

Chromium coated nuclear fuel claddings for enhanced Accident Tolerant LWRs Fuel (DBA-LOCA and slightly beyond DBA-LOCA conditions

J. C. Brachet, M. Le Saux, T. Guilbert, S. Urvoy, E. Rouesne, M. Le Flem, I. Idarraga-Trujillo, F. Schuster, F. Lomello, Alain Billard, et al.

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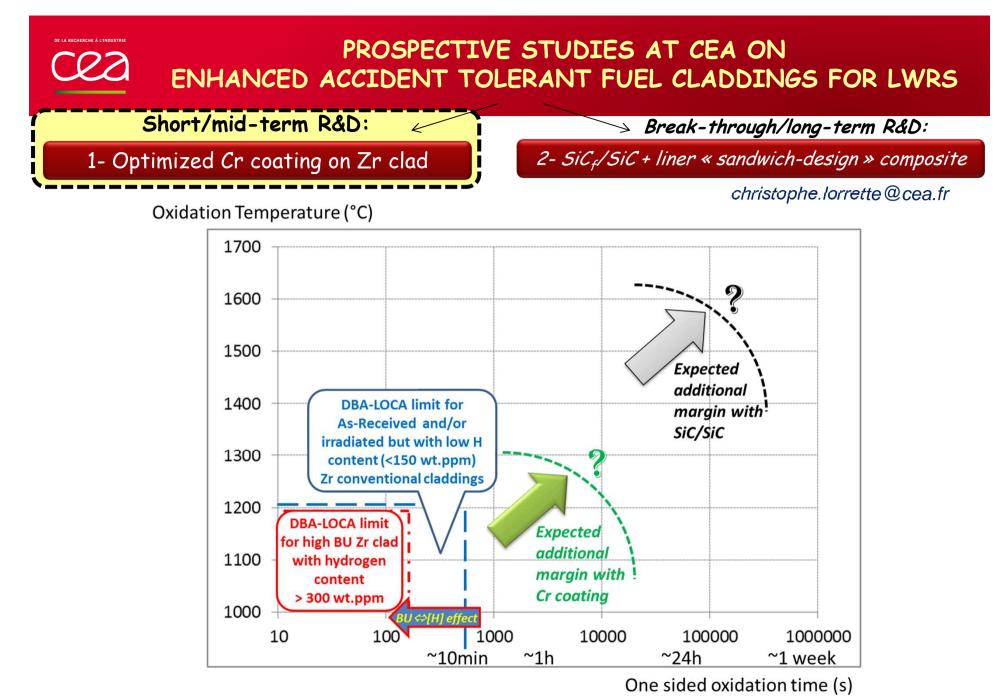
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TARGETS AND MAIN CHALLENGES...

Specific targets for optimized coating on Zr-based claddings:

Decrease the HT oxidation rate with steam water at HT:

- > to slow down/delay the gaseous hydrogen release (mitigation of explosion risk)
- To increase the HT oxidation time/temperature margin vs. quenching and postquenching embrittlement limit (avoid multi-fragmentation...)
- → potential positive effect on Peak Cladding Temperature (PCT) by reducing the contribution of the (HT oxidation) exothermic reaction on the cladding temperature incursion => has to be quantified

Points not addressed by such clad concept:

- \rightarrow Impact on HT mechanical properties => probably low, but has to be quantified
- \rightarrow Margin to melting temperature => Eutectic formation between Zr and Cr at ~1330°C
- Well-known Zr cladding materials and **no significant change of the fuel sub-assembly** geometry nor of the mechanical/thermal properties

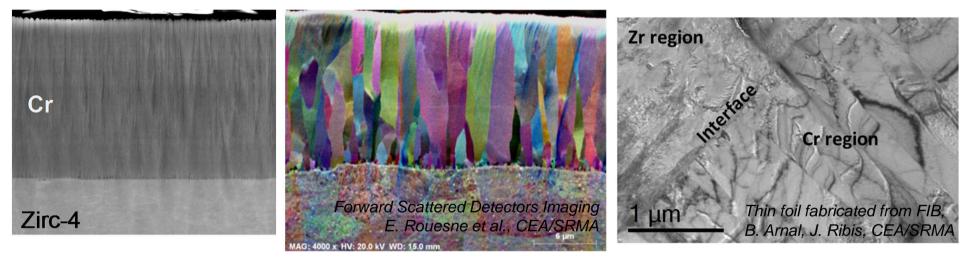
Main challenges:

- Fabricate an homogeneous and HT protective coating with suitable process
- Effects of neutron irradiation, thermal cycling, stress/strain cycles (PCI, fatigue...), more severe corrosive environment... on the coating adhesion and efficiency... ?
- Fabrication issues (longer cladding tubes, production rate, non destructive quality control, weldability of end plugs, etc...)



⇒ Process derived from Physical Vapor Deposition (PVD) ⇒ Typical PVD coating thickness range = $5-20\mu m$



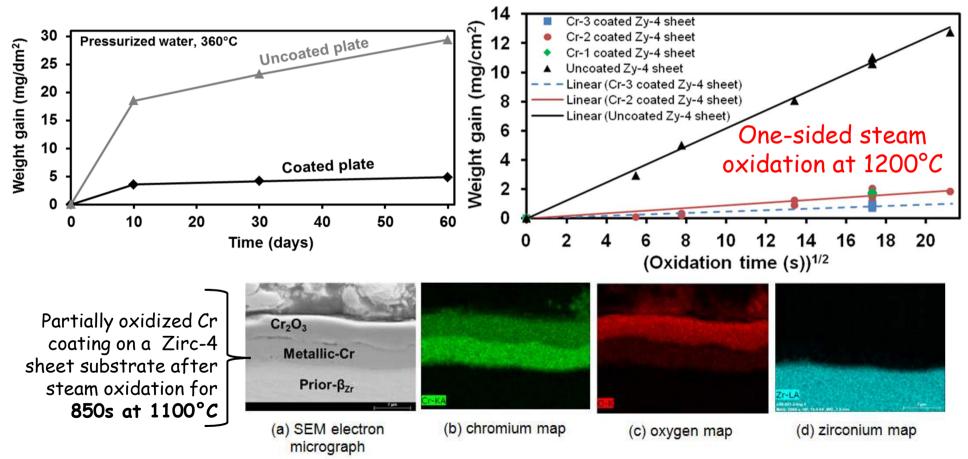


SEM and TEM electron micrographs of the cross-section of as-received chromium coatings on a Zirc-4 substrate

⇒ <u>Dense and protective</u> PVD Cr coatings have been obtained with <u>no indication of damage, i.e., no voids/microcracks</u> both within the bulk Cr coating <u>and</u> at the Zr/Cr interface

Cea BEHAVIOR IN NOMINAL CONDITIONS

- More than 20 coatings (including ceramics, and metals) investigated at CEA on <u>sheets</u> <u>Zirc-4</u> base substrate => among all, <u>optimized Cr-based coatings are promizing</u>
- → <u>Significant decrease in the oxidation kinetics for both in-service (out-of-pile)</u> conditions and at HT (1000-1200°C) typical of DBA-LOCA conditions



Previous work on Cr coated Zirc-4 sheets for beyond DBA-LOCA Cea oxidation times: 1000°C, 15000s => "post-breakaway" Coated: Uncoated: Fractographs: 1000 Coated Zy-4 900 Uncoated Zy-4 800 700 Stress (MPa) 600 500 400 Ref. I. Idarraga et al., proceedings 2013 LWR Fuel Performance 300 Meeting/Topfuel 200 100 **Oxidized** plates 0.4 0.6 0,8 1,4 1,6 1,8 0.2 1.2 Elongation (%) (a) Engineering tensile curves (b) SEM fractographs (Room Temperature)

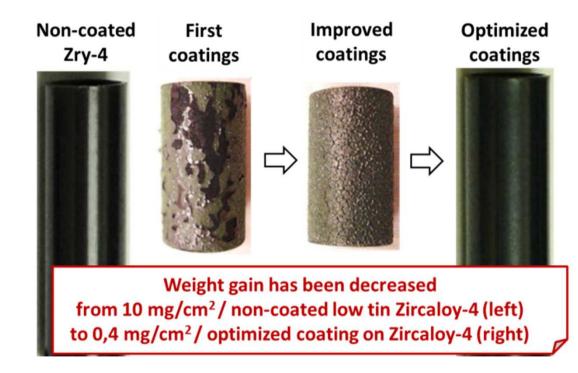
Figure 5 – PQ mechanical behavior of coated and uncoated 1 mm thick sheet (Zircaloy-4) specimen after two-side steam oxidation at 1000°C for 15000 s [1]

High embrittlement of the uncoated Zirc-4 sheet due to important hydriding: Post-HT-oxidation:[H_{uncoated}] ~ 2000wt.ppm vs. [H_{coated}] < 100wt.ppm

Coating applied on tubular claddings geometry and complementary tests for beyond DBA-LOCA conditions

- First tests performed on flat (sheet) Zirc-4 specimens (1 mm thick)
- Transposition of the coating process to tubular clad geometry (up to 20 cm in length):

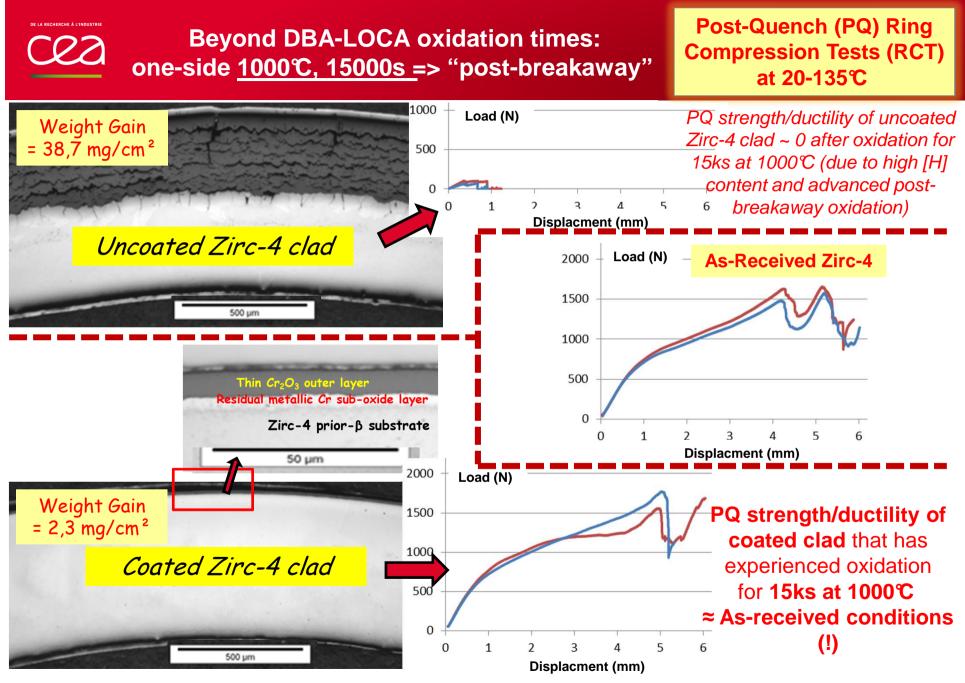
Ex. : One-side steam oxidation for 300s at 1200°C + direct water guenching from 1200°C



More recent results:

 Tests performed on optimized Cr coated (tubular) clad segments for beyond DBA-LOCA oxidation times

=> NEXT SLIDES ...



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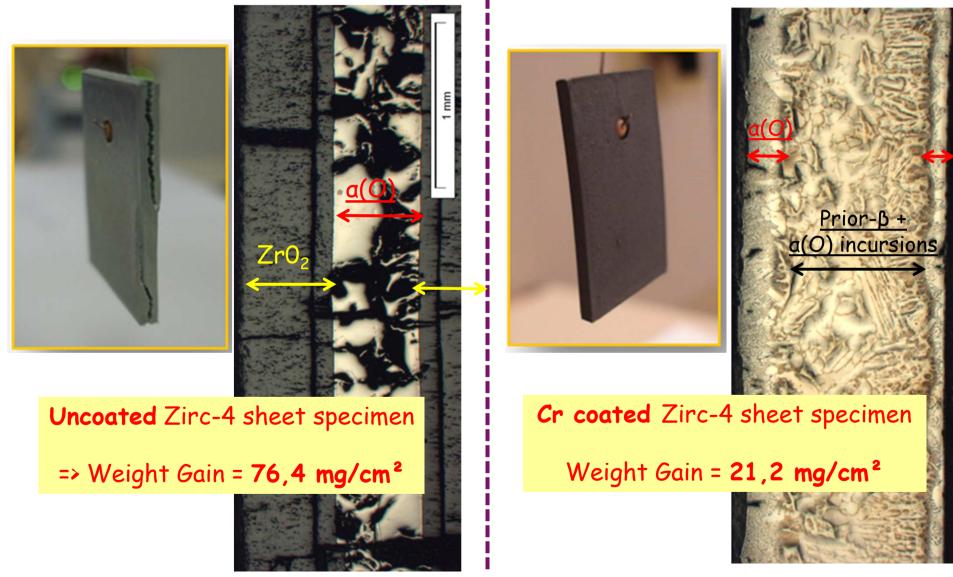
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Beyond DBA-LOCA oxidation times: one side, <u>1200℃, 6000s</u>



Partial conclusions: Optimized Cr coatings improve the capacity of the HT oxidized cladding to maintain some residual ductility and to survive to the final quenching for beyond DBA-LOCA HT oxidation times, thus providing significant additional "grace period" $\underbrace{Complementary results for beyond DBA-LOCA temperature conditions:}_{\text{``TGA'' oxidation in oxygen+helium environment up to } 1300^{\circ} - 5600 \text{ s}}_{\text{of Zirc-4 sheets with Cr coatings initial thickness ranging from 15 to } 20 \mu m}$



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Status of the R&D on innovative coated Zr cladding for PWR at CEA:

- Optimized PVD Cr-based coatings on Zirc-4 gave encouraging results so far
 - Decrease in weight gain in nominal, DBA-LOCA and slightly beyond DBA-LOCA conditions
 - Resistance to quench and post-quench ductility are enhanced by avoiding/limiting brittle ZrO₂ and Zr-a(O) layers formation and by reducing the oxygen content in the prior-beta layer
 - Delay the Breakaway phenomena (at ~1000°C)
 => decreases the associated hydrogen pick up
- Homogeneous/protective Cr coatings on Zirc-4 clad segment (up to ~20 cm)

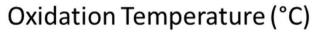
On-going and further work, pending issues:

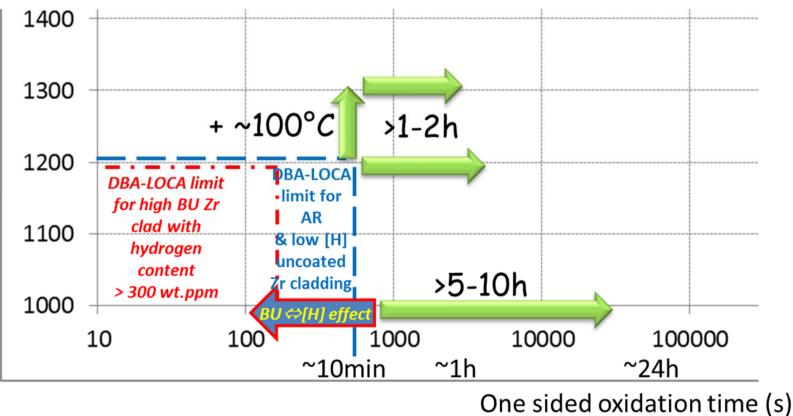
- Assessment of Cr coatings behavior in other relevant nominal and accidental scenarii, including **under neutron irradiation**, **thermal/stress/strain cycles**...
- Assessment of the potential additional margins achieved by relevant LOCA transients calculations (adapt existing codes and methodology)
- Fabrications issues on-going efforts to qualify the coating process for longer tubes, non destructive quality control, etc...

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Expected additional HT steam oxydation time/temperature margins for ~5-20µm thick optimized Cr coated Zirc-4 claddings vs. current DBA-LOCA limits

> Remark: of course, additional « grace timetemperature » which can be achieved depends on the initial Cr coating thickness





Cea THANK YOU FOR YOUR ATTENTION

Some CEA references:

[1] I. Idarraga et al., "Assessment at CEA of coated nuclear fuel cladding for LWRs with increased margins in LOCA and beyond LOCA conditions", in proceedings of 2013 LWR Fuel Performance Meeting/TopFuel, Charlotte, USA, (Sept. 15-19, 2013)

[2] M. Le Flem, et al., "Evaluation of potential alternative nuclear fuel cladding materials for LWRs - CEA studies", proceeding of OECD-NEA ATF Workshop, Dec. 10th, 2012, Issy-les-Moulineaux, France

[3] J.C. Brachet et al. "CEA studies on nuclear fuel claddings for LWRs enhanced accident tolerant fuel: recent results, pending issues and prospects", Proceedings of OECD-NEA Meeting on "Increased Accident Tolerance of Fuels for LWRs", Issy-les-Moulineaux, France, (Oct. 28-29, 2013)

[4] T. Forgeron et al., "ATF R&D status and Perspectives (I3P- Three Parties Institute CEA-EDF-AREVA)", Proceeding of OECD-NEA Start-up meeting of the Expert Group on Accident Tolerant Fuels of LWRs, OECD-HQ, Issy les Moulineaux, France, (April, 28-29, 2014)

[5] J.C. Brachet, Ch. Lorrette, et al., "CEA studies on advanced nuclear fuel claddings for enhanced Accident Tolerant LWRs Fuel (LOCA and beyond LOCA conditions)", Proceedings of the 8th. Int. Symposium "Fontevraud 8" on "Contribution of Materials Investigations and Operating Experience to LWRs' Safety, Performance and Reliability", Avignon, France, (Sept. 14-18 2014)

