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OUT-OF-PILE R&D ON COATED NUCLEAR FUEL ZIRCONIUM BASED CLADDINGS FOR ENHANCED ACCIDENT TOLERANCE IN LWRS

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INTRODUCTION / MAIN OBJECTIVES

Objective of R&D on coatings in the framework of the French CEA-EDF-AREVA Collaborative Program: to protect the cladding against corrosion and reduce hydrogen uptake during normal operating and accidental conditions and then to provide **additional margins on the nuclear fuel clad behavior in LWRs accidental conditions** such as LOCA by **decreasing the HT oxidation rate with steam at HT** (and the associated heat production):





- (1) Early studies scoping tests on different types of coatings including ceramic and metallic, mono or multilayered ones => <u>chromium</u> coatings have been selected as the most promising for next R&D steps
- (2) Improvement of the HT oxidation resistance of Cr coatings (DBA-LOCA)
- (3) Cr-coated Zr nuclear fuel clad behavior upon nominal conditions: \Rightarrow as-received microstructure/properties
- \Rightarrow corrosion resistance in PWR environment at 360°C & steam (100 bar) at 415°C
- (4) Cr-coated Zr nuclear fuel clad behavior upon steam oxidation at 1200°C: \Rightarrow numerous tests performed so far
- \Rightarrow determination of the Cr-coating oxidation/consumption mechanism
- \Rightarrow quantification/modelisation of the overall coating consumption kinetics
- (*) Remark: Additional studies/results for beyond LOCA conditions already published, including:
- "post-breakaway" conditions, i.e., oxidation at 1000°C up to > 4 hours;
- HT oxidation testing up to 1300°C, > 1 hour;
- => Assessment of the final quenching resistance and Post-Quenching (PQ) ductility ...

(1.1) EARLY STUDIES / SCOPING TESTS OF DIFFERENT TYPES OF COATINGS ON ZIRCALOY-4 SHEET SUBSTRATE

The different coatings studied (PVD - DC Magnetron sputtering):

Coating type	notation	Architecture / period (λ^*)	Period number	Total thickness ±0,2 μm
TiN	TiN	Singlelayered	/	2,6
CrN	CrN	Singlelayered	/	3
TiN and AITiN	TiN/AITiN	Multilayered, $\lambda = 2 \times 8 \text{ nm}$	>200	3,4
CrN and AlTiN	CrN/AITiN	Multilayered, $\lambda = 2 \times 8 \text{ nm}$	>200	3,2
Nb _{82%} V _{18%} (at%)	NbV	Singlelayered	/	5
Nb _{67%} Cr _{10%} Ti _{23%} (at%)	NbCrTi	Singlelayered	/	4
Cr	Cr	Singlelayered	/	1 and 5
	Cr/Cr	Multipass, $\lambda = 500 \text{ nm}$	14	7
Cr and Nb _{67%} Cr _{10%} Ti _{23%} (at%)	Cr/NbCrTi M10	Multilayered, $\lambda = 2 \times 5 \text{ nm}$	>500	6
	Cr/NbCrTi M100	Multilayered, $\lambda = 2 \times (50 \text{ to } 80) \text{ nm}$	40	5,5
	Cr/NbCrTi M600	Multilayered, $\lambda = 2 \times 300 \text{ nm}$	10	6
	Cr/NbCrTi M800	Multilayered, $\lambda = 2 \times 400 \text{ nm}$	5	4



* The period corresponds to the number of "layers", i.e. 1 layer of Cr in the case of Cr/Cr coating and 2 successive layers of different materials for TiN/AITiN, CrN/AICrN and Cr/NbCrTi coatings

(1.2) EARLY STUDIES / SCOPING TESTS OF DIFFERENT TYPES OF COATINGS ON ZIRCALOY-4 SHEET SUBSTRATE

Autoclave tests, 360°C, pressurized water, PWR chemistry (M. Tupin, P. Billaud et al., CEA):



- Poor behavior of TiN and NbV coatings

- Loss of weight gain observed for AI containing multilayered coatings, probably due to outward migration of aluminum and potential release in the pressurized water as observed by some other authors (Alat et al. - 2015, R. Van Nieuwenhove - this conference, etc...)

- Best behavior observed for CrN, Cr, Cr/Cr and multilayered Cr/NbCrTi coatings

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"18th Int. Symp. on Zr in Nucl. Ind.", Hilton Head, SC, (May 2016) | PAGE 5

(1.3) EARLY STUDIES / SCOPING TESTS OF DIFFERENT TYPES OF COATINGS ON ZIRCALOY-4 SHEET SUBSTRATE



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"18th Int. Symp. on Zr in Nucl. Ind.", Hilton Head, SC, (May 2016) PAGE 6

(2.1) IMPROVEMENT OF THE HT OXIDATION RESISTANCE OF CR BASED COATINGS (DBA-LOCA):

<u>1st R&D step</u>: improvement of the HT steam oxidation resistance of Cr-coated materials conducted on Zircaloy-4 <u>sheet samples</u> and focusing on DBA-LOCA conditions at first *(i.e., max. PCT = 1200°C, max. ECR (BJ) = 17% => oxidation time <600s for one-sided oxidation)*

One-sided steam oxidation at 1200°C for 300s (ECR (BJ) ~ 10%):



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"18th Int. Symp. on Zr in Nucl. Ind.", Hilton Head, SC, (May 2016) PAGE 7

(2.2) IMPROVEMENT OF THE HT OXIDATION RESISTANCE (DBA-LOCA) 2nd R&D step: Cr-coatings on tubular geometry (One-sided steam oxidation at 1200°C for 300s - ECR (BJ) ~ 10%): Uncoated-reference Zircaloy-4: Cladding oxidation due to peeling Blistering of coating of coating Weight-Gain~11mg/cm² eight-Gain-11 mg/cm Weight-Gain~1-2mg/cm² 45 mm 45 mm 30 mm Ni coating Blistering Peeling Ni coating ZrO₂ ZrO₂ Cr₂O₃ Cr Internal coating oxidation Zr-α(0) $Zr-\alpha(O)$ Zr-α(O) Zy-4 Zy-4 Zy-4 100 um 100 µm 50 µm Last generation Cr- Cr_2O_3 coated Zircaloy-4: Cr2O3 Cr Cr Weight-Gain<1mg/cm² Zy-4 Zy-4 ZrCr₂ layer 30 mm 🗕 2 µm 50 um

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(3.1) (OUT-OF-PILE) NOMINAL BEHAVIOR OF (LAST GENERATION) CR-COATED ZR:

As-received µstructure:

- Thicknesses studied: 1-20µm
- Multi-scale characterization:
 Optical microscopy, XRD,
 SEM-EBSD, EPMA, TEM
 (High resolution mode)
 on thin foils (FIB)...

Last generation of Cr-coatings: => Dense coatings, with no cracks, good Zr-Cr bonding with no interfacial defects



Additionally, nano-hardness, scratch and preliminary fretting/wear tests have been done => Good adhesion properties and good resistance to fretting for as-received conditions

(3.2) (OUT-OF-PILE) NOMINAL BEHAVIOR OF (LAST GENERATION) CR-COATED ZR:

Corrosion properties

415°C pressurized steam (100 days) 360°C PWR pressurized water 80 Ni deposit (Sample prep) Zv4 ref 70 ZrO₂ layer ZrO₂ Zv4 coated tubes ~10µm 60 Zv4 coated 2 sides Weight gain (mg/dm²) 50 **Uncoated Zy4** 40 Ni deposit (Sample prep) 30 Cr₂O₃ layer Cr ~100nm 20 10 0 / Cr-coated Zy4 0 50 100 150 200 Time (days)

- Cr-coated samples exhibit significantly reduced weight gain in autoclave

- (1-3 mg/dm²) with very little variation with time
- No dissolution or delamination of Cr-coating observed
- No ZrO_2 « pockets » formed under the Cr-coating = coating is very protective

(3.3) HT STEAM OXIDATION BEHAVIOR CR-COATED ZR => CHROMIUM COATING EVOLUTION/CONSUMPTION KINETICS UNDER STEAM ENVIRONMENT AT HT:



1D diffusion calculation performed (semi-infinite volume assumption): $C(x,t) - C_o = (C_s - C_o)(1 - erf(x/(2/Dt)))$

- D : thermal (volumic) diffusion coefficient in prior- β_{Zr} (from the litterature) = 6,29 $\mu m^2/s$ at 1200°C,

- C(x, t) is the Cr concentration diffusing into prior- β_{Zr} at distance x from Zr-Cr interface, and for diffusion time t,

- C_o is the initial Cr concentration in the Zircaloy-4 substrate (~0.1 wt%)

- C_s the concentration at the Cr/Zr interface assuming local thermodynamic equilibrium

=> calculated thanks to CEA-"Zircobase" thermodynamic database => $\sim 3 \text{ wt}$ % at 1200°C (solubility limit of Cr in β_{Zr})

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"18th Int. Symp. on Zr in Nucl. Ind.", Hilton Head, SC, (May 2016) PAGE 11

(3.4) HT STEAM OXIDATION BEHAVIOR OF CR-COATED ZR:

Cr coating consumption mechanism during steam oxidation at HT

Two main contributions:
(1) Growth/thickening of outer Cr₂O₃ scale;
(2) Diffusion of Cr from the Zr/Cr interface into the prior-β substrate.
For coating thicknesses of at least a few micrometers and as far as DBA-LOCA oxidation times are considered, the two basic kinetics are parabolic:



(3.5) HT STEAM OXIDATION BEHAVIOR OF CR-COATED ZR:





 In the framework of the French CEA-EDF-AREVA Collaborative Program, zirconium based <u>coated claddings</u> are developed as potential <u>Enhanced Accident Tolerant Fuels</u> (EATF) materials <u>for LWRs</u> applications:

=> Mid-term (~10 years) R&D with negligible/limited impact on the geometry, mechanical, neutronic and thermal properties of the nuclear fuel assembly => <u>Easier/faster licensing</u>

- From the early scoping tests performed on different types of coatings (including metallic, ceramics, multi-layered...), <u>chromium</u> coatings have been selected and further <u>improvement of its HT oxidation resistance</u> has been conducted;
- Last generation of chromium coated zirconium based nuclear fuel clad behavior shows:
 (1) very good nominal corrosion resistance (PWR environment at 360°C & steam at 415°C)
 (2) attractive HT steam oxidation resistance for DBA-LOCA and beyond (up to 1300°C);

=> for prototypical DBA-LOCA oxidation times, Cr-coating coating consumption mechanism and kinetics established at 1200°C *(can be extended at other oxidation temperatures)*;

- <u>Ballooning & burst</u> tests under <u>internal pressure at HT</u> (LOCA) are on-going showing encouraging behavior (protective effect maintained, no peeling ... cf. Topfuel 2016)
- Ions and neutron <u>irradiation</u> experiments are on-going/planned (cf. poster A. Wu - this conference, Halden-IFA-796 project...)

THANK YOU FOR YOUR ATTENTION

Some CEA references:

[1] I. Idarraga et al., "Assessment at CEA of coated nuclear fuel cladding for LWRs with increased margins in LOCA and beyond LOCA conditions", in proceedings of 2013 LWR Fuel Performance Meeting/TopFuel, Charlotte, USA, (Sept. 15-19, 2013)

[2] J.C. Brachet, Ch. Lorrette, et al., "CEA studies on advanced nuclear fuel claddings for enhanced Accident Tolerant LWRs Fuel (LOCA and beyond LOCA conditions)", Proceedings of the 8th. Int. Symposium "Fontevraud 8" on "Contribution of Materials Investigations and Operating Experience to LWRs' Safety, Performance and Reliability", Avignon, France, (Sept. 14-18 2014)

[3] "J.C. Brachet, et al., "On-going studies at CEA on chromium coated zirconium based nuclear fuel claddings for enhanced tolerant LWRs fuel", proceeding of TopFuel 2015, (13-19 Sept. 2015), Zurich, Switzerland

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Additional/back slides

IMPACT OF PRE-EXISTING CR COATING DEFECTS

The potential negative impact of a pre-existing defect (i.e., asreceived crack) within the Cr coating on the subsequent coated clad oxidation is an important issue:

⇒ Results obtained so far show only a limited an very localized impact of a pre-existing thruwall Cr layer crack on the subsequent oxidation behavior for both nominal and accidental (LOCA) conditions (i.e., neither local accelerated corrosion, nor coating spallation)

⇒ However, further work is necessary: longer oxidation times...



HT BEHAVIOR IN STEAM - DBA-LOCA TEMPERATURE RANGE

- 1-SIDED OXIDATION KINETICS



HT BEHAVIOR IN STEAM - DBA-LOCA TEMPERATURE RANGE

- POST-QUENCHING (PQ) MECHANICAL PROPERTIES

- 1000°C FOR « POST-BREAKAWAY » 1-SIDED OXIDATION TIME = 15000S

Ring Compression Test at 135°C:



Expected additional HT steam oxidation time/temperature margins for ~5-20µm thick optimized Cr coated Zirc-4 claddings vs. current DBA-LOCA limits

> Remark: of course, additional « grace timetemperature » which can be achieved depends on the initial Cr coating thickness

Oxidation Temperature (°C)

Cea

HT BEHAVIOR IN STEAM – BEYOND DBA-LOCA TEMPERATURE RANGE

1300°C, oxidation time = 5400s

