

The transport and settling of platinum-group-metal particles in glass melts

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Society of Glass Technology Centenary Conference SGT100 & ESG2016

Sheffield UK, September 4th-8th 2016

Waste vitrification

THE TRANSPORT AND SETTLING OF PLATINUM-GROUP-METAL PARTICLES IN GLASS MELTS

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Short overview of the High-Level-Waste confinement process

- Ensure long-time confinement of High-Level-Waste (HLW) of spent nuclear fuel reprocessing
- Vitrification: atomic-scale incorporation of HLW in a glass melt
- Most recently developed technology used for vitrification
 Induction-heated cold crucible



Full scale prototype at CEA



Free surface of the melt in the crucible



Glass pouring

Particles encountered in nuclear waste vitrification processes

- Nuclear glass melt
 - Homogeneous borosilicate glass matrix
 - Seeded with small particles (~ $10\mu m$)
- Particles are Platinum-Group-Metals with a **much higher density than glass**
 - **—** Ruthenium dioxide (RuO_2) needles
 - Palladium (Pd) spheres











Numerical simulation of the vitrification process in cold crucibles





Full thermo-hydraulic numerical simulation of the cold crucible

- Numerical simulation of coupled models
 - Fluid mechanics for stirred glass: Navier-Stokes equations
 - Temperature transport: Energy equation
 - Electromagnetics for induction: Maxwell equations

- But PGM particles have significant effects when their local volume fraction increases (because of settling)
 - Viscosity increase
 - Electrical conductivity increase

Need for a supplementary model to compute particles local volume fraction in the melt C(x, y, z, t)

Transport model for particles



Resulting in a single unsteady evolution equation for the particles volume fraction C

$$\frac{\partial C}{\partial t} + \mathbf{u} \cdot \nabla C + \nabla \cdot (\tau \hat{\mathbf{g}} C) = \nabla \cdot (\kappa \nabla C)$$

Accounting for temperature-dependent transport properties

4 transport parameters, 2 of them strongly depend on glass viscosity η



Glass melting processes are non-isothermal and glass viscosity depends on temperature



Laboratory settling experiments with a nuclear waste simulant (1)

- Initially homogeneous nuclear waste simulant seeded with PGM particles Heat-treated during prescribed settling time without any
 - stirring $(\mathbf{u} = 0)$

Crucible vertically sliced after cooling Laser Induced Breakdown Spectroscopy (LIBS) to measure vertical concentration profiles C(z,t)





Sliced glass sample in the

Comparison to the transport model (solved numerically)

 $\frac{\partial C}{\partial t} + \mathbf{u} \cdot \nabla C + \nabla \cdot (\tau \hat{\mathbf{g}} C) = \nabla \cdot (\kappa \nabla C)$

Laboratory settling experiments with a nuclear waste simulant (2)



Structure of the sludge

- Exp. 3: sludge obtained from long-term settling
 - Sample height 4.9cm
 - Uniform temperature 1200°C
 - Both ruthenium dioxide and palladium particles 0.7%vol
 - Heating time 211h

- Intricate percolated network
- Relatively low volume fraction 3.5%vol





Conclusion

- Accurate transport model with low computational cost
- Able to account for PGM settling in glass melts, with temperature stratification
- Precise 3d imaging gives an insight into the sludge structure

- Transport model included in 3d simulations of waste vitrification in cold crucibles along with:
 - "Cloud settling" through Rayleigh-Taylor instabilities
 - Percolation model for the electrical conductivity of PGM-laden glass

THANK YOU FOR YOUR ATTENTION

Detailed transport model

Hindering function

$$\frac{\partial C}{\partial t} + \mathbf{u} \cdot \nabla C + \nabla \cdot \left(\frac{\tau_0}{f_\eta} \,\widehat{\mathbf{g}} C \left(1 - \frac{C}{C_a}\right)^\beta\right) = \nabla \cdot \left(\frac{\kappa_0}{f_\eta} \,\nabla C\right)$$

Stokes terminal velocity
$$\tau_0 = \frac{d_p^2(\rho_p - \rho_f)g}{18\eta_0}$$

Diffusivity $\kappa_0 = \frac{d_p^2 \alpha}{18\eta_0}$
Vogel-Fulcher-Tamman viscosity law $f_\eta \propto \exp\left(\frac{B}{R(T-T_v)}\right)$