333} + \frac{2}{1-\nu} b_{2}\bar{x}_{,123} \right) x_{3,\nu}' \right]$$
(A.4)

$$\varepsilon_{13}(\bar{\boldsymbol{x}}) = \frac{1}{8\pi} \sum_{\alpha=1}^{N_{q}} w_{\alpha} \left[\frac{1}{2} \left(-b_{1}\bar{x}_{,222} - b_{1}\bar{x}_{,332} - b_{1}\bar{x}_{,112} + b_{2}\bar{x}_{,221} - \frac{1+\nu}{1-\nu} b_{2}\bar{x}_{,331} + b_{2}\bar{x}_{,111} + \frac{2}{1-\nu} b_{3}\bar{x}_{,123} \right) x_{11}' + \frac{1}{1-\nu} \left(b_{1}\bar{x}_{,331} - b_{3}\bar{x}_{,113} \right) x_{2,\nu}' + \frac{1}{2} \left(b_{3}\bar{x}_{,222} + b_{3}\bar{x}_{,332} + b_{3}\bar{x}_{,112} - b_{2}\bar{x}_{,223} - b_{2}\bar{x}_{,333} + \frac{1+\nu}{1-\nu} b_{2}\bar{x}_{,113} - \frac{2}{1-\nu} b_{1}\bar{x}_{,123} \right) x_{3,\nu}' \right]$$
(A.5)

$$\varepsilon_{12}(\bar{\boldsymbol{x}}) = \frac{1}{8\pi} \sum_{\alpha=1}^{N_{q}} w_{\alpha} \left[\frac{1}{2} \left(b_{1}\bar{x}_{,333} + b_{1}\bar{x}_{,113} + b_{1}\bar{x}_{,223} - b_{3}\bar{x}_{,331} - b_{3}\bar{x}_{,111} + \frac{1+\nu}{1-\nu} b_{3}\bar{x}_{,221} - \frac{2}{1-\nu} b_{2}\bar{x}_{,123} \right) x_{1,\nu}' + \frac{1}{2} \left(-b_{2}\bar{x}_{,333} - b_{2}\bar{x}_{,113} - b_{2}\bar{x}_{,223} + b_{3}\bar{x}_{,332} - \frac{1+\nu}{1-\nu} b_{3}\bar{x}_{,112} + b_{3}\bar{x}_{,222} + \frac{2}{1-\nu} b_{1}\bar{x}_{,123} \right) x_{2,\nu}' + \frac{1}{1-\nu} \left(b_{2}\bar{x}_{,112} - b_{1}\bar{x}_{,221} \right) x_{3,\nu}' \right].$$
(A.6)

Expressions of $\bar{x}_{,ijk}$ are given, for example, in Ref. [34].

Appendix B. Definition of spherical harmonics, solid harmonics and recursion formulas

⁶²³ The definition of spherical harmonics used here is the same as in Ref. [24]:

$$Y_l^m(\theta,\varphi) = \sqrt{\frac{(l-|m|)!}{(l+|m|)!}} P_l^{|m|}(\cos\theta)e^{im\varphi},\tag{B.1}$$

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 $_{\rm 624}$ $\,$ where P_l^m are the associated Legendre polynomials, which are defined by

$$P_l^m(x) = (-1)^m (1 - x^2)^{m/2} \frac{\mathrm{d}^m}{\mathrm{d}x^m} P_l(x).$$
(B.2)

In Eq. (B.2), P_l are the Legendre polynomials. From Eq. (B.1) we deduce that $(Y_l^m(\theta, \varphi))^* = Y_l^{-m}(\theta, \varphi)$. The normalisation is such that

$$\int_0^{2\pi} \int_0^{\pi} Y_l^m(\theta,\varphi) (Y_{l'}^{m'}(\theta,\varphi))^* \sin\theta \,\mathrm{d}\theta \mathrm{d}\varphi = \frac{4\pi}{2l+1} \delta_{ll'} \delta_{mm'}.$$
 (B.3)

627 For example we have:

$$Y_1^0(\theta,\varphi) = \cos\theta \tag{B.4}$$

$$Y_1^1(\theta,\varphi) = -\frac{1}{\sqrt{2}}\sin\theta e^{i\varphi}.$$
(B.5)

Regular solid harmonics are defined by $R_l^m(x, \theta, \varphi) = x^l Y_l^m(\theta, \varphi)$ and irregular solid harmonics are defined by $I_l^m(x, \theta, \varphi) = Y_l^m(\theta, \varphi)/x^{l+1}$. Irregular solid harmonics can also be written as [68, 27]

$$I_l^m(x,\theta,\varphi) = (-1)^m A_l^m \left(\frac{\partial}{\partial x_1} + i\frac{\partial}{\partial x_2}\right)^m \left(\frac{\partial}{\partial x_3}\right)^{l-m} \left(\frac{1}{x}\right), \tag{B.6}$$

631 with

$$A_l^m = \frac{(-1)^l}{\sqrt{(l-m)!(l+m)!}}$$
(B.7)

(the factor $(-1)^m$ in Eq. (B.6) is missing in [27]). We note

$$B_l^m = \sqrt{(l+m)(l-m)},\tag{B.8}$$

$$C_l^m = \sqrt{(l+m)(l+m-1)},$$
 (B.9)

635 and

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$$s(m) = \begin{cases} 1 & \text{if } m \ge 0\\ -1 & \text{if } m < 0. \end{cases}$$
(B.10)

636 We can write recursion formulas under the form:

$$B_{l+1}^m Y_{l+1}^m - (2l+1)Y_1^0 Y_l^m + B_l^m Y_{l-1}^m = 0$$
(B.11)

$$s(m-1)C_{l+1}^{-m+1}Y_{l+1}^{m-1} + \sqrt{2}(2l+1)Y_1^{-1}Y_l^m - s(m-1)C_l^mY_{l-1}^{m-1} = 0$$
(B.12)

$$s(m)C_{l+2}^{m}Y_{l+1}^{m+1} - \sqrt{2}(2l+1)Y_{1}^{1}Y_{l}^{m} - s(m)C_{l}^{-m}Y_{l-1}^{m+1} = 0.$$
(B.13)

⁶³⁷ Using these formulas, it can be shown that products $x_k I_l^m(x, \theta, \varphi)$ can be ⁶³⁸ decomposed as functions of $I_p^q(x, \theta, \varphi)$ and $x^2 I_p^q(x, \theta, \varphi)$:

$$x_{1}I_{l}^{m}(x,\theta,\varphi) = -\frac{1}{2(2l+1)}s(m)C_{l+2}^{m}x^{2}I_{l+1}^{m+1}(x,\theta,\varphi) + \frac{1}{2(2l+1)}s(m-1)C_{l+1}^{-m+1}x^{2}I_{l+1}^{m-1}(x,\theta,\varphi) + \frac{1}{2(2l+1)}s(m)C_{l}^{-m}I_{l-1}^{m+1}(x,\theta,\varphi) - \frac{1}{2(2l+1)}s(m-1)C_{l}^{m}I_{l-1}^{m-1}(x,\theta,\varphi)$$
(B.14)

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$$x_{2}I_{l}^{m}(x,\theta,\varphi) = \frac{i}{2(2l+1)}s(m-1)C_{l+1}^{-m+1}x^{2}I_{l+1}^{m-1}(x,\theta,\varphi) + \frac{i}{2(2l+1)}s(m)C_{l+2}^{m}x^{2}I_{l+1}^{m+1}(x,\theta,\varphi) - \frac{i}{2(2l+1)}s(m)C_{l}^{-m}I_{l-1}^{m+1}(x,\theta,\varphi) - \frac{i}{2(2l+1)}s(m-1)C_{l}^{m}I_{l-1}^{m-1}(x,\theta,\varphi)$$
(B.15)

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$$x_3 I_l^m(x,\theta,\varphi) = \frac{B_{l+1}^m}{2l+1} x^2 I_{l+1}^m(x,\theta,\varphi) + \frac{B_l^m}{2l+1} I_{l-1}^m(x,\theta,\varphi).$$
(B.16)

Similar expressions can be deduced for $x_k R_l^m(x, \theta, \varphi)$, using Eqs (B.14)–(B.16) and the relation $R_l^m(x, \theta, \varphi) = x^{2l+1} I_l^m(x, \theta, \varphi)$:

$$x_{1}R_{l}^{m}(x,\theta,\varphi) = -\frac{1}{2(2l+1)}s(m)C_{l+2}^{m}R_{l+1}^{m+1}(x,\theta,\varphi) + \frac{1}{2(2l+1)}s(m-1)C_{l+1}^{-m+1}R_{l+1}^{m-1}(x,\theta,\varphi) + \frac{1}{2(2l+1)}s(m)C_{l}^{-m}x^{2}R_{l-1}^{m+1}(x,\theta,\varphi) - \frac{1}{2(2l+1)}s(m-1)C_{l}^{m}x^{2}R_{l-1}^{m-1}(x,\theta,\varphi)$$
(B.17)

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$$x_{2}R_{l}^{m}(x,\theta,\varphi) = \frac{i}{2(2l+1)}s(m)C_{l+2}^{m}R_{l+1}^{m+1}(x,\theta,\varphi) + \frac{i}{2(2l+1)}s(m-1)C_{l+1}^{-m+1}R_{l+1}^{m-1}(x,\theta,\varphi) - \frac{i}{2(2l+1)}s(m)C_{l}^{-m}x^{2}R_{l-1}^{m+1}(x,\theta,\varphi) - \frac{i}{2(2l+1)}s(m-1)C_{l}^{m}x^{2}R_{l-1}^{m-1}(x,\theta,\varphi)$$
(B.18)

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$$x_3 R_l^m(x,\theta,\varphi) = \frac{B_{l+1}^m}{2l+1} R_{l+1}^m(x,\theta,\varphi) + \frac{B_l^m}{2l+1} x^2 R_{l-1}^m(x,\theta,\varphi).$$
(B.19)

Appendix C. Translation and conversion of expansions for a biharmonic function

In this section the results of Gumerov and Duraiswami [39] are given for our definition of spherical harmonics. Let $\psi(\boldsymbol{x}) = \phi(\boldsymbol{x}) + x^2 \omega(\boldsymbol{x})$ be a multipole or local expansion of a biharmonic function. We want to set the origin of the expansion at \boldsymbol{t} , *i.e.* we consider the new function (Fig. C1)

$$\hat{\psi}(\boldsymbol{x}) = \psi(\boldsymbol{x} + \boldsymbol{t})$$

$$= \phi(\boldsymbol{x} + \boldsymbol{t}) + (\boldsymbol{x} + \boldsymbol{t}) \cdot (\boldsymbol{x} + \boldsymbol{t})\omega(\boldsymbol{x} + \boldsymbol{t})$$

$$= \hat{\phi}(\boldsymbol{x}) + (x^2 + 2\boldsymbol{x} \cdot \boldsymbol{t} + t^2)\hat{\omega}(\boldsymbol{x}), \qquad (C.1)$$

where $\hat{\phi}$ and $\hat{\omega}$ are the same expansions as ϕ and ω , but with their origin at t, *i.e.* their coefficients are deduced from the expansions ϕ and ω by translation or conversion operators of harmonic functions [24, 25]. We want to set Eq. (C.1) under the form

$$\hat{\psi}(\boldsymbol{x}) = \tilde{\phi}(\boldsymbol{x}) + x^2 \tilde{\omega}(\boldsymbol{x}), \qquad (C.2)$$

where $\tilde{\phi}$ and $\tilde{\omega}$ remain to be found. Operators which transform the coefficients of the expansion (ϕ, ω) into the coefficients of the expansion $(\tilde{\phi}, \tilde{\omega})$ are the translation/conversion operators which are used in the FMM.



Figure C1. Change of center of expansion.

⁶⁵⁷ Two cases can be distinguished:

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• For a multipole-to-multipole (M2M) transformation (translation), $\hat{\phi}(\boldsymbol{x})$ and $\hat{\omega}(\boldsymbol{x})$ are expanded with irregular solid harmonics. Eqs (B.14)–(B.16) can be used to

write the term
$$2\boldsymbol{x} \cdot \boldsymbol{t}\hat{\omega}(\boldsymbol{x})$$
 as contributions to $\tilde{\phi}(\boldsymbol{x})$ and $\tilde{\omega}(\boldsymbol{x})$. We obtain

$$\begin{split} \tilde{\phi}_{l}^{m} &= \hat{\phi}_{l}^{m} + t^{2} \hat{\omega}_{l}^{m} - \hat{\omega}_{l+1}^{m+1} \frac{t_{1} + it_{2}}{2l + 3} s(m) C_{l+1}^{m+1} + \hat{\omega}_{l+1}^{m} \frac{2t_{3}}{2l + 3} B_{l+1}^{m} + \hat{\omega}_{l+1}^{m-1} \frac{t_{1} - it_{2}}{2l + 3} s(m-1) C_{l+1}^{-m+1} \frac{t_{1} - it_{2}}{2l + 3} s(m-1) C_{l+1}^{-m+1} \frac{(C.3)}{2l} \\ \tilde{\omega}_{l}^{m} &= \hat{\omega}_{l}^{m} + \hat{\omega}_{l-1}^{m+1} \frac{t_{1} + it_{2}}{2l - 1} s(m) C_{l}^{-m} + \hat{\omega}_{l-1}^{m} \frac{2t_{3}}{2l - 1} B_{l}^{m} - \hat{\omega}_{l-1}^{m-1} \frac{t_{1} - it_{2}}{2l - 1} s(m-1) C_{l+1}^{m-1} . \end{split}$$

$$(C.4)$$

These expressions correspond to expression 57 in Ref. [39] obtained with a different definition of spherical harmonics.

• For a multipole-to-local (M2L) or a local-to-local (L2L) transformation, $\hat{\phi}(\boldsymbol{x})$ and

664 665 $\hat{\omega}(\boldsymbol{x})$ are expanded with regular solid harmonics. Eqs (B.17)–(B.19) can be used to write the term $2\boldsymbol{x} \cdot \boldsymbol{t}\hat{\omega}(\boldsymbol{x})$ as contributions to $\tilde{\phi}(\boldsymbol{x})$ and $\tilde{\omega}(\boldsymbol{x})$. We obtain

$$\tilde{\phi}_{l}^{m} = \hat{\phi}_{l}^{m} + t^{2}\hat{\omega}_{l}^{m} + \hat{\omega}_{l-1}^{m+1}\frac{t_{1} + it_{2}}{2l - 1}s(m)C_{l}^{-m} + \hat{\omega}_{l-1}^{m}\frac{2t_{3}}{2l - 1}B_{l}^{m} - \hat{\omega}_{l-1}^{m-1}\frac{t_{1} - it_{2}}{2l - 1}s(m - 1)C_{l+1}^{m-1}$$
(C.5)

$$\tilde{\omega}_{l}^{m} = \hat{\omega}_{l}^{m} - \hat{\omega}_{l+1}^{m+1} \frac{t_{1} + it_{2}}{2l+3} s(m) C_{l+1}^{m+1} + \hat{\omega}_{l+1}^{m} \frac{2t_{3}}{2l+3} B_{l+1}^{m} + \hat{\omega}_{l+1}^{m-1} \frac{t_{1} - it_{2}}{2l+3} s(m-1) C_{l+1}^{-m+1}.$$
(C.6)

⁶⁶⁶ These expressions correspond to expression 56 in Ref. [39].

⁶⁶⁷ Appendix D. Derivatives of irregular solid harmonics

Derivatives of irregular solid harmonics are used in the local expansion of the strain field of a cavity (Eq. (25)) and a dislocation segment (Eqs (32)-(33)). They are also used to obtain the derivatives of regular solid harmonics (Appendix E). To determine them, it is convenient to use expression (B.6). The dependency of I_l^m functions on x, θ, φ is dropped for clarity.

- 673 First derivatives
 - *m* = 0

$$\frac{\partial}{\partial x_1}(I_l^0) = \frac{1}{2}\sqrt{(l+1)(l+2)}(I_{l+1}^1 + I_{l+1}^{-1})$$
(D.1)

$$\frac{\partial}{\partial x_2}(I_l^0) = -\frac{i}{2}\sqrt{(l+1)(l+2)}(I_{l+1}^1 - I_{l+1}^{-1})$$
(D.2)

$$\frac{\partial}{\partial x_3}(I_l^0) = -(l+1)I_{l+1}^0$$
(D.3)

• $m \ge 1$ $\frac{\partial}{\partial x_1}(I_l^m) = \frac{1}{2} \left(\sqrt{(l+m+2)(l+m+1)}I_{l+1}^{m+1} - \sqrt{(l-m+2)(l-m+1)}I_{l+1}^{m-1} \right)$ (D.4) $\frac{\partial}{\partial x_2}(I_l^m) = -\frac{i}{2} \left(\sqrt{(l+m+2)(l+m+1)}I_{l+1}^{m+1} + \sqrt{(l-m+2)(l-m+1)}I_{l+1}^{m-1} \right)$

$$\frac{\partial}{\partial t_{l+1}} = \frac{2}{2} \left(\sqrt{(l+m+2)(l+m+1)} \frac{1}{l+1} + \sqrt{(l-m+2)(l-m+1)} \frac{1}{l+1} \right)$$
(D.5)

$$\frac{\partial}{\partial x_3}(I_l^m) = -\sqrt{(l+m+1)(l-m+1)}I_{l+1}^m$$
(D.6)

674 Second derivatives

$$m = 0$$

$$\frac{\partial^2}{\partial x_1^2} (I_l^0) = \frac{1}{4} \sqrt{(l+1)(l+2)(l+3)(l+4)} I_{l+2}^2 - \frac{1}{2}(l+1)(l+2)I_{l+2}^0$$

$$+ \frac{1}{4} \sqrt{(l+1)(l+2)(l+3)(l+4)} I_{l+2}^{-2} \quad (D.7)$$

$$\frac{\partial^2}{\partial x_2^2}(I_l^0) = -\frac{1}{4}\sqrt{(l+1)(l+2)(l+3)(l+4)}I_{l+2}^2 - \frac{1}{2}(l+1)(l+2)I_{l+2}^0 - \frac{1}{4}\sqrt{(l+1)(l+2)(l+3)(l+4)}I_{l+2}^{-2} \quad (D.8)$$

$$\frac{\partial^2}{\partial x_3^2} (I_l^0) = (l+1)(l+2)I_{l+2}^0 \tag{D.9}$$

$$\frac{\partial^2}{\partial x_1 \partial x_2} (I_l^0) = -\frac{i}{4} \sqrt{(l+1)(l+2)(l+3)(l+4)} I_{l+2}^2 + \frac{i}{4} \sqrt{(l+1)(l+2)(l+3)(l+4)} I_{l+2}^{-2} \quad (D.10)$$

$$\frac{\partial^2}{\partial x_2 \partial x_3} (I_l^0) = \frac{i}{2} (l+1) \sqrt{(l+2)(l+3)} (I_{l+2}^1 - I_{l+2}^{-1})$$
(D.11)

$$\frac{\partial^2}{\partial x_1 \partial x_3} (I_l^0) = -\frac{1}{2} (l+1) \sqrt{(l+2)(l+3)} (I_{l+2}^1 + I_{l+2}^{-1})$$
(D.12)

•
$$m = 1$$

$$\frac{\partial^2}{\partial x_1^2}(I_l^1) = \frac{1}{4}\sqrt{(l+2)(l+3)(l+4)(l+5)}I_{l+2}^3 - \frac{1}{2}\sqrt{l(l+1)(l+2)(l+3)}I_{l+2}^1 - \frac{1}{4}\sqrt{l(l+1)(l+2)(l+3)}I_{l+2}^{-1} \quad (D.13)$$

$$\frac{\partial^2}{\partial x_2^2} (I_l^1) = -\frac{1}{4} \sqrt{(l+2)(l+3)(l+4)(l+5)} I_{l+2}^3 - \frac{1}{2} \sqrt{l(l+1)(l+2)(l+3)} I_{l+2}^1 + \frac{1}{4} \sqrt{l(l+1)(l+2)(l+3)} I_{l+2}^{-1} \quad (D.14)$$

$$\frac{\partial^2}{\partial x_3^2}(I_l^1) = \sqrt{l(l+1)(l+2)(l+3)}I_{l+2}^1$$
(D.15)

$$\frac{\partial^2}{\partial x_1 \partial x_2} (I_l^1) = -\frac{i}{4} \sqrt{(l+2)(l+3)(l+4)(l+5)} I_{l+2}^3 - \frac{i}{4} \sqrt{l(l+1)(l+2)(l+3)} I_{l+2}^{-1} \quad (D.16)$$

$$\frac{\partial^2}{\partial x_2 \partial x_3} (I_l^1) = \frac{i}{2} \sqrt{l(l+2)(l+3)(l+4)} I_{l+2}^2 + \frac{i}{2}(l+2) \sqrt{l(l+1)} I_{l+2}^0$$
(D.17)

$$\frac{\partial^2}{\partial x_1 \partial x_3} (I_l^1) = -\frac{1}{2} \sqrt{l(l+2)(l+3)(l+4)} I_{l+2}^2 + \frac{1}{2}(l+2) \sqrt{l(l+1)} I_{l+2}^0 \qquad (D.18)$$

• $m \ge 2$

$$\frac{\partial^2}{\partial x_1^2}(I_l^m) = \frac{1}{4}\sqrt{(l+m+1)(l+m+2)(l+m+3)(l+m+4)}I_{l+2}^{m+2} - \frac{1}{2}\sqrt{(l-m+2)(l-m+1)(l+m+1)(l+m+2)}I_{l+2}^m + \frac{1}{4}\sqrt{(l-m+1)(l-m+2)(l-m+3)(l-m+4)}I_{l+2}^{m-2} \quad (D.19)$$

$$\frac{\partial^2}{\partial x_2^2} (I_l^m) = -\frac{1}{4} \sqrt{(l+m+1)(l+m+2)(l+m+3)(l+m+4)} I_{l+2}^{m+2} -\frac{1}{2} \sqrt{(l-m+1)(l-m+2)(l+m+1)(l+m+2)} I_{l+2}^m -\frac{1}{4} \sqrt{(l-m+1)(l-m+2)(l-m+3)(l-m+4)} I_{l+2}^{m-2}$$
(D.20)

$$\frac{\partial^2}{\partial x_3^2}(I_l^m) = \sqrt{(l-m+1)(l-m+2)(l+m+1)(l+m+2)}I_{l+2}^m$$
(D.21)

$$\frac{\partial^2}{\partial x_1 \partial x_2} (I_l^m) = -\frac{i}{4} \sqrt{(l+m+1)(l+m+2)(l+m+3)(l+m+4)} I_{l+2}^{m+2} + \frac{i}{4} \sqrt{(l-m+1)(l-m+2)(l-m+3)(l-m+4)} I_{l+2}^{m-2} \quad (D.22)$$

$$\frac{\partial^2}{\partial x_2 \partial x_3} (I_l^m) = \frac{i}{2} \sqrt{(l-m+1)(l+m+1)(l+m+2)(l+m+3)} I_{l+2}^{m+1} + \frac{i}{2} \sqrt{(l-m+1)(l-m+2)(l-m+3)(l+m+1)} I_{l+2}^{m-1} \quad (D.23)$$

$$\frac{\partial^2}{\partial x_1 \partial x_3} (I_l^m) = -\frac{1}{2} \sqrt{(l-m+1)(l+m+1)(l+m+2)(l+m+3)} I_{l+2}^{m+1} + \frac{1}{2} \sqrt{(l-m+1)(l-m+2)(l-m+3)(l+m+1)} I_{l+2}^{m-1} \quad (D.24)$$

675 Third derivatives

•
$$m = 0$$

$$\frac{\partial^3}{\partial x_1^3}(I_l^0) = \frac{1}{8}\sqrt{(l+1)(l+2)(l+3)(l+4)(l+5)(l+6)}(I_{l+3}^3 + I_{l+3}^{-3}) - \frac{3}{8}(l+1)(l+2)\sqrt{(l+3)(l+4)}(I_{l+3}^1 + I_{l+3}^{-1}) \quad (D.25)$$

$$\begin{aligned} \frac{\partial^3}{\partial x_2^3}(I_l^0) &= \frac{i}{8}\sqrt{(l+1)(l+2)(l+3)(l+4)(l+5)(l+6)}(I_{l+3}^3 - I_{l+3}^{-3}) \\ &+ \frac{3i}{8}(l+1)(l+2)\sqrt{(l+3)(l+4)}(I_{l+3}^1 - I_{l+3}^{-1}) \end{aligned} \tag{D.26}$$

$$\frac{\partial^3}{\partial x_3^3}(I_l^0) = -(l+1)(l+2)(l+3)I_{l+3}^0 \tag{D.27}$$

$$\frac{\partial^3}{\partial x_1^2 \partial x_2} (I_l^0) = -\frac{i}{8} \sqrt{(l+1)(l+2)(l+3)(l+4)(l+5)(l+6)} (I_{l+3}^3 - I_{l+3}^{-3}) + \frac{i}{8} (l+1)(l+2) \sqrt{(l+3)(l+4)} (I_{l+3}^1 - I_{l+3}^{-1})$$
(D.28)

$$\frac{\partial^3}{\partial x_1^2 \partial x_3} (I_l^0) = -\frac{1}{4} (l+1) \sqrt{(l+2)(l+3)(l+4)(l+5)} (I_{l+3}^2 + I_{l+3}^{-2}) + \frac{1}{2} (l+1)(l+2)(l+3) I_{l+3}^0 \quad (D.29)$$

$$\frac{\partial^3}{\partial x_1 \partial x_2^2} (I_l^0) = -\frac{1}{8} \sqrt{(l+1)(l+2)(l+3)(l+4)(l+5)(l+6)} (I_{l+3}^3 + I_{l+3}^{-3}) - \frac{1}{8} (l+1)(l+2) \sqrt{(l+3)(l+4)} (I_{l+3}^1 + I_{l+3}^{-1})$$
(D.30)

$$\frac{\partial^3}{\partial x_2^2 \partial x_3} (I_l^0) = \frac{1}{4} (l+1) \sqrt{(l+2)(l+3)(l+4)(l+5)} (I_{l+3}^2 + I_{l+3}^{-2}) + \frac{1}{2} (l+1)(l+2)(l+3) I_{l+3}^0 \quad (D.31)$$

$$\frac{\partial^3}{\partial x_1 \partial x_3^2} (I_l^0) = \frac{1}{2} (l+1)(l+2)\sqrt{(l+3)(l+4)} (I_{l+3}^1 + I_{l+3}^{-1})$$
(D.32)

$$\frac{\partial^3}{\partial x_2 \partial x_3^2} (I_l^0) = -\frac{i}{2} (l+1)(l+2)\sqrt{(l+3)(l+4)} (I_{l+3}^1 - I_{l+3}^{-1})$$
(D.33)

$$\frac{\partial^3}{\partial x_1 \partial x_2 \partial x_3} (I_l^0) = \frac{i}{4} (l+1) \sqrt{(l+2)(l+3)(l+4)(l+5)} (I_{l+3}^2 - I_{l+3}^{-2})$$
(D.34)
• $m = 1$

$$\begin{aligned} \frac{\partial^3}{\partial x_1^3}(I_l^1) &= \frac{1}{8}\sqrt{(l+2)(l+3)(l+4)(l+5)(l+6)(l+7)}I_{l+3}^4 \\ &\quad -\frac{3}{8}\sqrt{l(l+1)(l+2)(l+3)(l+4)(l+5)}I_{l+3}^2 \\ &\quad +\frac{3}{8}(l+2)(l+3)\sqrt{l(l+1)}I_{l+3}^0 \\ &\quad -\frac{1}{8}\sqrt{l(l+1)(l+2)(l+3)(l+4)(l+5)}I_{l+3}^{-2} \end{aligned} \tag{D.35}$$

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$$\begin{aligned} \frac{\partial^3}{\partial x_2^3}(I_l^1) &= \frac{i}{8}\sqrt{(l+2)(l+3)(l+4)(l+5)(l+6)(l+7)}I_{l+3}^4 \\ &+ \frac{3i}{8}\sqrt{l(l+1)(l+2)(l+3)(l+4)(l+5)}I_{l+3}^2 \\ &+ \frac{3i}{8}(l+2)(l+3)\sqrt{l(l+1)}I_{l+3}^0 \\ &+ \frac{i}{8}\sqrt{l(l+1)(l+2)(l+3)(l+4)(l+5)}I_{l+3}^{-2} \end{aligned} \tag{D.36}$$

$$\frac{\partial^3}{\partial x_3^3}(I_l^1) = -(l+2)\sqrt{l(l+1)(l+3)(l+4)}I_{l+3}^1$$
(D.37)

$$\frac{\partial^{3}}{\partial x_{1}^{2} \partial x_{2}} (I_{l}^{1}) = -\frac{i}{8} \sqrt{(l+2)(l+3)(l+4)(l+5)(l+6)(l+7)} I_{l+3}^{4} + \frac{i}{8} \sqrt{l(l+1)(l+2)(l+3)(l+4)(l+5)} I_{l+3}^{2} + \frac{i}{8} (l+2)(l+3) \sqrt{l(l+1)} I_{l+3}^{0} - \frac{i}{8} \sqrt{l(l+1)(l+2)(l+3)(l+4)(l+5)} I_{l+3}^{-2}$$
(D.38)

$$\begin{aligned} \frac{\partial^3}{\partial x_1^2 \partial x_3} (I_l^1) &= -\frac{1}{4} \sqrt{l(l+2)(l+3)(l+4)(l+5)(l+6)} I_{l+3}^3 \\ &+ \frac{1}{2}(l+2) \sqrt{l(l+1)(l+3)(l+4)} I_{l+3}^1 \\ &+ \frac{1}{4}(l+2) \sqrt{l(l+1)(l+3)(l+4)} I_{l+3}^{-1} \end{aligned}$$
(D.39)

$$\begin{aligned} \frac{\partial^3}{\partial x_1 \partial x_2^2} (I_l^1) &= -\frac{1}{8} \sqrt{(l+2)(l+3)(l+4)(l+5)(l+6)(l+7)} I_{l+3}^4 \\ &- \frac{1}{8} \sqrt{l(l+1)(l+2)(l+3)(l+4)(l+5)} I_{l+3}^2 \\ &+ \frac{1}{8} (l+2)(l+3) \sqrt{l(l+1)} I_{l+3}^0 \\ &+ \frac{1}{8} \sqrt{l(l+1)(l+2)(l+3)(l+4)(l+5)} I_{l+3}^{-2} \end{aligned}$$
(D.40)

$$\frac{\partial^3}{\partial x_2^2 \partial x_3} (I_l^1) = \frac{1}{4} \sqrt{l(l+2)(l+3)(l+4)(l+5)(l+6)} I_{l+3}^3 + \frac{1}{2}(l+2)\sqrt{l(l+1)(l+3)(l+4)} I_{l+3}^1 - \frac{1}{4}(l+2)\sqrt{l(l+1)(l+3)(l+4)} I_{l+3}^{-1} - \frac{1}{4}(l+2)\sqrt{l(l+1)(l+3)(l+4)} I_{l+3}^{-1}$$
(D.41)

$$\frac{\partial^3}{\partial x_1 \partial x_3^2} (I_l^1) = \frac{1}{2} \sqrt{l(l+1)(l+2)(l+3)(l+4)(l+5)} I_{l+3}^2 - \frac{1}{2} (l+2)(l+3) \sqrt{l(l+1)} I_{l+3}^0 \quad (D.42)$$

$$\frac{\partial^3}{\partial x_2 \partial x_3^2} (I_l^1) = -\frac{i}{2} \sqrt{l(l+1)(l+2)(l+3)(l+4)(l+5)} I_{l+3}^2 - \frac{i}{2}(l+2)(l+3) \sqrt{l(l+1)} I_{l+3}^0 \quad (D.43)$$

$$\frac{\partial^3}{\partial x_1 \partial x_2 \partial x_3} (I_l^1) = \frac{i}{4} \sqrt{l(l+2)(l+3)(l+4)(l+5)(l+6)} I_{l+3}^3 + \frac{i}{4} (l+2) \sqrt{l(l+1)(l+3)(l+4)} I_{l+3}^{-1} \quad (D.44)$$

• m = 2

$$\begin{aligned} \frac{\partial^3}{\partial x_1^3}(I_l^2) &= \frac{1}{8}\sqrt{(l+3)(l+4)(l+5)(l+6)(l+7)(l+8)}I_{l+3}^5 \\ &\quad -\frac{3}{8}\sqrt{(l-1)l(l+3)(l+4)(l+5)(l+6)}I_{l+3}^3 \\ &\quad +\frac{3}{8}\sqrt{(l-1)l(l+1)(l+2)(l+3)(l+4)}I_{l+3}^1 \\ &\quad +\frac{1}{8}\sqrt{(l-1)l(l+1)(l+2)(l+3)(l+4)}I_{l+3}^{-1} \quad (D.45) \end{aligned}$$

$$\begin{aligned} \frac{\partial^3}{\partial x_2^3}(I_l^2) &= \frac{i}{8}\sqrt{(l+3)(l+4)(l+5)(l+6)(l+7)(l+8)}I_{l+3}^5 \\ &+ \frac{3i}{8}\sqrt{(l-1)l(l+3)(l+4)(l+5)(l+6)}I_{l+3}^3 \\ &+ \frac{3i}{8}\sqrt{(l-1)l(l+1)(l+2)(l+3)(l+4)}I_{l+3}^1 \\ &- \frac{i}{8}\sqrt{(l-1)l(l+1)(l+2)(l+3)(l+4)}I_{l+3}^{-1} \quad (D.46) \end{aligned}$$

$$\frac{\partial^3}{\partial x_3^3}(I_l^2) = -\sqrt{(l-1)l(l+1)(l+3)(l+4)(l+5)}I_{l+3}^2 \tag{D.47}$$

$$\begin{aligned} \frac{\partial^3}{\partial x_1^2 \partial x_2} (I_l^2) &= -\frac{i}{8} \sqrt{(l+3)(l+4)(l+5)(l+6)(l+7)(l+8)} I_{l+3}^5 \\ &+ \frac{i}{8} \sqrt{(l-1)l(l+3)(l+4)(l+5)(l+6)} I_{l+3}^3 \\ &+ \frac{i}{8} \sqrt{(l-1)l(l+1)(l+2)(l+3)(l+4)} I_{l+3}^1 \\ &+ \frac{i}{8} \sqrt{(l-1)l(l+1)(l+2)(l+3)(l+4)} I_{l+3}^{-1} \end{aligned}$$
(D.48)

$$\begin{aligned} \frac{\partial^3}{\partial x_1^2 \partial x_3} (I_l^2) &= -\frac{1}{4} \sqrt{(l-1)(l+3)(l+4)(l+5)(l+6)(l+7)} I_{l+3}^4 \\ &+ \frac{1}{2} \sqrt{(l-1)l(l+1)(l+3)(l+4)(l+5)} I_{l+3}^2 \\ &- \frac{1}{4} (l+3) \sqrt{(l-1)l(l+1)(l+2)} I_{l+3}^0 \quad (\text{D.49}) \end{aligned}$$

$$\begin{aligned} \frac{\partial^3}{\partial x_1 \partial x_2^2} (I_l^2) &= -\frac{1}{8} \sqrt{(l+3)(l+4)(l+5)(l+6)(l+7)(l+8)} I_{l+3}^5 \\ &\quad -\frac{1}{8} \sqrt{(l-1)l(l+3)(l+4)(l+5)(l+6)} I_{l+3}^3 \\ &\quad +\frac{1}{8} \sqrt{(l-1)l(l+1)(l+2)(l+3)(l+4)} I_{l+3}^1 \\ &\quad -\frac{1}{8} \sqrt{(l-1)l(l+1)(l+2)(l+3)(l+4)} I_{l+3}^{-1} \end{aligned}$$
(D.50)

$$\frac{\partial^3}{\partial x_2^2 \partial x_3} (I_l^2) = \frac{1}{4} \sqrt{(l-1)(l+3)(l+4)(l+5)(l+6)(l+7)} I_{l+3}^4 + \frac{1}{2} \sqrt{(l-1)l(l+1)(l+3)(l+4)(l+5)} I_{l+3}^2 + \frac{1}{4} (l+3) \sqrt{(l-1)l(l+1)(l+2)} I_{l+3}^0 \quad (D.51)$$

$$\frac{\partial^3}{\partial x_1 \partial x_3^2} (I_l^2) = \frac{1}{2} \sqrt{(l-1)l(l+3)(l+4)(l+5)(l+6)} I_{l+3}^3 - \frac{1}{2} \sqrt{(l-1)l(l+1)(l+2)(l+3)(l+4)} I_{l+3}^1 \quad (D.52)$$

$$\frac{\partial^3}{\partial x_2 \partial x_3^2} (I_l^2) = -\frac{i}{2} \sqrt{(l-1)l(l+3)(l+4)(l+5)(l+6)} I_{l+3}^3 - \frac{i}{2} \sqrt{(l-1)l(l+1)(l+2)(l+3)(l+4)} I_{l+3}^1 \quad (D.53)$$

$$\frac{\partial^3}{\partial x_1 \partial x_2 \partial x_3} (I_l^2) = \frac{i}{4} \sqrt{(l-1)(l+3)(l+4)(l+5)(l+6)(l+7)} I_{l+3}^4 - \frac{i}{4}(l+3)\sqrt{(l-1)l(l+1)(l+2)} I_{l+3}^0 \quad (D.54)$$

$$\begin{aligned} \frac{\partial^3}{\partial x_1^3}(I_l^m) &= \frac{1}{8}\sqrt{(l+m+1)(l+m+2)(l+m+3)(l+m+4)(l+m+5)(l+m+6)}I_{l+3}^{m+3} \\ &- \frac{3}{8}\sqrt{(l-m+1)(l-m+2)(l+m+1)(l+m+2)(l+m+3)(l+m+4)}I_{l+3}^{m+1} \\ &+ \frac{3}{8}\sqrt{(l-m+1)(l-m+2)(l-m+3)(l-m+4)(l+m+1)(l+m+2)}I_{l+3}^{m-1} \\ &- \frac{1}{8}\sqrt{(l-m+1)(l-m+2)(l-m+3)(l-m+4)(l-m+5)(l-m+6)}I_{l+3}^{m-3} \\ &\qquad (D.55) \end{aligned}$$

$$\begin{aligned} \frac{\partial^3}{\partial x_2^3}(I_l^m) &= \frac{i}{8}\sqrt{(l+m+1)(l+m+2)(l+m+3)(l+m+4)(l+m+5)(l+m+6)}I_{l+3}^{m+3} \\ &+ \frac{3i}{8}\sqrt{(l-m+1)(l-m+2)(l+m+1)(l+m+2)(l+m+3)(l+m+4)}I_{l+3}^{m+1} \\ &+ \frac{3i}{8}\sqrt{(l-m+1)(l-m+2)(l-m+3)(l-m+4)(l+m+1)(l+m+2)}I_{l+3}^{m-1} \\ &+ \frac{i}{8}\sqrt{(l-m+1)(l-m+2)(l-m+3)(l-m+4)(l-m+5)(l-m+6)}I_{l+3}^{m-3} \\ &\qquad (D.56) \end{aligned}$$

$$\frac{\partial^3}{\partial x_3^3}(I_l^m) = -\sqrt{(l-m+1)(l-m+2)(l-m+3)(l+m+1)(l+m+2)(l+m+3)}I_{l+3}^m$$
(D.57)

$$\frac{\partial^{3}}{\partial x_{1}^{2} \partial x_{2}} (I_{l}^{m}) = -\frac{i}{8} \sqrt{(l+m+1)(l+m+2)(l+m+3)(l+m+4)(l+m+5)(l+m+6)} I_{l+3}^{m+3} + \frac{i}{8} \sqrt{(l-m+1)(l-m+2)(l+m+1)(l+m+2)(l+m+3)(l+m+4)} I_{l+3}^{m+1} + \frac{i}{8} \sqrt{(l-m+1)(l-m+2)(l-m+3)(l-m+4)(l+m+1)(l+m+2)} I_{l+3}^{m-1} - \frac{i}{8} \sqrt{(l-m+1)(l-m+2)(l-m+3)(l-m+4)(l-m+5)(l-m+6)} I_{l+3}^{m-3}$$

$$(D.58)$$

$$\frac{\partial^3}{\partial x_1 \partial x_3^2} (I_l^m) = \frac{1}{2} \sqrt{(l-m+1)(l-m+2)(l+m+1)(l+m+2)(l+m+3)(l+m+4)} I_{l+3}^{m+1} - \frac{1}{2} \sqrt{(l-m+1)(l-m+2)(l-m+3)(l-m+4)(l+m+1)(l+m+2)} I_{l+3}^{m-1}$$
(D.62)

$$\frac{\partial^3}{\partial x_2 \partial x_3^2} (I_l^m) = -\frac{i}{2} \sqrt{(l-m+1)(l-m+2)(l+m+1)(l+m+2)(l+m+3)(l+m+4)} I_{l+3}^{m+1} - \frac{i}{2} \sqrt{(l-m+1)(l-m+2)(l-m+3)(l-m+4)(l+m+1)(l+m+2)} I_{l+3}^{m-1}$$
(D.63)

$$\frac{\partial^3}{\partial x_1 \partial x_2 \partial x_3} (I_l^m) = \frac{i}{4} \sqrt{(l-m+1)(l+m+1)(l+m+2)(l+m+3)(l+m+4)(l+m+5)} I_{l+3}^{m+2} - \frac{i}{4} \sqrt{(l-m+1)(l-m+2)(l-m+3)(l-m+4)(l-m+5)(l+m+1)} I_{l+3}^{m-2}$$
(D.64)

676 Appendix E. Derivatives of regular solid harmonics

Derivatives of regular solid harmonics are useful for the multipole expansion of the strain field of a cavity (Eq. (23)) and a dislocation segment (Eqs. (28)-(29)). They can be obtained by writing $R_l^m(x, \theta, \varphi) = x^{2l+1} I_l^m(x, \theta, \varphi)$, then using Eq. (B.6) and recursion formulas (B.12) and (B.13). First and second derivatives are given in Ref. [27]. They are recalled here for the sake of completeness.

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682 First derivatives

• m = 0

$$\frac{\partial}{\partial x_1} (R_l^0) = \frac{1}{2} \sqrt{l(l-1)} \left(R_{l-1}^{-1} + R_{l-1}^1 \right)$$
(E.1)

$$\frac{\partial}{\partial x_2}(R_l^0) = -\frac{i}{2}\sqrt{l(l-1)} \left(R_{l-1}^1 - R_{l-1}^{-1}\right)$$
(E.2)

$$\frac{\partial}{\partial x_3}(R_l^0) = lR_{l-1}^0 \tag{E.3}$$

 $\bullet \ m \geq 1$

$$\frac{\partial}{\partial x_1}(R_l^m) = \frac{1}{2} \left(\sqrt{(l-m)(l-m-1)} R_{l-1}^{m+1} - \sqrt{(l+m)(l+m-1)} R_{l-1}^{m-1} \right) \quad (E.4)$$

$$\frac{\partial}{\partial x_2}(R_l^m) = -\frac{i}{2} \left(\sqrt{(l-m)(l-m-1)} R_{l-1}^{m+1} + \sqrt{(l+m)(l+m-1)} R_{l-1}^{m-1} \right)$$
(E.5)

$$\frac{\partial}{\partial x_3}(R_l^m) = \sqrt{(l+m)(l-m)}R_{l-1}^m \tag{E.6}$$

683 Second derivatives

•
$$m = 0$$

$$\frac{\partial^2}{\partial x_1^2}(R_l^0) = \frac{1}{4}\sqrt{l(l-1)(l-2)(l-3)}\left(R_{l-2}^2 + R_{l-2}^{-2}\right) - \frac{1}{2}l(l-1)R_{l-2}^0$$
(E.7)

$$\frac{\partial^2}{\partial x_2^2}(R_l^0) = -\frac{1}{4}\sqrt{l(l-1)(l-2)(l-3)}\left(R_{l-2}^2 + R_{l-2}^{-2}\right) - \frac{1}{2}l(l-1)R_{l-2}^0$$
(E.8)

$$\frac{\partial^2}{\partial x_3^2} (R_l^0) = l(l-1)R_{l-2}^0$$
(E.9)

$$\frac{\partial^2}{\partial x_1 \partial x_2} (R_l^0) = -\frac{i}{4} \sqrt{l(l-1)(l-2)(l-3)} \left(R_{l-2}^2 - R_{l-2}^{-2} \right)$$
(E.10)

$$\frac{\partial^2}{\partial x_2 \partial x_3} (R_l^0) = -\frac{i}{2} l \sqrt{(l-1)(l-2)} \left(R_{l-2}^1 - R_{l-2}^{-1} \right)$$
(E.11)

$$\frac{\partial^2}{\partial x_1 \partial x_3} (R_l^0) = \frac{1}{2} l \sqrt{(l-1)(l-2)} \left(R_{l-2}^1 + R_{l-2}^{-1} \right)$$
(E.12)

• m = 1

$$\frac{\partial^2}{\partial x_1^2}(R_l^1) = \frac{1}{4}\sqrt{(l-1)(l-2)(l-3)(l-4)}R_{l-2}^3 - \frac{1}{2}\sqrt{(l+1)l(l-1)(l-2)}R_{l-2}^1 - \frac{1}{4}\sqrt{(l+1)l(l-1)(l-2)}R_{l-2}^{-1} \quad (E.13)$$

$$\frac{\partial^2}{\partial x_2^2}(R_l^1) = -\frac{1}{4}\sqrt{(l-1)(l-2)(l-3)(l-4)}R_{l-2}^3 - \frac{1}{2}\sqrt{(l+1)l(l-1)(l-2)}R_{l-2}^1 + \frac{1}{4}\sqrt{(l+1)l(l-1)(l-2)}R_{l-2}^{-1} \quad (E.14)$$

$$\frac{\partial^2}{\partial x_3^2} (R_l^1) = \sqrt{(l+1)l(l-1)(l-2)} R_{l-2}^1$$
(E.15)

$$\frac{\partial^2}{\partial x_1 \partial x_2} (R_l^1) = -\frac{i}{4} \sqrt{(l-1)(l-2)(l-3)(l-4)} R_{l-2}^3 - \frac{i}{4} \sqrt{(l+1)l(l-1)(l-2)} R_{l-2}^{-1}$$
(E.16)

$$\frac{\partial^2}{\partial x_2 \partial x_3} (R_l^1) = -\frac{i}{2} \sqrt{(l+1)(l-1)(l-2)(l-3)} R_{l-2}^2 - \frac{i}{2} \sqrt{(l+1)l(l-1)^2} R_{l-2}^0$$
(E.17)
$$\frac{\partial^2}{\partial x_1 \partial x_3} (R_l^1) = \frac{1}{2} \sqrt{(l+1)(l-1)(l-2)(l-3)} R_{l-2}^2 - \frac{1}{2} \sqrt{(l+1)l(l-1)^2} R_{l-2}^0$$
(E.18)

 $\bullet \ m \geq 2$

$$\frac{\partial^2}{\partial x_1^2} (R_l^m) = \frac{1}{4} \sqrt{(l-m)(l-m-1)(l-m-2)(l-m-3)} R_{l-2}^{m+2} - \frac{1}{2} \sqrt{(l+m)(l+m-1)(l-m)(l-m-1)} R_{l-2}^m + \frac{1}{4} \sqrt{(l+m)(l+m-1)(l+m-2)(l+m-3)} R_{l-2}^{m-2}$$
(E.19)

$$\frac{\partial^2}{\partial x_2^2} (R_l^m) = -\frac{1}{4} \sqrt{(l-m)(l-m-1)(l-m-2)(l-m-3)} R_{l-2}^{m+2} -\frac{1}{2} \sqrt{(l+m)(l+m-1)(l-m)(l-m-1)} R_{l-2}^m -\frac{1}{4} \sqrt{(l+m)(l+m-1)(l+m-2)(l+m-3)} R_{l-2}^{m-2}$$
(E.20)

$$\frac{\partial^2}{\partial x_3^2}(R_l^m) = \sqrt{(l+m)(l+m-1)(l-m)(l-m-1)}R_{l-2}^m$$
(E.21)

$$\frac{\partial^2}{\partial x_1 \partial x_2} (R_l^m) = -\frac{i}{4} \sqrt{(l-m)(l-m-1)(l-m-2)(l-m-3)} R_{l-2}^{m+2} + \frac{i}{4} \sqrt{(l+m)(l+m-1)(l+m-2)(l+m-3)} R_{l-2}^{m-2} \quad (E.22)$$

$$\frac{\partial^2}{\partial x_2 \partial x_3} (R_l^m) = -\frac{i}{2} \sqrt{(l+m)(l-m)(l-m-1)(l-m-2)} R_{l-2}^{m+1} - \frac{i}{2} \sqrt{(l+m)(l+m-1)(l+m-2)(l-m)} R_{l-2}^{m-1} \quad (E.23)$$

$$\frac{\partial^2}{\partial x_1 \partial x_3} (R_l^m) = \frac{1}{2} \sqrt{(l+m)(l-m)(l-m-1)(l-m-2)} R_{l-2}^{m+1} - \frac{1}{2} \sqrt{(l+m)(l+m-1)(l+m-2)(l-m)} R_{l-2}^{m-1} \quad (E.24)$$

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$$m = 0$$

$$\frac{\partial^{3}}{\partial x_{1}^{3}}(R_{l}^{0}) = \frac{1}{8}\sqrt{l(l-1)(l-2)(l-3)(l-4)(l-5)} \left(R_{l-3}^{3} + R_{l-3}^{-3}\right)$$

$$-\frac{3}{8}l(l-1)\sqrt{(l-2)(l-3)} \left(R_{l-3}^{1} + R_{l-3}^{-1}\right) \quad (E.25)$$

$$\frac{\partial^3}{\partial x_2^3} (R_l^0) = \frac{i}{8} \sqrt{l(l-1)(l-2)(l-3)(l-4)(l-5)} \left(R_{l-3}^3 - R_{l-3}^{-3} \right) \\ + \frac{3i}{8} l(l-1) \sqrt{(l-2)(l-3)} \left(R_{l-3}^1 - R_{l-3}^{-1} \right)$$
(E.26)
$$\frac{\partial^3}{\partial x_2^3} (R_l^0) = l(l-1) \sqrt{(l-2)(l-3)} \left(R_{l-3}^1 - R_{l-3}^{-1} \right)$$
(E.26)

$$\frac{\partial^3}{\partial x_3^3}(R_l^0) = l(l-1)(l-2)R_{l-3}^0 \tag{E.27}$$

$$\frac{\partial^3}{\partial x_1^2 \partial x_2} (R_l^0) = -\frac{i}{8} \sqrt{l(l-1)(l-2)(l-3)(l-4)(l-5)} \left(R_{l-3}^3 - R_{l-3}^{-3} \right) \\ + \frac{i}{8} l(l-1) \sqrt{(l-2)(l-3)} \left(R_{l-3}^1 - R_{l-3}^{-1} \right)$$
(E.28)

$$\frac{\partial^3}{\partial x_1^2 \partial x_3} (R_l^0) = \frac{1}{4} \sqrt{l^2 (l-1)(l-2)(l-3)(l-4)} \left(R_{l-3}^2 + R_{l-3}^{-2} \right) - \frac{1}{2} l(l-1)(l-2) R_{l-3}^0$$
(E.29)

$$\frac{\partial^3}{\partial x_1 \partial x_2^2} (R_l^0) = -\frac{1}{8} \sqrt{l(l-1)(l-2)(l-3)(l-4)(l-5)} \left(R_{l-3}^3 + R_{l-3}^{-3} \right) -\frac{1}{8} l(l-1) \sqrt{(l-2)(l-3)} \left(R_{l-3}^1 + R_{l-3}^{-1} \right)$$
(E.30)

$$\frac{\partial^3}{\partial x_2^2 \partial x_3} (R_l^0) = -\frac{1}{4} \sqrt{l^2 (l-1)(l-2)(l-3)(l-4)} \left(R_{l-3}^2 + R_{l-3}^{-2} \right) - \frac{1}{2} l(l-1)(l-2) R_{l-3}^0$$
(E.31)

$$\frac{\partial^3}{\partial x_1 \partial x_3^2} (R_l^0) = \frac{1}{2} l(l-1) \sqrt{(l-2)(l-3)} \left(R_{l-3}^1 + R_{l-3}^{-1} \right)$$
(E.32)

$$\frac{\partial^3}{\partial x_2 \partial x_3^2} (R_l^0) = -\frac{i}{2} l(l-1) \sqrt{(l-2)(l-3)} \left(R_{l-3}^1 - R_{l-3}^{-1} \right)$$
(E.33)

$$\frac{\partial^3}{\partial x_1 \partial x_2 \partial x_3} (R_l^0) = -\frac{i}{4} \sqrt{l^2 (l-1)(l-2)(l-3)(l-4)} \left(R_{l-3}^2 - R_{l-3}^{-2} \right)$$
(E.34)
• $m = 1$

$$\frac{\partial^3}{\partial x_1^3}(R_l^1) = \frac{1}{8}\sqrt{(l-1)(l-2)(l-3)(l-4)(l-5)(l-6)}R_{l-3}^4$$
$$-\frac{3}{8}\sqrt{(l+1)l(l-1)(l-2)(l-3)(l-4)}R_{l-3}^2$$
$$+\frac{3}{8}\sqrt{(l+1)l}(l-1)(l-2)R_{l-3}^0$$
$$-\frac{1}{8}\sqrt{(l+1)l(l-1)(l-2)(l-3)(l-4)}R_{l-3}^{-2} \quad (E.35)$$

$$\begin{aligned} \frac{\partial^3}{\partial x_2^3}(R_l^1) &= \frac{i}{8}\sqrt{(l-1)(l-2)(l-3)(l-4)(l-5)(l-6)}R_{l-3}^4 \\ &+ \frac{3i}{8}\sqrt{(l+1)l(l-1)(l-2)(l-3)(l-4)}R_{l-3}^2 \\ &+ \frac{3i}{8}\sqrt{(l+1)l}(l-1)(l-2)R_{l-3}^0 \\ &+ \frac{i}{8}\sqrt{(l+1)l(l-1)(l-2)(l-3)(l-4)}R_{l-3}^{-2} \end{aligned} \tag{E.36}$$

$$\frac{\partial^3}{\partial x_3^3}(R_l^1) = \sqrt{(l+1)l(l-1)^2(l-2)(l-3)}R_{l-3}^1$$
(E.37)

$$\frac{\partial^3}{\partial x_1^2 \partial x_2} (R_l^1) = -\frac{i}{8} \sqrt{(l-1)(l-2)(l-3)(l-4)(l-5)(l-6)} R_{l-3}^4 + \frac{i}{8} \sqrt{(l+1)l(l-1)(l-2)(l-3)(l-4)} R_{l-3}^2 + \frac{i}{8} \sqrt{(l+1)l}(l-1)(l-2) R_{l-3}^0 - \frac{i}{8} \sqrt{(l+1)l(l-1)(l-2)(l-3)(l-4)} R_{l-3}^{-2}$$
(E.38)

$$\begin{aligned} \frac{\partial^3}{\partial x_1^2 \partial x_3} (R_l^1) &= \frac{1}{4} \sqrt{(l+1)(l-1)(l-2)(l-3)(l-4)(l-5)} R_{l-3}^3 \\ &\quad -\frac{1}{2} \sqrt{(l+1)l(l-1)^2(l-2)(l-3)} R_{l-3}^1 \\ &\quad -\frac{1}{4} \sqrt{(l+1)l(l-1)^2(l-2)(l-3)} R_{l-3}^{-1} \end{aligned}$$
(E.39)

$$\begin{aligned} \frac{\partial^3}{\partial x_1 \partial x_2^2}(R_l^1) &= -\frac{1}{8}\sqrt{(l-1)(l-2)(l-3)(l-4)(l-5)(l-6)}R_{l-3}^4 \\ &\quad -\frac{1}{8}\sqrt{(l+1)l(l-1)(l-2)(l-3)(l-4)}R_{l-3}^2 \\ &\quad +\frac{1}{8}\sqrt{(l+1)l(l-1)(l-2)}R_{l-3}^0 \\ &\quad +\frac{1}{8}\sqrt{(l+1)l(l-1)(l-2)(l-3)(l-4)}R_{l-3}^{-2} \end{aligned}$$
(E.40)

$$\frac{\partial^3}{\partial x_2^2 \partial x_3} (R_l^1) = -\frac{1}{4} \sqrt{(l+1)(l-1)(l-2)(l-3)(l-4)(l-5)} R_{l-3}^3 - \frac{1}{2} \sqrt{(l+1)l(l-1)^2(l-2)(l-3)} R_{l-3}^1 + \frac{1}{4} \sqrt{(l+1)l(l-1)^2(l-2)(l-3)} R_{l-3}^{-1} \quad (E.41)$$

$$\frac{\partial^{3}}{\partial x_{1}\partial x_{3}^{2}}(R_{l}^{1}) = \frac{1}{2}\sqrt{(l+1)l(l-1)(l-2)(l-3)(l-4)}R_{l-3}^{2} - \frac{1}{2}\sqrt{(l+1)l}(l-1)(l-2)R_{l-3}^{0}$$
(E.42)
$$\frac{\partial^{3}}{\partial x_{2}\partial x_{3}^{2}}(R_{l}^{1}) = -\frac{i}{2}\sqrt{(l+1)l(l-1)(l-2)(l-3)(l-4)}R_{l-3}^{2} - \frac{i}{2}\sqrt{(l+1)l}(l-1)(l-2)R_{l-3}^{0}$$
(E.43)
$$\frac{\partial^{3}}{\partial x_{2}\partial x_{3}^{2}}(R_{l}^{1}) = -\frac{i}{2}\sqrt{(l+1)(l-1)(l-2)(l-3)(l-4)}R_{l-3}^{2} - \frac{i}{2}\sqrt{(l+1)l}(l-1)(l-2)R_{l-3}^{0}$$

$$\frac{\partial^{2}}{\partial x_{1}\partial x_{2}\partial x_{3}}(R_{l}^{1}) = -\frac{i}{4}\sqrt{(l+1)(l-1)(l-2)(l-3)(l-4)(l-5)}R_{l-3}^{3} - \frac{i}{4}\sqrt{(l+1)l(l-1)^{2}(l-2)(l-3)}R_{l-3}^{-1} \quad (E.44)$$

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$$m = 2$$

$$\frac{\partial^{3}}{\partial x_{1}^{3}}(R_{l}^{2}) = \frac{1}{8}\sqrt{(l-2)(l-3)(l-4)(l-5)(l-6)(l-7)}R_{l-3}^{5}$$

$$-\frac{3}{8}\sqrt{(l+2)(l+1)(l-2)(l-3)(l-4)(l-5)}R_{l-3}^{3}$$

$$+\frac{3}{8}\sqrt{(l+2)(l+1)l(l-1)(l-2)(l-3)}R_{l-3}^{1}$$

$$+\frac{1}{8}\sqrt{(l+2)(l+1)l(l-1)(l-2)(l-3)}R_{l-3}^{-1}$$
(E.45)

$$\frac{\partial^{3}}{\partial x_{2}^{3}}(R_{l}^{2}) = \frac{i}{8}\sqrt{(l-2)(l-3)(l-4)(l-5)(l-6)(l-7)}R_{l-3}^{5} + \frac{3i}{8}\sqrt{(l+2)(l+1)(l-2)(l-3)(l-4)(l-5)}R_{l-3}^{3} + \frac{3i}{8}\sqrt{(l+2)(l+1)l(l-1)(l-2)(l-3)}R_{l-3}^{1} - \frac{i}{8}\sqrt{(l+2)(l+1)l(l-1)(l-2)(l-3)}R_{l-3}^{-1} \quad (E.46)$$

$$\frac{\partial^{3}}{\partial x_{2}^{3}}(R_{l}^{2}) = \sqrt{(l-2)(l-1)l(l-2)(l-3)}R_{l-3}^{-1} \quad (E.46)$$

$$\frac{\partial^3}{\partial x_3^3}(R_l^2) = \sqrt{(l+2)(l+1)l(l-2)(l-3)(l-4)}R_{l-3}^2$$
(E.47)

$$\frac{\partial^3}{\partial x_1^2 \partial x_2} (R_l^2) = -\frac{i}{8} \sqrt{(l-2)(l-3)(l-4)(l-5)(l-6)(l-7)} R_{l-3}^5 + \frac{i}{8} \sqrt{(l+2)(l+1)(l-2)(l-3)(l-4)(l-5)} R_{l-3}^3 + \frac{i}{8} \sqrt{(l+2)(l+1)l(l-1)(l-2)(l-3)} R_{l-3}^1 + \frac{i}{8} \sqrt{(l+2)(l+1)l(l-1)(l-2)(l-3)} R_{l-3}^1 + \frac{i}{8} \sqrt{(l+2)(l+1)l(l-1)(l-2)(l-3)} R_{l-3}^1$$
(E.48)

$$\frac{\partial^3}{\partial x_1^2 \partial x_3} (R_l^2) = \frac{1}{4} \sqrt{(l+2)(l-2)(l-3)(l-4)(l-5)(l-6)} R_{l-3}^4 - \frac{1}{2} \sqrt{(l+2)(l+1)l(l-2)(l-3)(l-4)} R_{l-3}^2 + \frac{1}{4} \sqrt{(l+2)(l+1)l(l-1)(l-2)^2} R_{l-3}^0$$
(E.49)

$$\frac{\partial^3}{\partial x_1 \partial x_2^2} (R_l^2) = -\frac{1}{8} \sqrt{(l-2)(l-3)(l-4)(l-5)(l-6)(l-7)} R_{l-3}^5 -\frac{1}{8} \sqrt{(l+2)(l+1)(l-2)(l-3)(l-4)(l-5)} R_{l-3}^3 +\frac{1}{8} \sqrt{(l+2)(l+1)l(l-1)(l-2)(l-3)} R_{l-3}^1 -\frac{1}{8} \sqrt{(l+2)(l+1)l(l-1)(l-2)(l-3)} R_{l-3}^{-1}$$
(E.50)

$$\frac{\partial^3}{\partial x_2^2 \partial x_3} (R_l^2) = -\frac{1}{4} \sqrt{(l+2)(l-2)(l-3)(l-4)(l-5)(l-6)} R_{l-3}^4$$
$$-\frac{1}{2} \sqrt{(l+2)(l+1)l(l-2)(l-3)(l-4)} R_{l-3}^2$$
$$-\frac{1}{4} \sqrt{(l+2)(l+1)l(l-1)(l-2)^2} R_{l-3}^0 \quad (E.51)$$

$$\frac{\partial^3}{\partial x_1 \partial x_3^2} (R_l^2) = \frac{1}{2} \sqrt{(l+2)(l+1)(l-2)(l-3)(l-4)(l-5)} R_{l-3}^3 - \frac{1}{2} \sqrt{(l+2)(l+1)l(l-1)(l-2)(l-3)} R_{l-3}^1 \quad (E.52)$$

$$\frac{\partial^3}{\partial x_2 \partial x_3^2} (R_l^2) = -\frac{i}{2} \sqrt{(l+2)(l+1)(l-2)(l-3)(l-4)(l-5)} R_{l-3}^3 - \frac{i}{2} \sqrt{(l+2)(l+1)l(l-1)(l-2)(l-3)} R_{l-3}^1 \quad (E.53)$$

$$\frac{\partial^3}{\partial x_1 \partial x_2 \partial x_3} (R_l^2) = -\frac{i}{4} \sqrt{(l+2)(l-2)(l-3)(l-4)(l-5)(l-6)} R_{l-3}^4 + \frac{i}{4} \sqrt{(l+2)(l+1)l(l-1)(l-2)^2} R_{l-3}^0 \quad (E.54)$$

• $m \ge 3$

$$\begin{aligned} \frac{\partial^3}{\partial x_1^3}(R_l^m) &= \frac{1}{8}\sqrt{(l-m)(l-m-1)(l-m-2)(l-m-3)(l-m-4)(l-m-5)}R_{l-3}^{m+3} \\ &- \frac{3}{8}\sqrt{(l+m)(l+m-1)(l-m)(l-m-1)(l-m-2)(l-m-3)}R_{l-3}^{m+1} \\ &+ \frac{3}{8}\sqrt{(l+m)(l+m-1)(l+m-2)(l+m-3)(l-m)(l-m-1)}R_{l-3}^{m-1} \\ &- \frac{1}{8}\sqrt{(l+m)(l+m-1)(l+m-2)(l+m-3)(l+m-4)(l+m-5)}R_{l-3}^{m-3} \\ &\qquad (E.55) \end{aligned}$$

$$\frac{\partial^3}{\partial x_3^3}(R_l^m) = \sqrt{(l+m)(l+m-1)(l+m-2)(l-m)(l-m-1)(l-m-2)}R_{l-3}^m$$
(E.57)

$$\begin{aligned} \frac{\partial^3}{\partial x_1^2 \partial x_2} (R_l^m) &= -\frac{i}{8} \sqrt{(l-m)(l-m-1)(l-m-2)(l-m-3)(l-m-4)(l-m-5)} R_{l-3}^{m+3} \\ &+ \frac{i}{8} \sqrt{(l+m)(l+m-1)(l-m)(l-m-1)(l-m-2)(l-m-3)} R_{l-3}^{m+1} \\ &+ \frac{i}{8} \sqrt{(l+m)(l+m-1)(l+m-2)(l+m-3)(l-m)(l-m-1)} R_{l-3}^{m-1} \\ &- \frac{i}{8} \sqrt{(l+m)(l+m-1)(l+m-2)(l+m-3)(l+m-4)(l+m-5)} R_{l-3}^{m-3} \\ &\qquad (E.58) \end{aligned}$$

$$\frac{\partial^3}{\partial x_1^2 \partial x_3} (R_l^m) = \frac{1}{4} \sqrt{(l+m)(l-m)(l-m-1)(l-m-2)(l-m-3)(l-m-4)} R_{l-3}^{m+2} - \frac{1}{2} \sqrt{(l+m)(l+m-1)(l+m-2)(l-m)(l-m-1)(l-m-2)} R_{l-3}^m + \frac{1}{4} \sqrt{(l+m)(l+m-1)(l+m-2)(l+m-3)(l+m-4)(l-m)} R_{l-3}^{m-2} \quad (E.59)$$

$$\frac{\partial^3}{\partial x_2^2 \partial x_3} (R_l^m) = -\frac{1}{4} \sqrt{(l+m)(l-m)(l-m-1)(l-m-2)(l-m-3)(l-m-4)} R_{l-3}^{m+2} - \frac{1}{2} \sqrt{(l+m)(l+m-1)(l+m-2)(l-m)(l-m-1)(l-m-2)} R_{l-3}^m - \frac{1}{4} \sqrt{(l+m)(l+m-1)(l+m-2)(l+m-3)(l+m-4)(l-m)} R_{l-3}^{m-2}$$
(E.61)

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$$\frac{\partial^3}{\partial x_1 \partial x_3^2} (R_l^m) = \frac{1}{2} \sqrt{(l+m)(l+m-1)(l-m)(l-m-1)(l-m-2)(l-m-3)} R_{l-3}^{m+1} - \frac{1}{2} \sqrt{(l+m)(l+m-1)(l+m-2)(l+m-3)(l-m)(l-m-1)} R_{l-3}^{m-1}$$
(E.62)

$$\frac{\partial^3}{\partial x_2 \partial x_3^2} (R_l^m) = -\frac{i}{2} \sqrt{(l+m)(l+m-1)(l-m)(l-m-1)(l-m-2)(l-m-3)} R_{l-3}^{m+1} - \frac{i}{2} \sqrt{(l+m)(l+m-1)(l+m-2)(l+m-3)(l-m)(l-m-1)} R_{l-3}^{m-1} \quad (E.63)$$

$$\frac{\partial^3}{\partial x_1 \partial x_2 \partial x_3} (R_l^m) = -\frac{i}{4} \sqrt{(l+m)(l-m)(l-m-1)(l-m-2)(l-m-3)(l-m-4)} R_{l-3}^{m+2} + \frac{i}{4} \sqrt{(l+m)(l+m-1)(l+m-2)(l+m-3)(l+m-4)(l-m)} R_{l-3}^{m-2} \quad (E.64)$$

⁶⁸⁵ Appendix F. Correction ε^0 to the dipolar field

The field which must be subtracted from the discrete sum of the dipolar field over neighbour boxes, in order to get a periodic solution, has an analytical form if the sum is performed over a cuboid domain centered on the similation box [48]. We recall here its expression for the sake of completeness. Let the dimensions of the summation domain be l_1 , l_2 and l_3 . We have for the two components ε_{11}^0 and ε_{12}^0 (other terms are obtained by cyclic permutation of indices):

$$\varepsilon_{11}^{0} = \frac{p_{11}}{2\pi(1-\nu)\mu} \left[2(1-2\nu)A(l_1,l_2,l_3) + B(l_1,l_2,l_3) \right] \\ - \frac{p_{22}}{2\pi(1-\nu)\mu} E(l_2,l_1,l_3) - \frac{p_{33}}{2\pi(1-\nu)\mu} E(l_3,l_1,l_2) \quad (F.1)$$

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$$\varepsilon_{12}^{0} = \frac{p_{12}}{2\pi(1-\nu)\mu} \left[2(1-\nu) \left(A(l_1, l_2, l_3) + A(l_2, l_1, l_3) \right) - \left(E(l_1, l_2, l_3) + E(l_2, l_1, l_3) \right) \right].$$
(F.2)

⁶⁹³ Functions A, B and E are given by

$$A(x, y, z) = \arctan\left(\frac{yz}{x\sqrt{x^2 + y^2 + z^2}}\right)$$
(F.3)

$$B(x, y, z) = \frac{xyz(2x^2 + y^2 + z^2)}{(x^2 + y^2)(x^2 + z^2)\sqrt{x^2 + y^2 + z^2}}$$
(F.4)

$$E(x, y, z) = \frac{xyz}{(x^2 + y^2)\sqrt{x^2 + y^2 + z^2}}.$$
 (F.5)

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