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Article

Insight into the Crystal Structures and Physical Properties of the Uranium Borides $UB_{1.78\pm 0.02}$, $UB_{3.61\pm 0.041}$ and $UB_{11.19\pm 0.13}$

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Abstract: In this study we reported the synthesis of three polycrystalline uranium borides $UB_{1.78\pm 0.02}$, $UB_{3.61\pm 0.041}$, and $UB_{11.19\pm 0.13}$ and their analyses using chemical analysis, X-ray diffraction, SQUID magnetometry, solid-state NMR, and Fourier transformed infrared spectroscopy. We discuss the effects of stoichiometry deviations on the lattice parameters and magnetic properties. We also provide their static and MAS-NMR spectra showing the effects of the 5f-electrons on the ^{11}B shifts. Finally, the FTIR measurements showed the presence of a local disorder.

Keywords: uranium borides; magnetic susceptibility; ^{11}B NMR; FTIR



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1. Introduction

Borides have been studied for their properties such as hardness, stability to radiolytic decay, chemical inertness, and magnetism. Three main applications made them of particular interest: (i) their consideration as an alternative intermediate storage form for actinide elements due to their high refractory properties [1], (ii) their potential formation in sodium cooled fast reactors because of a core melt during reaction of the B_4C control rods with the UO_2 fuel [2], and (iii) their possibility as candidate constituents for multi-phase accident tolerant fuel [3]. In the uranium-boron phase diagram, three compounds have been reported to exist: UB_2 , UB_4 , and UB_{12} [4,5].

They are mostly synthesized by arc-melting elemental uranium and boron in stoichiometric amounts. Nevertheless, due to boron evaporation, non-stoichiometric UB_X ($X = 2 \pm x$, $4 \pm x$, or $12 \pm x$) phases and additional UB_X or UO_2 phases are often detected. Brewer et al. [6] reported the detection of UB_2 in samples containing 25 to 75 atomic %B and UB_4 in samples containing 75 to 85 atomic %B. The magnetic properties of the UB_X ($X = 2, 4, 12$) are intensively studied in the literature. Indeed, UB_{2-x} is a Pauli paramagnet and a compensated metal with closed Fermi surfaces [7]. UB_{4-x} is a magnetic compensated metal, but also a moderate heavy fermion where a cross-over is observed between itinerant 5f electrons at low temperatures and localized 5f electrons at high temperature with a Curie–Weiss behavior [8,9]. Finally, UB_{12-x} is a Pauli paramagnet and compensated metal, with speculations of superconductivity below 0.4 K [10,11]. Nevertheless, the slight composition/impurities effects on the magnetism has been reported [12] but not thoroughly discussed.

Here, we present the synthesis of $UB_{1.78\pm 0.02}$, $UB_{3.61\pm 0.041}$, and $UB_{11.19\pm 0.13}$ by arc melting, their room temperature crystal structure by XRD and low temperature single crystal XRD (100 K to 300 K), and their magnetic susceptibility and their local structure determined by ^{11}B nuclear magnetic resonance (NMR) and Fourier transformed infrared

(FTIR) spectroscopy. We discuss our results in comparison with the literature and clarify the differences observed based mostly on composition effects.

2. Materials and Methods

The samples were prepared by arc melting of the constituent elements, uranium metal and boron, under a high purity argon atmosphere (6N), on a water-cooled copper hearth. Metallic zirconium in the chamber acts as a getter for oxygen. The UB_x ingots were melted and turned several times to achieve homogenous samples. To minimise oxidation, the samples were stored under high vacuum ($\sim 10^{-6}$ mbar). Chemical analyses to determine the B/U ratio were done using Inductively coupled plasma mass spectrometry (ICPMS). The oxygen content was measured via a direct combustion using the infrared absorption detection technique with an ELTRA ONH-2000 instrument. The detection limit for oxygen was determined by measuring a blank 10 times, and the standard deviation of these measurements multiplied by 2.33 was considered as the detection limit. Powder X-ray diffraction analyses were performed on a Bruker Bragg-Brentano D8 advanced diffractometer (Cu $K\alpha_1$ radiation at a wavelength of 1.5406 Å) equipped with a Ge (111) monochromator and a Lynxeye linear position sensitive detector. The powder patterns were recorded at room temperature using a step size of 0.01973° with an exposure of 4 s across the angular range $15^\circ \leq 2\theta \leq 120^\circ$. Operating conditions were 40 kV and 40 mA. The Rietveld refinement was implemented using the program Topas version 4.1. Single crystals of UB_x selected from the as-cast samples were measured between 100 and 300 K on a Bruker APEX II Quazar diffractometer (Mo- $K\alpha$ radiation, graphite monochromator, $\lambda = 0.71073$ Å) to follow the lattice parameters versus the temperature. Infrared spectra were recorded on a Bruker Alpha-P FT-IR spectrometer equipped with a “platinum” attenuated total reflection sample module. Using the Origin 2021 software, each main peak was selected. All the samples were analyzed on a 9.4T Bruker NMR spectrometer (^{11}B frequency at 128.38 MHz) adapted for the study of nuclear materials. To avoid skin-depth effects and for better RF penetration, powders were used by crushing the bulk samples. A 1.3 mm probe was used and the samples were spun at 40 kHz. A one pulse experiment was performed with a 90° pulse of $1\mu\text{s}$ with optimised recycling delays of 0.5 s to 2 s. For $UB_{11.19}$ and $UB_{3.61}$, the full static spectra were obtained using the variable offset cumulative spectrum (VOCS) technique [13,14]. The samples were referenced to 1 M H_3BO_3 (liq.) as an external reference at 19.6 ppm. All the spectra were fitted using the dmfit software [15] and home-built software [16].

3. Results and Discussion

3.1. Chemical Analyses and X-ray Diffraction

After preparation by arc melting, each sample was analyzed by chemical analyses to define the U/B ratio and eventual oxygen content. Table 1 summarizes the results. Analyses of O content in the sample were successfully done and the quantity was below 200 ppm (detection limit of the instrument). This enabled us to demonstrate the quality of the samples prepared by arc-melting. In the following section, to underline the non-stoichiometry, we call the samples $UB_{1.78}$, $UB_{3.61}$, and $UB_{11.19}$, implying that the composition uncertainty is included. The atomic percentage loss in boron goes from 11, 10 to 7 at% for the samples with formal composition UB_2 , UB_4 , and UB_{12} , respectively. The atomic% of boron falls within the composition range of UB_2 and UB_4 described by Brewer et al. [6]. Usually, excess of 5 to 10 wt% in boron is added to reach a pure stoichiometry but, as the different phase structure exists within a range of composition of U/B ratio [6], we did not compensate for the boron loss. We also wanted to avoid the formation of additional phases in the samples.

Table 1. U/B ratio obtained by ICPMS and oxygen content determined by direct combustion—infrared absorption.

Samples	UB_{2-x}	UB_{4-x}	UB_{12-x}
U/B (at/at)	1.780 ± 0.020	3.613 ± 0.041	11.19 ± 0.13
Oxygen content (mg/g)		<200 *	

* Detection limit of the method.

The XRD patterns reported in Figure 1 and Figure S1 prove well crystallized samples—narrow peaks with high intensities. One crystalline phase was detected for $UB_{1.78}$ and $UB_{11.19}$, unlike $UB_{3.68}$ where an admixture of 5 wt% of UB_{2-x} was detected. The corresponding lattice parameters, crystal structures, and phase compositions are reported in Table 2. In the U–B phase diagram, UB_2 crystallizes in the hexagonal crystal structure, UB_{12} in the cubic crystal structure, and UB_4 in the tetragonal crystal structure (insets Figure 1).

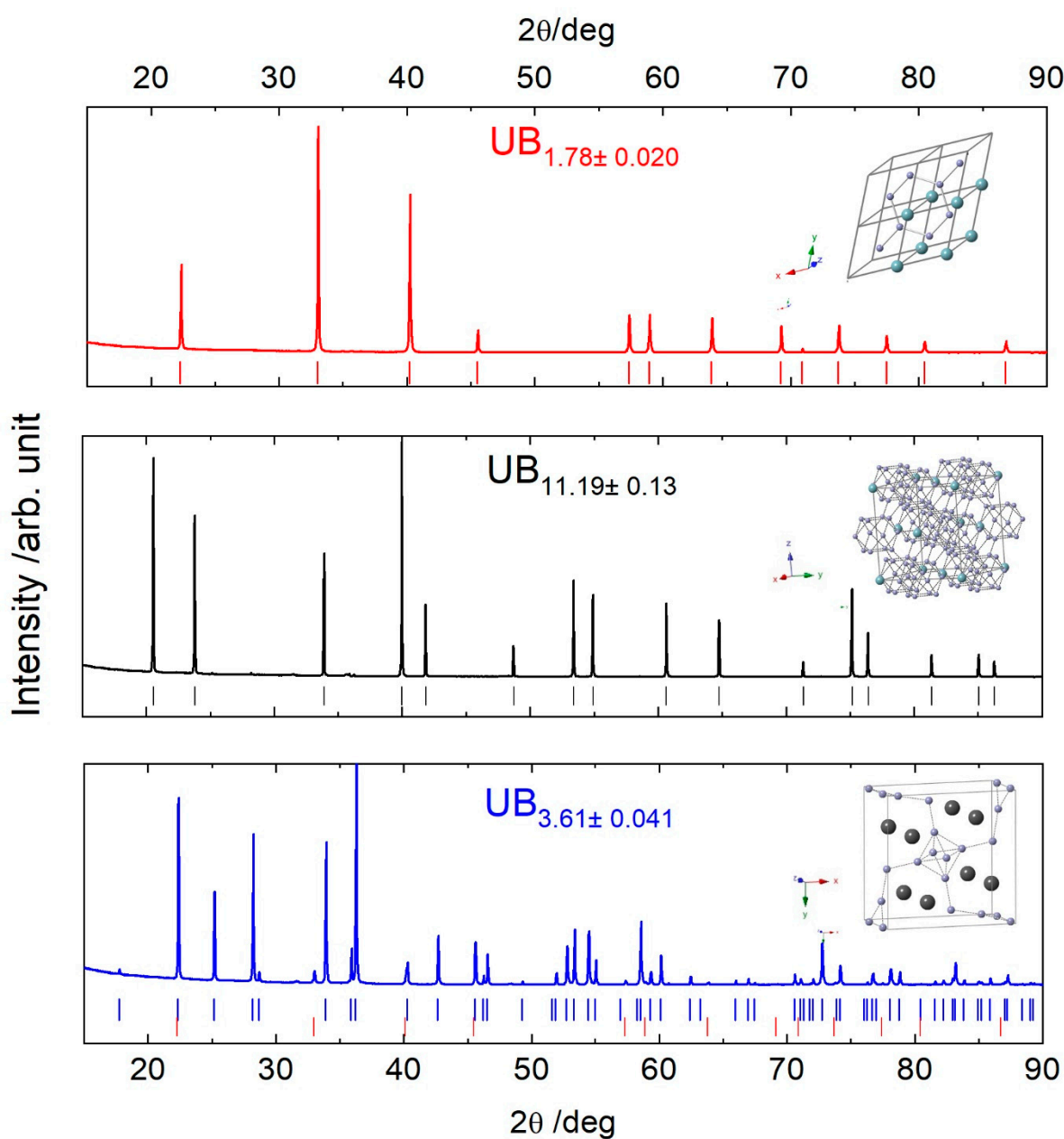
**Figure 1.** X-ray diffraction patterns and structures of the uranium borides. The uranium atoms are bigger than the B atoms.

Table 2. Structural parameters.

Name	UB _{1.78}	UB _{11.19}	UB _{3.61}
Structure	hexagonal	cubic	tetragonal
Space group n°	P6/mmm	Fm-3m	P4/mbm
a = b/Å	3.1225 (1)	7.4715 (1)	7.0777 (4)
c/Å	3.9858 (2)	-	3.9783 (2)
α/deg	90	90	90
β/deg	-	-	-
γ/deg	120	-	-
Rwp	8.38	10.24	9.01
Gof	2.81	3.16	3.05
Composition/%	100	100	95.1 (+UB ₂)
Shortest U-U spacing Å	3.12	5.28	3.65

Despite the numerous studies of UB_{2±x}, UB_{4-x}, and UB_{12-x}, and the known range of non-stoichiometry, only few studies in the open literature link the effects of lattice parameters with composition. According to the previously stated Brewer definition, each UB_x exists over a range of stoichiometry. Nevertheless, it is clear from the literature that this lower boron content can directly be seen on the lattice parameters. The lattice parameters of the present study and those found in the open literature are given in Table 3. With its lower boron content, UB_{1.78}, possesses a lower a lattice parameter compared to UB_{1.79} (a = 3.1309 (5) Å, c = 3.9837 (5) Å) [17] and UB_{2.02} (a = 3.133 (1) Å, c = 3.9860 (1) Å). Despite the close stoichiometry between our sample and UB_{1.79}, the lattice parameters are slightly different, probably due to the impurities indicated by the previous authors (U and UB₄), which might modify the overall UB_{2-x} stoichiometry. For UB_{11.78}, the only reported UB_{12-x} stoichiometry in the open literature was UB₁₆ [18] (a = 7.475 Å), which indeed differs from ours. Finally, the UB_{3.98} (a = 7.0764 Å, c = 3.9811 Å) sample synthesized by Menovsky et al. [19] is the closest to UB₄ stoichiometry that we could find in the open literature. Compared to our present UB_{4-x} sample, the lattice parameters a decrease whereas c slightly increase.

Table 3. Lattice parameters of UB_{2±x}, UB_{4-x}, and UB_{12-x} reported in the present study (P.S.) and from the literature.

UB ₂		UB _x		UB ₄		UB _x		UB ₁₂		UB _x	
a (Å)	c (Å)	Ref.	X	a (Å)	c (Å)	Ref.	X	a (Å)	Ref.	X	
1	3.1225	3.9858	P.S.	1.78 (2)	7.0573	3.9008	[20]	7.468	[21] ¹		
2	3.1302 (3)	3.9878 (3)	[20]		7.0777	3.9783	P.S.	3.613 (41)	7.4715	P.S.	11.19 (13)
3	3.132	3.986	[7] ²		7.080	3.978	[21]		7.473	[22]	
4	3.139	3.994	[23]		7.079 (1)	3.983 (1)	[24]		7.474	[10] ³ , [11]	
5	3.133	3.986	[25]	2.02	7.0795	3.9794	[26] ⁴		7.475	[18,20]	16
6	3.084	4.020	[27]		7.0764	3.9811	[19]	3.98			
7	3.136 (6)	3.988 (8)			7.08	3.98	[9]				
8	3.1309 (5)	3.9837 (5)	[17]	1.79 (6)							

¹ +UB₁₂ diamagnetic; ² +impurities; ³ +UB₄; ⁴ +amorphous carbon.

In addition to the room temperature XRD, we also performed single crystal measurements at low temperatures. Figure 2 shows the variation of lattice parameters and volume with temperature. For a direct and relative comparison, we used 0.2 Å length scale for all the lattice parameters. At room temperature, we observed a difference between the values recorded on the powders and the four circle. For UB_{2-x} a = 3.1444 (62) c = 4.0035 (79); for UB_{12-x} a = 7.4789 (15); and for UB_{4-x} a = 7.1036 (40), c = 4.009 (42). This difference can be explained by a higher experimental uncertainty of the four-circle at room temperature and to the different sampling used between both measurements. Indeed, the stoichiometry in the powder sample average out all the lattice parameters values. We could fit both lattice

parameters and volumes using linear equations given in Table 4. For UB_{2-x} (Figure 2a) and UB_{4-x} (Figure 2c), the lattice parameters are decreasing with decreasing temperatures. For UB_{12-x} (Figure 2b), the lattice parameters are slightly increasing with decreasing temperatures. We calculated the lattice thermal expansion using the formula in ref. [28] (Figure 2b, bottom) but did not observe a negative minimum such as in LuB_{12} or YB_{12} [29]. This steep negative thermal expansion might exist at lower temperatures, but such analysis was not possible with our four-circle equipment.

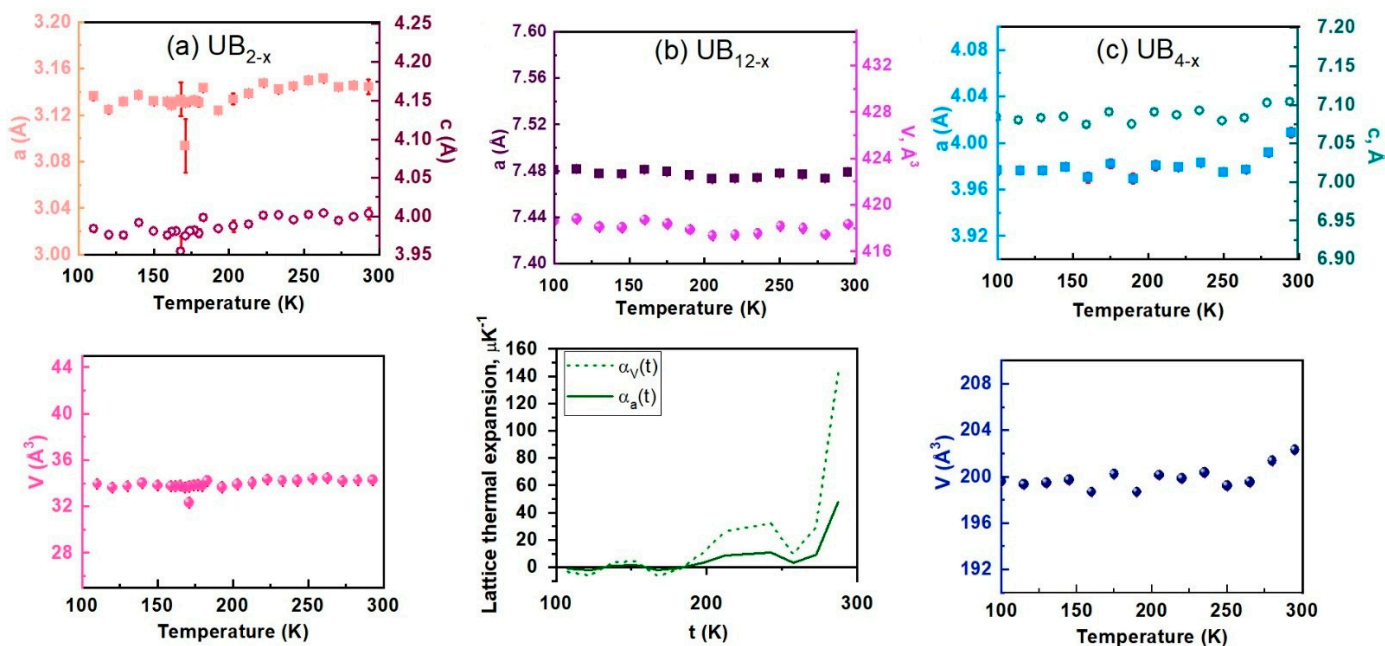


Figure 2. Lattice parameters and volume expansion of (a) UB_{2-x} , (b) UB_{4-x} , and (c) UB_{12-x} in the temperature range (100–300) K. For UB_{12-x} , the lattice thermal expansion is also presented.

Table 4. Linear fit of the lattice parameters and volumes with the temperature.

Compounds	UB_{2-x}			UB_{4-x}			UB_{12-x}		
Parameter	a	c	V	a	c	V	a	V	
slope	$(1.0648 \pm 1.8127) \cdot 10^{-5}$	$(1.3111 \pm 0.2538) \cdot 10^{-4}$	0.0043 ± 0.0016	$(7.3755 \pm 2.9109) \cdot 10^{-5}$	$(7.1264 \pm 3.0399) \cdot 10^{-5}$	$(8.83 \pm 3.64) \cdot 10^{-3}$	$(-2.3 \pm 1) \cdot 10^{-5}$	$-4.82 \pm 1.53) \cdot 10^{-3}$	
intercept	3.1172 ± 0.0040	3.9641 ± 0.0057	33.0589 ± 0.2981	7.072 ± 0.006	3.9662 ± 0.0061	198.087 ± 0.7423	7.4814 ± 0.0021	419.0161 ± 0.2836	
Adj-R ²	0.5727	0.5067	0.2054	0.2942	0.2570	0.2725	0.2206	0.4076	

3.2. Magnetic Susceptibility

The temperature dependencies of the magnetic susceptibilities (χ) for the three uranium borides are presented in Figure 3. For $UB_{1.78}$, the sample presents a temperature independent magnetic susceptibility—Pauli paramagnetism—with $\chi_{UB_{1.78}} \approx 0.55$ m emu/mol. A small anomaly is observed at approximately 50 K and is attributed to oxygen present in the SQUID magnetometer capsule. The field-dependence of the magnetization (inset of Figure 3a) is linear, and the curves measured at different temperatures have the same slope, as expected from a Pauli paramagnet. In previous work, Chachkhiani et al. [30] ($\chi_{UB_{2-x}} = 0.56$ m emu/mol) and Yamamoto et al. [7,8] ($\chi_{UB_{2-x}} = 0.55$ m emu/mol, Table 3 number 3) both described similar intrinsic Pauli magnetism on UB_2 powders and single crystals, respectively. The second authors additionally described an increase in the susceptibility below 80 K ascribed to unknown magnetic impurities. This, together with other experimental results (e.g., De Haas–Van Alphen) and band calculations indicate that 5f electrons in UB_2 have a very itinerant character. The close agreement of our data with UB_2 single crystals shows that the physical properties are still retained in our bulk sample, despite the slightly lower boron content compared to UB_2 ($UB_{1.78}$). In addition, and contrary to Yamamoto et al., our sample does not present any visible magnetic impurity besides

the slight oxygen signal (coming from the equipment). Our results therefore confirm the temperature-independent character of the magnetic susceptibility of UB_2 below 80 K and down to 2 K.

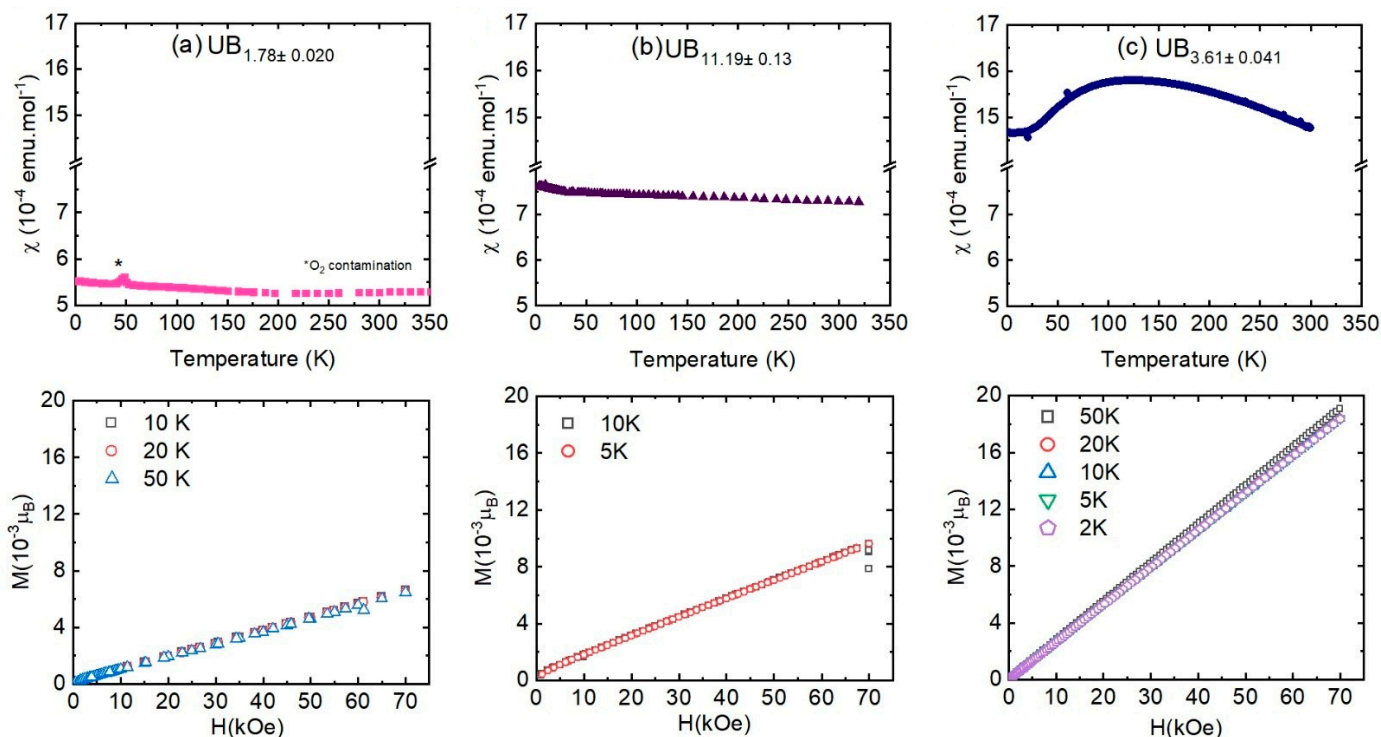


Figure 3. Magnetic susceptibility curves against temperature (top) and magnetization against field (bottom) for (a) $UB_{1.78}$, (b) $UB_{11.19}$, and (c) $UB_{3.61}$.

We find the magnetic susceptibility of $UB_{11.19}$ to be nearly temperature independent, indicative of Pauli paramagnetism with $\chi_{UB_{11.19}} = 0.76$ memu/mol. The magnetization varies linearly versus applied magnetic field (inset of Figure 3b) and the curves at different temperatures superpose, as expected for a Pauli paramagnet. In the literature, Kasaya [31] and Tróc [32] reported $\chi_{UB_{12-x}} = 0.75$ memu/mol, similar to the present study. A second and more recent study performed jointly by Tróc et al. [10] and Samsel-Czekala et al. [11] reported a slightly smaller value with $\chi_{UB_{12-x}} \approx 0.65$ m emu/mol attributed to the presence of UB_4 and paramagnetic impurities leading to a sharp increase at low temperature. Consistently, their magnetization vs. applied magnetic field curves are non-linear at 2 and 6 K and their slopes changes up to 14 K. We did not notice such behaviours (Figure 3b). Blum and Bertaut [21] published the smallest UB_{12-x} lattice parameter and reported a diamagnetic behaviour; this can be safely ruled out.

The magnetic susceptibility at room and low temperatures amounts to $\chi_{UB_{3.61}}^{\min} \sim 1.46$ memu/mol and displays a broad maximum ($\chi_{UB_{3.61}}^{\max} \sim 1.58$ memu/mol) in the temperature region 110–140 K. The magnetization varies linearly versus applied magnetic field (Figure 3c, lower), with a much larger slope than in $UB_{1.78}$ and $UB_{11.19}$, consistent with the behaviour of their magnetic susceptibilities. The curves superpose from 2 to 20 K and the slope slightly changes at 50 K. From the magnetic susceptibilities published on UB_{4-x} , most authors have different results due to the range of non-stoichiometry (or impurities). Our results compare well with the magnetic susceptibility of Galatanu et al. [9] measured along the [100] direction of a single crystal, suggesting the possible occurrence of preferential orientation along the a-axis in our bulk sample. Their lattice parameters are similar to ours (Table 3, number 7). Additionally, our magnetization vs. magnetic field curve complements their data, as they recorded the curves for 2 K, 100 K, and higher temperatures. We can confirm the increase in linear slope at 50 K, which also occurs at

100 K. Our data also agree relatively well with those of Menovski et al. [19] on $UB_{3.98}$ single crystals. They nonetheless reported a sharper maximum at 114 K with no magnetic ordering. The agreement between these previous studies and the present findings confirms the occurrence of the cross-over maximum and the robustness of this feature despite 5 wt% UB_{2-x} as a second phase. Overall, UB_{4-x} has delocalized electrons at low temperatures with a cross-over at approximately 110 K to localized 5f electrons at high temperatures [9] (Curie–Weiss law with an effective moment $\mu_{\text{eff}} \approx 3.3\mu_B/U$). In contrast, the polycrystalline sample measured by Wallash et al. [33] does not exhibit preferential orientation and no maximum, and Chachkhiani et al. [30] attributed the peak in the susceptibility in terms of antiferromagnetic ordering.

To sum up, we showed the influence of the boron content and additional boride phases on the magnetic susceptibilities.

3.3. Nuclear Magnetic Resonance

The NMR spectra recorded in static and magic angle spinning (MAS) (40 kHz) conditions, and their corresponding fits are presented in Figure 4.

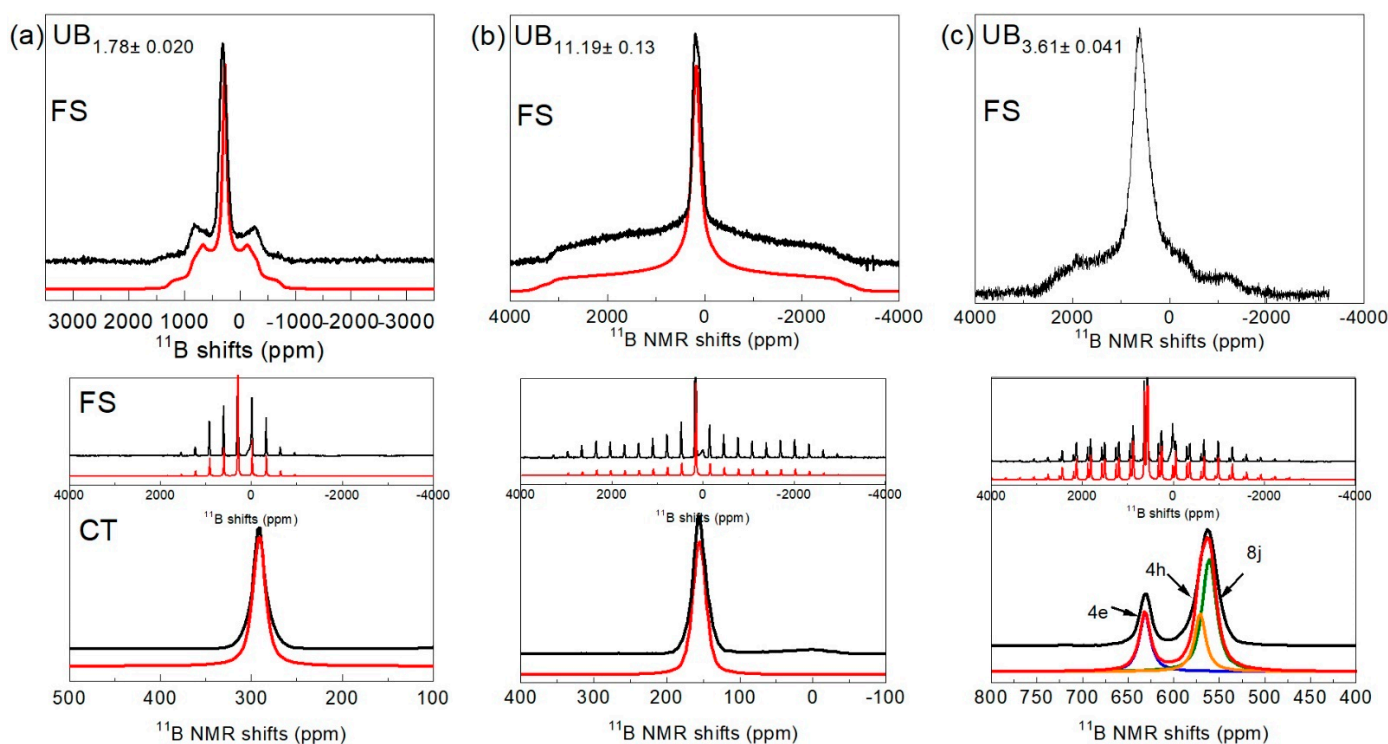


Figure 4. ^{11}B static (upper) and MAS-NMR spectra (lower) for (a) $UB_{1.78}$, (b) $UB_{11.19}$, and (c) $UB_{3.61}$ showing the central transition (CT) and the Full spectra (FS). The fits are the red lines.

The static ^{11}B NMR spectra are defined by a typical powder pattern for a nuclear spin $I = 3/2$ in the presence of first order quadrupolar interaction. The ^{11}B MAS-NMR spectra are characterized by central peaks and their associated spinning sidebands due to the satellite transitions ($\pm 3/2 \leftrightarrow \pm 1/2$). All the NMR parameters are given in Table 5. The shifts have, as expected, values largely above the common range expected for diamagnetic boron (~ -3 to ~ 20 ppm) [34,35]. As expected from their crystal structure, $UB_{1.78}$ and $UB_{11.19}$ present one NMR peak in both static and MAS conditions. For $UB_{3.78}$, in addition to the peaks corresponding to the central transition, we could detect the signal of UB_{2-x} (3%), as shown by XRD. The crystal structure of UB_4 possesses three different crystal B sites. The static NMR spectrum presents only one unresolved peak similar to the work published by Fukushima et al. [36]. By spinning the sample, the spectral resolution increases and we can identify two main peaks at 631.5 and ~ 563 ppm. Nevertheless, the deconvolution

of the peak at 563 ppm was not possible using only one contribution due to a strong asymmetry. This peak was therefore fitted with two contributions at 571 and 561 ppm. Although the ^{11}B peak at 561 ppm is clearly attributed to B3 (8j) due to its intensity being twice the one of the two other peaks, it is less straightforward for the B1 (4e) and B2 (4h) sites. To differentiate them, we proceeded as suggested by Creyghton et al. [37] and used the quadrupolar parameters. In fact, these authors stated that due to their similar crystal structures and the small c/a ratio [38,39] differences between NdB_4 (0.568 [40]) and LaB_4 (0.571 [41]), the quadrupolar parameters must follow the same sequence. In our case, the c/a ratio of UB_4 (0.5624 [42]) is close enough to that of YB_4 (0.5654 [43]) to apply the same principle. Therefore, as the peak at 631 ppm has the smallest C_Q it can be attributed to the B1 (4e) site similarly to YB_4 [44], and the one at 571 ppm to B2 (4h). We want to further underline that for NdB_4 , there is an inversion between B2 and B3 in the crystal structures reported in the literature. In fact, the P4/mbm structures B2 has the Wyckoff Symbol 4g and B3 8i.

Table 5. ^{11}B NMR parameters Knight shifts ($^{11}\text{B}K$), quadrupolar coupling constant (C_Q), and asymmetry parameter (η_Q) of the uranium borides compounds.

Compound Name		$^{11}\text{B}K$ (ppm)	FWHM (ppm)	C_Q (kHz)	η_Q
$\text{UB}_{1.78}$	B1	303.6 ^a	–	298.4 ^a	0.2 ^a
		291.6 ^b	16.8		
$\text{UB}_{3.61}$	B1 (4e)	631.5	14.2	390 ^b	1 ^b
	B2 (4h)	571	14	560 ^b	0.9 ^b
	B3 (8j)	561.2	18	560 ^b	0.8 ^b
$\text{UB}_{11.19}$	B1	164.3 ^a	–	774.7 ^a	0.8 ^a
		155.4 ^b	21.2		
AlB_2 [18]		-10 ± 5		1.08	0
ZrB_2 [18]		–29		–	–
MgB_2 [45]		40 ± 10		1.67	0
	4e	34.7		1.14	
YB_4 [44,46]	4h	12.6		1.46	
	8j	5.4		1.04	
	4e	42		0.69	0
LaB_4 [37,47]	4h	47		1.1	0
	8j	18		0.8	0.5
	4e	3300 (± 10 –15%)		0.84	0
NdB_4 [37,46]	4h	2300 (± 10 –15%)		0.89	0.5
	8j	2600 (± 10 –15%)		1.244	0
ZrB_{12} [18]		10		1.083	0.98
YB_{12} [18]		25		1.08	0.93

The parameters were obtained ^a static and ^b 40 kHz.

We compared in Table 5 the NMR parameters of the UB_x with their isostructural MB_x : $\text{UB}_{1.78}$ with the Pauli paramagnets AlB_2 and ZrB_2 [18] and the superconductor ($T_c = 39$ K [46]) MgB_2 [48]; $\text{UB}_{3.61}$ with the diamagnetic YB_4 [44,46], the diamagnetic at room temperature [49] LaB_4 [37,47], and the antiferromagnetic (at 7 K) [49] NdB_4 [37,46]; finally, $\text{UB}_{11.19}$ with the metallic ZrB_{12} [18] and YB_{12} [18,50]. All the UB_x possess higher shifts than their MB_x counterpart, most probably linked to the hybridization between the U5f and B2p orbitals. All the C_Q values in the UB_x are smaller compared to their MB_x counterparts, but they have similar asymmetry parameters.

To conclude the NMR observations, we found that despite the sample's lower boron content, only the expected number of ^{11}B NMR peaks were detected. Due to the quadrupolar nucleus, the disorder might not be excluded as it can be expressed in the linewidth (Table 5). This result contrasts with the uranium carbides where additional ^{13}C (spin 1/2) signals were detected as a fingerprint of the non-stoichiometry [51].

3.4. Infrared Spectroscopy

The FTIR spectra of the uranium borides are presented in Figure 5 and the main peaks positions are given in Table 6. According to the factor group analysis, the expected IR active modes for the idealized crystal structures at the Brillouin zone centre are ($A_{2u} + E_{1u}$) for UB_2 , ($3A_{2u} + 9E_u$) for UB_4 , and ($3T_{1u}$) for UB_{12} [52]. All these optical phonons represent vibrations of the boron sublattice only.

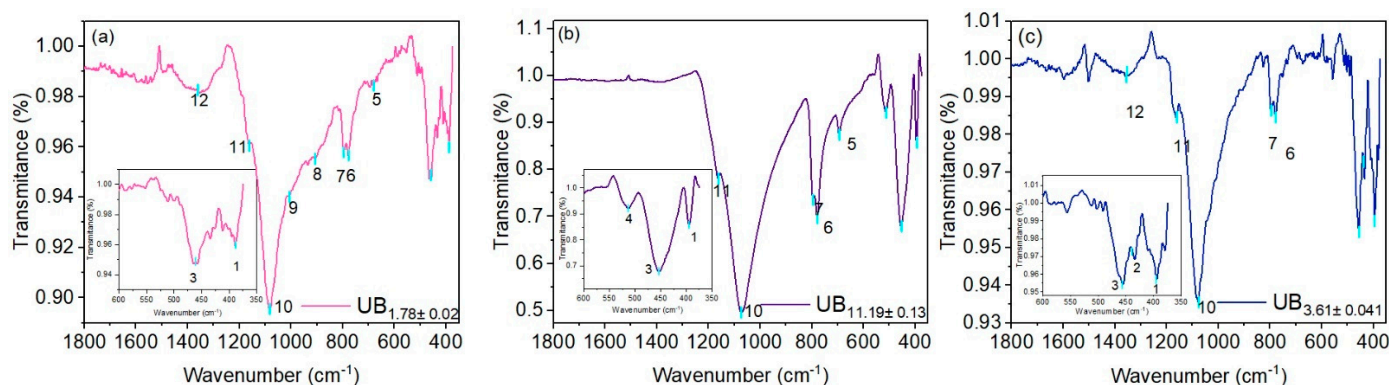


Figure 5. Infrared spectra of (a) $UB_{1.78}$, (b) $UB_{11.19}$, and (c) $UB_{3.61}$. The blue spots correspond to the main bands.

Table 6. Values of the main FTIR bands (cm^{-1}). For UB_4 extracted from the literature, dd corresponds to doubly degenerate mode. For ZrB_{12} , the numbers in bold are the active IR bands defined by the authors, with a star are the active IR bands defined by the Bilbao crystallographic database, and in italic are the active Raman bands. The frequencies presented for UB_{12} from ref. [52] correspond to Raman modes.

	Present Study			Literature							
	$UB_{1.78}$	$UB_{3.61}$	$UB_{11.19}$	UB_4 [53]	ZrB_{12} [54]	UB_{12} [54]	UB_2 [23]		MgB_2		
1	388	395	394	dd	315	T_{1u}	0	E_{1u}	393	E_{1u} [55]	322/327 #
2	–	438			350	T_{2u}	198.3	A_{2u}	481	A_{2u}	394/405 #
3	463	456	453	dd	358	T_{1u}	206.6	E_{2g}	598		
4	500	–	513		399	T_{1g}	478.1	B_{1g}	794	E_{1u} [56]	335 #
5	680	–	693	dd	608	E_u	498.2			A_{2u}	401 #
6	778	778	778	dd	722	A_{2g}	518.8				
7	796	796	795	dd	796	E_g	639.9	621		E_{1u} [57]	320 #
8	907	–	–			T_{2u}	661.6			A_{2u}	390 #
9	1002	–	–			T_{1g}	693.6				
10	1083	1077	1073			T_{1u}	729.1				
11	1162	1162	1162			T_{2g}	801.7	748			
12	1360	1354	–			T_{1u}	854.2				
						A_{2u}	982.4				
						E_g	1010.1	971			
						A_{1g}	1078.3	1039			
						T_{2g}	1084.4	1088			

calculated values for MgB_2 .

To further analyze the data, we considered the calculated phonon spectra of UB_2 and extracted the frequency of the optical modes (Table 6) [23]. Jossou et al. [23] calculated the doubly degenerate E_{1u} mode (B and U planes sliding along x, y) and the A_{2u} (B and U planes moving against each other) modes at 393 and 481 cm^{-1} , respectively. These results are similar to the values obtained for the modes at 388 cm^{-1} and 465 cm^{-1} . The difference might be due to different lattice parameters (Table 3 number 4) and characteristics of the different stoichiometries. The additional peaks can be attributed to a loss of the local

symmetry and the Raman forbidden modes can be relaxed, which indicates a reduction of symmetry [58]. The high frequency modes manifested as broad intense bands centred at 1002 cm^{-1} , 1162 cm^{-1} , and 1360 cm^{-1} might be a combination of multiple modes located near the 463 cm^{-1} mode [58,59]. It is worth comparing $\text{UB}_{1.78}$ with MgB_2 as, due to its supraconductivity [48], its optical properties have been extensively studied. The values of the IR active phonon modes in UB_2 are relatively similar to the one reported experimentally [58,59] and theoretically [55–57] for MgB_2 (Table 6).

We compared UB_{12} with the modes previously calculated for ZrB_{12} [54] (Table 6), as they should qualitatively give similar FTIR spectra. We believe that the correct IR active modes are the 3T_{1u} . The first T_{1u} mode at 206.6 cm^{-1} is not visible in our IR spectrum, but the two other T_{1u} modes, at 729.1 cm^{-1} and 854.2 cm^{-1} , might correspond to the modes at 778 cm^{-1} and 795 cm^{-1} (Table 6). Similar to the FTIR from the MB_{12} series [54], silent modes appear on our spectrum as the local distortions can break the local symmetry, and the selection rules are then lifted. This peculiar behaviour seems typical for metal borides. Thus, the modes at 1073 cm^{-1} and 1162 cm^{-1} can be attributed to the Raman forbidden modes E_g/A_{1g} and T_{2g} , respectively. It must be noted that based on the present data, an unambiguous attribution is not possible.

To our knowledge, Lopez-Bezanilla [53] calculated the only phonon spectra on UB_4 . The author reported 12 out of the 31 optical modes and did not discuss their nature. A recent work by Surucu et al. [60] on LaB_4 reported all the optical modes and determined a phonon range of $264\text{--}1072\text{ cm}^{-1}$, which therefore implies that a larger range will be expected for UB_4 . For this compound, we therefore cannot go much further in the spectral analysis but noted that the observed optical modes in the IR spectrum represent the whole frequency range of the calculated phonon spectrum ($315\text{--}796\text{ cm}^{-1}$) (Table 6). The mode at 1007 cm^{-1} has a similar value as the forbidden Raman mode B_{1g} predicted for $V = 199\text{ \AA}$ at $\sim 1000\text{ cm}^{-1}$ (see ref. [61]). Finally, it is worth noting that the characteristic bands observed for UB_{2-x} (388 cm^{-1} and 465 cm^{-1}) are also detected on the UB_{4-x} spectrum, in agreement with the XRD and NMR results.

From this FTIR study and the detection of additional modes, we can observe the expected structural defects that are lifting the selection rules stated from the idealized structure.

4. Conclusions

In the present study, we revisited the binary system U-B via its three samples with nominal composition UB_2 , UB_4 , and UB_{12} . Within the uncertainty of the chemical and oxygen content analyses, purity of the samples and their U/B ratio were well characterised. The data supported the view proposed by Brewer et al. about their range of non-stoichiometry described in the literature. The composition definition of each sample is essential to link the properties and likely to explain the difference observed in literature with the lattice parameters. The X-ray diffraction patterns show the purity of the samples for UB_{12-x} and UB_{2-x} , whereas a minor content of UB_{2-x} could be found for UB_{4-x} . Furthermore, their high intensity peaks show long range order on the arc melted as cast samples. In the low range of temperature $100\text{ K--}300\text{ K}$, XRD did not indicate a clear negative thermal expansion in UB_{12-x} as its counterpart LuB_{12} . Although influenced by the boron content, the magnetic properties agree to a certain extent with the literature data. The NMR analyses show the influence of the hybridisation between U5f and B2p. The FTIR spectra confirm the local composition effects and the disorder. We believe that our study will trigger the curiosity of theoreticians to draw models with clear references.

Supplementary Materials: The following supporting information can be downloaded at: <https://www.mdpi.com/article/10.3390/min12010029/s1>, Figure S1: Full X-ray diffraction patterns and corresponding fits for the uranium borides.

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NMR experiments. All authors contributed equally to writing the paper. All authors have read and agreed to the published version of the manuscript.

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References

1. Lupinetti, A.J.; Fife, J.L.; Garcia, E.; Dorhout, P.K.; Abney, K.D. Low-temperature synthesis of uranium tetraboride by solid-state metathesis reactions. *Inorg. Chem.* **2002**, *41*, 2316–2318. [\[CrossRef\]](#)
2. Gossé, S.; Alpettaz, T.; Géneau, C.; Allegri, P. High temperature thermochemistry of (U-Pu)O₂ MOX fuel with B₄C absorber—Application to severe accidents in SFR. In Proceedings of the International Conference on Fast Reactors and Related Fuel Cycles: Safe Technologies and Sustainable Scenarios, Paris, France, 4–7 March 2013.
3. Kardoulaki, E.; White, J.T.; Byler, D.D.; Frazer, D.M.; Shivprasad, A.P.; Saleh, T.A.; Gong, B.; Yao, T.; Lian, J.; McClellan, K.J. Thermophysical and mechanical property assessment of UB₂ and UB₄ sintered via spark plasma sintering. *J. Alloys Compd.* **2020**, *818*, 153216. [\[CrossRef\]](#)
4. Howlett, B.W. A Note on the Uranium-Boron Alloy System. *J. Inst. Met.* **1959**, *88*, 91–92.
5. Chevalier, P.Y.; Fischer, E. Thermodynamic modelling of the C-U and B-U binary systems. *J. Nucl. Mater.* **2001**, *288*, 100–129. [\[CrossRef\]](#)
6. Brewer, L.; Sawyer, D.L.; Templeton, D.H.; Dauben, C.H. A study of the refractory borides. *J. Am. Ceram. Soc.* **1951**, *34*, 173–179. [\[CrossRef\]](#)
7. Yamamoto, E.; Honma, T.; Haga, Y.; Inada, Y.; Aoki, D.; Suzuki, N.; Settai, R.; Sugawara, H.; Sato, H.; Yoshichika, Ō. Electrical and thermal properties of UB₂. *J. Phys. Soc. Jpn.* **1999**, *68*, 972. [\[CrossRef\]](#)
8. Yamamoto, E.; Honma, T.; Haga, Y.; Inada, Y.; Aoki, D.; Tokiwa, Y.; Suzuki, N.; Miyake, K.; Yoshichika, Ō. Magnetoresistance and de Haas Van Alphen effect in UB₂. *J. Phys. Soc. Jpn.* **1999**, *68*, 3347–3351. [\[CrossRef\]](#)
9. Galatanu, A.; Yamamoto, E.; Hagab, Y.; Onuki, Y. Magnetic behaviour of UB₄ at high temperatures. *Phys. B Condens. Matter.* **2006**, *378*, 999–1000. [\[CrossRef\]](#)
10. Tróc, R.; Wawryk, R.; Pikul, A.; Shitsevalova, N. Physical properties of cage-like compound UB₁₂. *Philos. Mag.* **2015**, *95*, 2343–2363. [\[CrossRef\]](#)
11. Samsel-Czekala, M.; Talik, E.; Tróc, R.; Shitsevalova, N. Electronic structure of cage-like compound UB₁₂—Theory and XPS experiment. *J. Alloys Compd.* **2014**, *615*, 446–450. [\[CrossRef\]](#)
12. Chipaux, R. Contribution à l'étude de la cristallographie et des propriétés électroniques des borures dactinides. Ph.D. Thesis, Université d'Aix-Marseille II, Marseille, France, 1988.
13. Massiot, D.; Farnan, I.; Gautier, N.; Trumeau, D.; Trokner, A.; Coutures, J.P. ⁷¹Ga and ⁶⁹Ga nuclear magnetic resonance study of β-Ga₂O₃: Resolution of four—and six-fold coordinated Ga sites in static conditions. *Solid State Nucl. Magn. Reson.* **1995**, *4*, 241–248. [\[CrossRef\]](#)
14. Martel, L.; Griveau, J.-C.; Eloirdi, R.; Selfslag, C.; Colineau, E.; Caciuffo, R. Ferromagnetic ordering in NpAl₂: Magnetic susceptibility and ²⁷Al nuclear magnetic resonance. *J. Magn. Magn. Mater.* **2015**, *387*, 72–76. [\[CrossRef\]](#)
15. Massiot, D.; Fayon, F.; Capron, M.; King, L.; Le Calvé, S.; Alonso, B.; Durand, J.O.; Bujoli, B.; Gan, Z.; Hoatson, G. Modelling one—and two-dimensional solid-state NMR spectra. *Magn. Reson. Chem.* **2002**, *40*, 70–76. [\[CrossRef\]](#)
16. Angeli, F.; Charpentier, T.; De Ligny, D.; Cailleteau, C. Boron speciation in soda-lime borosilicate glasses containing zirconium. *J. Am. Ceram. Soc.* **2010**, *93*, 2693–2704. [\[CrossRef\]](#)
17. Flotow, H.E.; Osborne, D.W.; O'Hare, P.A.G.; Settle, J.L.; Mrazek, F.C.; Hubbard, W.N. Uranium Diboride: Preparation, Enthalpy of Formation at 298.15°K, Heat Capacity from 1° to 350°K, and Some Derived Thermodynamic Properties. *J. Chem. Phys.* **1969**, *51*, 583–592. [\[CrossRef\]](#)
18. Akopov, G.; Mak, W.H.; Koumoulis, D.; Yin, H.; Owens-Baird, B.; Yeung, M.T.; Muni, M.H.; Lee, S.; Roh, I.; Sobell, Z.C.; et al. Synthesis and characterization of single-phase metal dodecaboride solid solutions: Zr_{1-x}Y_xB₁₂ and Zr_{1-x}U_xB₁₂. *J. Am. Chem. Soc.* **2019**, *141*, 9047–9062. [\[CrossRef\]](#)
19. Menovsky, A.; Franse, J.J.M.; Klaasse, J.C.P. The crystal growth of uranium tetraboride UB₄ from the melt. *J. Cryst. Growth* **1984**, *70*, 519–522. [\[CrossRef\]](#)
20. Dancausse, J.-P.; Gering, E.; Heathman, S.; Benedict, U.; Gerward, L.; Staun Olsen, S.; Hulliger, F. Compression study of uranium borides UB₂, UB₄ and UB₁₂ by synchrotron X-ray diffraction. *J. Alloys Compd.* **1992**, *189*, 205–208. [\[CrossRef\]](#)
21. Blum, P.; Bertaux, F. Contribution à l'étude des borures à teneur élevée en bore. *Acta Cryst.* **1954**, *7*, 81–86. [\[CrossRef\]](#)
22. Matkovich, V.I.; Economy, J.; Giese jr., R.F. Barrett, R. The structure of metallic dodecaborides. *Acta Cryst.* **1965**, *19*, 1056–1058. [\[CrossRef\]](#)
23. Jossou, E.; Oladimeji, D.; Malakkal, L.; Middleburgh, S.; Szpunar, B.; Szpunar, J. First-principles study of defects and fission product behavior in uranium diboride. *J. Nucl. Mater.* **2017**, *494*, 147–156. [\[CrossRef\]](#)

24. Chipaux, R.; Cecilia, G.; Beauvy, M.; Tróc, R. Capacité thermique a haute temperature de UBe_{13} , $ThBe_{13}$ et UB_4 . *J. Less Common Met.* **1986**, *121*, 347–351. [[CrossRef](#)]
25. Turner, J.; Martini, F.; Buckley, J.; Phillips, G.; Middleburgh, S.C.; Abram, T.J. Synthesis of candidate advanced technology fuel: Uranium diboride (UB_2) via carbo/borothermic reduction of UO_2 . *J. Nucl. Mater.* **2020**, *540*, 152388. [[CrossRef](#)]
26. Guo, H.; Wang, J.; Chen, D.; Tian, W.; Cao, S.; Chen, D.; Tan, C.; Deng, Q.; Qin, Z. Boro/carbothermal reduction synthesis of uranium tetraboride and its oxidation behavior in dry air. *J. Am. Ceram. Soc.* **2019**, *102*, 1049–1056. [[CrossRef](#)]
27. Evitts, L.J.; Middleburgh, S.C.; Kardoulaki, E.; Ipatova, I.; Rushton, M.J.D.; Lee, W.E. Influence of boron isotope ratio on the thermal conductivity of uranium diboride (UB_2) and zirconium diboride (ZrB_2). *J. Nucl. Mater.* **2020**, *528*, 151892. [[CrossRef](#)]
28. Okada, Y.; Tokumaru, Y. Precise determination of lattice parameter and thermal expansion coefficient of silicon between 300 and 1500 K. *J. Appl. Phys.* **1984**, *56*, 314–320. [[CrossRef](#)]
29. Czopnik, A.; Shitsevalova, N.; Pluzhnikov, V.; Krivchikov, A.; Paderno, Y.; Onuki, Y. Low-temperature thermal properties of yttrium and lutetium dodecaborides. *J. Phys. Condens. Matter.* **2005**, *17*, 5971–5985. [[CrossRef](#)]
30. Chachkhiani, Z.B.; Chachkhiani, L.G.; Chechernikov, V.I.; Slovyanskikh, V.K. Magnetic properties of thorium and uranium borides and $U_{1-x}Th_xB_4$ solid solutions. *Sov. Phys. J.* **1982**, *25*, 621–623. [[CrossRef](#)]
31. Kasaya, M.; Iga, F.; Katoh, K.; Takegahara, K.; Kasuya, T. Magnetic and transport properties of uranium borides, UB_{12} , Ulr_3B_2 and UOs_3B_2 . *J. Magn. Mater.* **1990**, *90*, 521–522. [[CrossRef](#)]
32. Troć, R.; Trzebiatowski, W.; Piprek, K. Magnetic properties of uranium borides and of uranium beryllide UBe_{13} Bull. *Acad. Polon. Sci. Ser. Sci. Chim.* **1971**, *19*, 427.
33. Wallash, A.; Crow, J.; Fisk, Z. Susceptibility, magnetization, and specific heat in the paramagnetic and ferromagnetic regions of $(Y_{1-x}U_x)B_4$. *J. Appl. Phys.* **1985**, *57*, 3143–3145. [[CrossRef](#)]
34. Mackenzie, K.J.D.; Smith, M.E. *Multinuclear Solid-State NMR of Inorganic Materials*; Cahn, R., Ed.; Elsevier: Amsterdam, The Netherlands, 2002.
35. Angel Wong, Y.-T.; Bryce, D.L. Recent Advances in ^{11}B Solid-State Nuclear Magnetic Resonance Spectroscopy of Crystalline Solids. *Annu. Rep. NMR Spectrosc.* **2018**, *93*, 213–279.
36. Fukushima, E.; Strubeing, V.O.; Hill, H.H. Induced Paramagnetism in Uranium Alloys: A Boron NMR Study of Alloys Formed between UB_4 and YB_4 . *J. Phys. Soc. Jpn.* **1975**, *39*, 921–926. [[CrossRef](#)]
37. Creighton, J.H.N.; Locher, P.R.; Buschow, K.H.J. Nuclear magnetic resonance of ^{11}B at the three boron sites in rare-earth tetraborides. *Phys. Rev. B* **1973**, *7*, 4829–4843. [[CrossRef](#)]
38. *Crystal Structures*, 2nd ed.; Interscience: New York, NY, USA, 1964; Volume 2.
39. Fisk, Z.; Cooper, A.S.; Schmidt, P.H.; Castellano, R.N. Preparation and lattice parameters of the rare earth tetraborides. *Mater. Res. Bull.* **1972**, *7*, 285–288. [[CrossRef](#)]
40. Roger, J.; Babizhetskyy, V.; Jardin, R.; Halet, J.-F.; Guérin, R. Solid state phase equilibria in the ternary Nd–Si–B system at 1270 K. *J. Alloys Compd.* **2006**, *415*, 73–84. [[CrossRef](#)]
41. Kato, K.; Kawada, I.; Oshima, C.; Kawai, S. Lanthanum tetraboride. *Acta Cryst. B* **1974**, *B30*, 2933–2934. [[CrossRef](#)]
42. Zalkin, A.; Templeton, D.H. The crystal structures of CeB_4 , ThB_4 and UB_4 . *Acta Cryst.* **1953**, *6*, 269–272. [[CrossRef](#)]
43. Giese, R.F., Jr.; Matkovich, V.I.; Economy, J. The crystal structure of YB_4 . *Z. Krist.* **1965**, *122*, 423–432. [[CrossRef](#)]
44. Jäger, B.; Paluch, S.; Wolf, W.; Herzig, P.; Żogał, O.J.; Shitsevalova, N.; Paderno, Y. Characterization of the electronic properties of YB_4 and YB_6 using ^{11}B NMR and first-principles calculations. *J. Alloys Compd.* **2004**, *383*, 232–238. [[CrossRef](#)]
45. Baek, S.H.; Suh, B.J.; Pavarini, E.; Borsa, F.; Barnes, R.G.; Bud'ko, S.L.; Canfield, P.C. NMR spectroscopy of the normal and superconducting states of MgB_2 and comparison to AlB_2 . *Phys. Rev. B* **2002**, *66*, 104510. [[CrossRef](#)]
46. Żogał, O.J.; Florian, P.; Massiot, D.; Paluch, S.; Shitsevalova, N.; Borshchevsky, D.F. ^{11}B and ^{89}Y MAS NMR and magnetic characterization of YB_4 . *Solid State Commun.* **2009**, *149*, 693–696. [[CrossRef](#)]
47. Schmitt, R.; Blaschkowski, B.; Eichele, K.; Meyer, H.-J. Calcium Tetraboride Does It Exist? Synthesis and Properties of a Carbon-Doped Calcium Tetraboride That Is Isotypic with the Known Rare Earth Tetraborides. *Inorg. Chem.* **2006**, *45*, 3067–3073.
48. Nagamatsu, J.; Nakagawa, N.; Muranaka, T.; Zenitani, Y.; Akimitsu, J. Superconductivity at 39 K in magnesium diboride. *Nature* **2001**, *410*, 63–64. [[CrossRef](#)]
49. Buschow, K.H.J.; Creighton, J.H.N. Magnetic Properties of Rare Earth Tetraborides. *J. Chem. Phys.* **1972**, *57*, 3910–3914. [[CrossRef](#)]
50. Fojud, Z.; Herzig, P.; Żogał, O.J.; Pietraszko, A.; Dukhnenko, A.; Jurga, S.; Shitsevalova, N. Electric-field-gradient tensor and boron site-resolved ^{11}B NMR in single-crystalline YB_{12} . *Phys. Rev. B* **2007**, *75*, 184102. [[CrossRef](#)]
51. Carvajal Nuñez, U.; Martel, L.; Prieur, D.; Lopez Honorato, E.; Eloirdi, R.; Farnan, I.; Vitova, T.; Somers, J. Coupling XRD, EXAFS, and ^{13}C NMR to study the effect of the carbon stoichiometry on the local structure of $UC_{1\pm x}$. *Inorg. Chem.* **2013**, *19*, 11669–11676. [[CrossRef](#)]
52. Kroumova, E.; Aroyo, M.I.; Perez Mato, J.M.; Kirov, A.; Capillas, C.; Ivantchev, S.; Wondratschek, H. Bilbao Crystallographic Server: Useful databases and tools for phase transitions studies. In *Phase Transitions 76*; Nos. 1-2; 2003; pp. 155–170. [[CrossRef](#)]
53. Lopez-Bezanilla, A. f-Orbital based Dirac states in a two-dimensional uranium compound. *J. Phys. Mater.* **2020**, *3*, 024002. [[CrossRef](#)]
54. Werheit, H.; Filipov, V.; Shirai, K.; Dekura, H.; Shitsevalova, N.; Schwarz, U.; Armbrüster, M. Raman scattering and isotopic phonon effects in dodecaborides. *J. Phys. Condens. Matter* **2011**, *23*, 065403. [[CrossRef](#)]

55. Bohnen, K.-P.; Heid, R.; Renker, B. Phonon Dispersion and Electron-Phonon Coupling in MgB_2 and AlB_2 . *Phys. Rev. Lett.* **2001**, *86*, 5771–5774. [[CrossRef](#)]
56. Kong, Y.; Dolgov, O.V.; Jepsen, O.; Andersen, O.K. Electron-phonon interaction in the normal and superconducting states of MgB_2 . *Phys. Rev. B* **2001**, *64*, 020501. [[CrossRef](#)]
57. Kortus, J.; Mazin, I.I.; Belashchenko, K.D.; Antropov, V.P.; Boyer, L.L. Superconductivity of metallic boron in MgB_2 . *Phys. Rev. Lett.* **2001**, *86*, 4656–4659. [[CrossRef](#)] [[PubMed](#)]
58. Alarco, J.A.; Chou, A.; Talbot, P.C.; Mackinnon, I.D.R. Phonon Modes of MgB_2 : Super-lattice Structures and Spectral Response. *Phys. Chem. Chem. Phys.* **2014**, *16*, 24443–24456. [[CrossRef](#)] [[PubMed](#)]
59. Sundar, C.S.; Bharathi, A.; Premila, M.; Kalavathi, S.; Reddy, G.L.N.; Sastry, V.S.; Hariharan, Y.; Radhakrishnan, T.S. Infrared absorption in superconducting MgB_2 . *arXiv* **2001**, arXiv:0104354.
60. Surucu, G.; Ozisik, H.; Deligoz, E.; Shein, I.R.; Matovnikov, A.V.; Mitroshenkov, N.V.; Morozov, A.V.; Novikov, V.V. Electronic and thermodynamic properties of lanthanum tetraboride on low-temperature experimental and ab-initio calculation data. *J. Alloys Compd.* **2020**, *862*, 158020. [[CrossRef](#)]
61. Werheit, H.; Filipov, V.; Shitsevalova, N.; Armbrüster, M.; Schwarz, U.; Ievdokimova, A.; Muratov, V.; Gurin, V.N.; Korsukova, M.M. Raman scattering in rare earths tetraborides. *Solid State Sci.* **2014**, *31*, 24–32. [[CrossRef](#)]