

## Context and Objectives

In the frame of **minor actinide recycling** in sodium cooled fast reactors, (**U, Am**)O<sub>2</sub> mixed oxides are promising transmutation targets. To assess the thermodynamic properties of the U-Am-O system, it is essential to have a thorough knowledge of the **binary phase diagrams**, which is difficult due to the lack of thermodynamic data on the Am-O system<sup>1</sup>. Nevertheless, an **Am-O phase diagram** modeling has been recently proposed by Gotcu *et al.*<sup>1</sup> in which a fluorite-type dioxide, a sesquioxide and an intermediate phase are reported for an O/Am ratio ranging from 2 to 1.5. Here, we show an investigation of the Am-O system coupling thermodynamic modeling based on the **CALPHAD** method and *in situ* high temperature X-Ray Diffraction (**XRD**).

## Methods

## Experimental Setup

Powder XRD was performed with a Bragg-Brentano  $\theta$ - $\theta$  Bruker D8 Advance X-ray diffractometer using Copper radiation and a LynX'Eye fast-counting PSD detector. The **heating stage** is constituted by a Pt strip and a radiant heating element. The **atmosphere is controlled** through a flowing gas whose partial oxygen pressure (**pO<sub>2</sub>**) is measured and adjusted at the inlet of the chamber with a *GenAir* device (probe and pump oxygen) in order to have the desired **oxygen potential  $\Delta G_{O_2}$** .

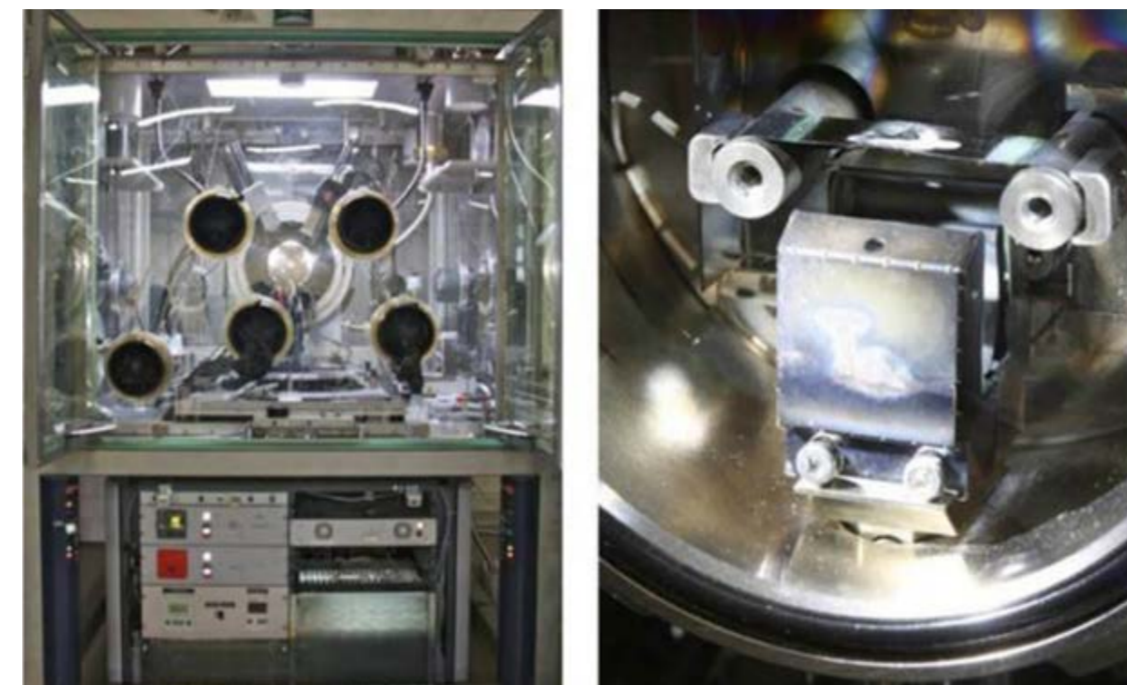
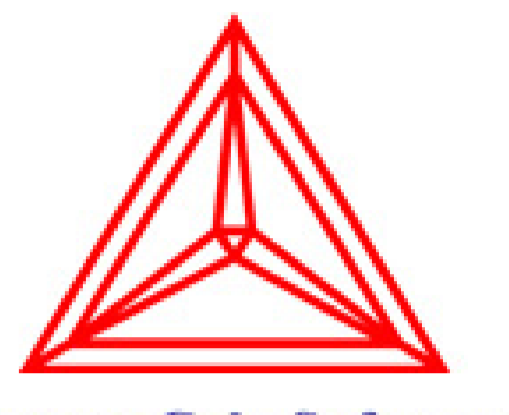


Fig. 1 The diffractometer glove box and the heating stage.

## Thermodynamic Calculations

In order to set up the most interesting experimental conditions (temperature  $T$  and  $\Delta G_{O_2}$ ), thermodynamic calculations based on the CALPHAD method were performed, using the **TAF-ID** (Thermodynamic of Advanced Fuels-International Database)<sup>2</sup> containing the model<sup>1</sup>.



Thermo-Calc Software

A new improved thermodynamic assessment of the Am-O system was also performed. Both computations and modeling were performed using the *Thermo-Calc* software.

## CALPHAD

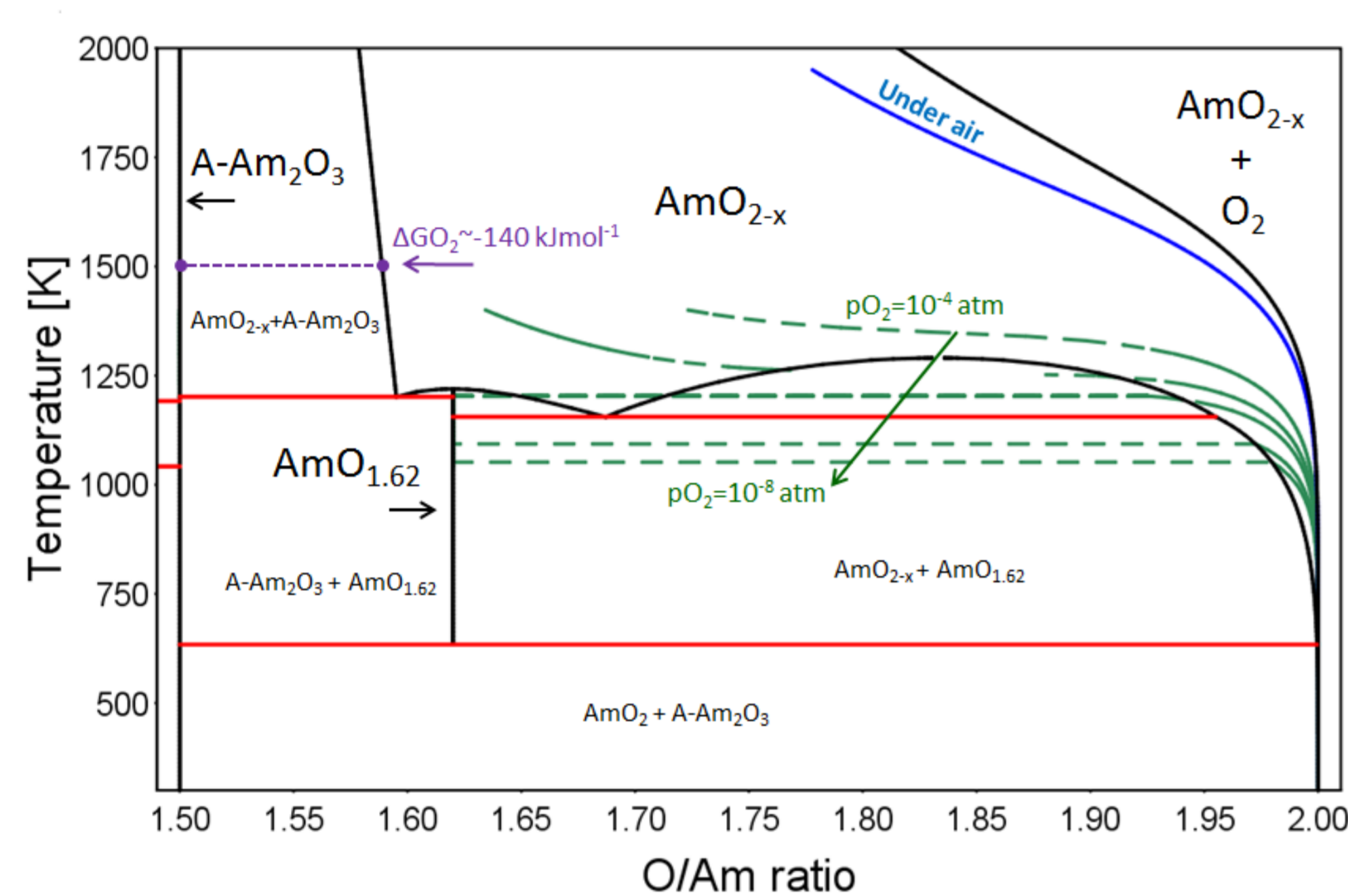
Calculations based on the literature model<sup>1</sup>

Fig. 2: The phase diagram computed with the available model

Thermodynamic calculations were performed in order to foresee the explored regions of the phase diagram:

- Equilibria under air
- - - Equilibria under He with constant pO<sub>2</sub>
- ⋯ AmO<sub>2-x</sub>/AmO<sub>2</sub>O<sub>3</sub> two phase region at 1500 K

## Improved CALPHAD assessment

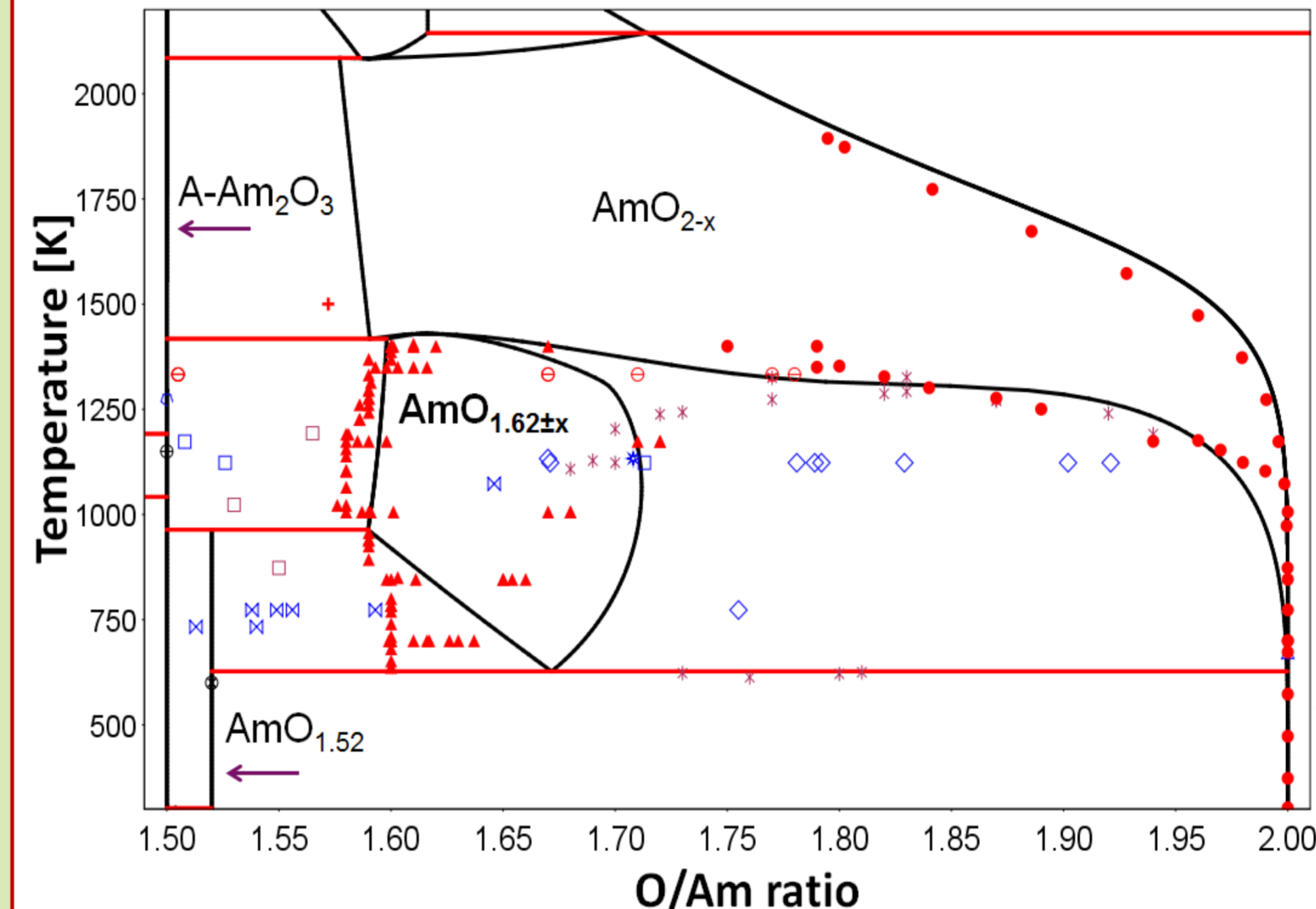


Fig. 7: The phase diagram computed with the new model

- The **AmO<sub>1.62</sub>** composition domain was described
- The **AmO<sub>1.52</sub>** phase was added
- The miscibility gap was not included in this model

## EXPERIMENTAL RESULTS

## Results in agreement with the literature model

The AmO<sub>2-x</sub> behavior at HT under air

- The slope change of the thermal expansion at **T≈1200K** indicates the starting of the **AmO<sub>2</sub> reduction under air**, in agreement with the model previsions

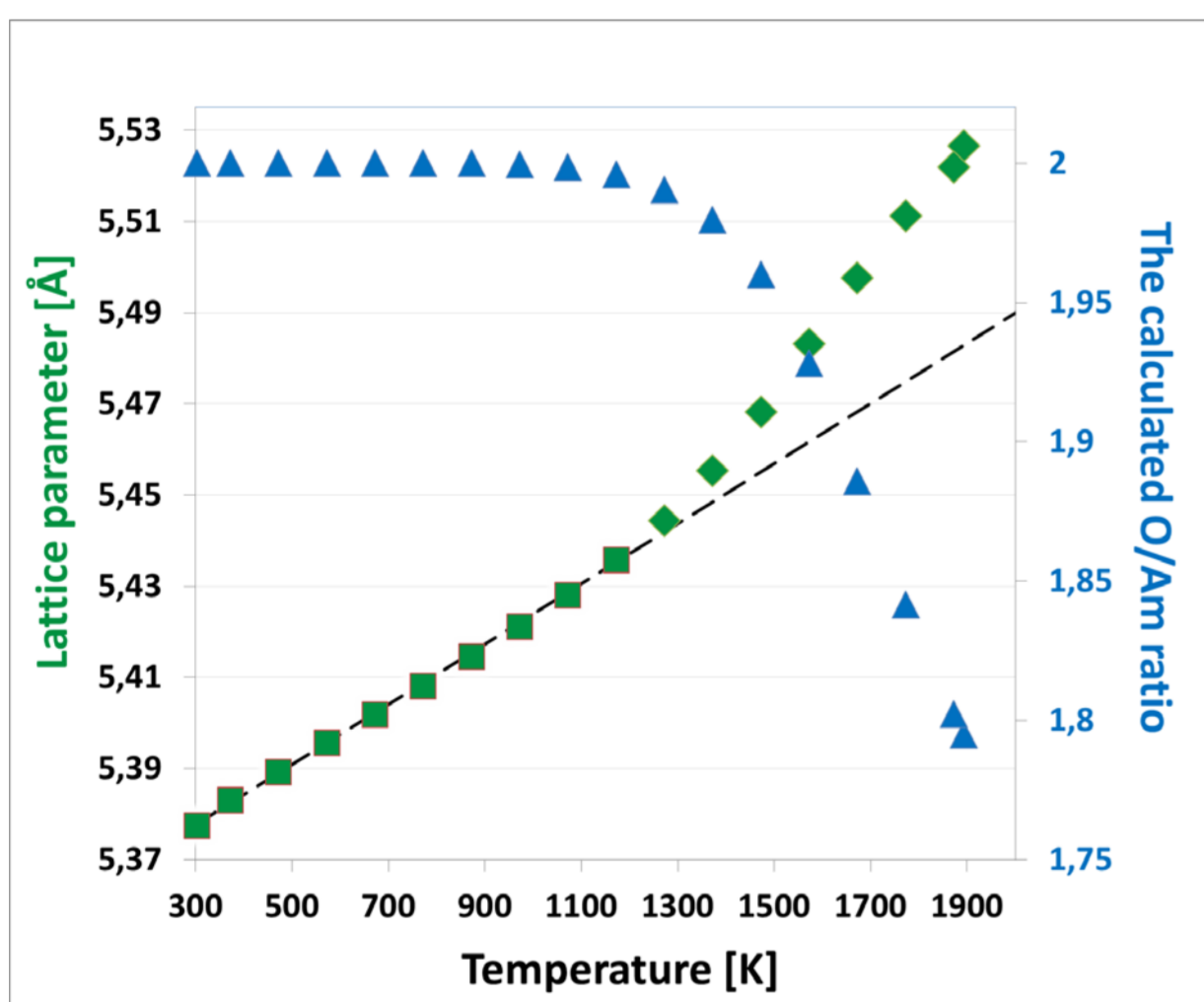
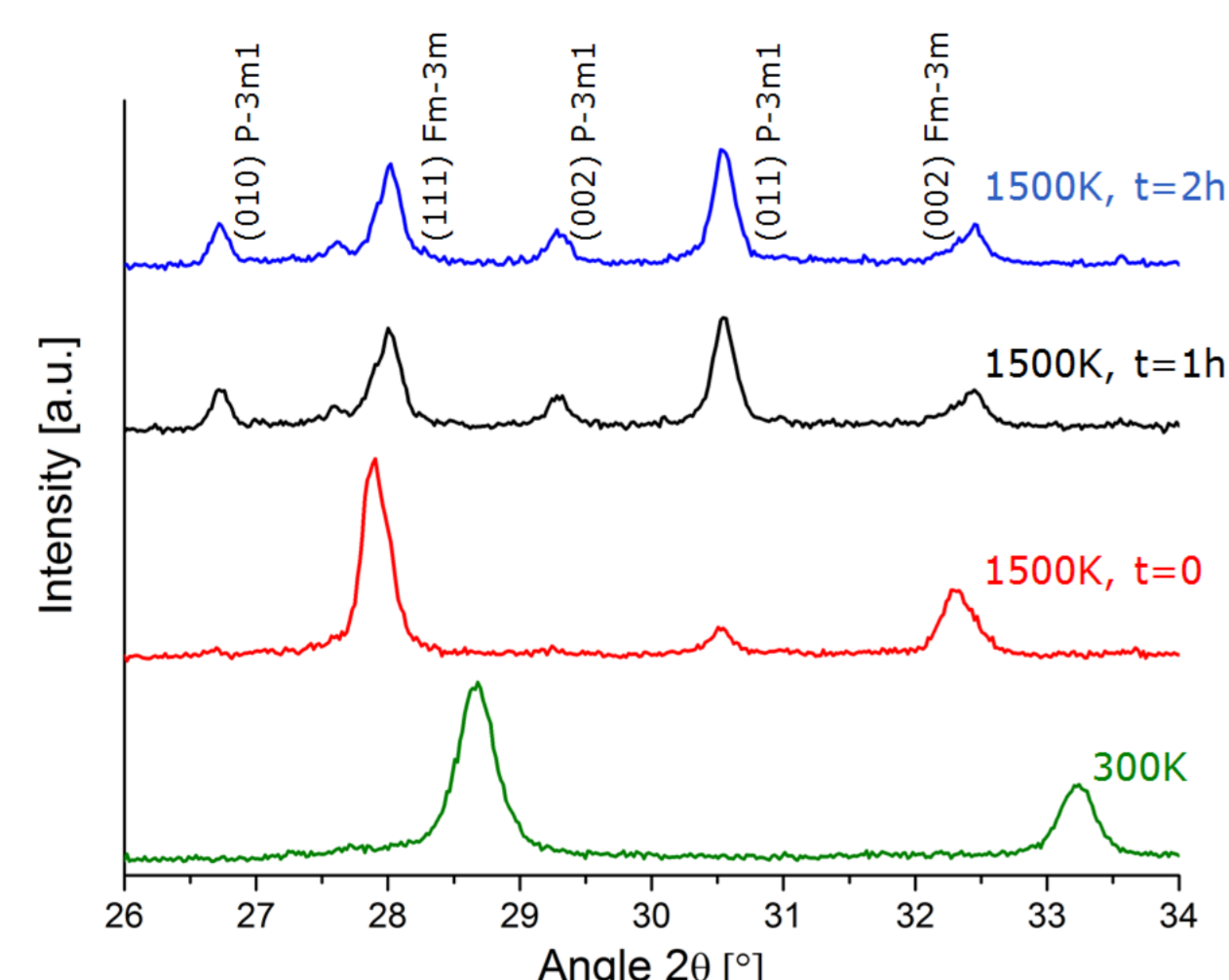


Fig. 3: Green points: measured lattice parameters during cooling from 1900K to RT under air; Blue points: the O/Am ratio calculated with ThermoCalc.

The AmO<sub>2-x</sub>/A-Am<sub>2</sub>O<sub>3</sub> two-phase region

- At **T=1500K**, the hexagonal **A-Am<sub>2</sub>O<sub>3</sub>** phase was observed by imposing the  **$\Delta G_{O_2}$**  expected by the model ( $\approx -140$  kJ/mol)

Fig. 4: Diffraction patterns acquired at RT and then at T=1500K for 2 hours under controlled atmosphere (pO<sub>2</sub>≈2 10<sup>-5</sup> atm).

The *reliable* data were fitted using the least square minimization in order to find a linear relation between the **Lattice Parameter a**, the **O/Am ratio** and **Temperature**.

This relation allowed a **coherent interpretation** of all the experimental data and lead to the **re-define of the phase boundaries** of the phase diagram.

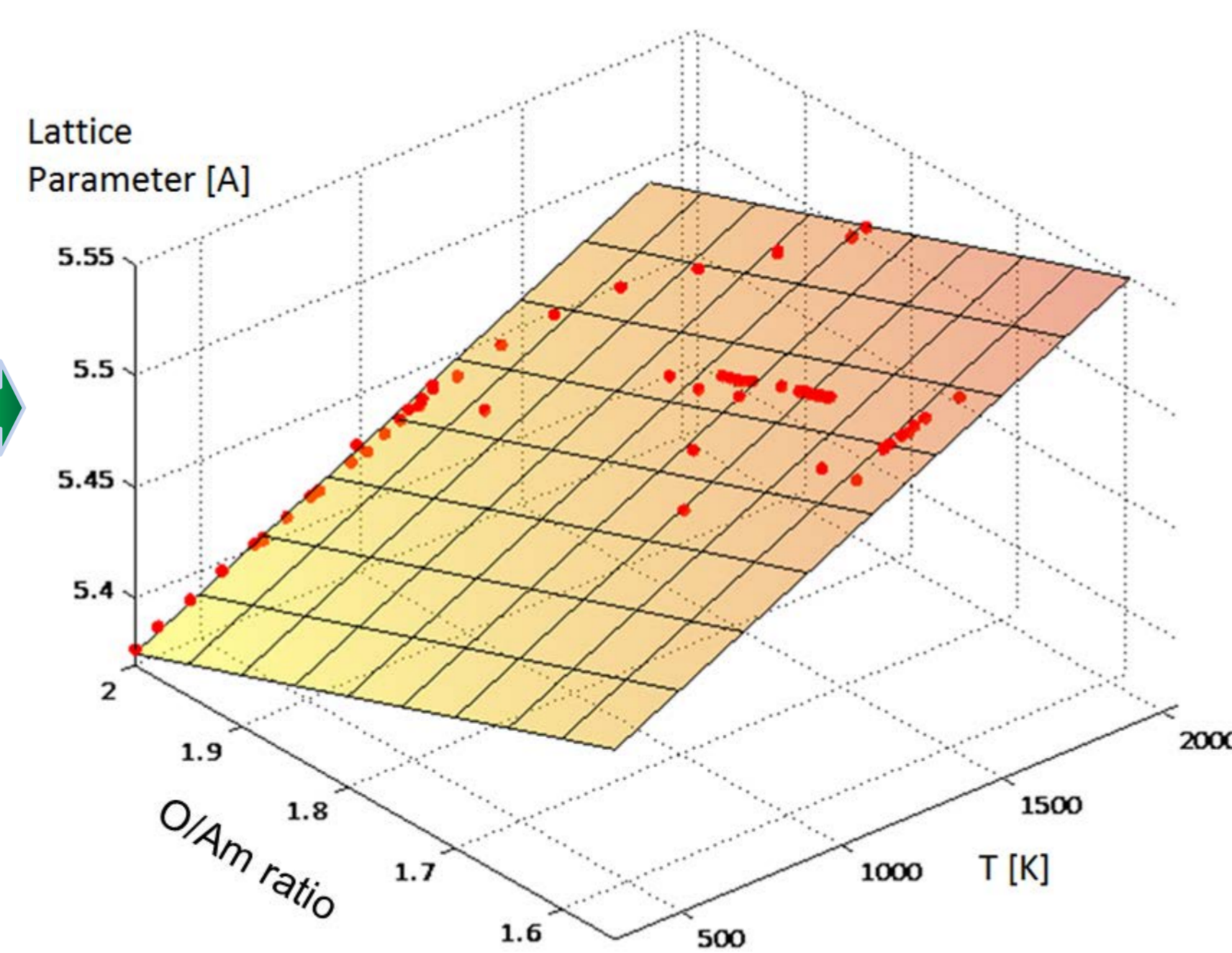
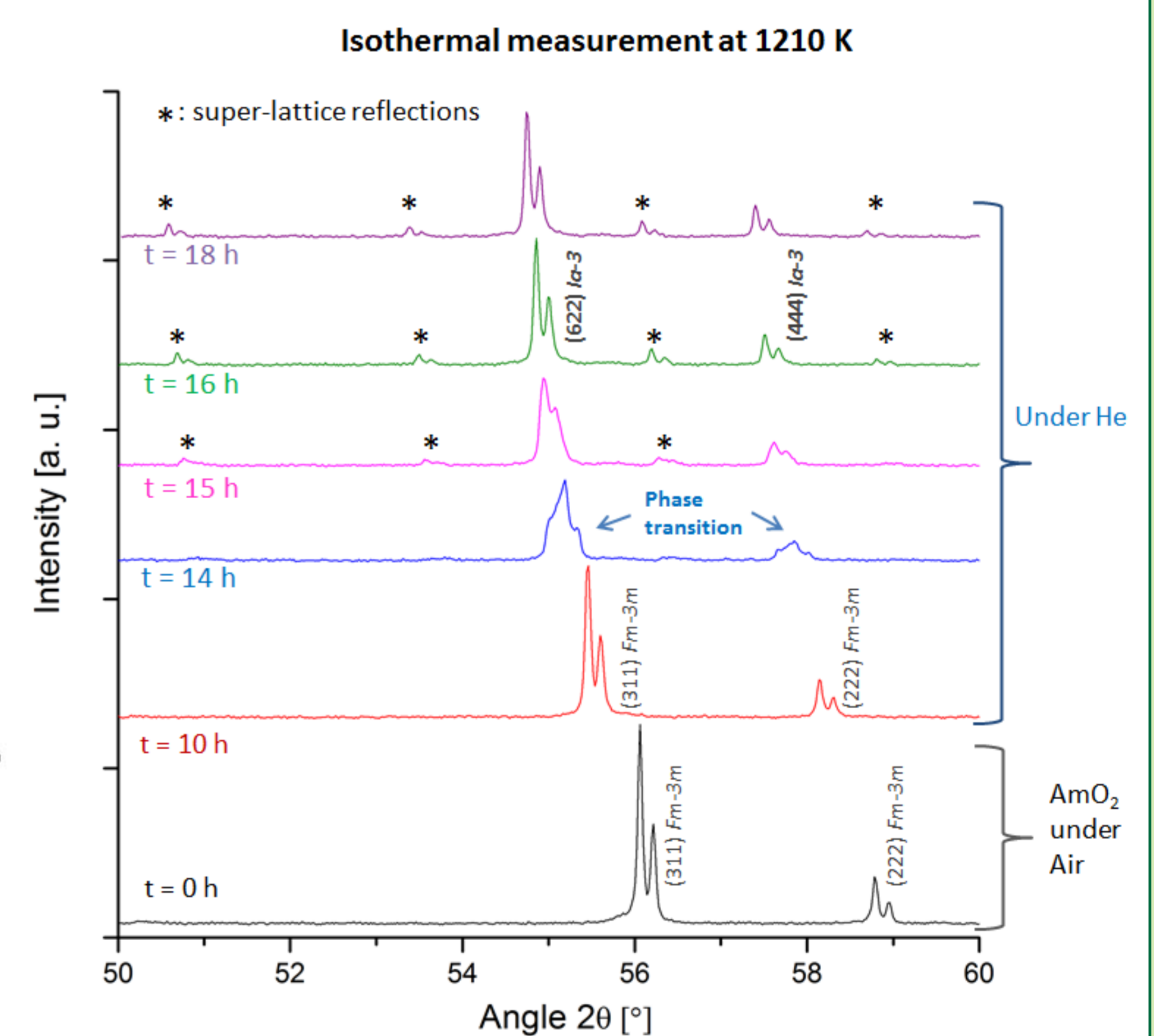


Fig. 5: Fit of the "reliable" data: a as a function of T and O/Am

$$a(T, O/Am) = C_1 + C_2 T - C_3 O/Am$$

## New insight

Contrary to the Pu-O system, Iso-T measurements at 1280 K and 1210 K have not shown the presence of a **miscibility gap**. The AmO<sub>2-x</sub> → AmO<sub>1.62±x</sub> phase transition was observed. **BCC AmO<sub>1.62±x</sub>** was found to have a wide **composition domain**.

Fig. 6: Diffraction patterns from an iso-T measurement at 1210 K under He. After 14 h, the AmO<sub>2-x</sub> → AmO<sub>1.62±x</sub> phase transition occurred. Then, the reduction of the AmO<sub>1.62±x</sub> was observed (shift of the peaks to lower angles), proving the existence of a composition domain

## Conclusions

In this work, an investigation of the Am-O system by HT-XRD was presented. The experimental conditions ( $T$ ,  $\Delta G_{O_2}$ ) were chosen according to thermodynamic computations based on the Gotcu Calphad model [1], with the aim of verify its accuracy. A good agreement between experimental results and calculations was found for the AmO<sub>2</sub> in the slightly hypo-stoichiometric region (experiments under air). The model has also proven to be able to predict the appearance of the A-Am<sub>2</sub>O<sub>3</sub> phase at HT (1500K), but the agreement between results and calculation was lost at lower temperature. Using the least square minimization method, a **lattice parameter - O/Am - T** relation was found, which allowed a coherent interpretation of all the data and the definition of the phase boundaries of a *new* BCC phase. Finally, thanks to the new data acquired, an improved **Calphad assessment** was performed and a **new phase diagram representation** was proposed.