

From arc-melted ingot to MTR fuel plate: a SEM/EBSD microstructural study of U_3Si_2

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1: 50^{èmes} Journées des Actinides 13th School on the Physics and Chemistry of the Actinides



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From arc-melted ingot to MTR fuel plate: a SEM/EBSD microstructural study of U₃Si₂

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50^{èmes} Journées des Actinides – 24/03/2021

Institut de recherche sur les systèmes nucléaires pour la production d'énergie bas carbone



Introduction

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- Conversion of Research Reactors to Low Enriched Uranium (<20% ²³⁵U).
- Development of new nuclear fuels with higher uranium density, for high performance MTRs.
- U_3Si_2 is one of the main candidates.

Fuel	Melting point (°C)	Physical density (gcm ⁻³)	Uranium Ioading (g cm ⁻³)			
U	1133	19.1	19.1			
U–7Mo	1145	18.4	17.1			
U–10Mo	1150	18.2	16.4			
U ₆ Mn	726	17.8	17.1			
U ₆ Fe	815	17.7	17.0			
U ₃ Si	930	15.6	15.0			
U ₃ Si ₂	1665	12.2	11.3			
USi	1580	10.96	9.8			
UAI ₂	1590	8.1	6.6			
UAI3	1350	6.8	5.0			
UAI ₄	731	6.1	4.2			
$U_{0.9}AI_4$	641	5.7	3.7			
UAI _x	NA	6.4	4.5			



- Objectives of this study
 - Material Testing Reactors (MTRs) → mainly nuclear fuel plates.
 - U₃Si₂/Al fuel plate manufacturing process



- Microstructural characteristics before irradiation important to understand the in-pile behaviour.
- EBSD is an efficient technique.
- U₃Si₂ microstructure reinvestigated after the **<u>3 main steps</u>** of the manufacturing process.



Experimental methods

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• U₃Si₂/Al fuel plate

Samples

Framatome-CERCA production
Fabrication in the frame of the CEA SHARE irradiation test (Leenaers *et al.*, 2008)



Arc-melted ingots and crushed ingots

 \circ Laboratory (CEA-Cadarache) production

Label	Cooling	U/Si atomic ratio	Weight (g)		
I1	"Very fast"	60/40	0.5		
I2	"Slow"	60/40	1		
I3	"Fast"	60/40 "low purity U"	1		
14	"Fast"	58/42	1		









- X-Ray Diffraction (XRD)
- Scanning Electron Microscopy (SEM) + Energy Dispersive Spectroscopy (EDS)
- Electron BackScattered Diffraction (EBSD)





Results and discussion

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U₃Si₂ 010



- Important fragmentation of particles.
- Single crystalline particles.







Map coloured in phase highlights the presence of Si rich secondary phases :

- U₃₄Si_{34.5} (3.5%)
- U₈Si₈X (0.5%)







SEM characterizations

• αU at grain boundaries.









Ceal "High purity" stoichiometric ingots - Cooling rate effect



Arc Vertical section Copper hearth

Ceal "High purity" stoichiometric ingots - Cooling rate effect



Grains size analysis



Slow cooling favours grain growth.

Impact on the crushed ingot



I1: fast cooled ingot



12: slow cooled ingot



- Mainly single crystalline particles.
- Grain boundaries in particles \rightarrow intragranular fragmentation (as evidenced by Metzger *et al.*, 2017).
- Influence of fast cooling:
 - $\rm U_3Si_2$ grains are smaller in the ingot
 - U_3Si_2 particles are larger after a given crushing sequence \rightarrow influence of the higher amount of grain boundaries on the mechanical resistance?



"Low purity" stoichiometric ingot

- Secondary phases indexed :
 - \circ U₈Si₈X
 - U₃₄Si_{34.5} (Le Bihan *et al.*, 1996)
- Solidification in spiral shape

I3 ingot









2%at. Silicon excess

- Secondary phase indexed as U₈Si₈X.
- Solidification in spiral shape.





\rightarrow Low purity uranium leads to an overestimation of uranium and deviation to stoichiometry



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Conclusion

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First EBSD examination performed on $U_3Si_2 \rightarrow$ new insights in characterization

Study of as-fabricated U_3Si_2/Al fuel plate and demonstration of the presence of:

o large grains and single crystalline particles (slow cooling rate on industrial ingots),

→ Laboratory production highlights that fast cooling rate >> grain size and >> mechanical resistance because of the intragranular fragmentation?

 \rightarrow Same fragmentation mechanism observed after U₃Si₂ hydrogenation.









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• two different Si rich secondary phases.

 \rightarrow U₈Si₈X supposingly stabilised by impurity (Laugier *et al.*, 1971) observed with high purity materials.

→ Thermodynamic calculations underway to understand its formation condition.

For further reading: Havette *et al.*, From arc-melted ingot to MTR fuel plate: A SEM/EBSD microstructural study of U₃Si₂, in *Journal of Nuclear Materials*, vol. 537, **2020**



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Thank you for your attention

JdA-2021: 50^{èmes} Journées des Actinides + 13th School on the Physics and Chemistry of the Actinides

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Appendix

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Table 1. Main impurities (in mg.kg⁻¹) in the two uranium batches used for the synthesis of U_3Si_2 by arc-melting.

U batches	Elements														
	Ag	AI	В	с	Со	Cr	Cu	Fe	Mn	Мо	ο	Ni	Pb	Si	Sn
"High purity"	<0.5	15	<0.2	300	≤1	3.5	2	25	3.5	<1	300	7	≤1	10	2
"Low purity"	-	90	-	300	-	12	15	85	5	-	1500	40	-	37	-



Cea Non stoichiometric ingots

Mechanism of spiral microstructure formation







Already observed in Zn-Zn+3%Mg eutectic (hexagonal) [9].



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C22 U-Si binary diagram

• Several compounds in the binary diagram:

