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Chapter 4

Surface Modifications of Nanodiamonds and Current Issues for Their Biomedical Applications

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Abstract Combining numerous unique assets, nanodiamonds are promising nanoparticles for biomedical applications. The present chapter focuses on the current knowledge of their properties. It shows how the control of their surface chemistry governs their colloidal behavior. This allows a fine tuning of their surface charge. Developments of bioapplications using nanodiamonds are summarized and further promising challenges for biomedicine are discussed.

4.1 Introduction

Nanoparticles (NPs) are currently designed and developed for nanomedecine applications which constitute an emerging research field [1]. Among the main stakes, cancer therapy expands with three main paradigms: drug delivery, biomarkers for diagnosis and therapeutics.

The main interest to use NPs for drug delivery is the ability to target cancer cells. Indeed, some tumors present a higher vascularisation which could favor a passive targeting of NPs via the EPR effect (Enhanced Permeability and Retention) [2]. The targeting can also be active when a specific grafting of NPs with specific markers is performed [3]. The use of NPs can enhance the treatment efficiency allowing the delivery of higher doses while it reduces significantly side effects. Polymeric NPs

[4] and liposomes are currently the most used NPs to encapsulate drugs. The polymer degradation permits their progressive delivery.

Second, NPs can also act as biomarkers for tumor imaging or tracking via their optical properties as quantum dots [5] or gold NPs via surface plasmon resonance [6]. However, some of these luminescent probes are limited either by their photostability or their cytotoxicity. Inorganic NPs can be used as contrast agents for magnetic resonance imaging [7, 8]. Lastly, C¹⁴ grafting on carbon nanotubes wasreported conferring radiolabeling properties for exploration of the in vivo biodistribution and pharmacokinetics [9].

Third, NPs can have therapeutic roles if they could be activated by an external stimulus. For example, under an external magnetic field, metallic NPs can generate temperature elevation called hyperthermia leading to cell death [10]. This activation can also be performed thanks to optical properties i.e. phototherapy for graphene oxide nanosheets [11, 12] or metallic NPs [13]. Efficiency of radiotherapy treatments can be enhanced by coupling with NPs. This constitutes a promising way to treat resistant tumors. Such approach may permit the delivery of lower doses for the same efficacy. A radiosensitization effect can be obtained under irradiation producing Reactive Oxygen Species (ROS) in cells. For this application, gold [14], platinum [15] or rare earths [16] NPs are used due to their high ratio of Auger electrons generated under irradiation.

The current trend which associates therapeutic and diagnostic properties is called theranostics [17, 18]. For this purpose, hybrid nanoparticles are designed combining properties of their core and shell: for example, superparamagnetic NPs embedded in mesoporous silica [19] or gold NPs in liposomes [20].

Nanodiamonds (NDs) constitute promising NPs for biomedicine applications. They possess many unique assets. They are scalable with sizes ranging from 100 nm down to 5 nm. One can expect the elimination by kidney for the smaller NDs (diameter lower than 6 nm) [21]. Nanodiamonds are also available in large quantities at reasonable prices (< \$1/g). Several studies reported their very low toxicity related to the high chemical inertia of diamond [22–25]. Recent research demonstrates they are even biocompatible with advantages for drug delivery or fluorescence labels [26, 27]. NDs enable covalent grafting of various chemical moieties on their surfaces [28] resulting in stable colloidal aqueous solutions. Covalent chemistry works similar way on NDs as for organics.

Their surface charge can be tuned allowing the electrostatic adsorption of biomolecules such as siRNA [27, 29]. They have been used for drug delivery applications [30–32].

Moreover, NDs can act as biomarkers for diagnosis. Indeed, color centers can be generated from nitrogen present in the diamond core by electron or helium irradiation [33] followed by a thermal annealing. Under green illumination, produced nitrogen-vacancy (NV) centers emits in the red wavelength region, they are highly photostable presenting neither photoblinking nor photobleaching [34].

Potential therapeutic behaviors of NDs related to their specific surface properties are currently under investigation. Investigations of nanodiamonds for biomedical applications started quite recently. Numerous parameters remain partially controlled such as their different surface chemistry, their surface charge, their colloidal stability and the role

of facets in their surface properties. The present chapter summarizes the current knowledge and the assets of NDs for nanomedecine. First, the different ways to produce nanodiamonds will be briefly presented (Part 4.2). Then, the characterization tools of their core and surface will be detailed focusing on the most relevant characterization methods (Part 4.3). The main surface modifications performed using different approaches (chemical, physical or plasma treatments) will be reviewed (Part 4.4). Electronic properties of modified nanodiamonds will be presented as well as their colloidal behaviors in water (Part 4.5). Finally, main assets of NDs for biomedical applications will be discussed before giving some future promising challenges (Part 4.6).

4.2 Production of Nanodiamonds

According to thermodynamics, graphite is the stable carbon form at ambient conditions while diamond is metastable [35, 36]. Since 1950s, synthetic diamond can be obtained by high pressure high temperature (HPHT) method using hydraulic press which reproduces the conditions of natural diamond formation [37]. In 1980s, chemical vapor deposition (CVD) diamond was discovered and developed [38]. Different techniques are presently used to fabricate nanodiamonds: either NDs could be obtained by milling of HPHT/CVD diamond or NDs could be synthetized using explosives or laser ablation of graphite precursors [39, 40].

Monocrystalline nanodiamonds with size down to 20 nm can be obtained by ball milling of micron-sized HPHT diamonds. High Resolution Transmission Electron Microscopy (HRTEM) revealed faceted shapes with sharp edges corresponding to fractured crystallographic planes. In few cases, quasi spherical nanodiamonds smaller than 10 nm were obtained by HPHT milling [41, 42]. More recently, 70–80 nm nanodiamonds were also produced from polycrystalline CVD diamond by bead assisted sonic disintegration (BASD) [43].

Alternatively, nanodiamonds could be synthetized by several dynamic processes from molecules of explosives and different carbon precursors [44] (Fig. 4.1). Three methods could be distinguished: the direct transformation of graphite by an external shock wave, the detonation of graphite mixed with explosives (hexogen/RDX) or the detonation of high energy explosives (TNT/hexogen/RDX). For detonation synthesis, the key parameters governing the yield of nanodiamonds are the heat capacity, the content and the reactivity of the medium, the cooling of detonation products [45]. Nanodiamonds produced by detonation usually exhibit smaller diameters ranging typically from 2 to 20 nm. Several models were proposed to explain their formation by detonation synthesis [45]. More details about ND synthesis by detonation methods could be found in previous reports [45, 46].

After detonation, the collected soot contains metals, metal oxides and carbides. In addition to carbon, its elemental composition reveals hydrogen, nitrogen and oxygen species. Indeed, nitrogen is initially present in HPHT diamond and in explosives used in the detonation synthesis. In most cases, a-C and/or sp² carbon are surrounding the diamond core. Detonation NDs are extracted after purification procedures using liquid oxidants to remove non-diamond carbon. Indeed, diamond and non-diamond carbon have different stability to oxidants. The efficiency of an ozone treatment was also reported to eliminate the non-diamond carbon [46]. Then, liquid-phase oxidation using HCl or other high boiling acids is applied to eliminate the non-carbon impurities. The resulting surface chemistry of detonation nanodiamonds will be described in the next paragraph (Part 4.3).

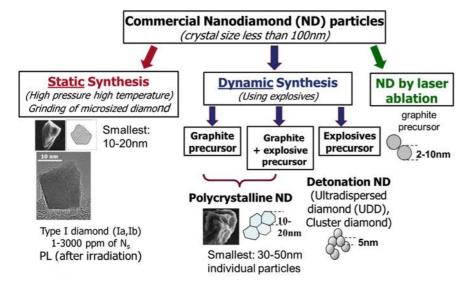


Fig. 4.1 The different natures of NDs [44] courtesy of O. Shenderova, International Technology Center, USA

Diamond nanoparticles have been classified according to their size [44]. Nanocrystalline particles (10–100 nm) could be either monocrystalline (HPHT milling) or polycrystalline after shock-wave compression of graphite or detonation of carbon/explosives mixture. Ultrananocrystalline NDs (2–10 nm) are mainly produced by detonation or laser ablation. Smallest particles behave to the family of diamondoids. Higher diamondoids (1–2 nm) are extracted from petroleum with hydrogen termination [47]. The smallest specie of cubic diamond, the adamantane, contains 10 carbon atoms ($C_{10}H_{16}$).

4.3 Characterization Tools

Depending on their synthesis methods (Part 4.2), nanodiamonds can behave different core-shell structures, morphologies and surface chemistry. Experimental and theoretical works reported multi-shell structures based on a diamond core which could be surrounded by defective diamond, amorphous carbon and/or several onion like outershells [48, 49]. According to calculations, this structure constitutes the most stable energetic form for sp³ nanometric clusters [50]. Characterization methods were developed to specifically probe the crystalline structure as well as the chemical composition of diamond core, outer shells and surface of nanodiamonds. The most relevant techniques to assess these characteristics are presented in Table 4.1.

Table 4.1 Characterization tools of nanodiamonds

Diamond core		
Particle size, crystalline structure	HRTEM, XRD, SAXS	
Morphology and shape	HRTEM	
Structural defects	HRTEM	
Chemical impurities	XPS, EELS/STEM	
Local environment	NEXAFS, ESR, NMR, ELNES	
NV centers	Time resolved PL, confocal PL	
Outer shells and surface		
a–C, sp ² carbon	Raman, HRTEM, XRD, XPS	
Surface chemistry and grafting	FTIR, XPS, TDMS	

4.3.1 Diamond Core

Its crystalline structure can be accurately investigated by high resolution transmission electron microscopy (HRTEM) in bright field mode. The signature of diamond lattice is also confirmed in the diffraction mode. Moreover, structural defects as twins or dislocations within diamond lattice can be resolved especially for nanodiamonds produced by detonation method [51]. The spherical or faceted shape of nanodiamonds which could greatly influence their reactivity could be characterized at the nanoscale using HRTEM. As an illustration, (111) facets with (100)-type truncations were identified even for detonation NDs of 2 nm [51] (Fig. 4.2). More recently, modern HRTEM instruments were developed to operate at low electron energy down to 40 keV. Lens aberrations are reduced allowing image resolution down to 50 pm [52]. In these experimental conditions, the graphitization of NDs which may occur under electron beam is completely avoided. Diamond core structure and size of nanodiamonds can be also investigated by X-ray Diffraction (XRD) [53] or Small Angle X-ray Scattering (SAXS) [54]. Contrary to XRD which is limited by diffraction effects leading to physical broadening, SAXS give access to sizes very close to HRTEM. Size distributions can even be extracted from SAXS data assuming a shape for NDs.

The concentration of chemical impurities (N, O, H, metals remaining from synthesis) within diamond core and their location in the diamond lattice is another important issue. For example, the nitrogen presence could confer optical properties to nanodiamonds via the formation of luminescent nitrogen vacancy (NV) centers. Atomic concentrations of impurities (excepted hydrogen) could be assessed from X-ray Photoemission Spectroscopy (XPS). Typically, the detection limit for oxygen and nitrogen is close to 0.5 at % [55]. Very few techniques such as SIMS or NMR are sensitive to hydrogen. During synthesis of nanodiamonds, hydrogen is likely to be inserted into diamond core. Indeed, hydrogen diffusion was previously reported in bulk diamond [56]. There is a crucial need to probe hydrogen into diamond core because it could confer new properties to nanodiamonds.

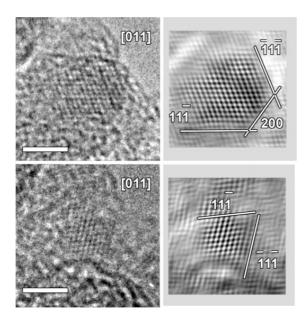


Fig. 4.2 HRTEM pictures of detonation nanodiamonds, scale bar 2 nm [51] courtesy of S. Turner, University of Antwerp

Electron spin resonance (ESR) is the most sensitive method to detect nitrogen (10¹² spins) [57, 58]. Indeed, single nitrogen in substitution in diamond lattice is paramagnetic [59]. Core loss electron energy loss spectroscopy (EELS) profiles can also probe light elements as nitrogen, oxygen, boron or carbon. Energy loss near edge structures (ELNES) technique provides a signature of the local binding of impurities. For example, the tetrahedral configuration can be deduced from the nitrogen K edge structure [51, 60]. Recent work shows this technique permits to confirm the tetrahedral insertion of boron atoms in diamond leading to p-doping [61, 62]. EELS combined with scanning TEM (STEM) can even provide mapping or profiles of impurities with an atomic resolution [63–65]. Near-edge X-ray absorption fine structure (NEXAFS) technique could probe the local bonding environment [66]. Nuclear Magnetic Resonance permits to probe hydrogen (¹H), nitrogen (¹⁵N) and carbon (¹³C) spins [67].

Photoluminescence (PL) spectroscopy can reveal signature of structural imperfection of diamond lattice (defects, impurities). Luminescent NV centers composed by a nitrogen atom in substitution combined with a neighbouring vacancy are characterized by PL. Under excitation at 531 nm, emission wavelengths at 575 and 638 nm are detected corresponding to different charge states NV⁰ and NV⁻, respectively [57]. NV centers exhibit a high stability without any photoblinking or photobleaching. Recently, the influence of NDs surface chemistry on NV luminescence was shown, especially a quenching of the NV⁻ luminescence when nanodiamond surface is hydrogenated [68].

4.3.2 Outer Shells and Surface Chemistry

Surface chemistry of nanodiamonds is strongly dependent on their synthesis method and the purification treatments (Part 4.2). For pristine NDs, the diamond core is usually coated with non sp³ carbon. These carbon outer-shells could be structured as onion-like shells [69] or could exhibit an amorphous structure (a–C) [70]. The nature of outer shells must be well characterized because it strongly affects the surface reactivity of nanodiamonds. Raman spectroscopy gives access to composition and structure for bulk carbon phases [71]. A signature at 1332 cm⁻¹ is obtained for diamond while D and G bands at 1410 and 1590 cm⁻¹ originate from amorphous and graphitic carbon, respectively. Nevertheless, a specific Raman interpretation is needed for detonation nanodiamonds (size lower than 6 nm) taking into account a phonon confinement leading to a downshift and an asymmetry of the diamond Raman peak [72, 73]. Some features like the one located at 1640 cm⁻¹ are still a matter of debate [36]. XPS is also sensitive to sp² and sp³ hybridization of carbon, it was recently used to monitor the surface graphitization of nanodiamonds [55]. However, artefacts due to nanomaterials especially the high fraction of surface versus bulk must be taken into account in the data interpretation [74, 75]. Auger Electron Spectroscopy (AES) is even more sensitive to carbon hybridization because two valence electrons are involved in the C KVV transition [70]. Using low loss EELS, a–C or graphite could also be discriminated [76]. HRTEM imaging provides the structure of the outer-shells at atomic scale [51, 75].

After their synthesis, nanodiamonds underwent oxidizing treatments to clean metallic impurities (Part 4.2). This leads to a heterogeneous surface chemistry involving different carbon-oxygen groups. To investigate the surface chemistry of nanodiamonds after purification or functionalization, Fourier Transformed InfraRed (FTIR) Spectroscopy is a powerful technique as diamond core is transparent to IR. It is sensitive to the vibrations of carbon-hydrogen (C–H_x) [77] and the different carbon-oxygen bonds (O–H, C–O, C–O–C, C=O). Moreover, it could detect

C-N or N-H vibrations [78]. Although FTIR is usually done in transmission mode, a diamond-coated ATR prism for infrared absorption spectroscopy was recently used to characterize surface-modified diamond nanoparticles [79]. Raman spectroscopy performed under UV light could provide signature of C-H surface bonds for hydrogenated diamond surface [80, 81]. High resolution electron energy loss spectroscopy (HREELS) can probe the surface bonds of the first atomic plane with a resolution of 5 meV. It was recently used to characterize the surface chemistry of hydrogenated nanodiamonds [76]. NMR experiments are sensitive to ozone-treated or hydroxylated nanodiamonds as well as functionalization with fluorine (¹⁹F) [67].

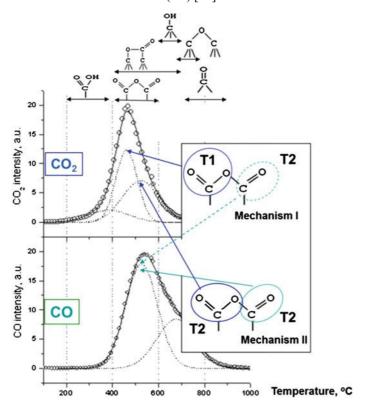


Fig. 4.3 Two possible desorption mechanisms of anhydrides on ozone treated NDs for two temperatures from TDMS [46] courtesy of O. Shenderova, International Technology Center, USA

Finally, as an indirect surface chemistry probe, let cite the thermal desorption mass spectrometry (TDMS) analysis which is a continuous measurement of the mass spectra of gases released from a sample under programmed heating [82, 83]. This technique permits a fine identification of functional groups according to their specific desorption temperature and gives information concerning their thermal stability. As an illustration, TDMS spectra can help to understand the anhydrides desorption from ozone purified NDs [46] (Fig. 4.3).

4.4 Surface Modifications of Nanodiamonds

Nanodiamond surface appears highly heterogeneous exhibiting different carbonoxygen functional groups (ether, hydroxyl, carbonyl, carbonyl, a—C and/or sp² carbon. This scattered surface chemistry is mainly related to purification steps (Part 4.3). Surface treatments were developed to get homogeneous chemistry allowing more efficient further graftings [28] either by chemical, thermal or plasma approaches. In addition, these controlled surface chemistries help to stabilize NDs in aqueous suspension limiting aggregation (Part 4.5). Indeed, detonation NDs tend to aggregate while many applications require isolated primary NDs [36]. Milling or BASD procedures [84] were successfully applied for deaggregation. Nevertheless, these treatments can induce either chemical contaminations or surface graphitization. The efficiency of surface treatments such as oxidation in air [85] or plasma hydrogenation [86] was reported. Indeed, tuning the surface chemistry can confer intrinsic electronic properties to NDs surface having consequences on their aggregation and colloidal behavior (Part 4.5). Thermal treatments permit to create hybrid nanoparticles constituted by a diamond core surrounded by sp² reconstructions [75]. Table 4.2 presents the major reported terminations and their uses.

Table 4.2 Surface terminations of NDs vs treatments

Surface terminations		Treatments	Objectives	
Hydrogenated		H ₂ plasma [70, 87]	positive ZP with SCL	
		annealing at 500 °C under H ₂ [88]	grafting	
Oxidized	Carboxylated	different acid treatments [28]	negative ZP	
		air annealing at 400-430 °C [89]	disaggregation	
		ozone treatment at 150–200 °C [46]	grafting of peptides	
	Hydroxylated	borane reduction [90]	grafting	
		Fenton reagent [91]	silanization	
		milling with beads [69, 92]		
		photochemistry [93]	1	
Fluorinated		F ₂ /H ₂ exposure at 150–470 °C [94]	grafting	
		CF ₄ plasma [95]		
Aminated		Cl-term NDs in gaseous ammonia [96]	positive ZP grafting	
		covalent grafting of amine derivative [84, 97, 98]		
Surface graphitised		annealing at 750 °C under vacuum [55, 99, 100]		
		long beads milling [101]	hybrid properties	

4.4.1 Surface Hydrogenation of Nanodiamonds

The aim is to generate homogeneous C–H bonds at nanodiamond surface. Hydrogenated intrinsic diamond surface behave specific electronic properties characterized by a negative electron affinity (NEA) [102]. After air exposure, it presents a surface conductivity by a transfer doping involving adsorbed species [103]. Hydrogen atoms may also diffuse into the diamond lattice where they are preferentially trapped on structural defects or chemical impurities [56].

Annealing treatments under H₂ flow at 850–900 °C were reported for diamond particles bigger than 100 nm [104, 105]. Nevertheless, this temperature range is very close to the one leading to graphitization for smaller nanodiamonds. Recent reports demonstrated that surface graphitization started at 750 °C for 5 nm detonation NDs [55, 99]. Similar treatments performed at lower temperature (typically 500 °C) seem to produce an efficient deaggregation of detonation NDs after centrifugation. Nevertheless, HRTEM reveals that onion like shells are still present after the annealing [88]. FTIR shows a partial coverage by C–H bonds conferring a hydrophobic behavior to treated nanodiamonds [88, 106]. These results suggest hydrogen is more likely bonded to sp² carbon. Another method for hydrogenation would use organic chemistry with metal catalyst such as Pd, Pt or Ni [107, 108]. However, in that case, catalysts must be further eliminated.

An interesting alternative consists to expose NDs to H₂ CVD plasma. Under microwave field or thermal cracking [97, 109], dihydrogen molecules are efficiently dissociated into atomic hydrogen. This specie has a high reactivity towards carbon and etches sp² carbon easier than sp³ carbon [110]. Thus, its interaction with nanodiamonds may allow the loss of non-diamond carbon, the reduction of oxygen species at the surface and the formation of C sp³-H bonds. This surface hydrogenation by CVD plasma is widely used for diamond layers to get hydrogen-terminated boron doped diamond for electrochemistry [111, 112]. Optimized CVD conditions for hydrogenation were determined for detonation nanodiamonds using a Microwave Plasma Chemical vapor Deposition (MPCVD) reactor connected to a UHV set-up [70]. Nanodiamonds were deposited on multilayer Silicon On Insulator (SOI) substrates by drop-casting and samples were analyzed without air exposure. A XPS sequential study allows the monitoring of the oxygen removal after MPCVD exposure. A temperature threshold of 700 °C is needed to fully remove oxygen from detonation NDs. The hydrogenation efficiency is closely related to the concentration of atomic hydrogen produced in the MPCVD plasma [70].

Proper conditions were then used in a home-made set-up allowing plasma hydrogenation of NDs in powder (Fig. 4.4). Nanodiamonds located in a quartz tube are directly treated in the gas phase [87]. A rotation of this tube allows the exposure of their whole surface to atomic hydrogen. This method is able to treat simultaneously a large amount of NDs (hundreds mg) which is a prerequisite to a meaningful surface functionalization. Both HPHT and detonation nanodiamonds which differ by their size and the graphitic shells surrounding the sp³ core, have been hydrogenated.

The surface chemistry of hydrogenated nanodiamonds (H-NDs) was then characterized using XPS and FTIR (Fig. 4.4b). For dried H-NDs, FTIR spectra well exhibit signals located around 2900 cm⁻¹ [87]. These bands have been previously assigned to C–H stretching on hydrogenated NDs surfaces [77].

The surface reactivity of HPHT H-NDs (50 nm) was investigated applying several surface modifications [93]: a selective oxidation under UV exposure and functionalization with alkenes and diazonium moieties. These reactions are common on hydrogenated diamond films towards efficient biological functionalizations (DNA, proteins grafting) [113–115] so their use on NDs may also lead to promising developments. However, the chemical mechanisms involved in these grafting routes require specific surface properties, involving charge transfer, which is only provided to the diamond surface by C–H hydrogen terminations [116].

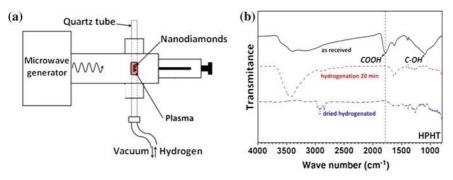


Fig. 4.4 a Set-up for plasma hydrogenation; b FTIR spectra of hydrogenated HPHT NDs [87]

On H-NDs, the oxidation procedure under UV gives rise to a surface hydroxylation as shown by the FTIR bands at 1050 and 3300 cm⁻¹. In addition, a new component appears in the XPS C1s core level [93]. A similar modification of the C1s peak was reported for a diamond surface after UV oxidation [117]. This oxidation treatment appears selective on H-NDs as neither carboxyl nor carbonyl groups were evidenced by XPS or FTIR. An inversion of the Zeta potential was measured after UV oxidation +30 mV compared to -45 mV for as received NDs. The origin of this positive Zeta potential will be discussed in the Part 4.5.

HPHT NDs were also reacted with pure undecylenic acid under 254 nm UV irradiation. This alkene was chosen for its long alkyl chain and its acid group, both functionalities easily identifiable by FTIR analysis with characteristic structures at 2800 cm⁻¹ and 1700 cm⁻¹ respectively (Fig. 4.5). Furthermore, the vinyl function is evidenced by a peak located around 900 cm⁻¹. For comparison, the same reaction was also conducted on as-received NDs.

The different signatures of undecylenic acid are observed only on the FTIR spectra for hydrogenated HPHT NDs (Fig. 4.5). Moreover, the essential role of UV light in the mechanisms of grafting was demonstrated. On diamond layers, numerous studies report that C–H strongly enhances chemical reaction [116] leading to the formation of covalent C–C bonds between the diamond surface and the grafted moiety. Indeed, hydrogenated terminations confer to the diamond layers specific electronic surface properties, such as a superficial conductive layer (SCL) [118] and a negative electron affinity (NEA)

[102, 119]. The latter is usually highlighted to explain the reactivity of hydrogenated diamond layers towards alkenes, by the mean of photo-excited electrons allowed to propagate out of the diamond into the nearby liquid phase, even with sub-bandgap excitation [120, 121]. Authors agree that hydrogen terminations are required for this reaction, and that electron transfer arises from sp³ surfaces and not from grain boundaries and/or sp² species at the interface. This constitutes a strong indirect proof of the efficiency of hydrogen plasma on NDs.

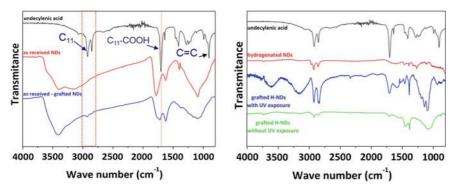


Fig. 4.5 FTIR spectra on (*left*) as received HPHT NDs (*right*) HPHT H-NDs [93]

The spontaneous grafting of a diazonium salt on hydrogenated HPHT NDs was also demonstrated [93]. On a diamond surface, this arises from specific electronic properties conferred by C–H terminations [122]. The covalent bonding goes through the creation of a phenyl radical by the mean of an electron transfer from the diamond surface to the aryldiazonium salt. This electron transfer is related to the presence of a superficial conductive p-type layer. The commonly accepted mechanism, so-called transfer doping [118] describes an accumulation of holes due to an

electron transfer from the valence band and a redox couple adsorbed on the surface when the diamond surface is exposed to wet atmosphere or immersed in an electrolyte. This phenomenon is only possible on hydrogenated layers which exhibit a low ionization potential, matching with the chemical potentials of the adsorbates.

These different surface modifications of hydrogenated HPHT NDs suggest their chemical reactivity is very similar to the one of hydrogenated diamond surfaces. In the Part 4.5, surface properties of hydrogenated nanodiamonds will be discussed.

4.4.2 Oxidation of Nanodiamonds

The surface chemistry of pristine nanodiamonds is highly heterogeneous (Part 4.3). The selective formation of C–O bonds at the surface by applying specific treatments is a major challenge to obtain more efficient graftings.

Among C–O groups, carboxylic acid termination has a specific interest. Indeed, its basic form (COO⁻) confers a negative charge to the NDs over a wide pH range in water [75]. This ensures stable colloidal suspensions (Part 4.5). Surface of NDs could be saturated with carboxyl functions using nitric (HNO₃), sulfuric (H₂SO₄) or perchloric (HCl) acids. A mixture of these acids in equal amounts reveals powerful [45, 89, 123]. "Piranha" water (sulfuric acid and hydrogen peroxide) constitutes an efficient alternative [124]. Air annealing of NDs at 400–450 °C leads to the formation of acid functions mainly carboxylic and to the etching of non-diamond carbon [89, 125, 126]. A quantitative analysis of COOH groups was performed using Boehm titration after acid treatment [127]. The mean amount corresponds to 0.85 COOH groups par nm² of ND surface. Recently, ozone purification treatment was reported leading to the formation of anhydrides [46].

Starting from pristine nanodiamonds, several procedures inducing hydroxyl terminations were reported. First, chemical reduction can be performed with borane (BH₃) or hydrides (LiAlH₄) starting from COOH [128, 129]. Second, C-OH bonds can also be formed using the "Fenton reaction" which takes place in a mixture of hydrogen peroxide and iron sulfate in strong acid solution [91, 130]. Third, a significant amount of hydroxyl groups can be generated during milling [69] or BASD [84] of NDs in water. More recently, a photochemical oxidation under UV was performed on NDs previously hydrogenated [93].

A recent review provides more details on the different oxidation procedures [28].

4.4.3 Amination, Fluorination or Chlorination of Nanodiamonds

Several attempts to generate fluorine, amine or chlorine terminations at NDs surface by annealing under controlled atmosphere were reported [36, 94, 131, 132]. Nevertheless, annealing temperatures are usually limited to avoid surface graphitization (Part 4.4.4). As a consequence, some oxidized groups and non-diamond carbon may remain at the surface. The selectivity of these gaseous treatments is often limited: after heating in ammonia flow, aminogroups, C=N and C=N-containing groups were identified at ND surface [132, 133]. In order to create more reactive species, a CF₄ atmospheric pressure plasma was used [95]. Chlorination of NDs was also successfully achieved by photochemical reaction of gaseous chlorine with NDs previously hydrogenated [96]. The interaction of these chlorinated NDs with ammonia led to the formation of NH₂ groups.

4.4.4 Surface Graphitization of Nanodiamonds

Diamond graphitization can be induced by different mechanisms: via thermal treatment under vacuum [134, 135] or exposure to reactive gas under ambient atmosphere [136–138]. It can be also obtained by beam irradiation (electron, ion, laser, gamma-ray) [139–141]. Finally, the graphitization could be catalyzed by metals as Fe or Co [142–144].

A recent review was devoted to diamond phase transitions at nanoscale [145]. For nanodiamond, the graphitization process can lead to a full transformation into onion like carbon (OLC) structures formed by concentric closed graphitic shells. Corresponding kinetics were extensively studied experimentally by annealing at temperatures included between 1100 and 1900 °C in different atmospheres (vacuum, oxygen, argon, hydrogen) [100, 111, 146–150]. Formations of curved graphite like structures and concentric-shells fullerenes on nanodiamonds were theoretically investigated [151–153]. For nanodiamonds 4–5 nm in size, the evolution of carbon bulk density measured by a gamma-ray attenuation method drops from 3.1 to 2.2 g cm⁻³ in the 900–1300 °C temperature range [154]. By comparison, the diamond/ graphite transition occurs above 1600 °C for bulk diamond [154, 155]. Starting from detonation nanodiamonds, HRTEM revealed the formation of fullerene like shells after an annealing at 1100–1300 °C [146]. At 1500 °C, nanodiamonds are completely converted into OLC structures while polygonal hollow onions are formed at temperatures higher than 1870 °C (Fig. 4.6). OLC have promising properties for energy storage, catalysis or composites [147–149, 156].

The graphitization mechanism is strongly sensitive to the presence of non-diamond carbon at the NDs surface [145]. Indeed, detonation NDs underwent purification procedures which can differ from a supplier to another (Part 4.2). This can explain the wide dispersion for the onset temperature of nanodiamonds graphitization reported in the literature, it varies from 670 to 1000 °C [157, 158]. The selective desorption of oxygen groups has also a significant effect on the graphitization onset. For HPHT NDs, a size effect is expected with a higher temperature suitable to initiate graphitization. Graphitization kinetics for submicron diamond and NDs were compared, different graphitization rates were determined [154].

Moreover, the previous report demonstrates that activation energies differ for {110} and {111} crystallographic facets [145, 159].

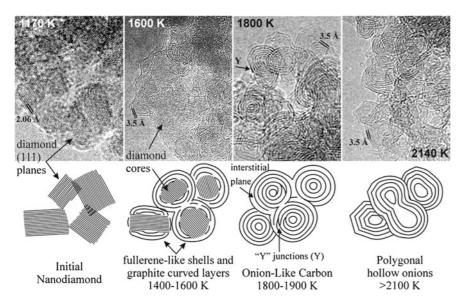


Fig. 4.6 Transformation of NDs into onion like carbon [146] courtesy of Y.V. Butenko, ESTEC, The Netherlands

Recently, several groups focused on the early stages of surface graphitization of detonation NDs using XPS, HRTEM, Raman or NMR characterizations [55, 75, 99, 160, 161]. The aim is to control the formation of hybrid nanoparticles with a sp² organized surface covering a sp³ core, thus avoiding the graphitization of the diamond core. Such hybrid NPs may have promising properties linked to fullerene or graphene assets as radiosensitization or photothermal therapy [162, 163]. In addition, surface graphitized nanodiamonds exhibit electrical conductivity sufficient for electrochemical applications [164, 165].

Surface graphitization of detonation NDs by annealing under UHV was monitored using XPS [55]. The evolution of the carbon core level (C1s) shows the existence of two regimes according to annealing temperature T (Fig. 4.7). For T included between 700 and 900 °C, surface graphitization is initiated from non-diamond carbon present at ND surface. Fullerene like reconstructions (FLRs) are formed as shown by HRTEM [75]. On the other hand, for T > 900 °C, the diamond core starts to graphitize leading to the formation of bucky diamond. It is thus possible to generate FLRs at the surface without altering the diamond core. The stability of FLRs at the NDs surface was previously demonstrated by ab initio calculations confirmed by x-ray absorption and emission [152]. The selective synthesis of a thin graphitic layer on the ND surface by annealing under vacuum at temperature lower than 900 °C gives rise to hybrid nanocarbons combining the intrinsic core properties of diamond with the surface reactivity of sp²-based nanomaterials. These results were recently confirmed by a Raman study of the sp³ to sp² conversion for detonation NDs [160]. NMR and EPR investigations detect the early stages of graphitization for annealing temperatures included between 600 and 800 °C [161].

Proper conditions determined for surface graphitization were then reported in a set-up where nanodiamonds are annealed under vacuum in a crucible. Annealed NDs are dispersable in water, their colloidal properties will be discussed in the Part 4.5. HRTEM observations confirm the structure modifications previously described showing FLRs formation [75]. A specific chemical reactivity was reported for detonation NDs after annealing at 750 °C exhibiting FLRs structures allowing arylation reactions [99].

In conclusion, surface chemistry of nanodiamonds can be tuned by thermal annealing, plasma exposure or chemical reactions. Hydrogenated and carboxylated NDs are currently well controlled whereas hydroxylation remains more difficult to achieve. These terminations are particularly suitable for the grafting of biomolecules or drugs on NDs [28, 36]. Sp² reconstructions can be generated at detonation NDs surface without altering the diamond core [55]. Arylation reactions allow their functionalization with complex organic moieties for bioapplications [99]. On the other hand, further progresses are needed to control the selectivity of amination, fluorination and chlorination of NDs.

The control of NDs surface charge is essential for electrostatic loading of drugs [27, 31]. Moreover, it greatly influences the interaction with cell membrane and the internalization pathways of NDs into cells [166]. The control of surface terminations allows a tuning of NDs surface charge in suspension from negative to positive versus pH. Colloidal properties of modified nanodiamonds in water will be now presented (Part 4.5). The link between their Zeta potential and their surface electronic properties will be investigated in details.

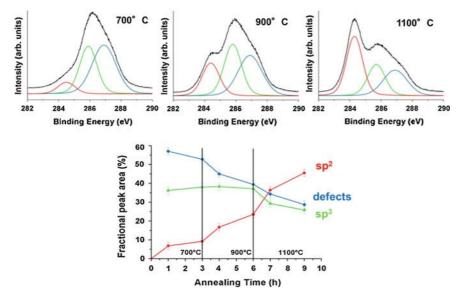


Fig. 4.7 C1s XPS spectra of NDs after sequential annealing treatments of 3 h at 700, 900, 1100 °C. Fitting components CI (*red*), CII (*green*), CIII (*blue*), related to sp² carbon-carbon bonds, sp³ carbon-carbon bonds, and defects, respectively, are plotted under the experimental curves. Evolution of the fractional peak areas versus annealing time [55]

4.5 Colloidal Properties of Modified Nanodiamonds

The colloidal stability of nanodiamonds in biologic media is of major interest for their biomedical applications. It strongly depends on their surface charge which controls hydrophilic/hydrophobic interactions with other NDs and moieties. Moreover, in the biological media, proteins adsorption leads to the formation of a corona [167] surrounding nanoparticles. The nature of proteins forming the corona is a key parameter for cell internalization pathways which occur by their specific recognition. In liquid media, nanodiamonds tend to form aggregates, this may be favored by opposite charged facets [168] or by graphitic carbon present at NDs surface [69, 128]. Several experimental methods were developed to reduce aggregation and obtain monodisperse colloidal suspensions. Bead assisted sonic disintegration (BASD) was performed using ceramic microbeads, nevertheless, it can produce chemical contaminations or surface graphitization [84, 169, 170]. Dry milling using salts and sugars avoiding contamination was also reported [171]. Modifications in NDs surface chemistry like plasma hydrogenation [86], surface graphitization [99], oxidation in air [85] or borane reduction [128] have also proved their efficiency for deaggregation.

The behavior of nanoparticles in liquid media could be investigated by Dynamic Light Scattering (DLS). From diffused intensity signal, this technique provides the measurement of the hydrodynamic diameter of stable NPs. This diameter is higher than the real one because it includes the solvated layer of adsorbed ions which is solvent and ionic force dependent [172]. The Zeta Potential (ZP) is defined from the double layer model (Stern and Gouy-Chapman layers) which describes the counterions adsorption. ZP is the potential at the surface of the electrical double layer called slipping plane. It is obviously related to the surface charge of NDs, very different values are obtained versus surface chemistry and size [24]. Zeta Potential is calculated from the electrophoretic mobility measured in DLS. Its evolution versus pH must be investigated. Indeed, functional groups present at NDs surface can switch from protonated to deprotonated states according to their pKa value. The colloidal stability will be then dependent on pH value. Indeed, the ZP governs the electrostatic repulsive interactions in liquid medium. Stable NPs in colloids usually exhibit ZP potential absolute value higher than 30 mV [173, 174].

The control of ND surface charge is essential for drug or biological moieties adsorption [27, 29, 31] it can also play a major role in internalization pathways and interactions with negatively charged cell membrane [166]. In this part, the surface reactivity of modified NDs will be discussed. Especially, stable NDs in aqueous suspensions exhibiting positive or negative ZP will be considered, mechanisms such as charge transfer leading to these specific charges will be also discussed.

4.5.1 Surface Reactivity of Modified NDs

The hydrophilicity of detonation nanodiamonds from the same origin was investigated versus their surface chemistry after plasma hydrogenation, air oxidation and surface graphitization [86]. Water adsorption isotherms measured by BET revealed a significantly higher hydrophilicity for H-NDs and sp²-NDs compared to COOH-NDs. Taking into account the specific surface area and assuming a monomolecular adsorption, hydrophilic sites were estimated to 2.2, 1.7 and 1.4 sites/nm² for sp²-NDs, H-NDs and COOH-NDs, respectively. These hydrophilic sites on H-NDs and sp²-NDs are likely to facilitate the dispersion in water of otherwise hydrophobic NDs (Part 4.5.2).

This high affinity of hydrogenated and surface graphitized NDs with oxygen is supported by XPS analysis performed after air exposure. Oxygen atomic concentrations of 6.5 at% and 4.0 at% were measured for sp²-NDs and H-NDs after air exposure [86] while it was completely removed after plasma hydrogenation or annealing under vacuum (Part 4.4) [55, 70].

4.5.2 Solubility, Stability in Colloids

Stable suspensions of individual detonation NDs in water were achieved by successive sonication and centrifugation [88, 92]. Sonication allows the breaking of Van der Waals interactions between aggregated NDs. This is an important issue for biomedical applications because detonation nanodiamonds (<10 nm) are expected to be easily eliminated by kidney (Part 4.6) [21]. The colloidal stability is closely related to surface charge of NDs because of electrostatic repulsive interactions. Indeed, these interactions can also promote aggregation as discussed previously. Among the literature, positive [129, 98] or negative [45, 88, 175–177] ZP were reported for nanodiamonds in suspension depending on their surface chemistry. These colloidal properties constitute an important asset for nanodiamonds compared to other carbon nanomaterials. Fullerenes or carbon nanotubes exhibit hydrophobic behavior and their stability in water suspension can only be obtained after their functionalization [178–180]. These colloids of NDs in water were also used to obtain homogeneous seeding of substrates for CVD growth of thin diamond films [181, 182]. Some groups used surfactants like sodium oleate [98] or oleylamine [183] to improve the colloidal stability of NDs. Nevertheless, aqueous suspensions are more suitable for bioapplications such as drug delivery.

4.5.3 Negatively Charged NDs

Carboxylated nanodiamonds obtained after annealing in air (Part 4.4.2) exhibit a strong negative Zeta potential included between -30 and -50 mV when dispersed in water for pH > 5 [45, 75]. The origin of this negative potential is related to the presence at the surface of carboxylate groups (COO⁻) the basic form of COOH. Below pH = 5, the formation of mainly carboxylic groups (COOH) leads to a drop of Zeta potential. This threshold pH value is close to the pKa of COOH/COO⁻ couple [184].

Several studies [69, 88, 185] reported a similar behavior i.e. negative ZP for pristine NDs. This has been explained by the presence of carboxylate groups at the NDs surface formed during purification treatments (Part 4.4).

4.5.4 Positively Charged NDs

Other "as received" nanodiamonds from different suppliers were stabilized inwater suspensions exhibiting a positive Zeta potential (> + 30 mV) [45, 175]. The origin of this positive ZP is not yet clearly understood. In the literature, colloidal aqueous suspensions of positively charged NDs (ZP > 30 mV) were also reported after different surface modifications: reduction with borane or LiAlH₄ [18, 129], UV hydroxylation [93], annealing under hydrogen flow [88], plasma hydrogenation [86] or surface graphitization [75]. These charge modifications may be related to graphitic carbon surrounding NDs core or to protonation of some chemical functional groups or lastly to intrinsic properties of hydrogenated diamond surface. Different likely mechanisms have been previously proposed in the literature and will be now discussed.

4.5.5 Presence of Graphitic Carbon at the NDs Surface

Gibson et al. reported positive Zeta potential for detonation nanodiamonds after two oxidization treatments CrO_2/H_2SO_4 followed by NaOH/H₂O₂ [177]. Authors attributed this positive ZP to electrostatic interactions between pyrone groups and π -electrons from sp²-carbon present at NDs surface corresponding to protonation of pyrone structures. Theoretical investigations supported this effect leading to carbon basicity [186, 187].

Williams et al. obtained a positive Zeta potential for detonation NDs exposed to hydrogen flow at 500 °C then dispersed into water solution [88]. Authors claimed it originates from the interaction of OLC structures (π bonding) remaining at NDs surface with oxonium ions (H_3O^+) present in water suspension at acidic pH. Their HRTEM pictures support the presence of OLC on hydrogen treated NDs. By analogy with carbon black, authors proposed an electron-donor-acceptor complex may be formed [188]. In both previous studies, a charge exchange between species from aqueous solution and sp² carbon remaining at NDs surface is involved.

More recently, positive Zeta potential was measured at pH < 8 on detonation NDs on which FLRs were intentionally generated by annealing under vacuum before their dispersion into water (Fig. 4.8) [75]. The origin of the positive charge was linked to the formation of endoperoxide groups by cycloaddition of O₂ molecules on FLRs. BET and XPS measurements well emphasize the high reactivity of sp²-NDs toward oxygen species after air exposure [86]. In situ XPS analysis after UHV annealing suggests a reversible oxidation similar to the one observed on graphene [189]. Moreover, after in situ desorption, the FTIR spectrum exhibits an intense band at 1100 cm⁻¹ well corresponding to C–O stretching of ether or epoxy groups [190]. This last result supports the formation of endoperoxide groups on FLRs structures. The protonation of these endoperoxides may be the explanation for oxygen hole doping [75].

4.5.5.1 Chemically Induced Positive ZP

Ozawa et al. reported on a positive Zeta potential switching after hydroxylation of carboxylated detonation NDs by a borane reduction [191]. Hydroxyl groups probed on their NDs by FTIR measurements, allowed further silanization coupling. Shenderova et al. obtained positive ZP for detonation NDs initially carboxylated after LiAlH₄ reduction [129]. Girard et al. also observed a ZP inversion (+30 mV at pH = 7 compared to -45 mV initially) after UV photochemical treatment of HPHT nanodiamonds [93]. Hydroxyl groups were identified from FTIR and XPS investigations. Nevertheless, any previous work provides an explanation for this positive charge and no link between positive ZP and hydroxylation has been currently established.

Finally, positive ZP potential (25 mV for pH between 2 and 3) was measured for nanodiamonds grafted with ethylenediamine (EDA) [176]. According to authors, protonation of amino groups may explain the basic properties in acidic environment.

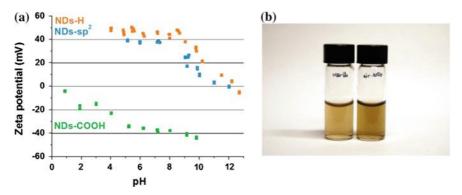


Fig. 4.8 a Zeta potential of H-NDs, COOH-NDs and sp²-NDs dispersed in water versus pH b Colloidal suspensions of sp²-NDs and COOH-NDs in water

4.5.5.2 Specific Surface Properties of Hydrogenated Nanodiamonds

Stable aqueous suspensions of detonation nanodiamonds previously plasma hydrogenated were recently reported [86]. Measured ZP is included between +45 and -50 mV for pH values in the range of 4–9 (Fig. 4.8). The experiment was carried out on initially negatively or positively charged NDs. According to authors, a transfer doping occurs onto 5 nm diamond nanoparticles suspended in water based on the semi-conductive behavior of the NDs core. These experiments show that an electrochemical exchange between residual CO_2 takes place between aqueous solution and hydrogenated diamond surface leading to holes accumulation similarly to the one observed for micro-sized diamond particles [192]. The key role played by oxygen was underlined by nitrogen flushing which provokes flocculation of H-NDs. A transfer doping similar to the one occurring on diamond surfaces also occurs onto 5 nm NDs suspended in water [86]. This effect was not obvious at nanoscale. Indeed, the band bending inducing the hole accumulation layer is of the same order of magnitude than the particle diameter. This transfer doping at the hydrogenated ND surface was recently confirmed by resistivity measurements [193]. Resistivity drops from 10^7 to 10^5 Ω cm for detonation nanodiamonds heated in hydrogen gas $(600-900\ ^{\circ}C)$. Oxidation of the hydrogenated sample at 300 $^{\circ}C$ recovers resistivity to its original value.

To conclude, Zeta potential of nanodiamonds can be tuned over a wide pH range applying different surface modifications. This behaviour is highly suitable for bioapplications. According to the literature, ZP switching can be explained by charge exchanges between NDs surface and species from aqueous suspensions. Some works emphasized the role played by sp² carbon at NDs surface. Other studies were performed using surface cleaned nanodiamonds i.e. Fenton reaction which clearly consumes a part of NDs or plasma hydrogenation. In the latter case, positive Zeta potential of H-NDs is due to specific surface properties of hydrogenated diamond surface. Neither chemical functional groups nor graphitic structures seem to be involved. This suggests mechanisms of different natures could explain this positive surface charge.

4.6 Nanodiamonds and Biomedical Applications

4.6.1 NDs Assets

Nanodiamonds (NDs) combined many required assets for biomedical applications: extremely low cytoxicity and genotoxicity, carbon chemistry allowing covalent grafting, photoluminescent color centers, tunable size (down to 5 nm). Let us discuss these advantages.

4.6.1.1 NDs Toxicity and Biodistribution

Several long-term in vitro [22, 194] and in vivo [195–197] previous studies demonstrated that NDs are non-cytotoxic, better tolerated by cells than other nanocarbon materials [198]. NDs introduced into *Caenorhabditis elegans* worm did not cause any detectable stress to worms [198]. In mice, intravenous injection of 50 nm NDs led to long-term entrapment in the liver and the lung [195], but no mice showed any abnormal symptoms. Similarly, subcutaneous and intraperitoneal injection of 100 nm NDs in rats led to accumulation of NDs in retention organs over 1 month with no impact on the rats' viability [197]. However, a slight surface-dependent genotoxicity of NDs was recently reported on embryonic stem cells [199].

Until now, no extensive study combining cytotoxicity and genotoxicity of NDs was reported. A recent study [25] focuses on in vitro cytotoxicity and genotoxicity of HPHT NDs perfectly characterized in terms of size (20 nm and 100 nm) and surface chemistry. The cellular induced effects of two sets of NDs were investigated in six human cell lines: HepG2 and Hep3B (liver), Caki-1 and Hek-293 (kidney), HT29 (intestine) and A549 (lung). The screening of ND cytotoxicity was carried out by measuring cell impedance (xCELLigence). This technique permits real-time monitoring of NPs effects on cell morphology, proliferation, adhesion and membrane potential [200]. Flow cytometry allowed the discrimination of viable cells, containing or not NDs, and dead cells. Finally, using immunofluorescence detection of nuclear γ-H2Ax foci, considered the most sensitive method for detecting DNA double-strand breaks [201], genotoxicity was also analyzed. According to the results, NDs do not induce any significant toxic effect on the six cell lines up to an exposure dose of 250 mg/mL (Fig. 4.9). As a comparison, 25 nm SiO₂ NPs are 9 times more genotoxic at a 10 times lower concentration [25].

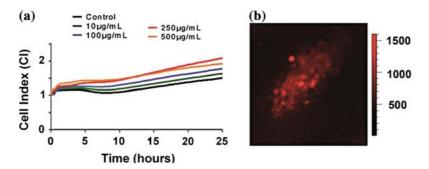


Fig. 4.9 a Cell index real-time monitoring of HepG2 cells exposed 24 h to 20 nm NDs; b Raman/ photoluminescence image on a HepG2 cell [25]

4.6.1.2 Grafting of Biological Moieties

We have previously underlined surface charge of NDs suspended in aqueous solution can be tuned over a wide pH range (Part 4.5) thanks to specific surface modifications (Part 4.4). Efficient surface treatments also permit to conjugate ND surfaces with fluorescent molecules [202–205], with DNA [206], siRNA [27], proteins [207, 208], lysozymes [208], growth hormones [209], antibodies [210, 211], anti-cancer drugs [30–32], as well as with dopamine derivatives [212].

Either covalent or non-covalent graftings were performed depending on expected bio-applications. These attempts have been recently reviewed in details [28]. The non-covalent grafting of large biomolecules based on electrostatic interactions was performed from well controlled NDs surface charge (Part 4.5). Cytochrome c [98] and bovine insulin [213] were immobilized on nanodiamonds. Furthermore, lysozymes [34, 214, 215] and apoobelin [216] were attached. Anticancer drugs like doxorubicine [217, 218], paclitaxel [30] or HCPT [32] were non-covalently adsorbed and then delivered. Some polymer coatings like polyethylenimine (PEI) have been also carried out on NDs to confer cationic surface charge for siRNA adsorption and delivery for gene therapy (Ewing's sarcoma) [27, 29, 206, 219]. In most cases, the functionality of biomolecules is preserved [214, 220] although significantly less efficiency has sometimes been observed. Cumulative effects were even obtained as lysozyme adsorption is combined with cytochrome c [215]. Depending on the functionality of the biologic unit, covalent grafting via organic chemistry could be suitable to get a specific binding site on the ND or to avoid modifications of the biomolecule conformation. Grafting routes which may involve a linker molecule could be performed from different surface terminations of NDs [28].

4.6.1.3 Photoluminescent NV Centers

Color centers could be efficiently generated in HPHT nanodiamonds by high-energy particle beam irradiation (He⁺ or electrons) leading to vacancy creation [124, 203]. During annealing, vacancies combined with nitrogen atoms to form NV centers which could be characterized by Photoluminescence (Part 4.3). HPHT NDs exhibited stable photoluminescence without bleaching or blinking in the red and near-infrared region corresponding to transmission window of tissues for size down to 7 nm [221]. These properties allowed the long-term tracking of fluorescent NDs into cells localized at subcellular scale by fluorescence and TEM microscopies. Expected applications concern biomedical imaging [203] and fluorescence energy transfer (FRET) with other fluorophores [222]. For detonation NDs, the limited nitrogen amount, the nanometric size (<5 nm) and structural defects present at the surface and in the diamond core prevent the observation of stable photoluminescence [65, 68].

4.6.1.4 Tunable Size

Nanodiamonds are scalable nanoparticles down to few nanometers [92]. For smallest NDs, in vivo clearance or kidney elimination could be expected according to the size of kidney capillarities. This constitutes an essential advantage for drug delivery or biomarkers applications. In addition, size effects on photoluminescence were reported which tune the wavelength of emitted photons [223].

4.6.2 Some Current Challenges

According to the previous assets, several challenges appear highly relevant and promising for the development of future bio-applications involving nanodiamonds: their labeling, their use for safer by design or their developments as a multifunctional platform for drug delivery.

4.6.2.1 Labeling

Labeling is inescapable for biodistribution and pharmacokinetics studies, for development of biomarkers or targeted drug delivery. Although NV centers are promising for in vitro imaging applications [203, 224, 225], their limited photoluminescence intensity is an obstacle to their detection in the body. In the literature, several groups reported the efficient tracking of NDs into cells by grafting of fluorophores [24, 98, 202, 226]. Bright blue luminescence was even obtained by octadecylamine covalently attached to NDs-COOH [227]. The main stake concerns the stability of fluorophores in cellular environment. An interesting alternative for tracking consists to combine NDs with contrast agents currently used for magnetic resonance imaging (MRI). The rich surface chemistry of NDs (Part 4.4) allows the covalent grafting of amine-functionalized Gd(III). Gd(III)-NDs conjugates exhibited ten times higher relaxivity and can be well detected by MRI [228]. Finally, contrary to carbon nanotubes [9], any radioactive labeling has been yet reported for nanodiamonds. This constitutes a major challenge because radioactive imaging is ordinarily used for biodistribution and pharmacokinetics studies of NPs.

4.6.2.2 Safer by Design

An important issue for drug delivery concerns the conception of safer nanoparticles [229, 230]. For this purpose, NPs physicochemical determinants such as size, shape, composition, surface chemistry or coating nature which may be critical for cytotoxic and genotoxic effects must be investigated. In vitro screening for assessing the possible reactivity, biomarkers of inflammation and cellular uptake can provide precious indications to chemists and

physicists to design "safe" nanomaterials. According to their extremely weak in vitro toxicity [25], nanodiamonds constitute excellent candidates to study the specific effects of physical and chemical parameters on toxicity.

4.6.2.3 Multifunctional Platform for Drug Delivery

Nanodiamonds combine biocompatibility, scalability and stability in aqueous solution (Part 4.5). Moreover, several studies reported their ability to carry and deliver different classes of drugs [30–32] or nucleic acids like siRNA [26, 27, 29, 219]. To build such platform, a stable labeling is required to allow tracking of NDs. Another important issue is related to the specific targeting of proteins or DNA. A recent study

[184] reports on 20 nm HPHT NDs covalently grafted with peptide nucleic acids (PNA). The original functionalization route is based on an optimized amidation of ND carboxylic acids groups. ND-PNA conjugates were validated through a successful recognition of complementary DNA in a mixture, showing their efficiency toward nucleic acid detection (Fig. 4.10). Such nucleic acid functionalized NDs open the way to a wide range of biomedical applications towards genetic diseases, genomic research or early cancer diagnosis.

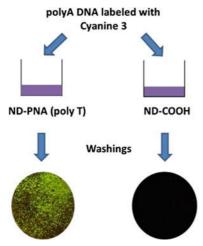


Fig. 4.10 Fluorescence of ND-PNA conjugates compared to ND-COOH after DNA hybridization observed at λ_{exc} = 550 nm and λ_{em} = 570 nm [184]

4.7 Conclusion

Nanodiamonds can behave very different surface properties according to their tunable surface chemistry. The current state of art of controlled surface terminations was summarized in this chapter. The link between surface chemistry, surface charge and colloidal properties of NDs was particularly emphasized. Indeed, it appears highly relevant for biomedical applications because it is a key parameter for drug adsorption or interactions with cell membrane. Among the potential surface chemistries, hydroxyl and halogen terminations (F, Cl, Br) offer specific reactivity. However, experimental protocols must be further improved to reach a higher selectivity.

Some NDs characteristics related to the diamond core remain also partially understood at the present state. In particular, it is essential to further investigate the impact of structural defects or chemical impurities on NDs properties. The presence of different crystallographic facets on HPHT and some detonation NDs should also significantly influence their surface reactivity. Moreover, consequences of hydrogen incorporated in nanodiamonds especially during detonation synthesis (Part 4.2) must be better understood. According to previous works performed on bulk diamond, hydrogen should be trapped by structural defects or chemical impurities [56]. Among further main challenges for biomedecine, the radioactive labeling appears as a priority to investigate in vivo the biodistribution and pharmacokinetics on NDs. Such achievement has not been yet reported. Mechanisms of drug adsorption/desorption on NDs must be deeper understood to achieve a controlled release. These phenomena are highly sensitive to size and surface chemistry. The combination of NDs with NV centers can permit novel developments in biology like the probing of ion channels in cells or the neuron imaging. Finally, the synthesis of hybrid NDs is also an essential challenge. For example, surface graphitized NDs may possess surface properties linked to fullerene or graphene assets required for radiosensitization or photothermal therapy. In conclusion, according to their multiple assets, nanodiamonds constitute excellent candidates to be used as active NPs with therapeutic behaviors for biomedical applications.

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