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DE LA RECHERCHE À L'INDUSTRIE







# STUDY OF THE TENSILE ANISOTROPY OF EXTRUDED ODS STEEL TUBES USING NOTCHED RINGS

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## **GEN IV SODIUM FAST REACTORS**

### **GEN IV Sodium Fast Reactors**

- More sustainable fuel cycle
- Improved safety
- Reduced waste production

Higher burnup leading to more severe fuel cladding sollicitations

- Temperature: 400-650°C
- Internal pressure:
- 120 bars (max at life end)
- Irradiation: 100-200 dpa
- Sodium environment effects

## > Oxide Dispersion Strengthened (ODS) steel

Good candidate material thanks to:

- high resistance to thermal creep
- Low irradiation swelling



**ODS STEEL TUBE MANUFACTURING** 

[MINOS 2012]



Studied material : tube

Process introducing anisotropic microstrucure



Ferritic 14%Cr ODS steelFe-14Cr-1W-0.3Ti- $0.25Y_2O_3$ Martensitic 9%Cr ODS steelFe-9Cr-1W- $0.3Ti-0.25Y_2O_3$ 







Equiaxed grains with no specific texture

# **CHARACTERIZING TUBES**

## □ Main objectives

- Testing the longitudinal and hoop properties of the tubes
- > Through tensile, uniaxial and internal pressure creep tests
- In order to predict the lifetime of fuel cladding

# **TILE AND RING TENSILE SPECIMENS**

## Main objectives

- Testing the longitudinal and hoop properties of the tubes
- > Through tensile, uniaxial and internal pressure creep tests
- In order to predict the lifetime of fuel cladding

## **Tensile** properties: **2 specimen geometries**

- Tile specimens for the longitudinal properties
- Notched ring specimens for the hoop properties

The ring test is non-standard [1-4], know to be dependent on :

- Load configuration
- Friction
- Length to width gauge ratio





# **TILE AND RING TENSILE SPECIMENS**

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Tile specimen





Ring specimen

In order to properly study material anisotropies, the effect of the specimen geometry must be analysed

## **TENSILE TESTS**



## **TEST RESULTS – ENGINEERING CURVES**



- In the 400-750°C range, the 14%Cr tube is significantly more resistant in the L direction
- For the 9%Cr tube, both directions are almost equivalent

## **TEST RESULTS – ENGINEERING CURVES**



- In the 400-750°C range, both 9%Cr and 14%Cr show anisotropic total elongation
- As the 9%Cr mechanical behaviour is considered to be isotropic, specimen geometry is a potential cause to investigate -> modelisation





## CONSTITUTIVE MODELLING AND FE SIMULATION



## MODELLING

## Choices for mechanical behaviour and damage laws

Tensile and creep tests suggest the following:

9%Cr tube: isotropic (viscoplastic) behaviour, anisotropic damage (to account for varying crack orientations)

□ 14%Cr tube: anisotropic (viscoplastic) behaviour, anisotropic damage

A continuum damage mechanics approach (Lemaitre-Chaboche) was used.

The following simulations work focus on the **tensile test** for the **9%Cr tube at 650°C**.

Understanding of the ring specimen influence on the overall behaviour will permit the proper analysis of the 14%Cr tube material anisotropies.

# **ESTIMATION OF THE FRICTION COEFFICIENT**

- Friction plays an important role and has to be estimated, lacking experimental means to evaluate it
- Digital Image Correlation is used to obtain displacement/strain fields on the outer surface of the gauge section of the ring
- Using a simple elastoplastic model the strain fields are compared for different friction coefficients.
- A value is chosen according to the best similarities (friction coefficient =0.1)





### DE LA RECHERCHE À L'INDUSTR

## **CONSTITUTIVE MODELLING – 9%CR**

Elasticity coupled with anisotropic	$\boldsymbol{\varepsilon}^{e} = \frac{1}{2G} (\boldsymbol{H}\boldsymbol{\sigma}'\boldsymbol{H})' + \frac{1}{9K} (g(\boldsymbol{H})\langle tr \boldsymbol{\sigma} \rangle_{+} + \langle tr \boldsymbol{\sigma} \rangle_{-}) \mathbb{1}$
damage	$\widetilde{\boldsymbol{\sigma}} = [\boldsymbol{H}\boldsymbol{\sigma}'\boldsymbol{H}]' + \frac{1}{2}(g(\boldsymbol{H})\langle tr \boldsymbol{\sigma} \rangle_{+} + \langle tr \boldsymbol{\sigma} \rangle_{-}) \mathbb{1}$
[Lemaitre et. al, 2000]	$\widetilde{\boldsymbol{\sigma}} = \mathbb{E}: \boldsymbol{\varepsilon}^{e}$

 $\tilde{\sigma}$ : Effective stress **D**: Damage 2<sup>nd</sup> order tensor  $\eta$ : hydrostatic sensitivity parameter K: bulk modulus G: shear modulus

$$H = (\mathbf{1} - \mathbf{D})^{\frac{-1}{2}}$$
$$g(H) = (1 - \eta) + \eta \left(\frac{1}{10}(tr(H))^2 + \frac{1}{30}tr(H)^2\right)$$

[Desmorat et. al, 2017]

> Anisotropic damage, crack opening/closing phenomenon on hydrostatic stress  $\tilde{\sigma}_H$ 

### DE LA RECHERCHE À L'INDUSTR

## **CONSTITUTIVE MODELLING – 9%CR**

Elasticity coupled with anisotropic damage [Lemaitre et. al, 2000]	$\boldsymbol{\varepsilon}^{e} = \frac{1}{2G} (\boldsymbol{H}\boldsymbol{\sigma}'\boldsymbol{H})' + \frac{1}{9K} (g(\boldsymbol{H})\langle tr \boldsymbol{\sigma} \rangle_{+} + \langle tr \boldsymbol{\sigma} \rangle_{-}) \mathbb{1}$ $\widetilde{\boldsymbol{\sigma}} = [\boldsymbol{H}\boldsymbol{\sigma}'\boldsymbol{H}]' + \frac{1}{3} (g(\boldsymbol{H})\langle tr \boldsymbol{\sigma} \rangle_{+} + \langle tr \boldsymbol{\sigma} \rangle_{-})$ $\widetilde{\boldsymbol{\sigma}} = \mathbb{E}: \boldsymbol{\varepsilon}^{e}$
Viscoplastic law coupled with damage	Elasticity criterion $f = \tilde{\sigma}_{eq} - R - \sigma_y$ $\varepsilon = \varepsilon^e + \varepsilon^p$ Isotropic hardening $R = R_{\infty}(1 - e^{-br})$ $\tilde{\sigma}_{eq}$ Von MisesFlow rule $\begin{bmatrix} \dot{e}^p = \dot{\lambda} \frac{\partial f}{\partial \tilde{\sigma}} = \frac{3}{2} \dot{\lambda} \frac{\tilde{\sigma}'}{\tilde{\sigma}_{eq}} \\ \dot{\varepsilon}^p = \dot{\lambda} \frac{\partial f}{\partial \sigma} = [H\dot{e}^p H]'$ ; $\dot{p} = \left(\frac{2}{3} \dot{\varepsilon}^p; \dot{\varepsilon}^p\right)^{1/2}$
	Norton viscosity law $\dot{r} = -\dot{\lambda} \frac{\partial f}{\partial R} = \dot{\lambda} = \left\langle \frac{f}{K_N} \right\rangle_+^N$

- $\sigma_y$ , : yield criterion
- $R_{\infty}$ , b : isotropic hardening parameters
- N, K<sub>N</sub> : Norton parameters

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## **CONSTITUTIVE MODELLING – 9%CR**

Elasticity coupled with anisotropic damage [Lemaitre et. al, 2000]	$\boldsymbol{\varepsilon}^{e} = \frac{1}{2G} (\boldsymbol{H}\boldsymbol{\sigma}'\boldsymbol{H})' + \frac{1}{9K} (g(\boldsymbol{H})\langle tr \boldsymbol{\sigma} \rangle_{+} + \langle tr \boldsymbol{\sigma} \rangle_{-}) \mathbb{1}$ $\widetilde{\boldsymbol{\sigma}} = [\boldsymbol{H}\boldsymbol{\sigma}'\boldsymbol{H}]' + \frac{1}{3} (g(\boldsymbol{H})\langle tr \boldsymbol{\sigma} \rangle_{+} + \langle tr \boldsymbol{\sigma} \rangle_{-}) \mathbb{1}$ $\widetilde{\boldsymbol{\sigma}} = \mathbb{E}: \boldsymbol{\varepsilon}^{e}$
Viscoplastic law coupled with damage	$f = \widetilde{\sigma}_{eq} - R - \sigma_y \qquad \varepsilon = \varepsilon^e + \varepsilon^p$ $R = R_{\infty} (1 - e^{-br})$ $\dot{e}^p = \dot{\lambda} \frac{\partial f}{\partial \widetilde{\sigma}} = \frac{3}{2} \dot{\lambda} \frac{\widetilde{\sigma}'}{\widetilde{\sigma}_{eq}}$ $\dot{\varepsilon}^p = \dot{\lambda} \frac{\partial f}{\partial \sigma} = [H\dot{e}^p H]'  ;  \dot{p} = \left(\frac{2}{3}\dot{\varepsilon}^p; \dot{\varepsilon}^p\right)^{1/2}$ $\dot{r} = -\dot{\lambda} \frac{\partial f}{\partial R} = \dot{\lambda} = \left\langle\frac{f}{K_N}\right\rangle_+$
Damage law $A, r_{threshold}$ damage parameters $\mathcal{H}$ Heaviside function	Stress triaxiality function $ \frac{\mathrm{d}}{\mathrm{dt}}(H^{-2}) = A\tilde{R}_{\nu} \mathcal{H}(r - r_{threshold})\langle \dot{\boldsymbol{e}}^{p} \rangle_{+} $ $ \tilde{R}_{\nu} = \frac{2}{3}(1 + \nu) + 3(1 - 2\nu) \left\langle \frac{\tilde{\sigma}_{H}}{\tilde{\sigma}_{eq}} \right\rangle_{+}^{2} ; \tilde{\sigma}_{H} = \frac{1}{3}tr(\tilde{\boldsymbol{\sigma}}) $ $ \frac{1}{17} = \frac{1}{3}tr(\tilde{\boldsymbol{\sigma}}) + \frac{1}{17} = \frac{1}{17}tr(\tilde{\boldsymbol{\sigma}}) + \frac{1}{17} = \frac{1}{17}tr(\tilde{\boldsymbol{\sigma}}) + \frac{1}{17}tr(\tilde{\boldsymbol{\sigma})} + \frac{1}{17}tr(\tilde$

## **CONSTITUTIVE MODELLING – 9%CR 650°C**



Model parameters identified with ring test data on a single element



Accumulated plastic strain r

Plastic strain is not homogeneous in the gauge section



Accumulated plastic strain r

- Plastic strain is not homogeneous in the gauge section
- Plastic strain varies through the thickness of the ring



Accumulated plastic strain r

- Plastic strain is not homogeneous in the gauge section
- Plastic strain varies through the thickness of the ring
- Plastic strain rate aslo varies through the thickness of the gauge section



Damage occurs first on the inside of the ring, at the center of the gauge length
 Damage location, linked to stress triaxiality, which is greater in the same area

Fracture may occur earlier than what the macroscopic strain would suggest

## **STRESS-STRAIN CURVE COMPARISON**



- For similar displacement, rupture occurs well before the experimental result
- Damage evolution is faster due to strain localization and high stress triaxiality

### > The criterion $r_{threshold}$ for damage initiation has to be higher

## **STRESS-STRAIN CURVE COMPARISON**



> Better prediction after  $r_{threshold} = 0.175$ 

- > Adjusted  $r_{threshold} < 0.25$  tile specimen damage threshold
  - > Other possible parameters to change :  $\eta$ , A (ongoing work)
  - Higher threshold values incompatible with ring creep tests
  - Possible anisotropic damage mechanisms

[Tanno et al.]



## CONCLUSIONS

- The notched ring tensile specimen, in the load conditions chosen, introduces perceived anisotropy in raw test results (total elongation)
- Constitutive modelling and FE simulation of the ring tensile test analysis allow a more rigorous parameter identification
- Macroscopic values such as total elongation can be adjusted
- The understanding of geometry-dependent anisotropy enables the study of material-dependent anisotropy in the 14%Cr ODS steel tube



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