



Diffusion of Chromium in Nickel Base Alloys Application to Stress Corrosion Cracking

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DE LA RECHERCHE À L'INDUSTRIE



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DIFFUSION OF CHROMIUM IN NICKEL BASE ALLOYS: APPLICATION TO STRESS CORROSION CRACKING

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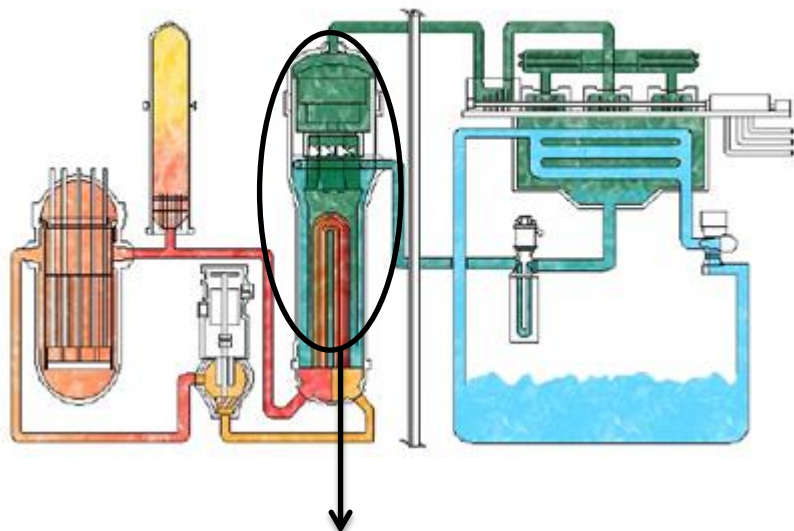
François JOMARD, Université de Versailles - GEMaC

25 JUNE 2015

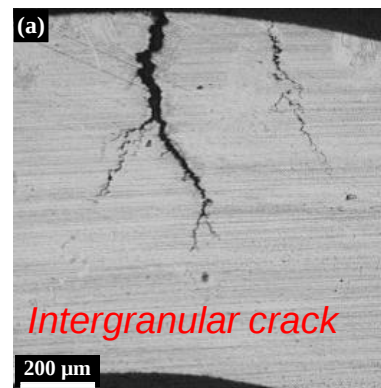
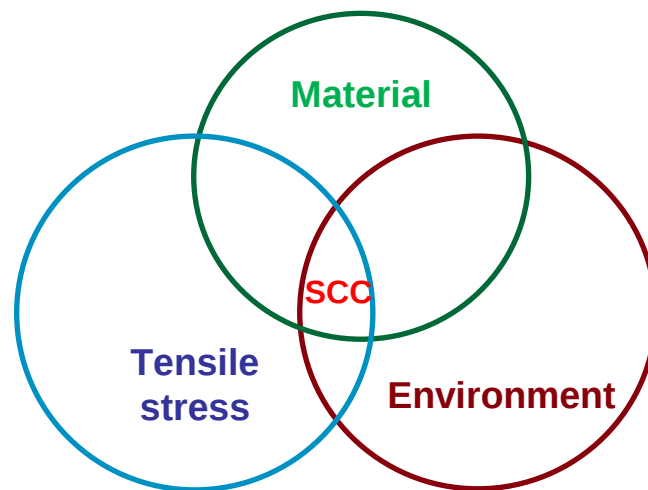
INDUSTRIAL CONTEXT

Stress corrosion cracking (SCC)

Pressurized Water Reactor (PWR)



Steam generator tubes : Nickel Base Alloy
A 600- $\rightarrow$ Ni-15Cr-Fe



Stress corrosion cracking of Alloy 600 formed on U bend specimen at 325°C during 1200 hr [1]

Intergranular crack

200 μm

Primary water conditions

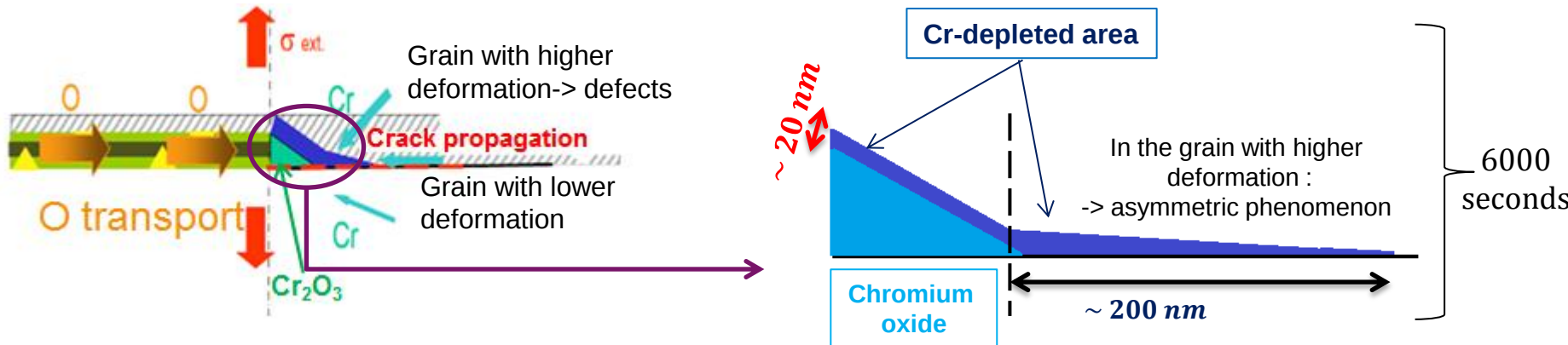
Temperature	285 - 325 °C
Pressure	155 bar
Boreen (H_3BO_3)	10 -1200 ppm
Lithium (LiOH)	0,7 - 2,2 ppm
H_2	25 - 50 $\text{cm}^3.\text{kg}^{-1}(\text{TPN})$
O_2	< 5 ppb
$\text{pH}_{300^\circ\text{C}}$	≈ 7

Life extension of PWR > 30 – 40 years

Mechanisms of Stress Corrosion Cracking (SCC) [1]

Asymmetric and selective oxidation

Crack tip : Cr-depleted area observed only in one grain along the grain boundary

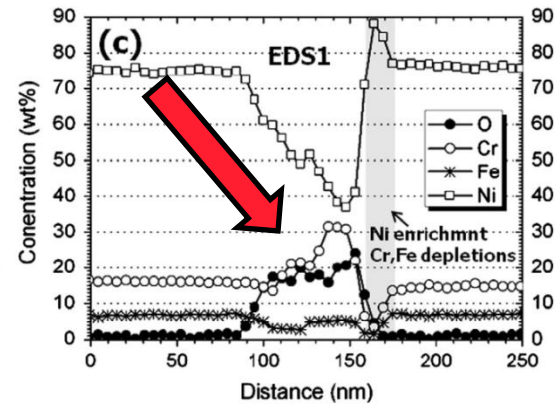
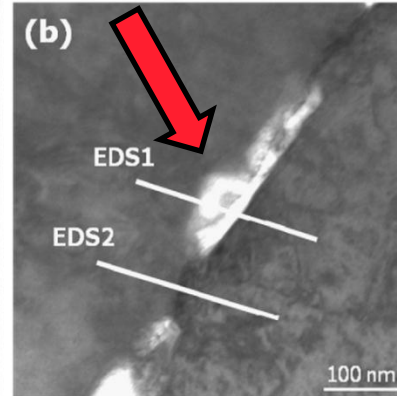
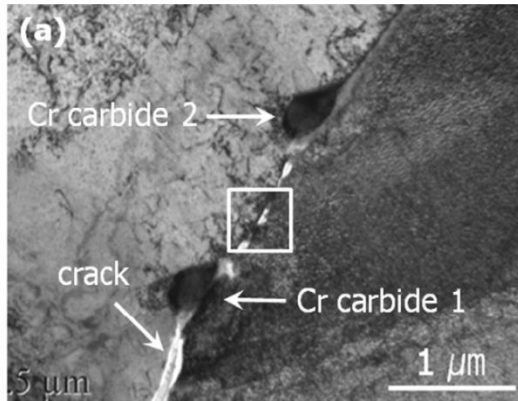


Mechanisms proposed by authors [1]

1. Oxygen transport is not the rate-limiting step
2. Strain maybe heterogeneous in the vicinity of the crack tip
3. Formation of the chromium depleted area occurs in about 6000 seconds.

Chromium diffusion in Alloy 600 at low temperatures (around 350°C) : one of the key steps of the SCC mechanism

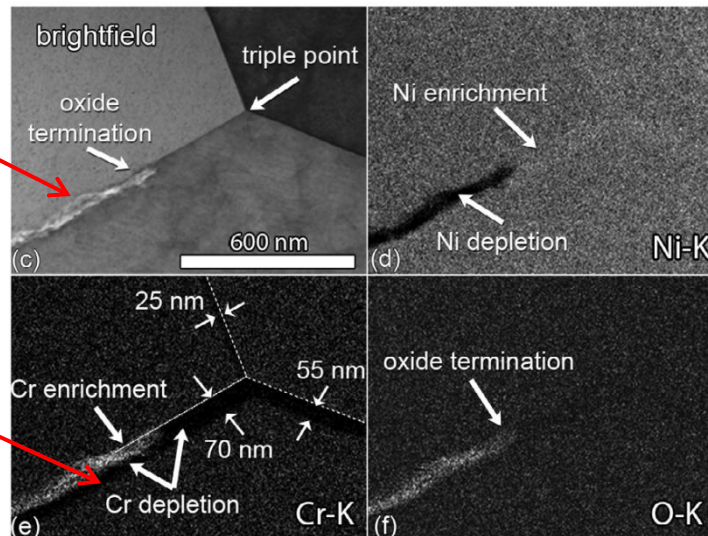
Intergranular oxide and chromium depleted area in literature



Lim Yun Soo, Kim Hong Pyo, and Hwang Seong Sik.
Journal of Nuclear Materials
440, no. 1-3 (2013) : 46-54.

Chromium oxide

Cr-depleted area

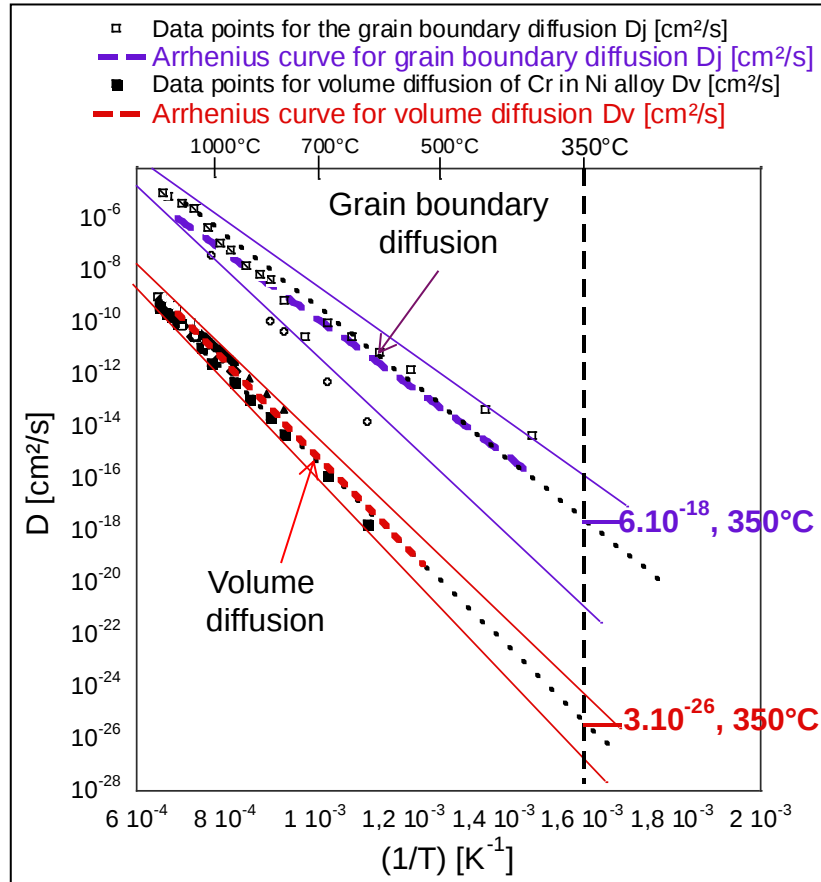


Cr-depleted area at metal/oxide interface and ahead of the intergranular oxide.

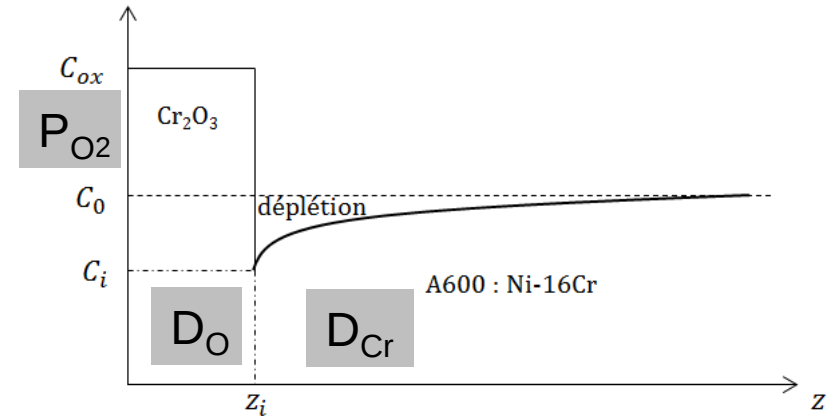
D.K. Schreiber, M.J. Olszta and S.M. Bruemmer, S., *Scripta Materiala* 89 (2014) 41-44

Volume and grain boundary diffusion of chromium : literature

Arrhenius [1,2,3,4,5] 400°C et 1270°C



Calculation of chromium depletion time in Alloy 600



$$\frac{C - C_o}{C_i - C_o} = \frac{\operatorname{erfc}\left(\frac{z}{2\sqrt{Dt}}\right)}{\operatorname{erfc}\left(\frac{z_i}{2\sqrt{Dt}}\right)}$$

z_i depend on :

- Oxygen diffusion coefficient
- Oxygen partial pressure
- Chromium diffusion coefficient

Results :

-Grain boundary diffusion

⇒ $t = 3$ days

-Volume diffusion

⇒ $t = 100$ years

- [1] D.D. Pruthi, M.S. Anand, and R.P. Agarwala, Journal of Nuclear Materials 64 (1977) 206-210
 [2] T.F. Chen, G.P. Tiwari, Y. Iijima, and K. Yamauchi. Materials Transactions 44 (2003) 40-46
 [3] K. Monma, O. Suto, and O. Hiroshi. The Japan Institute of Metals and Materials 28 (1964) 188-192
 [4] B. Million, J. Ruzickova, and J. Vrestal. Materials Science and Engineering 72 (1985) 85-100
 [5] Moulin P., Huntz A. M., and Lacombe P. Acta Metallurgica 77 (1979) : 1430-43.

These results are not sufficient to explain the chromium depleted area observed in literature (20 nm in 6000 seconds).

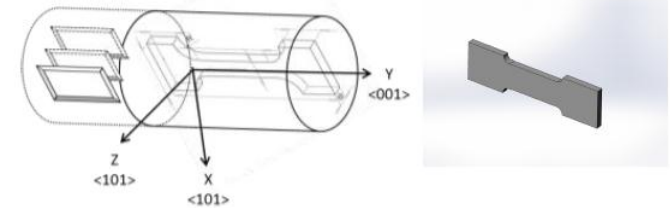
For which conditions is chromium diffusion enhanced ?

1. Materials

- ☐ Pure nickel - Single crystal - crystallographic orientation $\langle 101 \rangle$

2. Chromium deposition

- ☐ Sample polishing
- ☐ Evaporating chromium under vacuum
 - > 20 nm thin layer (powder ^{52}Cr - 99,6% purity)

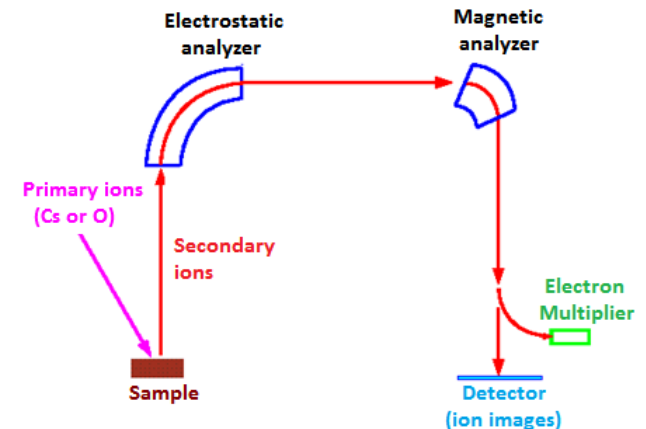


3. Diffusion treatments in secondary vacuum

4. Measurement of diffusion coefficients

- ☐ By Secondary Ion Mass Spectrometry (SIMS)

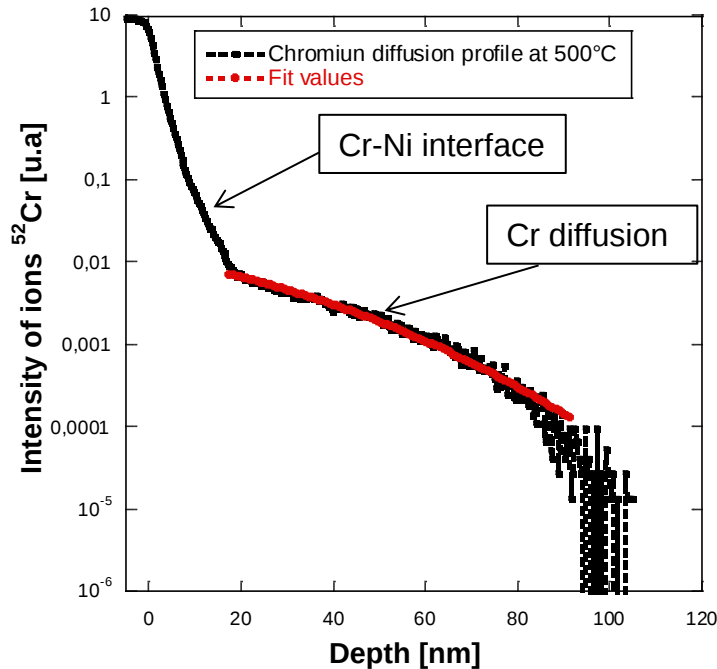
CAMECA - IMS 7F
Detection limit : of ppb to ppm
Lateral resolution : 0,5 μm
Depth resolution : environ 1 nm
Analysis conditions :
Primary ions : O_2^+
Partial pressure O_2 : $2 \cdot 10^{-6}$ mbar
Primary ions intensity : 50 nA
Crater Size : 150 μm * 150 μm
Analysis area : $\varnothing 33 \mu\text{m}$



- ☐ Measurement of the final crater depth by profilometry technique

RESULTS

Secondary Ion Mass Spectrometry (SIMS)



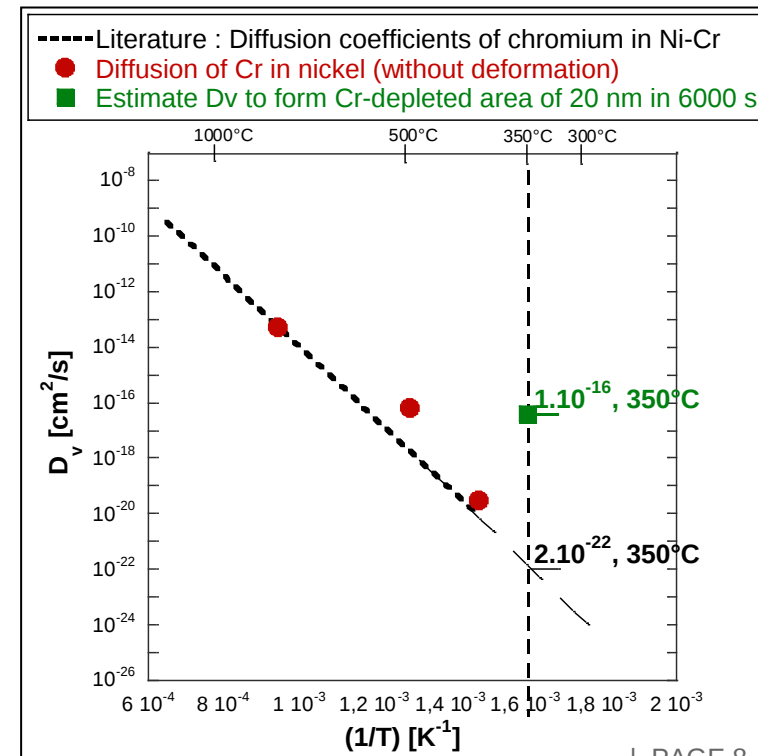
Fick's solution equation

$$C(x, t) = C_s \cdot \operatorname{erfc}\left(\frac{x}{2\sqrt{Dt}}\right)$$

Diffusion treatments	D [cm ² /s]
787°C, 5h	(5±1).10 ⁻¹⁴
500°C, 30h	(7±2).10 ⁻¹⁷
405°C, 1616h	(3±2).10 ⁻²⁰

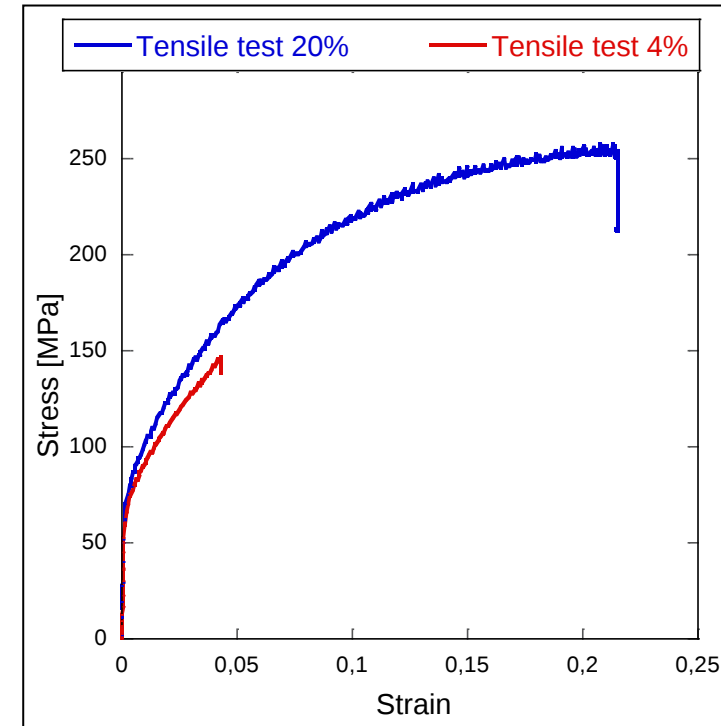
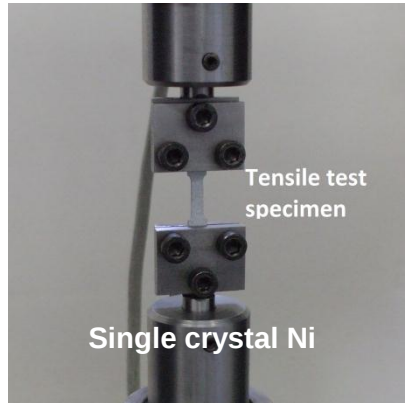
⇒ Results consistent with literature data.
⇒ Volume diffusion is not sufficient to explain the asymmetric Cr depletion observed in literature (20 nm in 6000 sec).

Effect of pre-deformation ?



Aim : influence of pre-deformation on diffusion coefficient

Diffusion in pre-deformed material : diffusion treatments after tensile tests



Samples conditions	Diffusion treatments	D [cm ² /s]
Non-deformed	30h, 500°C	$(7 \pm 2) \cdot 10^{-17}$
$\varepsilon_p = 4 \%$	30h, 487°C	$(3,5 \pm 0,9) \cdot 10^{-17}$
$\varepsilon_p = 20 \%$	30h, 487°C	$(3 \pm 3) \cdot 10^{-17}$

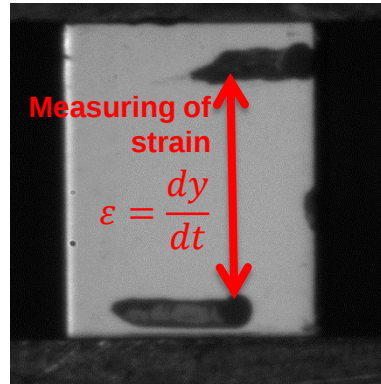
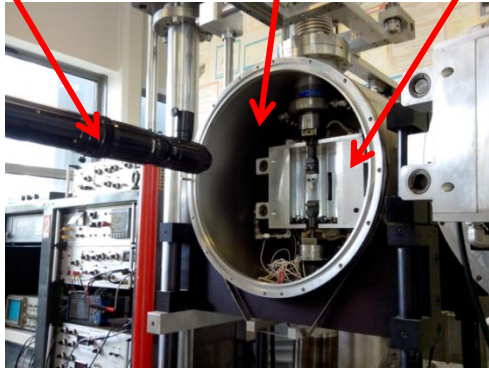
⇒ No significant effect of the pre-deformation on chromium diffusion.

⇒ Pre-deformation can not explain the asymmetric oxidation and depletion along grain boundary.

Coupling of diffusion and deformation ?

Aim : the effect of dislocations mobility

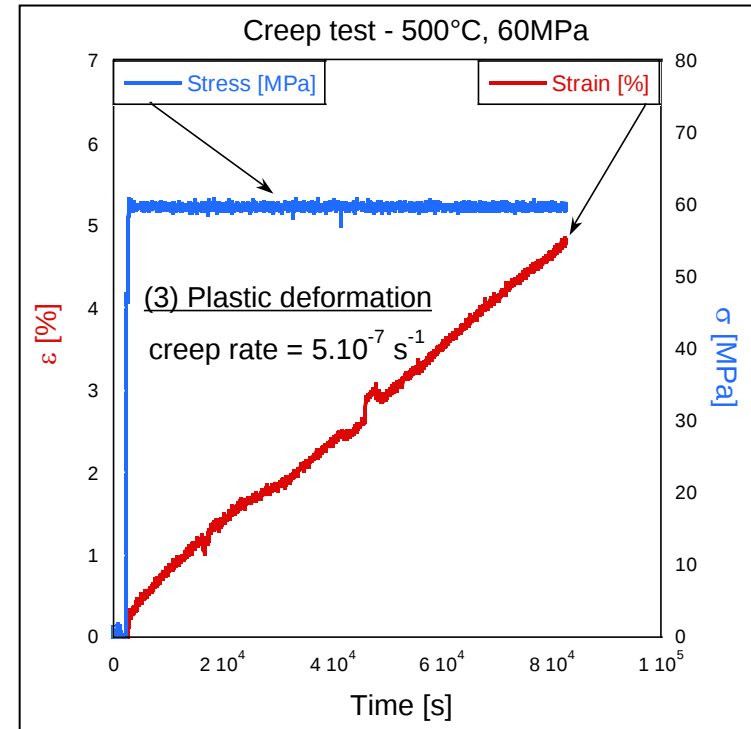
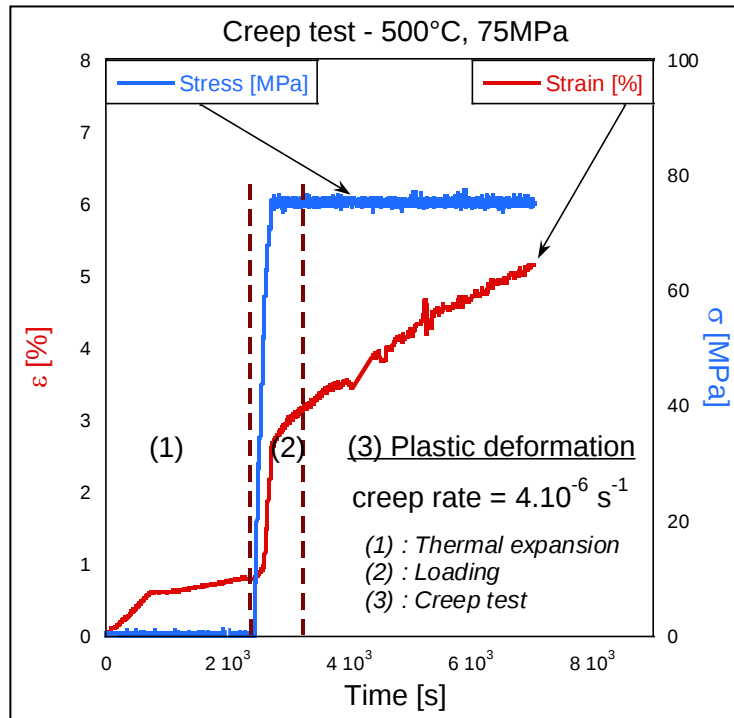
Camera Chamber Furnace



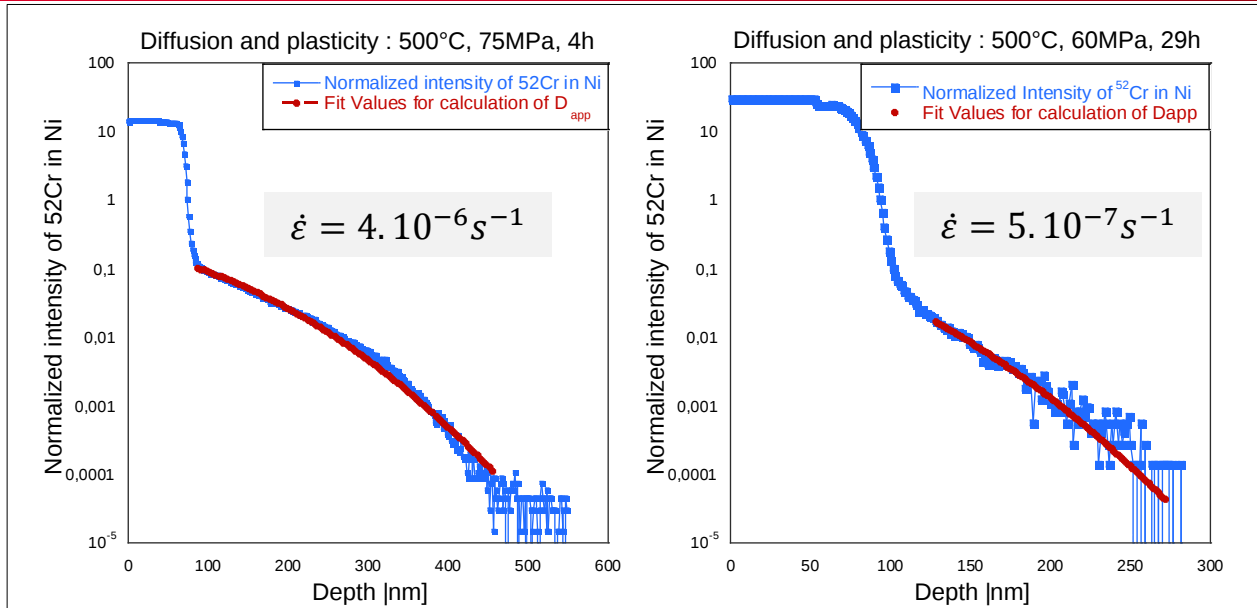
Creep tests

500°C, 60 MPa, 29h Creep rate : $5 \cdot 10^{-7} \text{s}^{-1}$

500°C, 75 MPa, 4h Creep rate : $4 \cdot 10^{-6} \text{s}^{-1}$



The effect of dislocations mobility at 500°C



Analytic solution [1] :

$$C(x) = C_o \left[\text{erfc} \left(\frac{x}{2\sqrt{D_{app} \times t}} \right) \right]$$

Experiments	$D_{app} [cm^2/s]$
500°C, 60 MPa, 29h Creep rate : $5.10^{-7} s^{-1}$	$2,8.10^{-16}$
500°C, 75 MPa, 4h Creep rate : $4.10^{-6} s^{-1}$	$6,4.10^{-15}$
(Without plasticity) at 500°C, 30h	$7,0.10^{-17}$

⇒ Enhanced chromium diffusion at 500°C with plasticity effect.

⇒ The diffusion coefficient increases with creep rate (or dislocations mobility v ; *Orowan's law* : $\dot{\epsilon} = \rho \cdot b \cdot v$).

Creep test at 350°C for the same dislocations mobility as at 500°C !

Creep law :

$$[1] \quad \dot{\epsilon}_s = S. \exp\left(\frac{-\Delta H}{RT} + \beta\sigma\right)$$

$S = \text{constant}$
 $\Delta H \sim 66 \text{ kcal/mol}$
 $\beta = 1.10^{-7}$

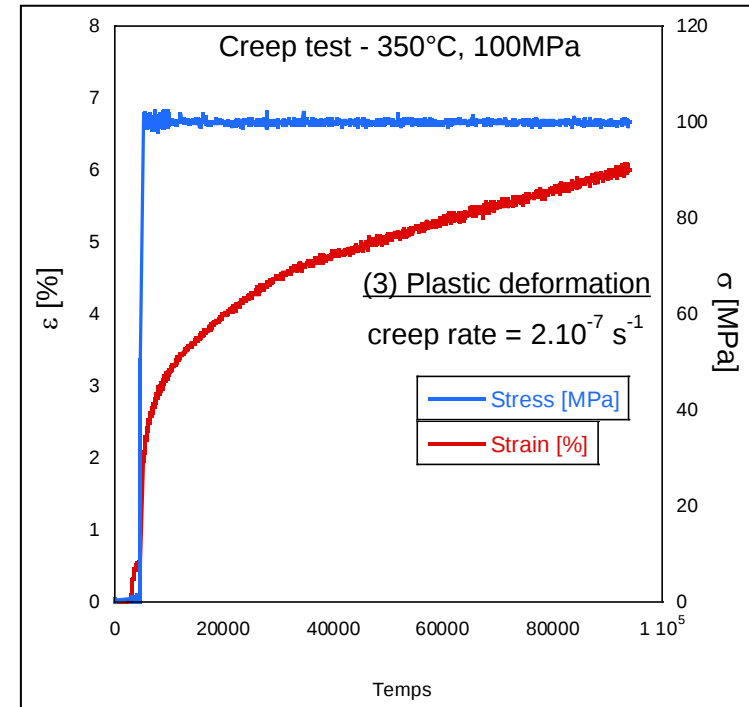
Estimation of stress at 350°C

$$\sigma_{350^\circ\text{C}} = 41 + 5 \ln\left(\frac{\dot{\epsilon}_{350^\circ\text{C}}}{\dot{\epsilon}_{500^\circ\text{C}}}\right) + \sigma_{500^\circ\text{C}} \text{ en MPa}$$

Same creep rate : $\dot{\epsilon}_{500^\circ\text{C}} = \dot{\epsilon}_{350^\circ\text{C}}$	
T = 500°C	$\sigma_{500^\circ\text{C}} = 60 \text{ MPa}$
T = 350°C	$\sigma_{350^\circ\text{C}} = 100 \text{ MPa}$

Creep test at 350°C

350°C, 100 MPa, 26h Creep rate : 2.10^{-7} s^{-1}

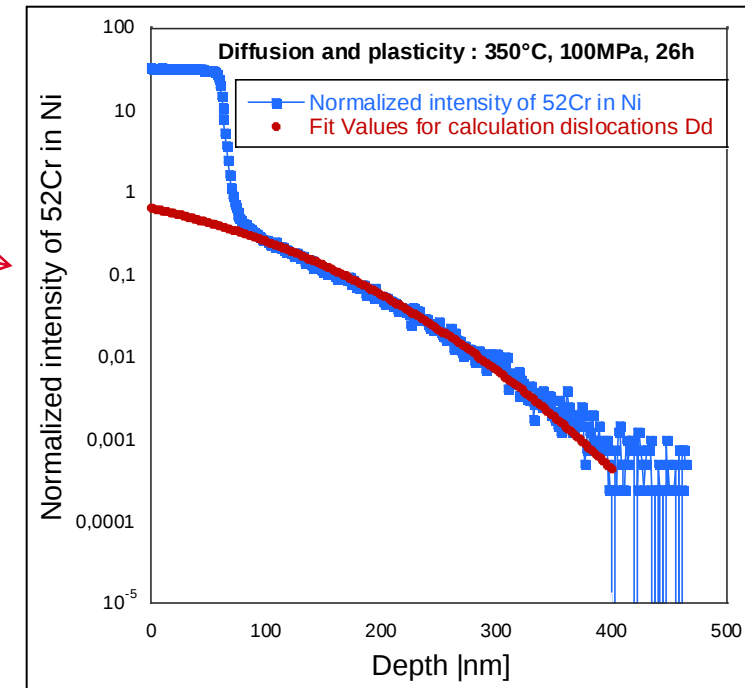
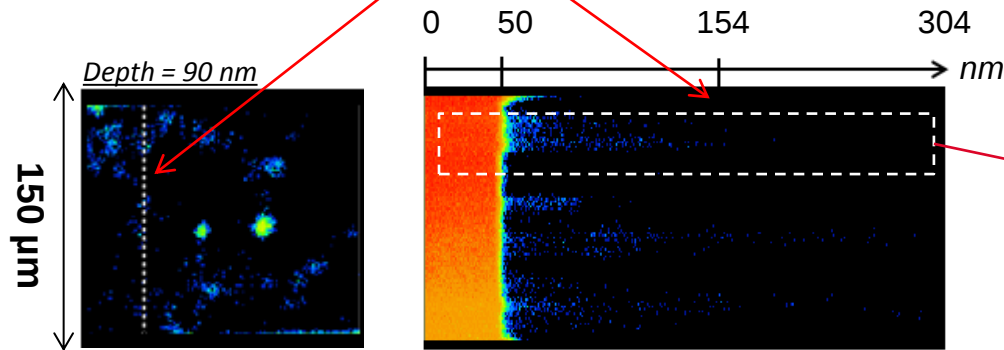


✓ Viscoplasticity is observed at 350°C in these conditions.

Measurement of diffusion coefficient of chromium after the creep test at 350°C.

Aim : the effect of dislocations mobility at 350°C

SIMS analysis : Cr-ionic image
at cross-section in the deep



Cr deposit

Chromium diffusion in nickel
up to 300 nm of depth.

SIMS
crater area



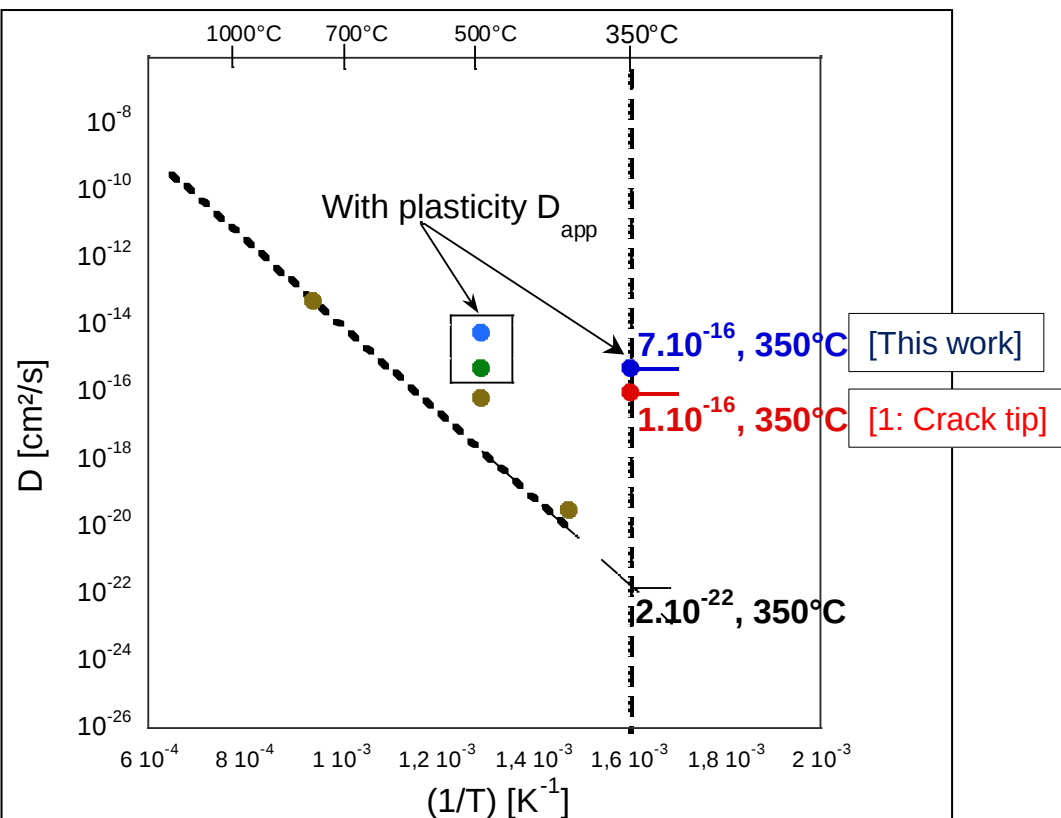
Optical image

Experiment	D [cm ² /s]
350°C, 100 MPa, 26h Creep rate : 2.10^{-7}s^{-1}	$7.4.10^{-16}$
(Without plasticity) at 350°C (extrapolated value)	$1.9.10^{-22}$
500°C, 60 MPa, 29h Creep rate : 5.10^{-7}s^{-1}	$2.8.10^{-16}$

⇒ Plasticity can accelerate the chromium diffusion at low temperatures.

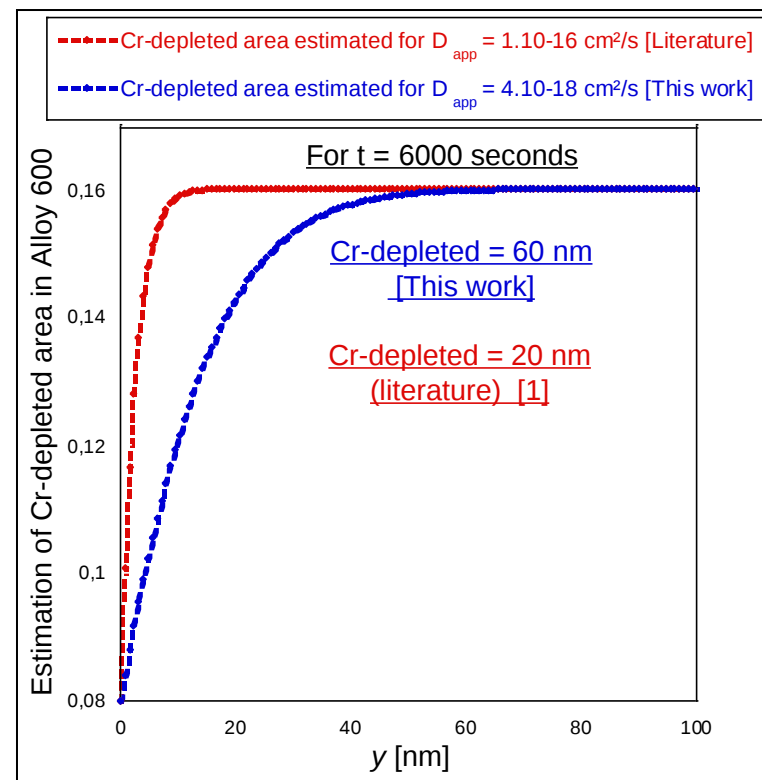
⇒ This phenomenon seems to depend on the creep rate.

Summary : Cr-depleted area at the SCC crack tip



----- Literature : Diffusion coefficients of Cr in alloy nickel D [cm²/s]

- Diffusion of Cr in nickel (without deformation) D [cm²/s]
- Diffusion coupling with plasticity : 4.10^{-6} s^{-1} - 500°C D_{app} [cm²/s]
- Diffusion coupling with plasticity : 5.10^{-7} s^{-1} - 500°C D_{app} [cm²/s]
- Diffusion coupling with plasticity : 2.10^{-7} s^{-1} - 350°C D_{app} [cm²/s]
- Estimate D to form Cr-depleted area of 20 nm in 6000 s



Chromium diffusion assisted by moving dislocations can explain Cr-depleted area, observed at SCC crack tip.

➤ Conclusions

- ❑ No effect of pre-deformation on chromium diffusion coefficient in nickel.
- ❑ The accelerating effect of plasticity on chromium diffusion in nickel has been showed at 500°C and 350°C. Diffusion is enhanced by dislocations mobility for the tested conditions.
- ❑ Diffusion along moving dislocations could take place at the SCC crack tip and could allow the formation of chromium depleted area.

➤ Prospects

- ❖ Perform more experiments at low temperatures (350°C) and for different dislocation velocities.
- ❖ Similar experiments with A600.
- ❖ Define a model which describes the formation of chromium depleted area at the SCC crack tip.

Thank you for your attention

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